

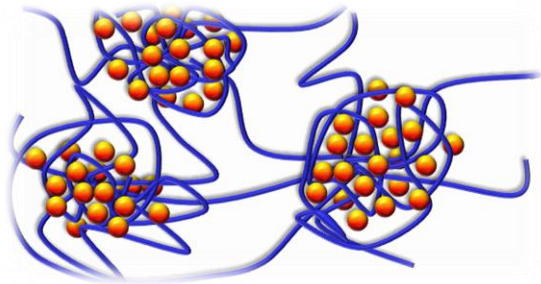
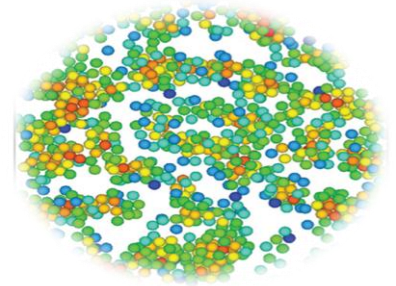
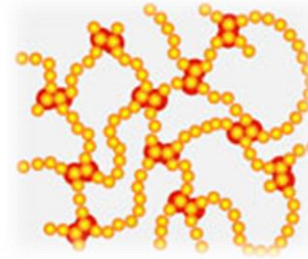
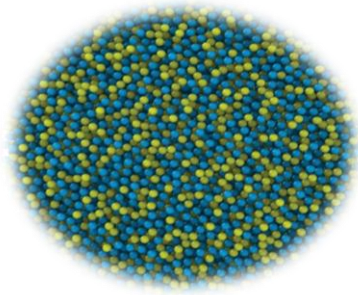
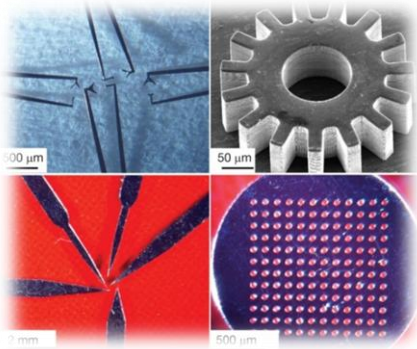
Atomic motion in glasses studied with coherent X-rays

Beatrice Ruta
ESRF – Grenoble, France

- ❑ Glassy systems
- ❑ X-ray Photon Correlation Spectroscopy
- ❑ Aging in metallic glasses
- ❑ Measurements *in operando* conditions
- ❑ Atomic dynamics in network & oxide glasses
- ❑ Conclusions and future perspectives

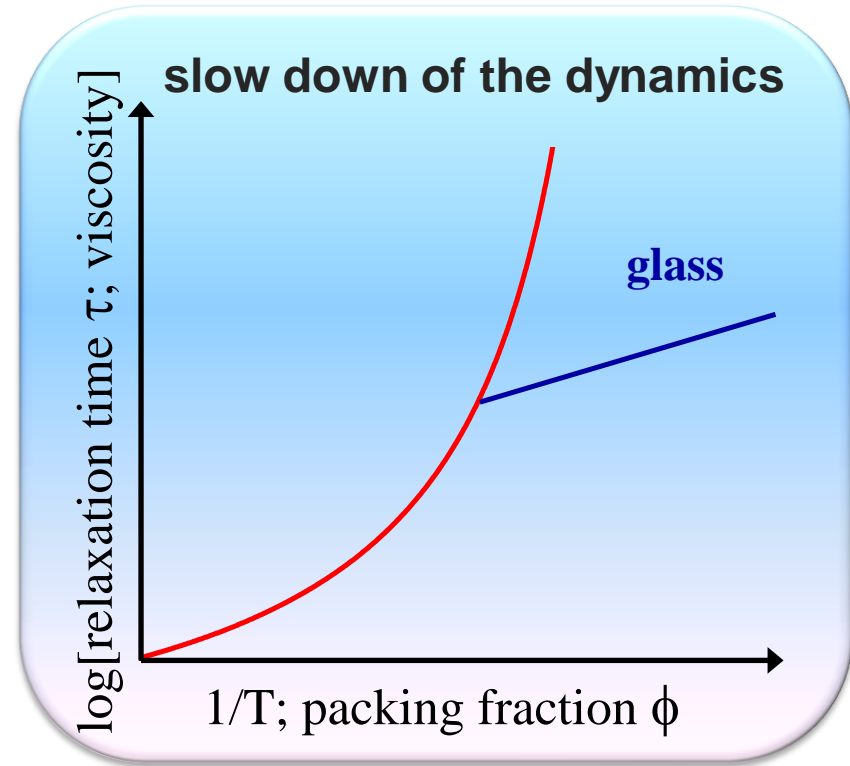
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Amorphous materials ...



... which can be driven in an arrested state ...

- **glasses** → by rapidly cooling the liquid
- **colloidal suspensions** → by increasing the packing fraction
- **granular materials** → by applying a shear
- **chemical gels** → spontaneously with time
- **physical gels** → by changing external parameters (i.e. temperature, pressure..)



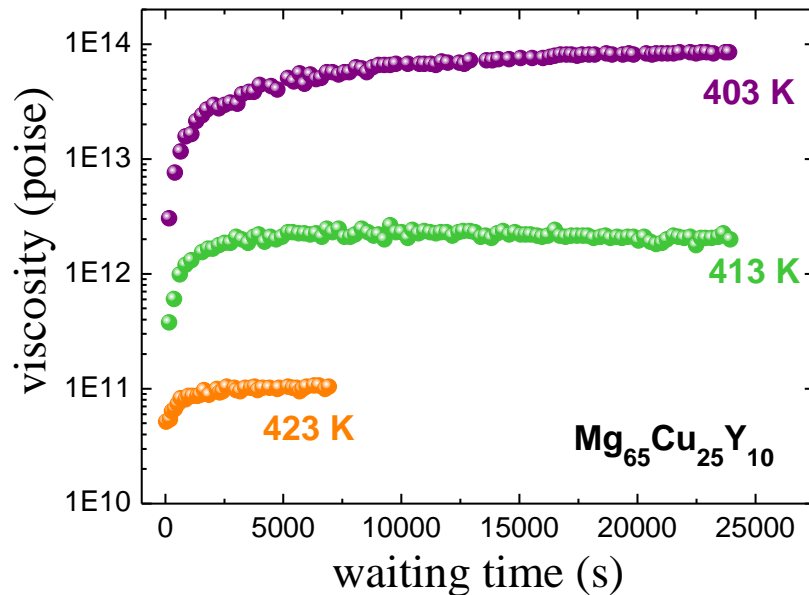
... where they display physical aging

→ every physical observables spontaneously evolve with time

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Temporal evolution

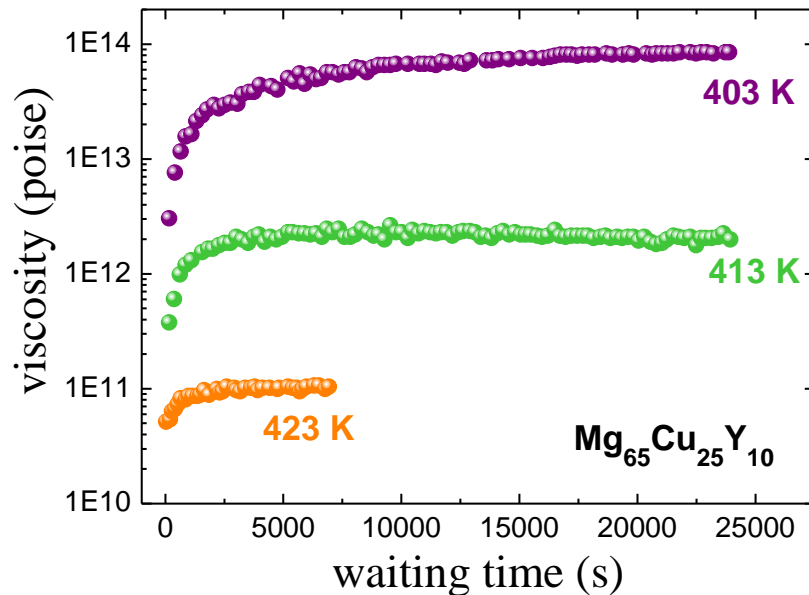


Busch et al. J. App. Phys. (1998)

... where they display physical aging

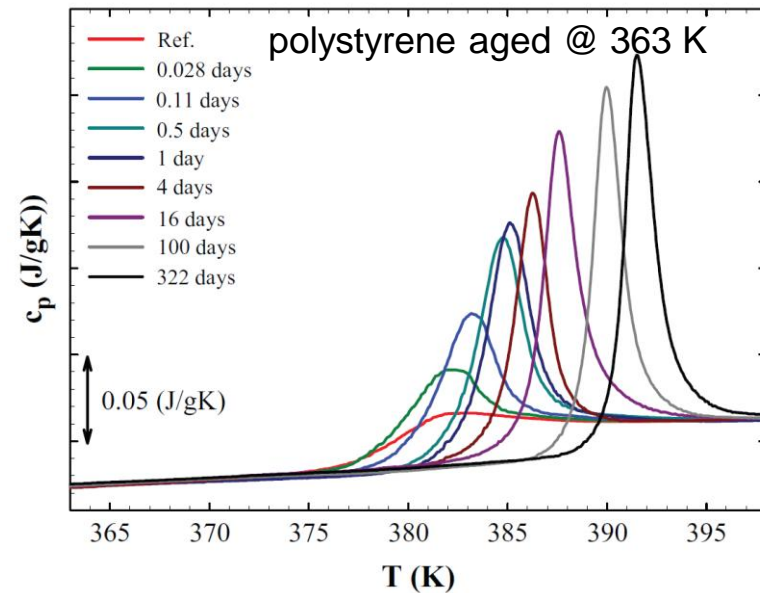
→ every physical observables spontaneously evolve with time

Temporal evolution



Busch et al. J. App. Phys. (1998)

Thermal history dependence

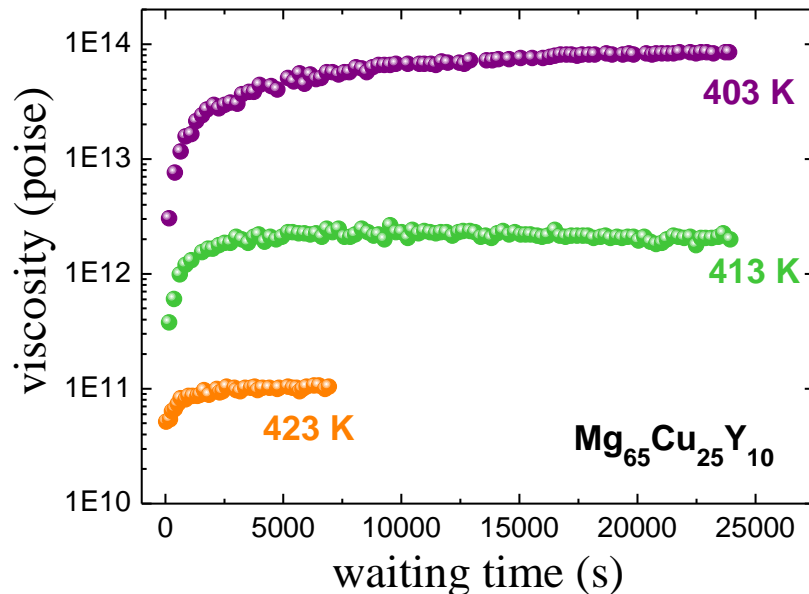


Cangialosi et al. Phys. Rev. Lett. (2012)

... where they display physical aging

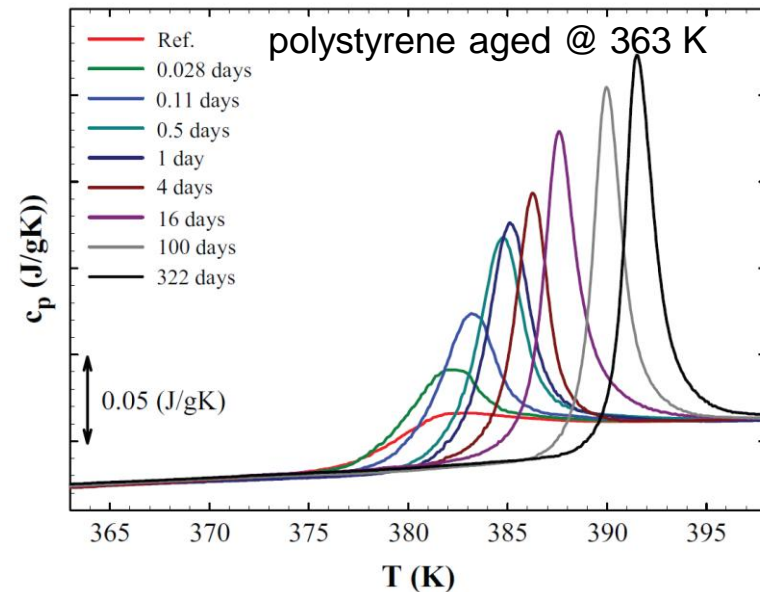
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Temporal evolution



Busch et al. J. App. Phys. (1998)

Thermal history dependence

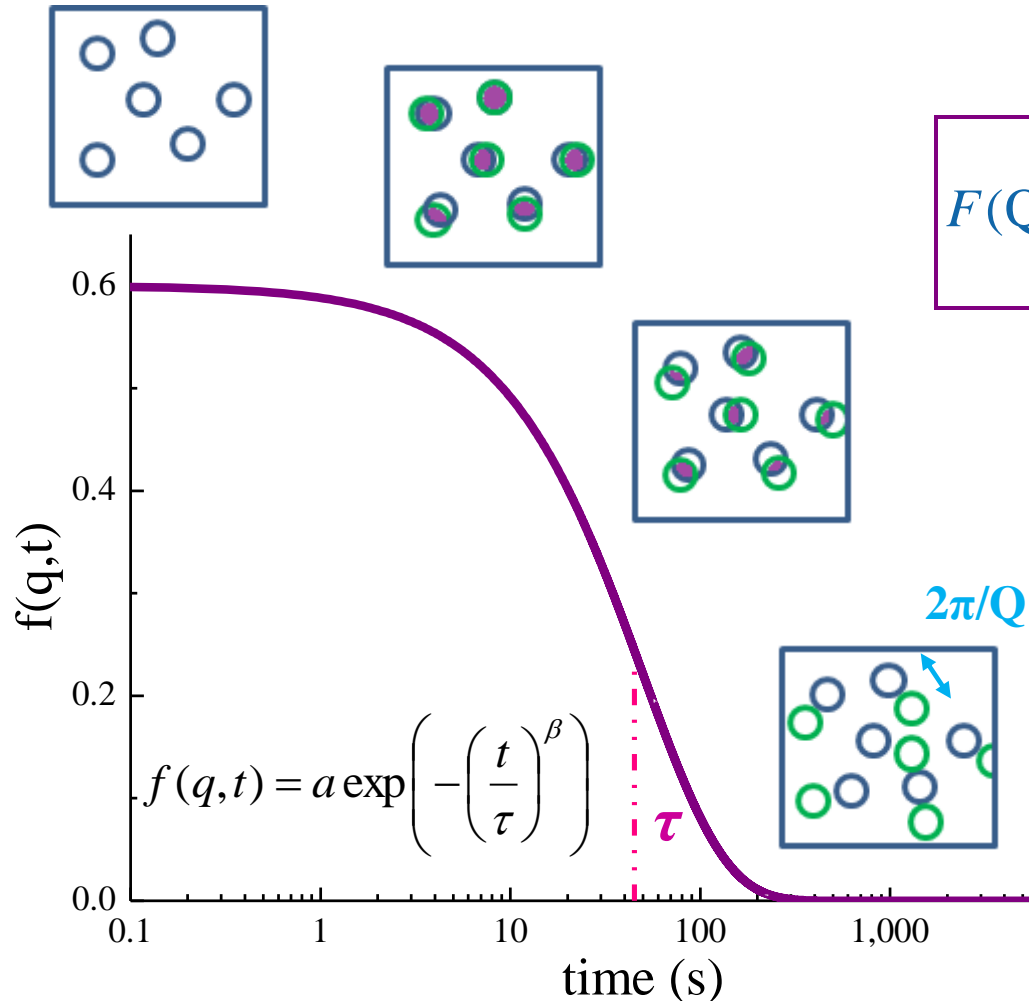


Cangialosi et al. Phys. Rev. Lett. (2012)

Understanding and controlling the physical properties requires information on the ongoing relaxation processes at the microscopic scale

RELAXATION DYNAMICS

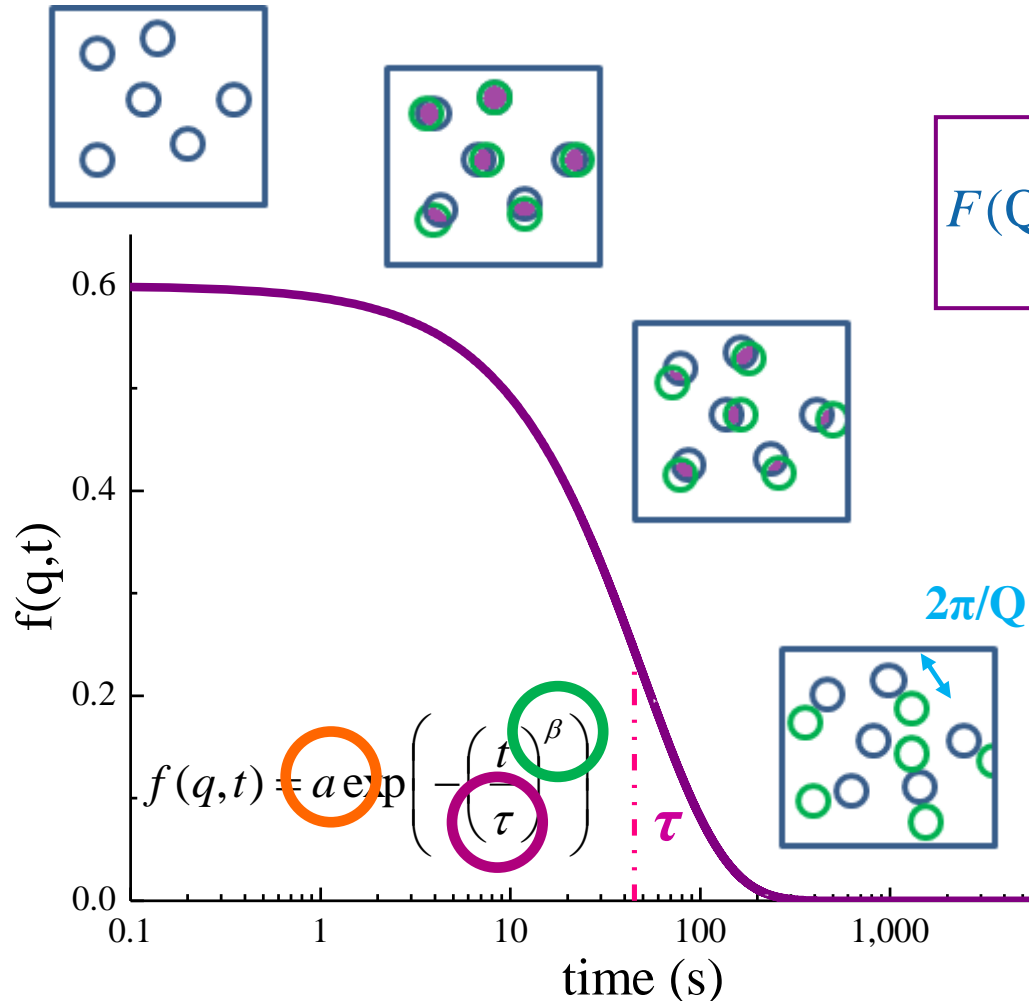
Information on the relaxation dynamics can be obtained from the decay of the **intermediate scattering function** on a scale $2\pi/Q$



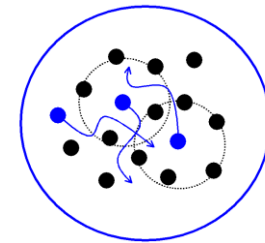
$$F(Q,t) = \frac{S(Q,t)}{S(Q)} = \frac{\langle \delta\rho_Q^*(0)\delta\rho_Q(t) \rangle}{\langle \delta\rho_Q^*(0)\delta\rho_Q(0) \rangle}$$

RELAXATION DYNAMICS

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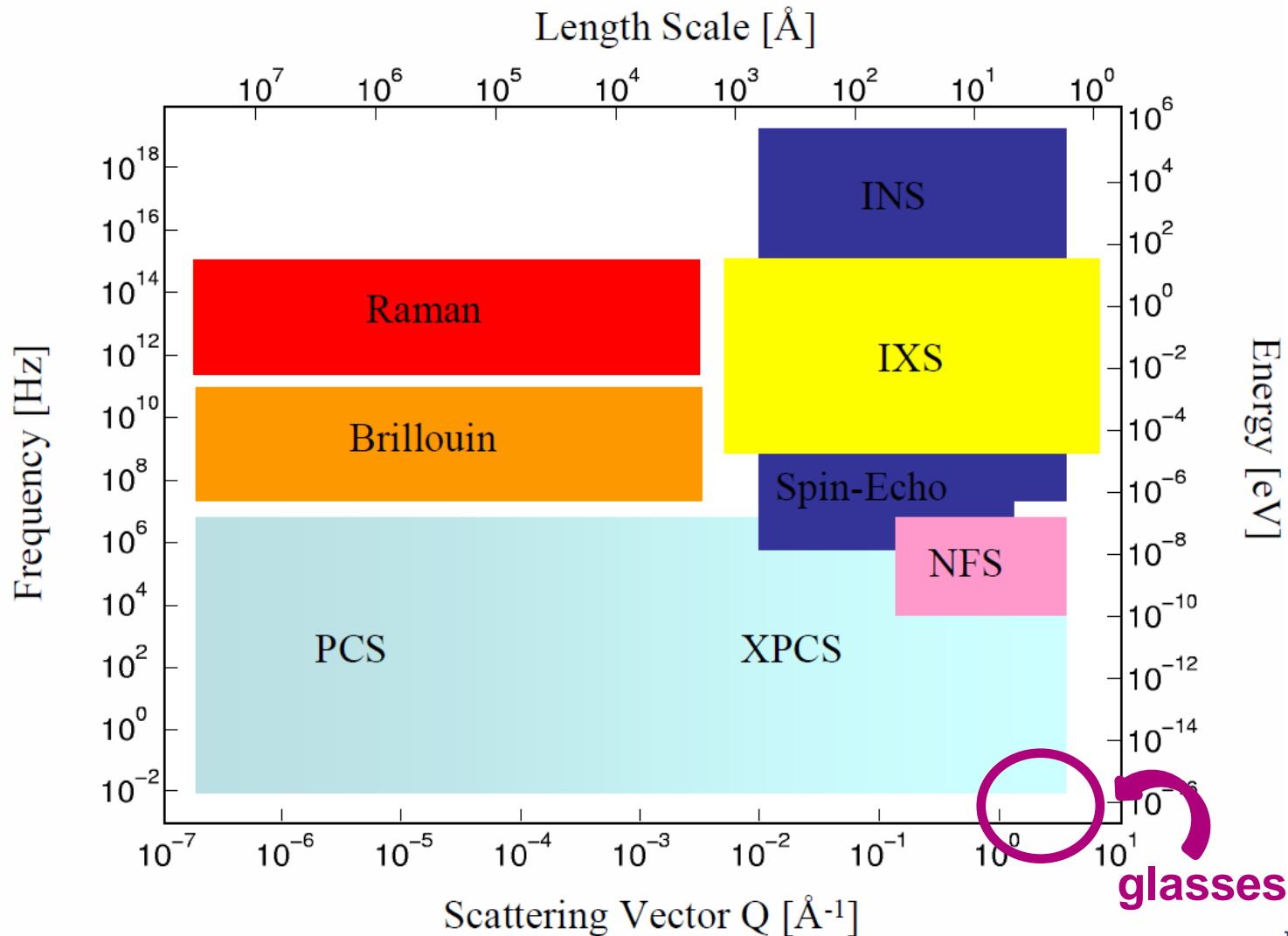
τ = time for structural rearrangements (on $\sim 2/3$ Å in the case of glasses)

How can we measure the relaxation dynamics in a glass?

- Glassy systems
- X-ray Photon Correlation Spectroscopy
- Aging in metallic glasses
- Measurements *in operando* conditions
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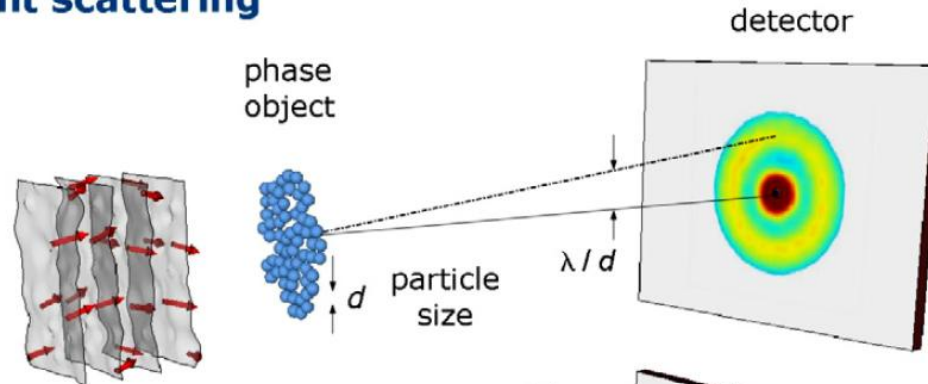
X-RAY PHOTON CORRELATION SPECTROSCOPY

X-ray Photon Correlation Spectroscopy is the **only technique** offering the potential to unravel the dynamics of glass formers around and below T_g ($\tau \sim 1-10^4$ s, $Q \sim 0.3-4 \text{ \AA}^{-1}$)

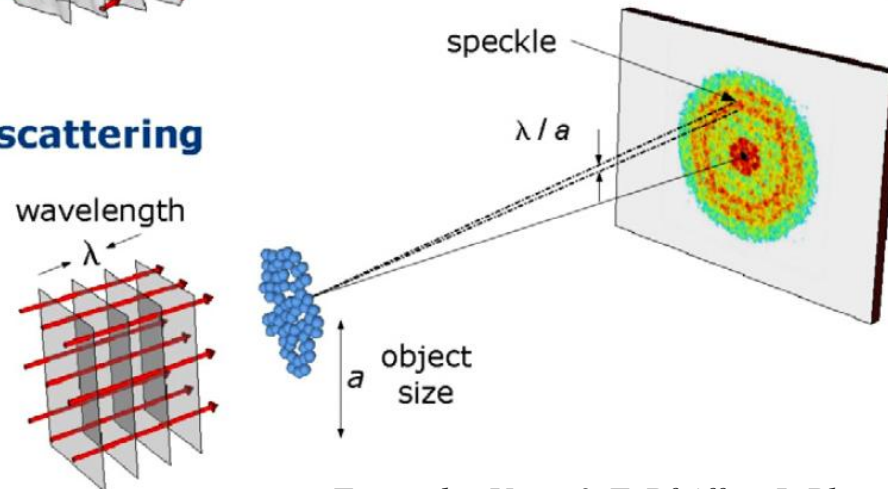


XPCS uses the partial coherent properties of X-rays at 3rd generation synchrotrons

Incoherent scattering



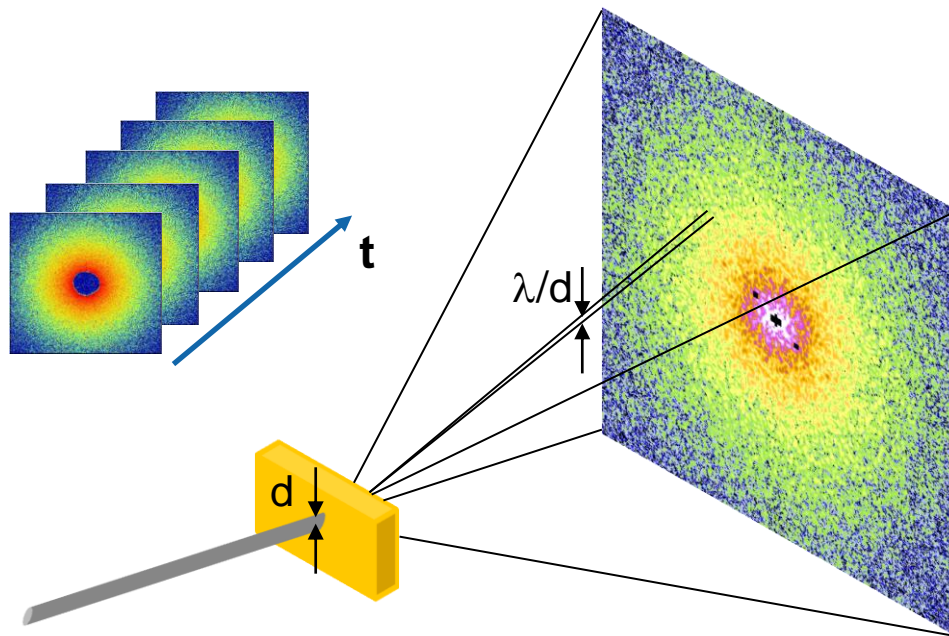
Coherent scattering



F. van der Veen & F. Pfeiffer, J. Phys. Cond. Mat. (1998)

X-RAY PHOTON CORRELATION SPECTROSCOPY

Information on the dynamics can be obtained by measuring a series of speckles patterns and quantifying **temporal correlations of intensity fluctuations** at a given wave-vector q



The intensity of the speckles is related to the **exact spatial arrangement** of the scatters inside the system

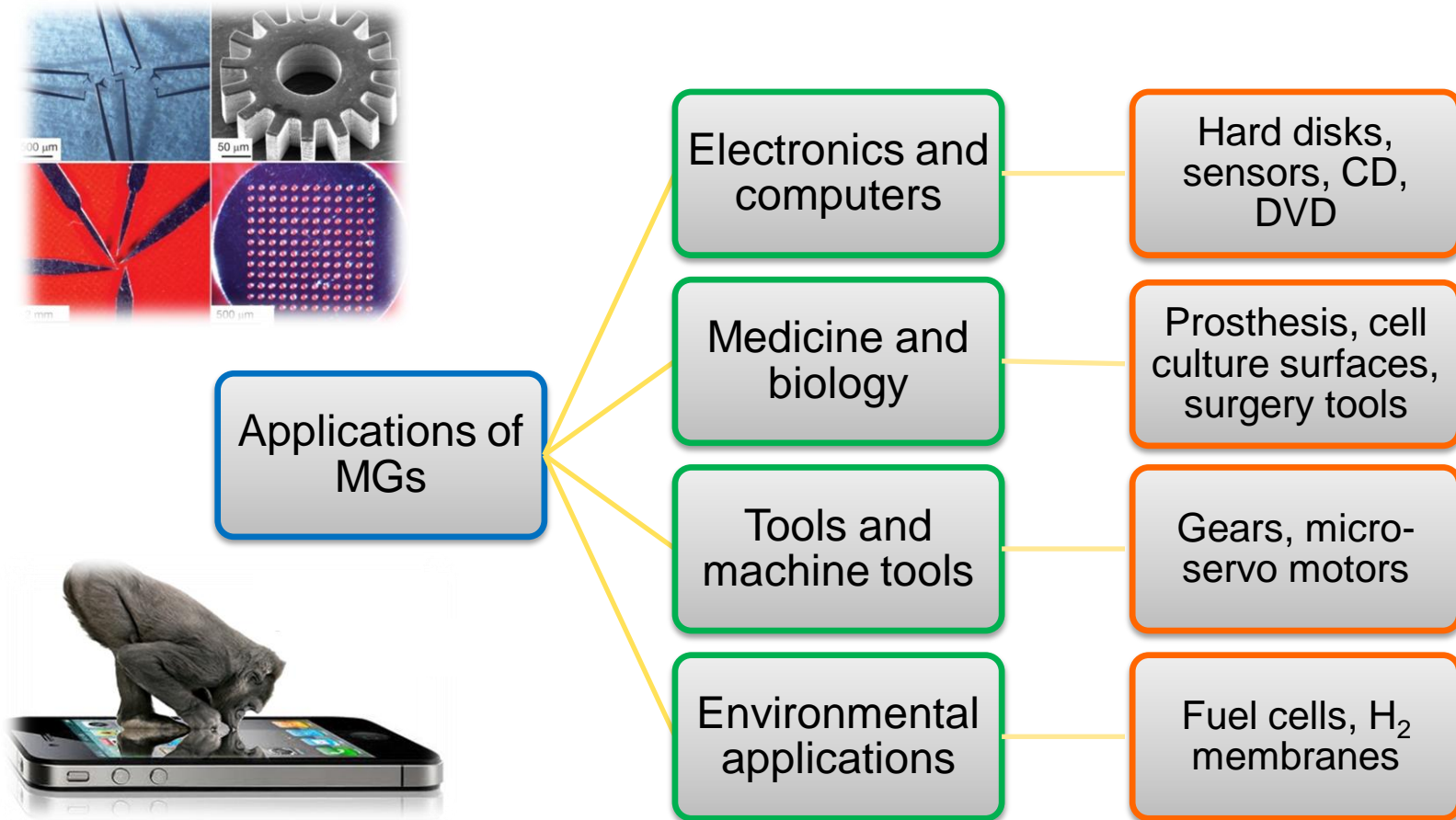
$$I(Q, t) \propto \left| \sum_n f_n(Q) \cdot e^{iQ \cdot r_n(t)} \right|^2$$

Information on the microscopic dynamics

- Glassy systems
- X-ray Photon Correlation Spectroscopy
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METALLIC GLASSES

MGs have outstanding properties and carry the promise of many applications.



Their widespread use is **however** blocked by the **difficulty of avoiding aging and crystallization** during processing (particularly strong for MGs)

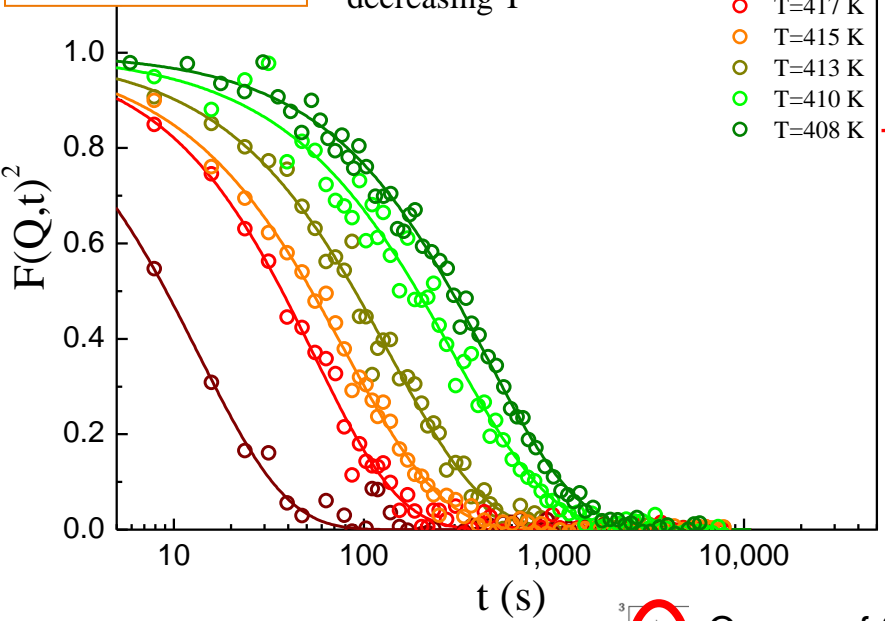
FIRST INVESTIGATION OF ATOMIC DYNAMICS IN MGs

$Mg_{65}Cu_{25}Y_{10}$

decreasing T

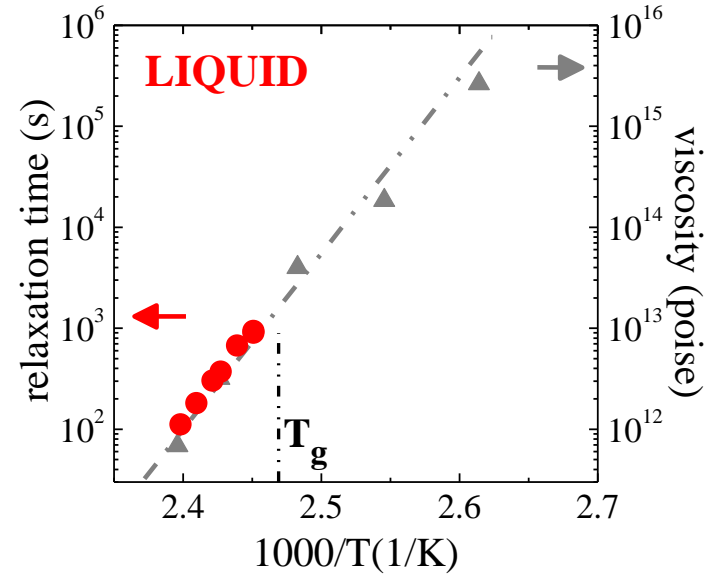
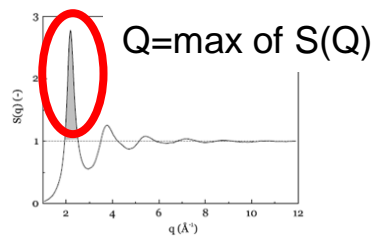
- T=418 K
- T=417 K
- T=415 K
- T=413 K
- T=410 K
- T=408 K

liquid

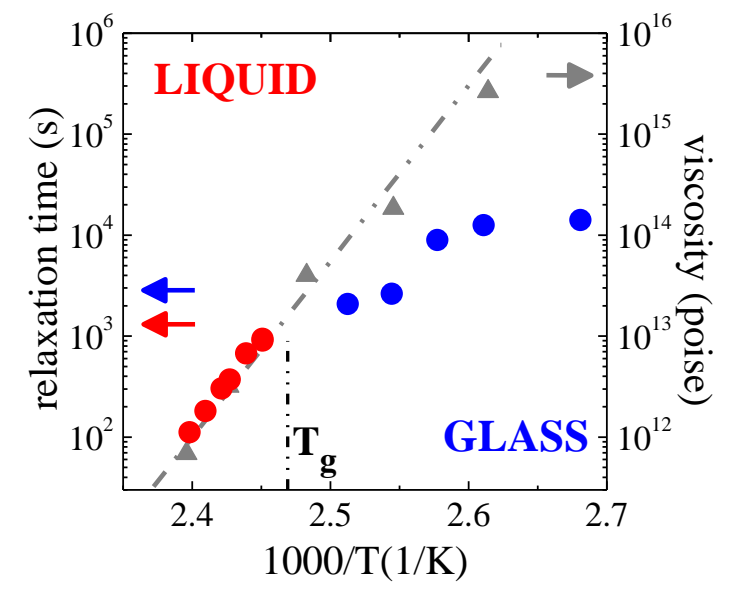
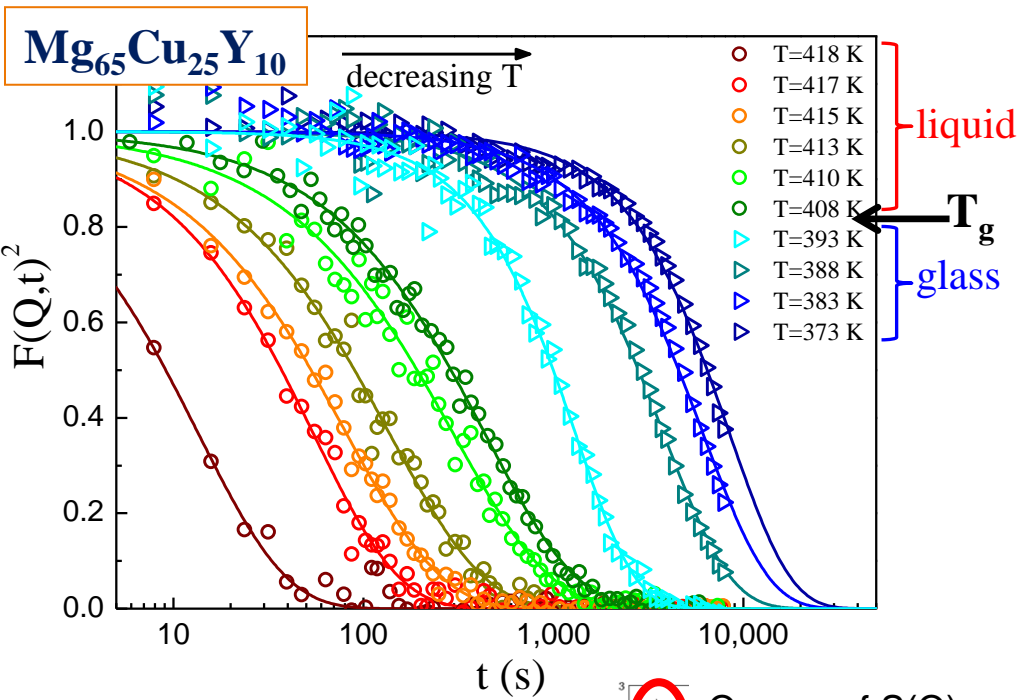


Fit model: KWW

$$F(Q,t) = \exp\left[-(t/\tau)^\beta\right]$$



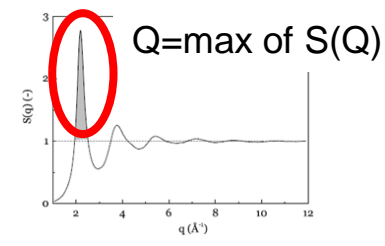
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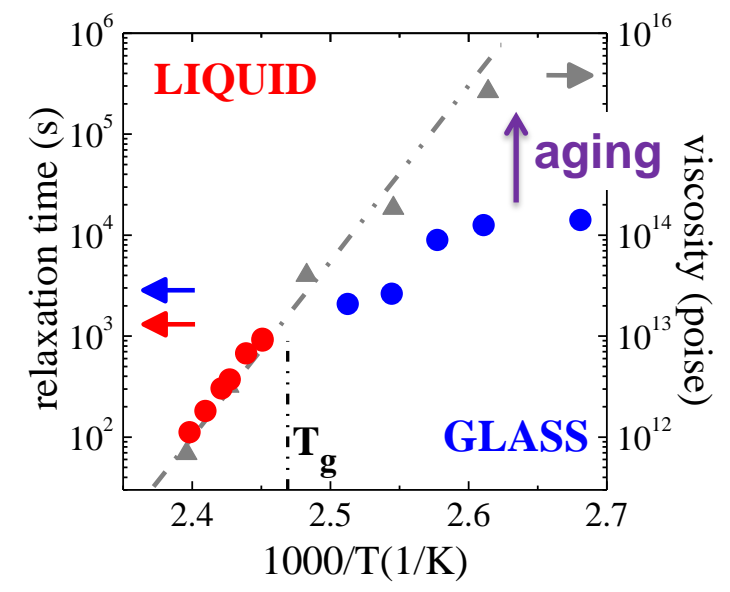
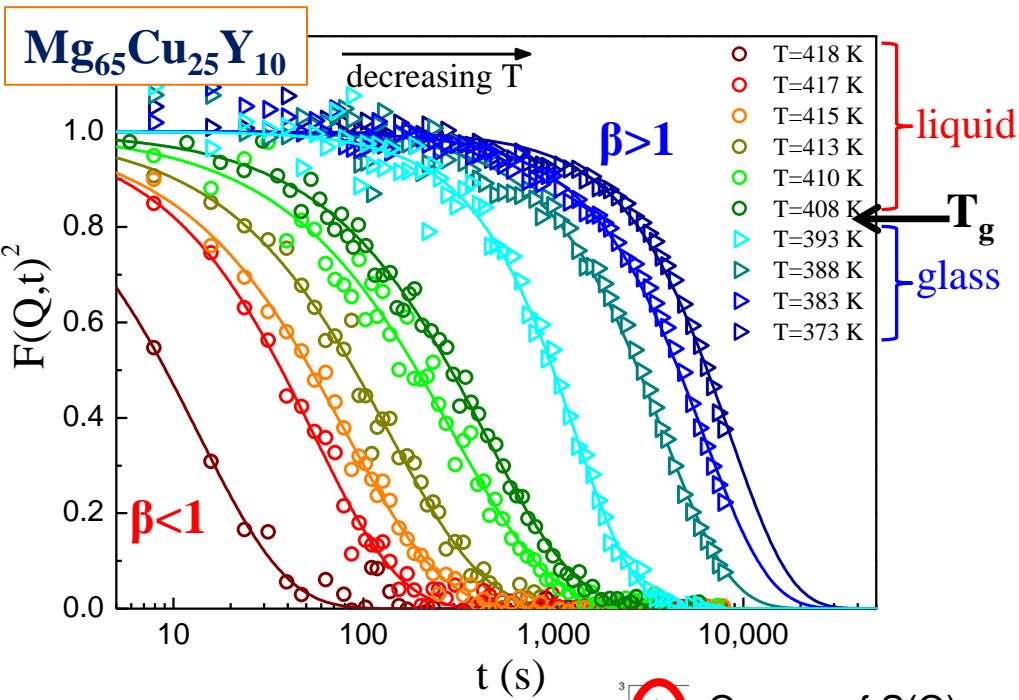
Departure from the liquid curve

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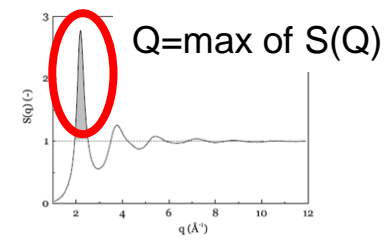
FIRST INVESTIGATION OF ATOMIC DYNAMICS IN MGs



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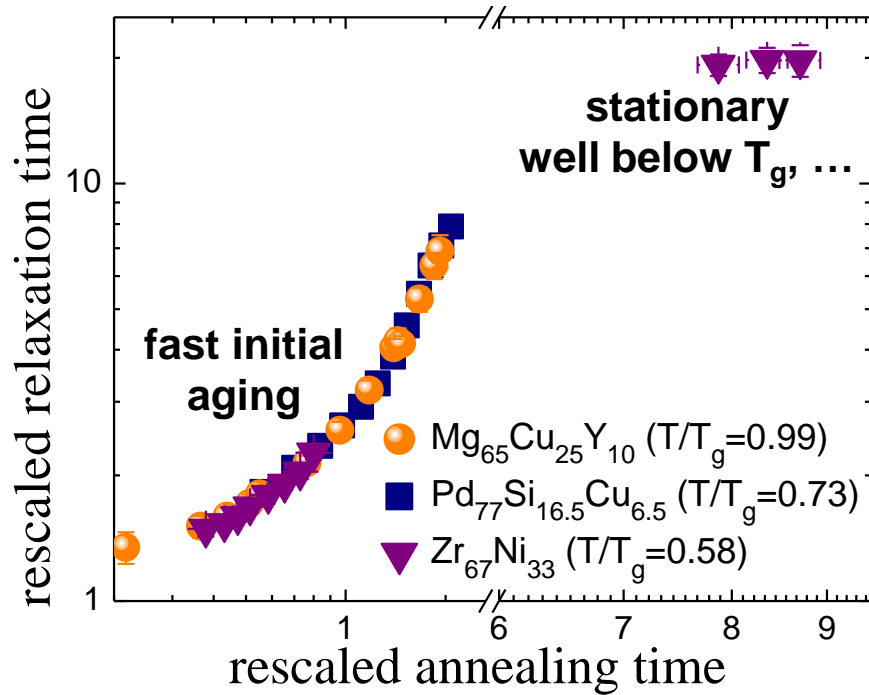
$$F(Q,t) = \exp \left[-(t / \tau)^\beta \right]$$



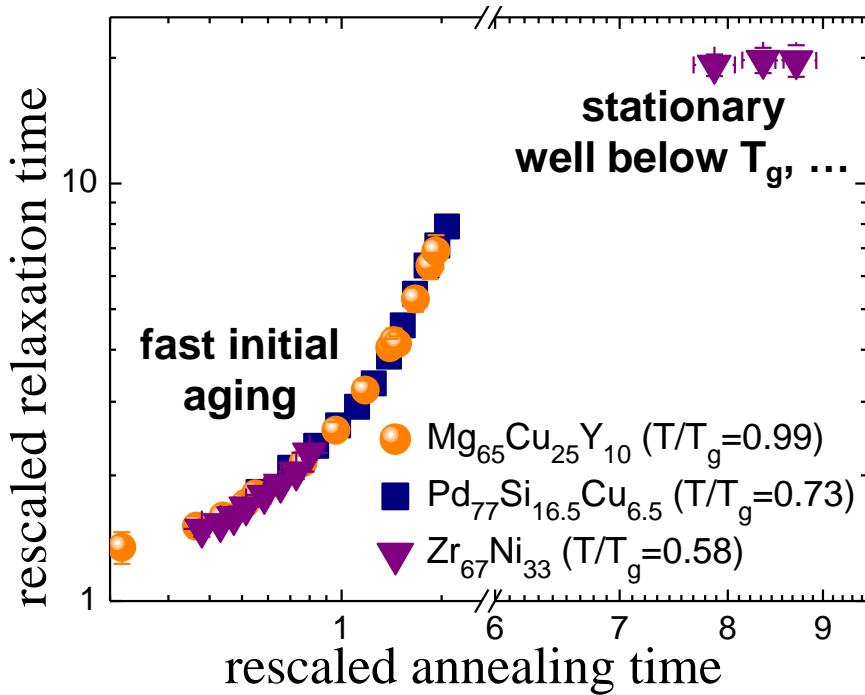
DYNAMICAL CROSSOVER AT T_g

$T > T_g$: stationary dynamics, $\beta < 1$, diffusive motion

$T < T_g$: aging, $\beta > 1$, Anomalous stress-dominated dynamics



Surprising hierarchy of “anomalous” dynamical regimes



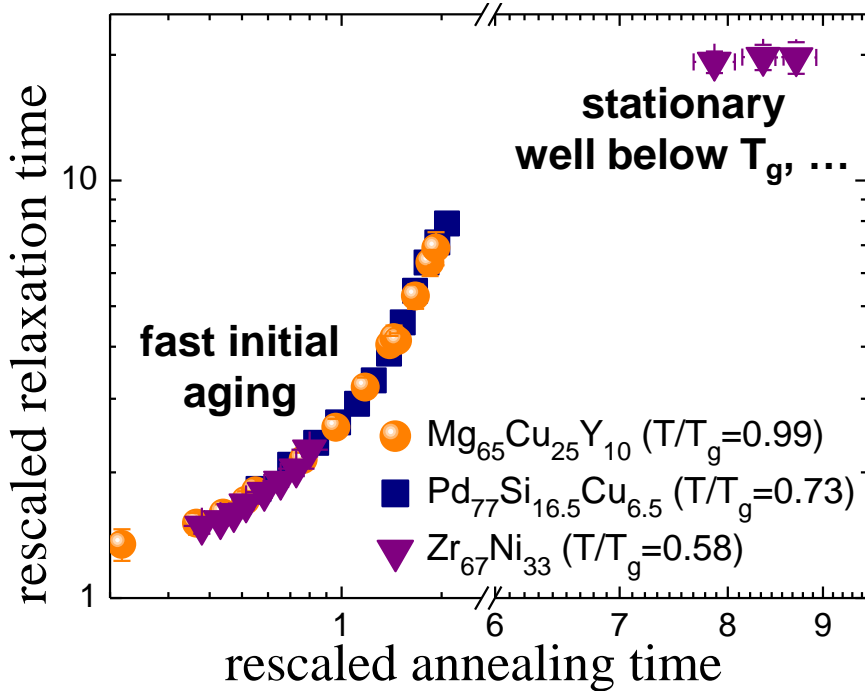
Surprising hierarchy of “anomalous” dynamical regimes

→ Contradiction with the continuous evolution of macroscopic measurements

Ruta et al. Phys. Rev. Lett. (2012)

Ruta et al. J. Chem. Phys. (2013)

Evenson, Ruta et al. Phys. Rev. Lett. (2015)



Surprising hierarchy of “anomalous” dynamical regimes

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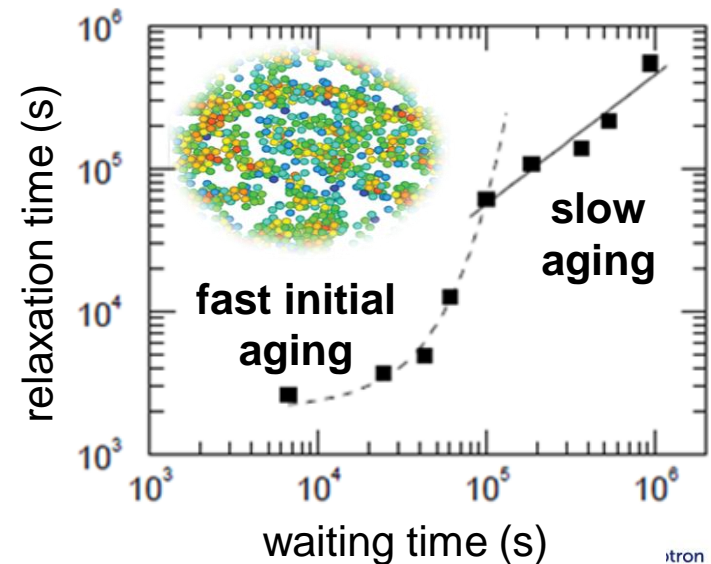
Ruta et al. J. Chem. Phys. (2013)

Evenson, Ruta et al. Phys. Rev. Lett. (2015)

Similar to what observed in soft materials (colloidal gels, emulsions, polymeric gels, ...)

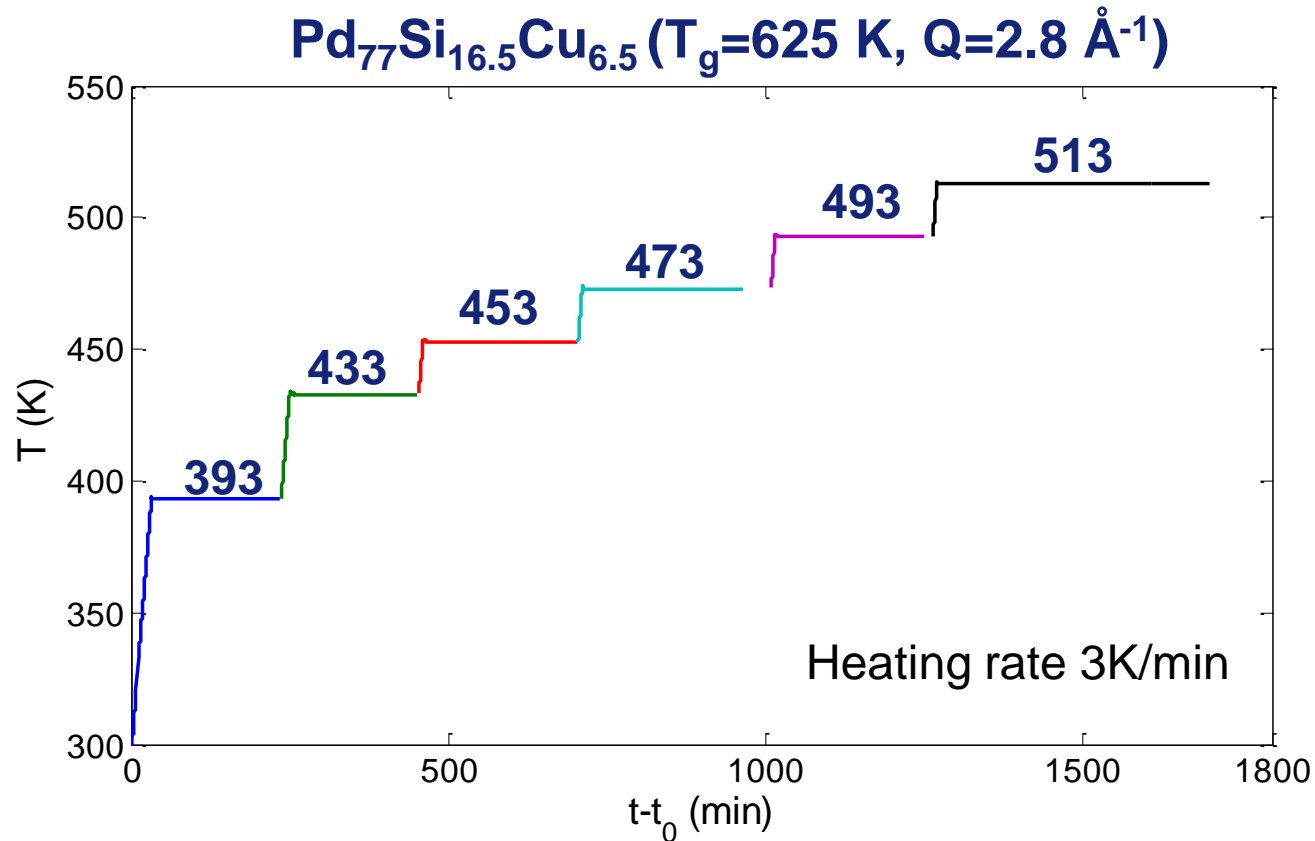
Cipelletti et al. Phys. Rev. Lett. (2000)

→ Universal stress-dominated dynamics?



What is the origin of the microscopic aging?

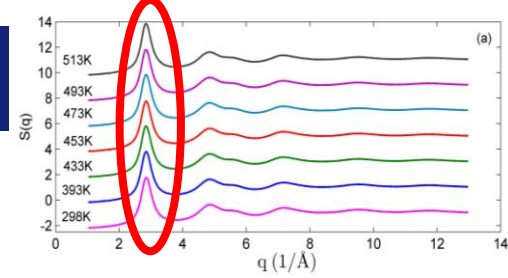
Combining XRD and XPCS → same thermal and temporal protocol



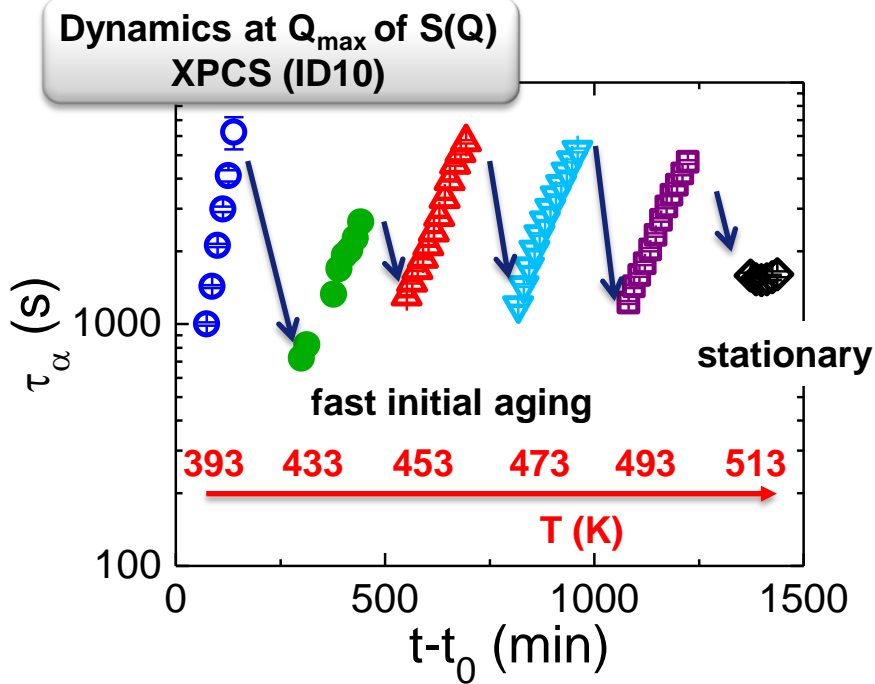
2 XRD experiments:

- i) high resolution around the FSDP
- ii) large q range (up to 25Å^{-1} to extract the pair distribution function $G(r)$)

STRUCTURE & DYNAMICS DURING AGING



Complementary dynamical (ID10) & structural (ID15) study

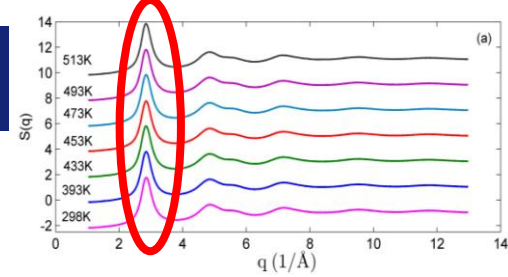


$$\tau(T, t_w) = \tau_0(T) \exp(t_w / \tau^*)$$

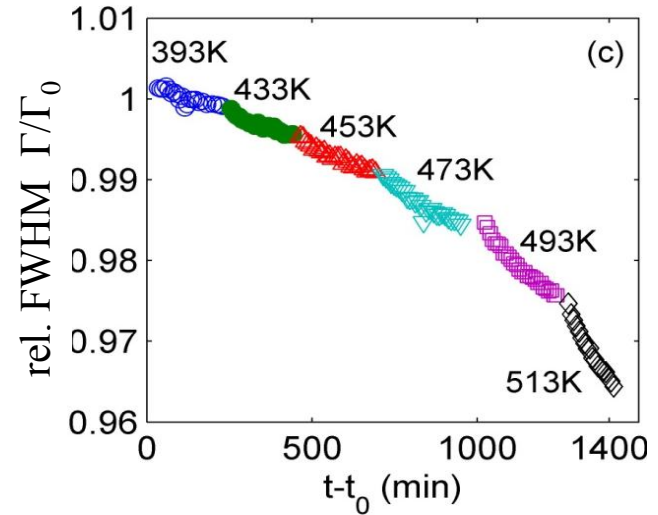
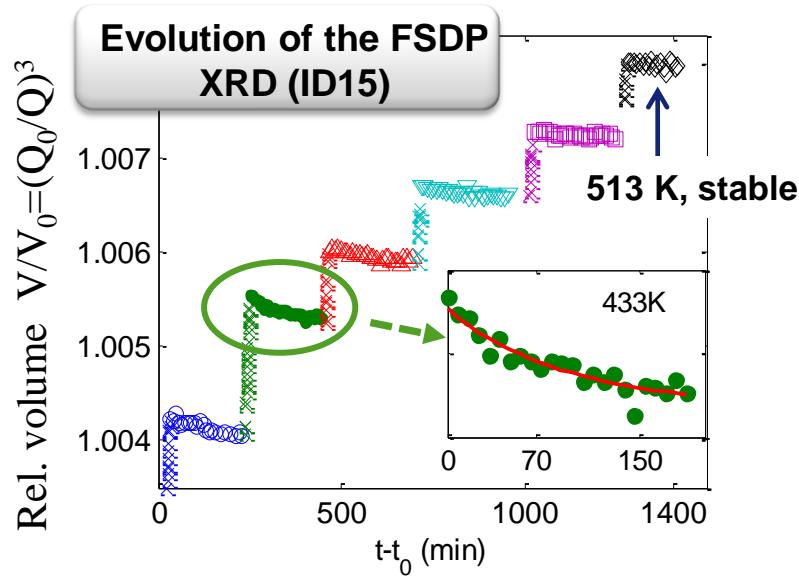
$$\tau^*(393 \rightarrow 493 \text{ K}) \sim 6000 \text{ s}$$

- Fast aging along isotherms, strong τ reduction at temperature jumps
- Frozen dynamics at the highest T (513 K)

STRUCTURE & DYNAMICS DURING AGING



Complementary dynamical (ID10) & structural (ID15) study



Isothermal volume relaxation

$$\frac{V}{V_0}(t, T) = \frac{V}{V_0}(0, T) \exp\left(-\frac{t}{\tau_V}\right)$$

Continuous FSDP sharpening

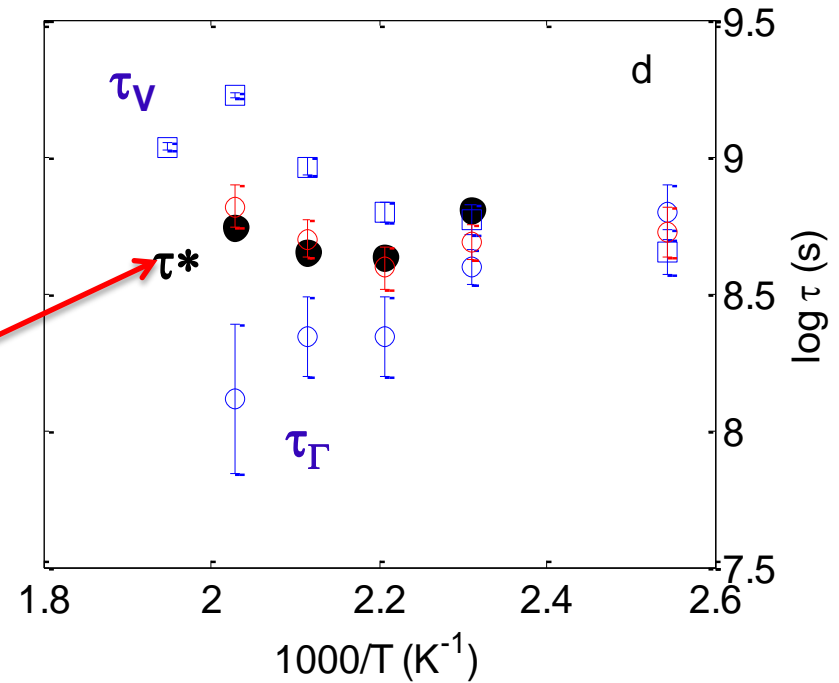
$$\frac{\Gamma}{\Gamma_0}(t, T) = \frac{\Gamma}{\Gamma_0}(0, T) \exp\left(-\frac{t}{\tau_\Gamma}\right)$$

Three characteristic aging times:

Aging time from $V/V_0 \rightarrow \tau_V$
 Aging time from $\Gamma/\Gamma_0 \rightarrow \tau_\Gamma$
 Aging time from $\tau_\alpha \rightarrow \tau^*$



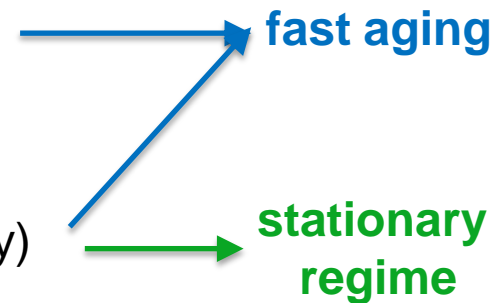
$$\tau^* = \frac{\tau_V + \tau_\Gamma}{2}$$



Two main processes controlling the aging:

1) volume shrinking (density changes)

2) medium range ordering (constant density)

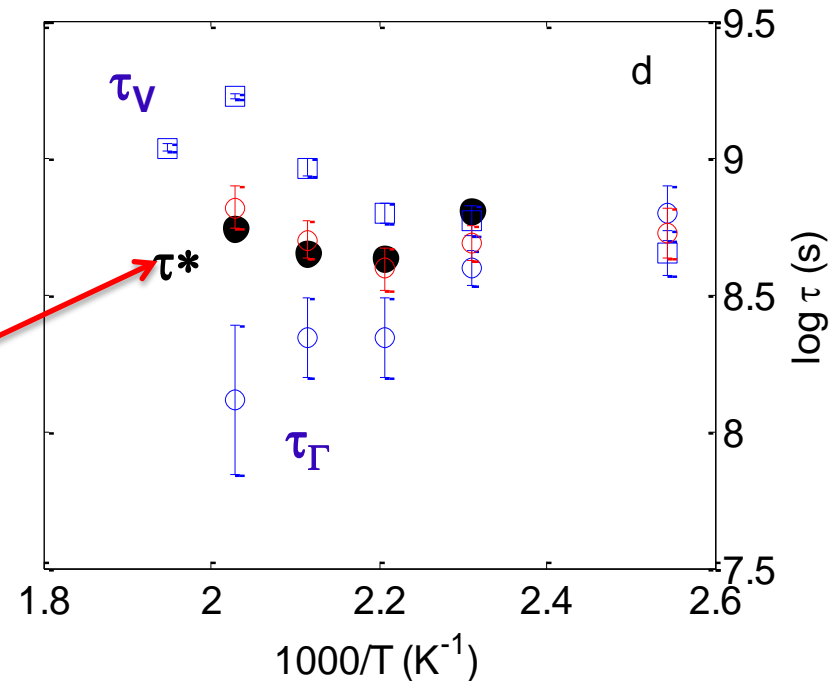


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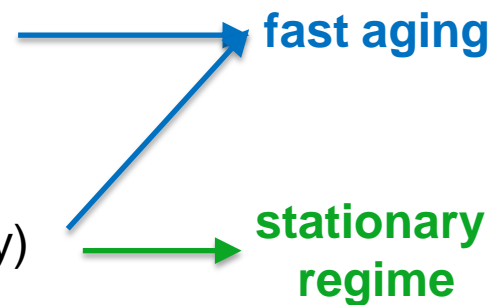


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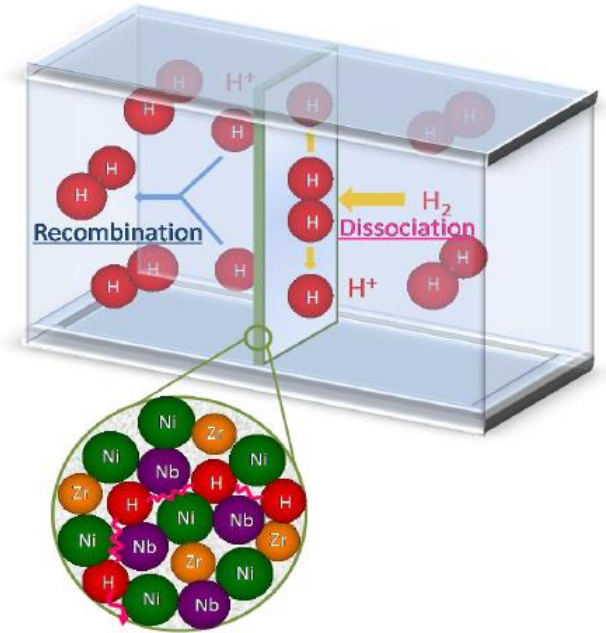
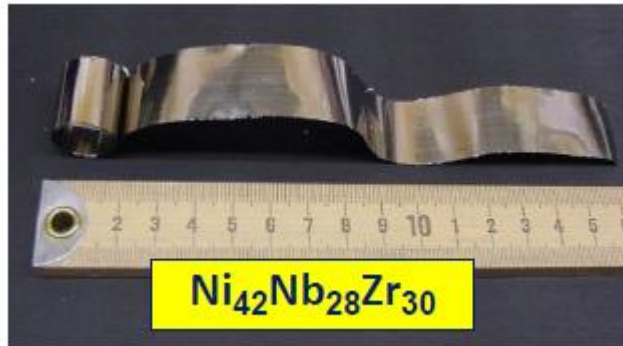
- 1) volume shrinking (density changes)
- 2) medium range ordering (constant density)



the evolution between the two could be related to a **ductile to brittle transition**

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MG MEMBRANES FOR H₂ SEPARATION FROM CO₂



Ni and Zr-based MGs:

→ **better H₂ permeability** than crystalline materials (high H₂ diffusivity due to the free volume)

→ **high H₂ solubilities and diffusivities**

→ **non-precious metals/alloys**

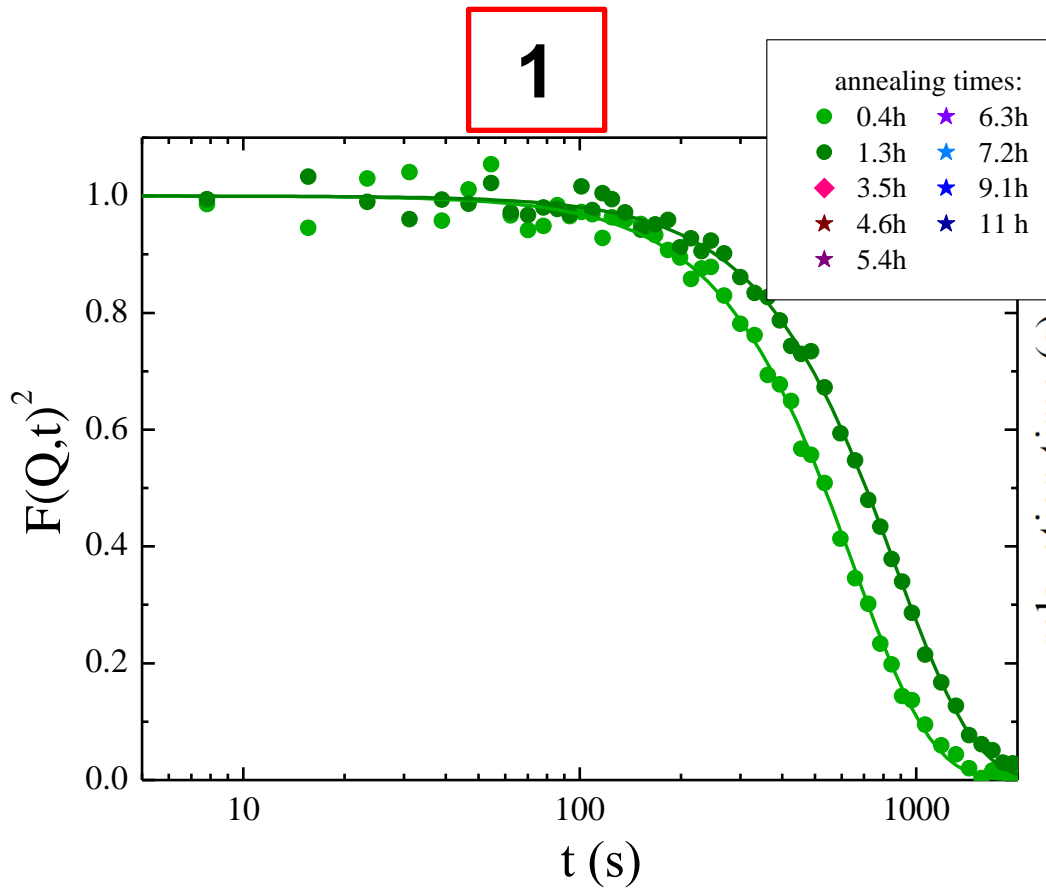
Pd: ~20000 euro/Kg

Group IV and V metals ~10-200 euro/Kg

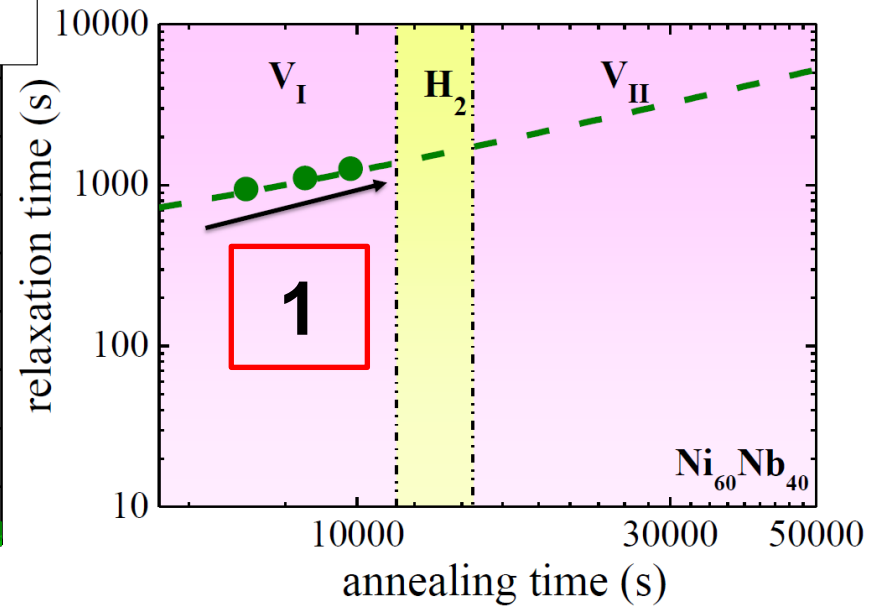
Collab. with Prof. D. Chandra
(DOE - NNSA Grant, US)



EFFECT OF H₂ ON THE DYNAMICS



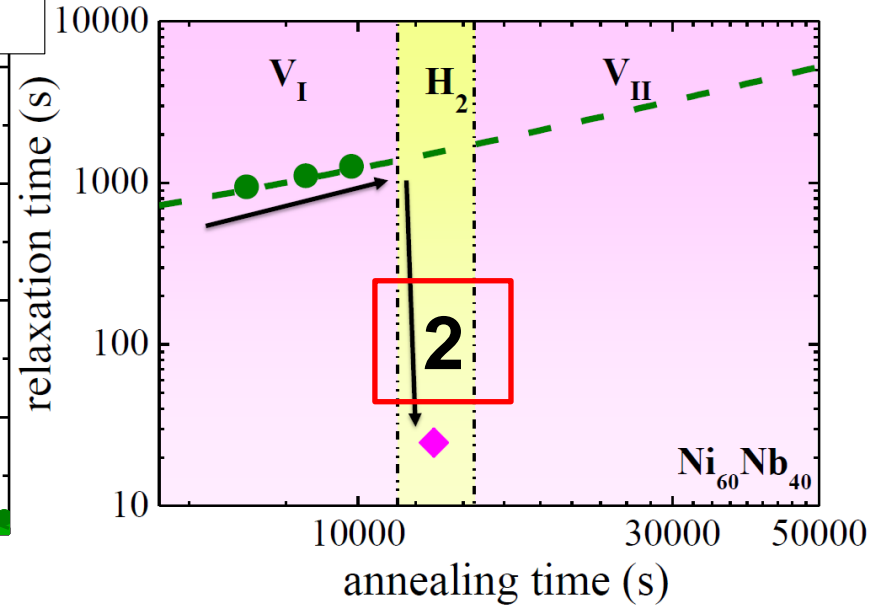
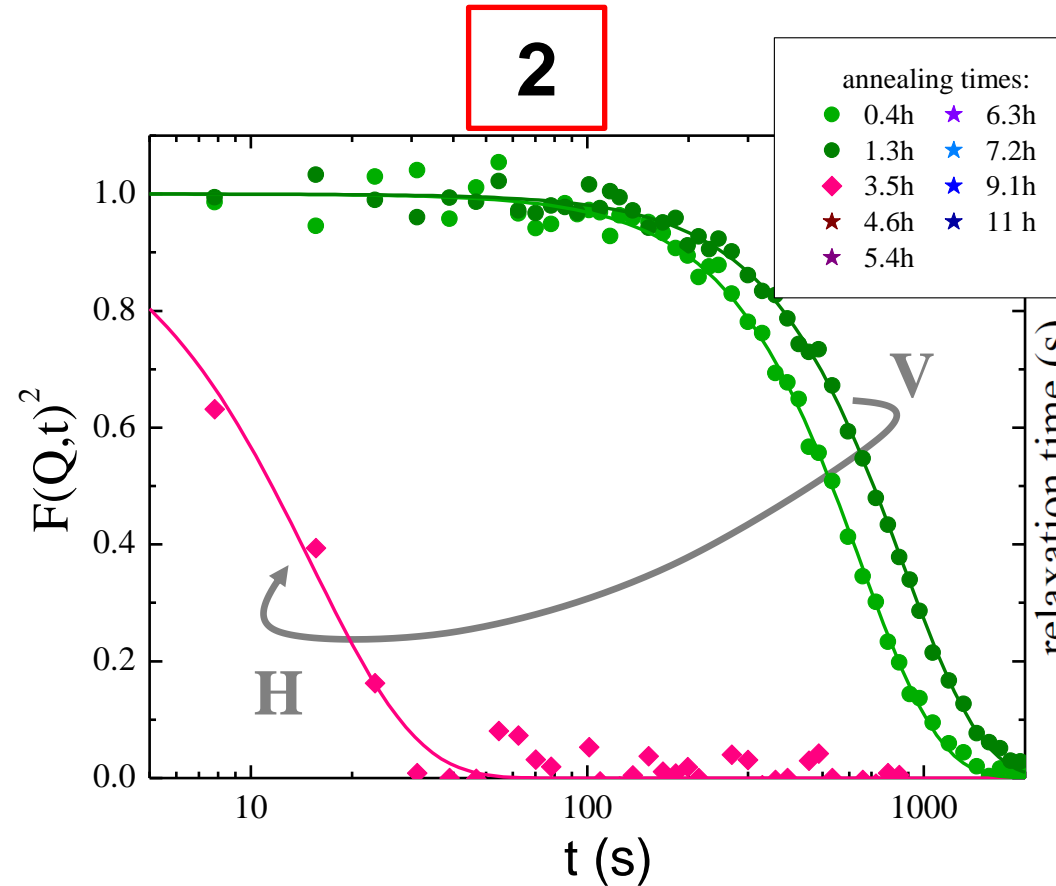
$\text{Ni}_{60}\text{Nb}_{40}$, $T=373$ K, $Q_p=2.7 \text{ \AA}^{-1}$
H @ 0.6 bar



EFFECT OF H₂ ON THE DYNAMICS

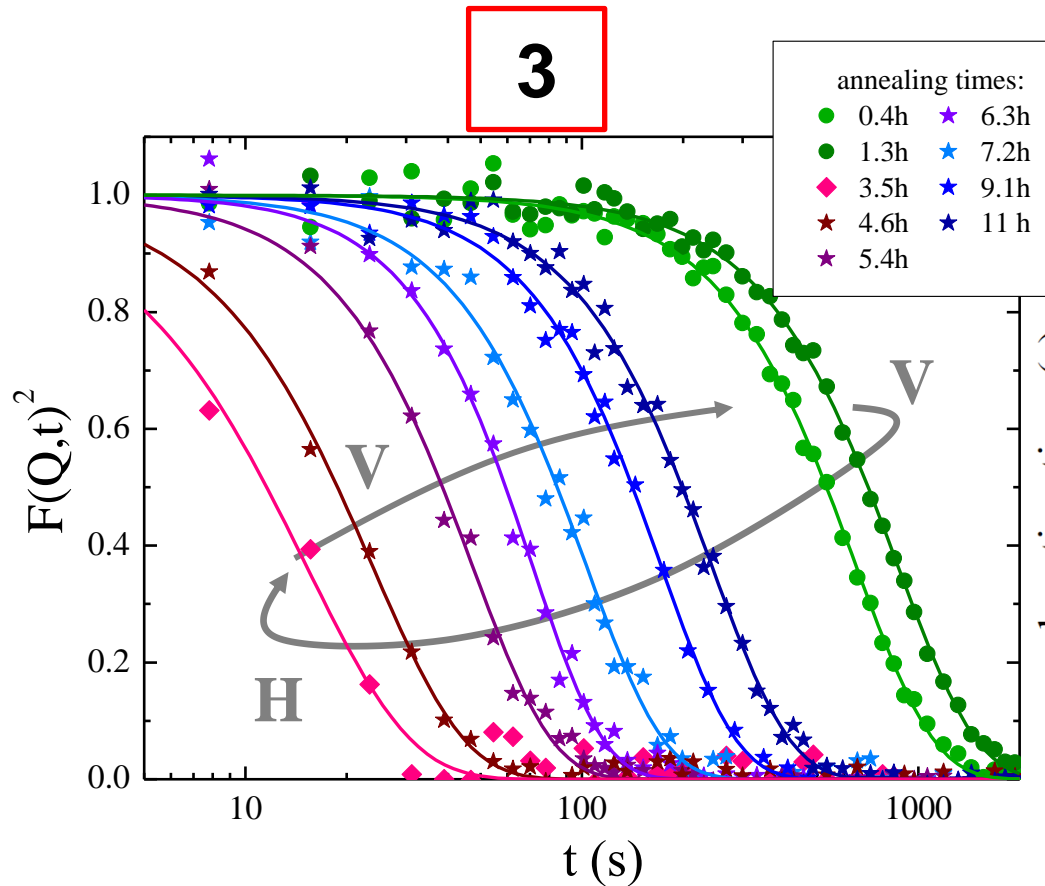
2

Ni₆₀Nb₄₀, T=373 K, Q_p=2.7 Å⁻¹
H @ 0.6 bar

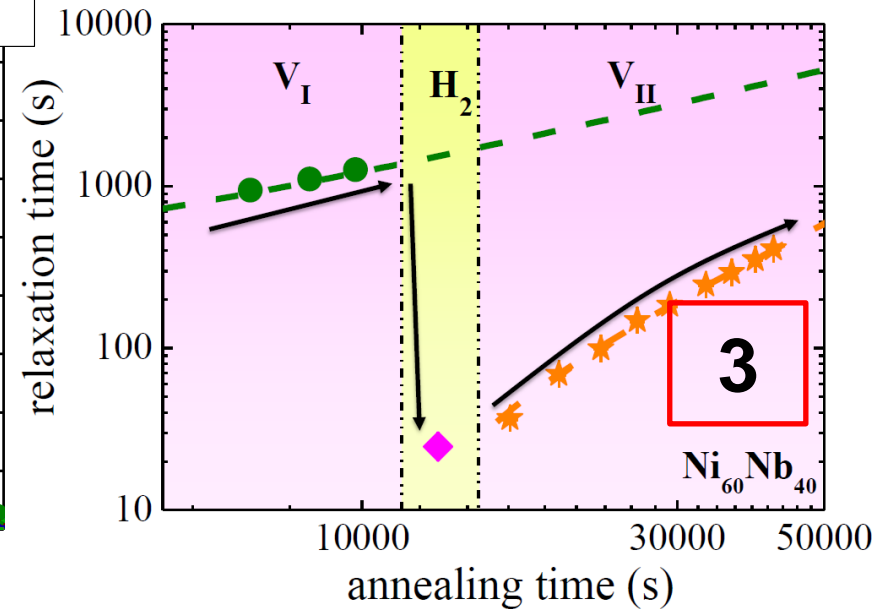


Dramatic acceleration of the dynamics due to the hydrogen atmosphere

EFFECT OF H₂ ON THE DYNAMICS



Ni₆₀Nb₄₀, T=373 K, $Q_p=2.7 \text{ \AA}^{-1}$
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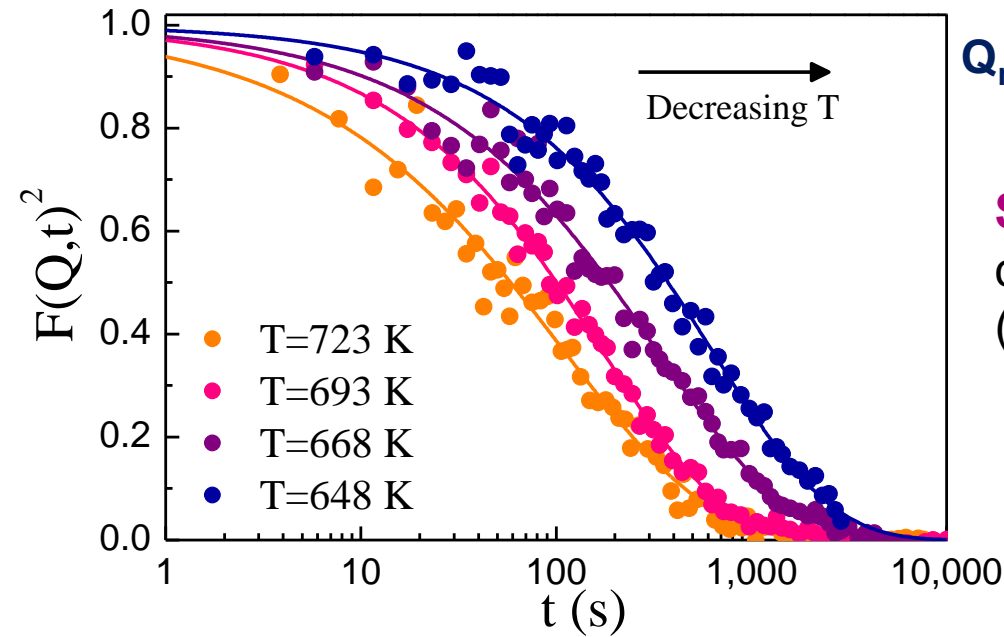


Dramatic acceleration of the dynamics due to the hydrogen atmosphere

Reversible transition: after removing the hydrogen, the dynamics slows down again but with a faster aging

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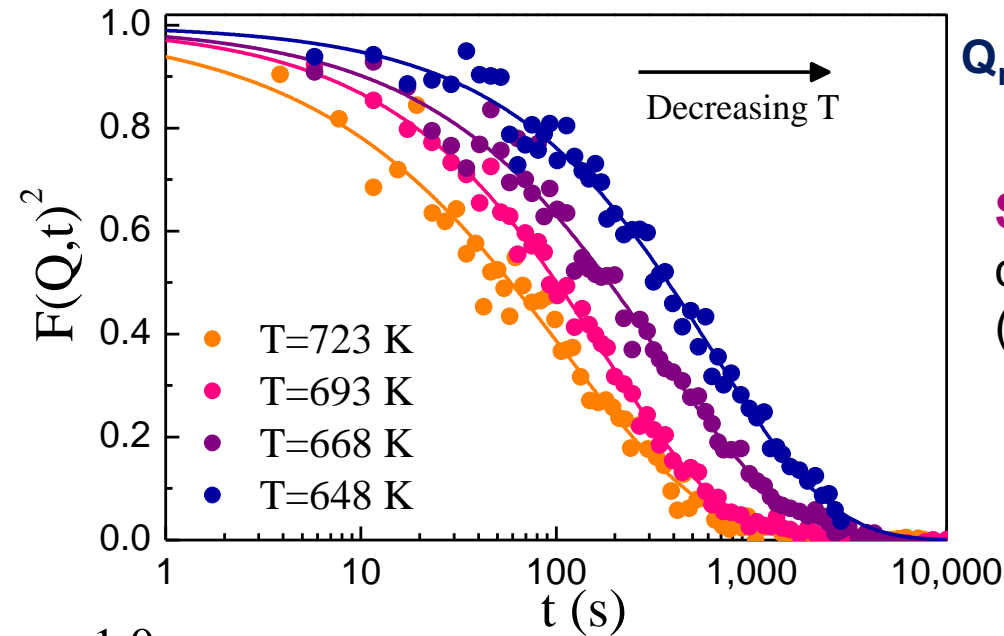
SILICATE GLASSES: 0.8SiO₂-0.2Na₂O (NS4, T_g=783 K)



$Q_{\max}=1.53 \text{ \AA}^{-1}$; measurements in the glass transition region

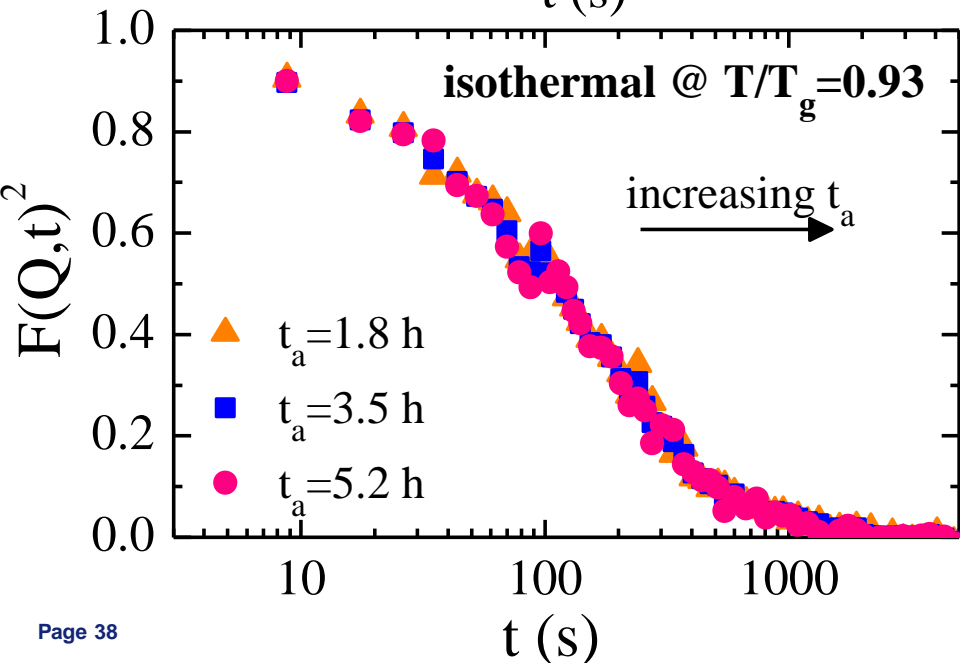
Stretched exponential decay of the correlation function as in the liquid phase ($\beta=0.67$) \rightarrow diffusive motion

SILICATE GLASSES: $0.8\text{SiO}_2\text{-}0.2\text{Na}_2\text{O}$ (NS4, $T_g=783$ K)



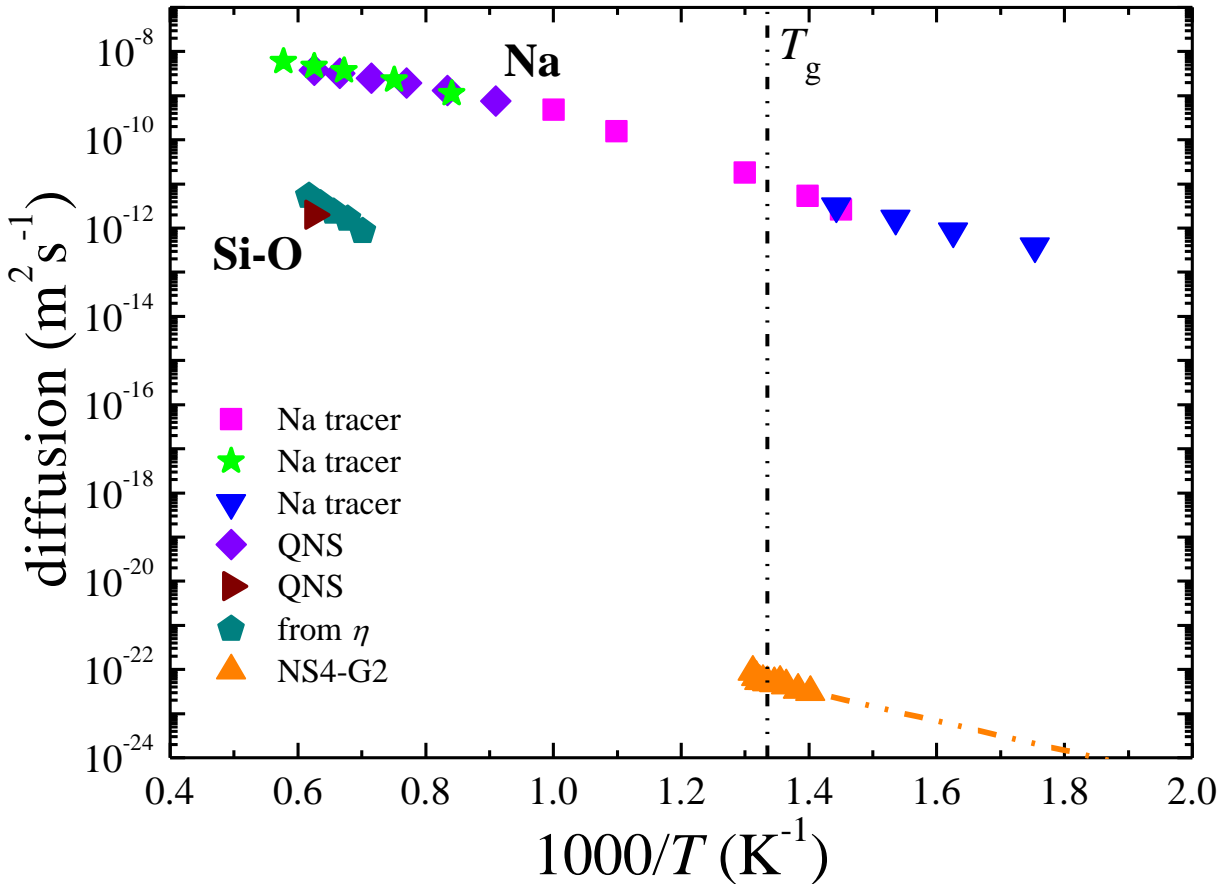
$Q_{\max}=1.53 \text{ \AA}^{-1}$; measurements in the glass transition region

Stretched exponential decay of the correlation function as in the liquid phase ($\beta=0.67$) \rightarrow diffusive motion



Absence of physical aging on the experimental time scale

COMPARISON WITH DIFFUSION DATA



XPCS data :

□ ~10 orders of magnitude slower than the Na diffusion

□ Closer to the low T extrapolation of the Si-O matrix

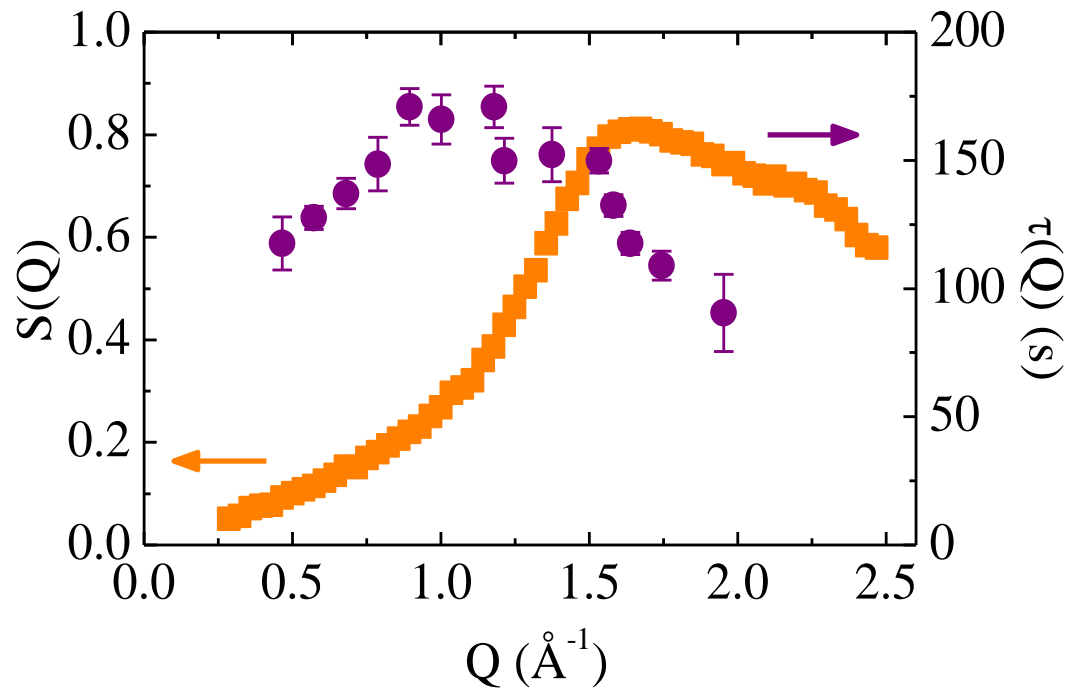
$$D_{\text{XPCS}} = 1/(\tau_{\text{incoh}} Q^2)$$

$$\tau_{\text{incoh}} = \tau_{\text{XPCS}} / S(Q)$$

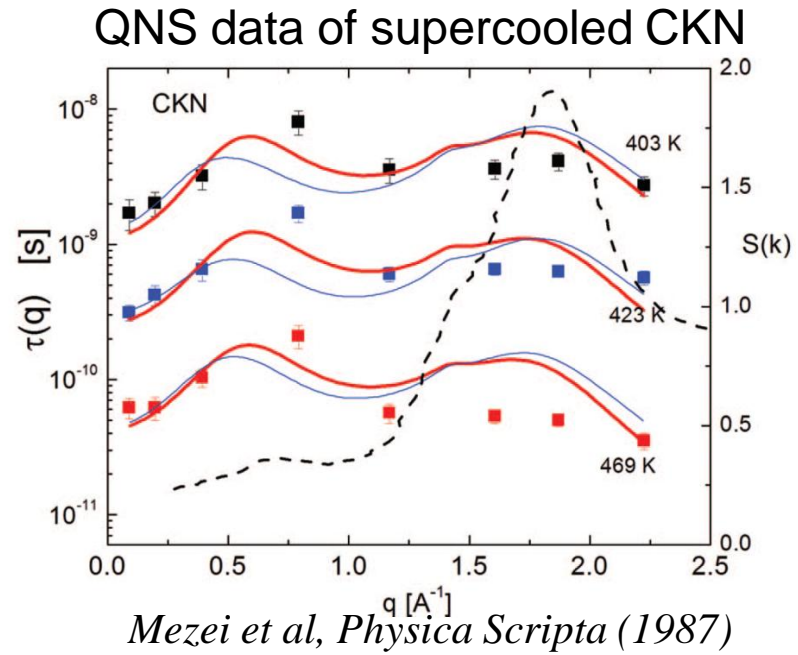
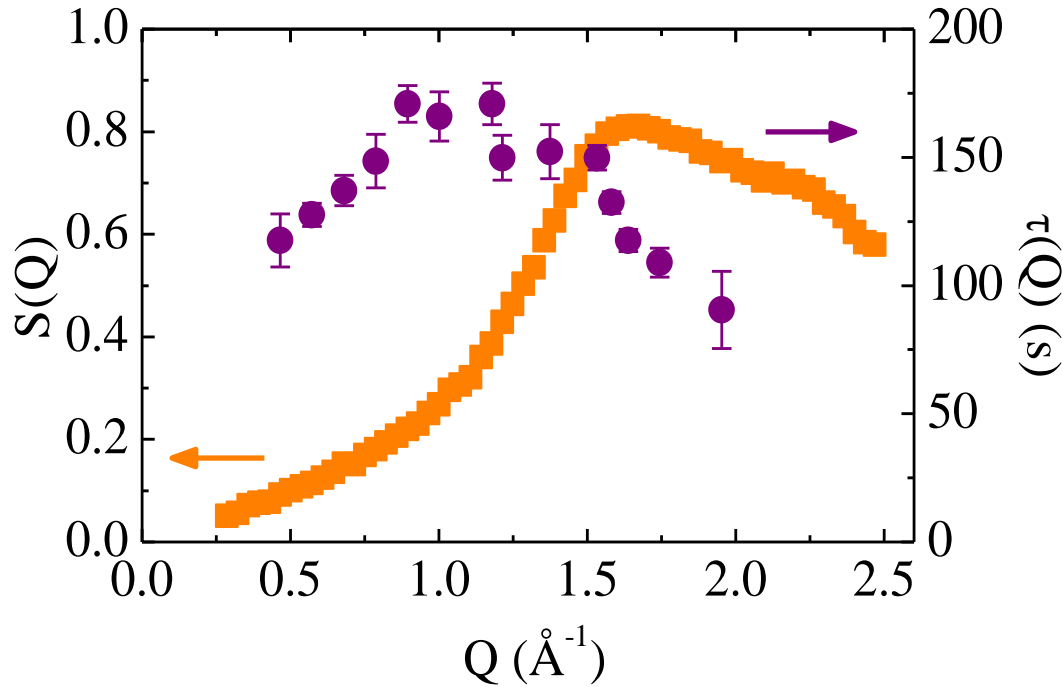
Hempelmann et al, *Z. Phys. B* (1994)

Kargl et al. *Phys. Rev. B*. (2006)

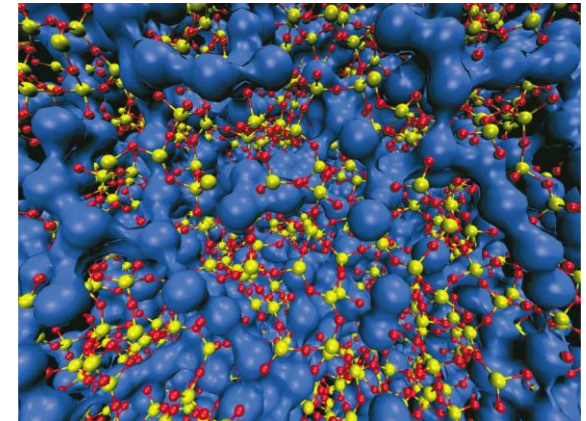
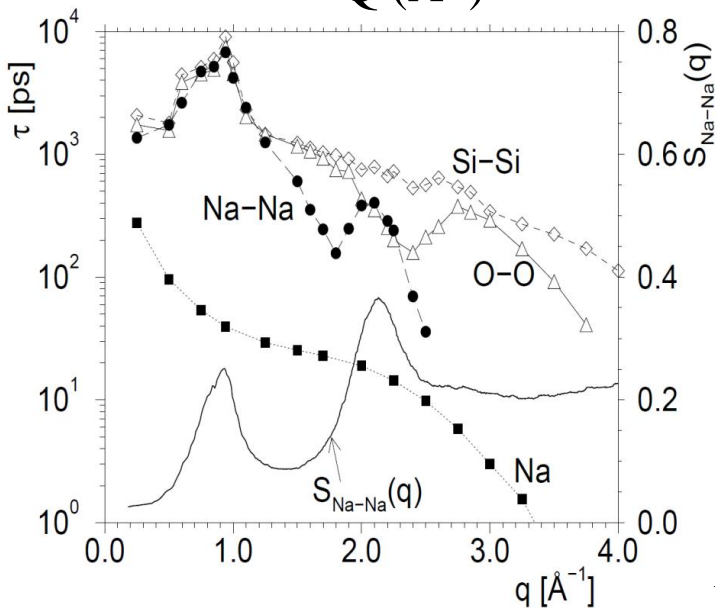
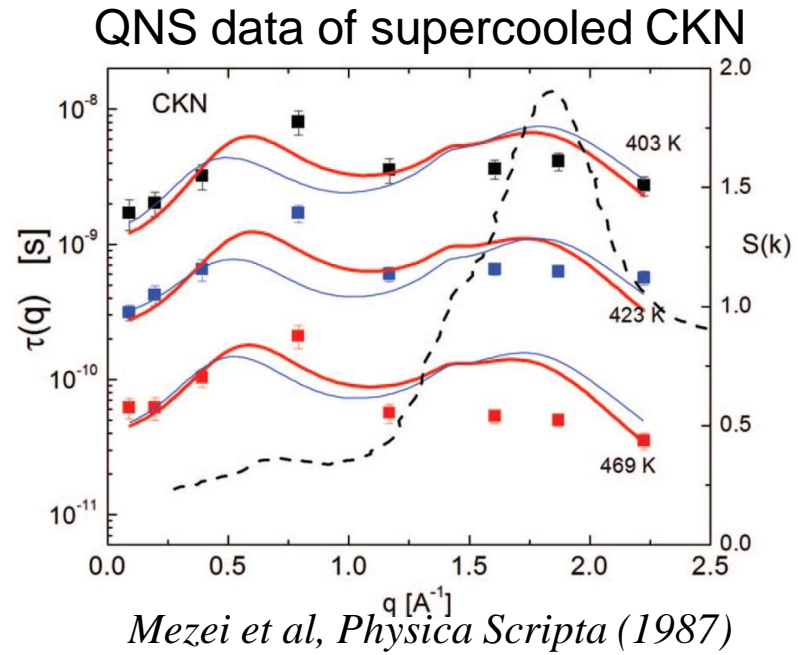
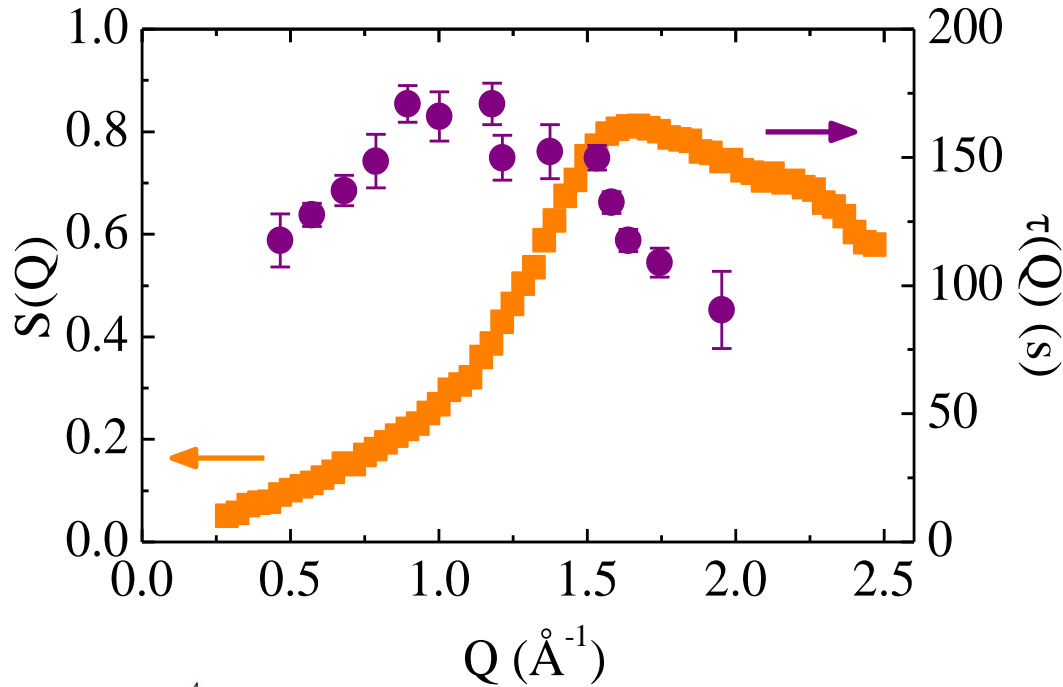
WAVE VECTOR DEPENDENCE



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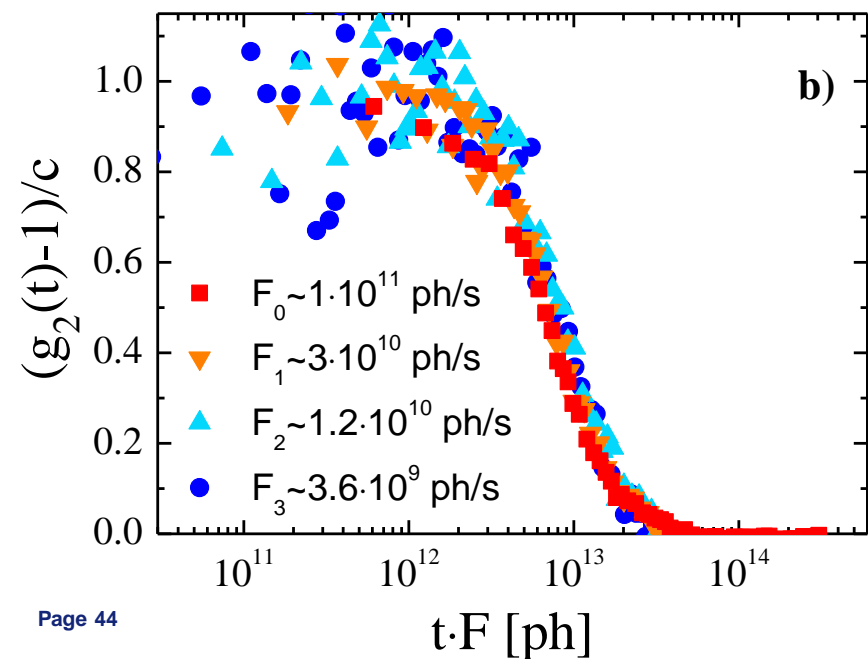
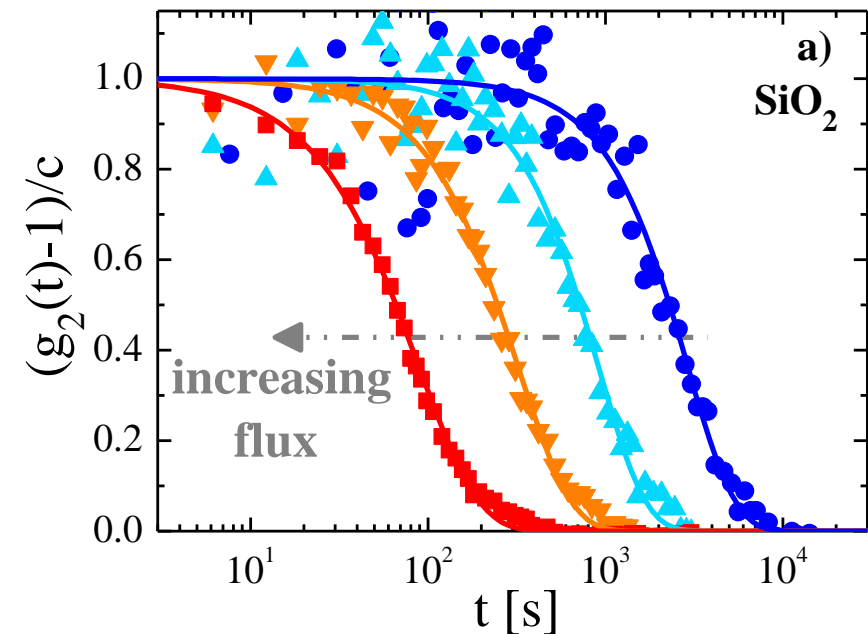


Meyer et al, Phys. Rev. Lett. (2004)

Horbach et al, Phys. Rev. Lett. (2002)

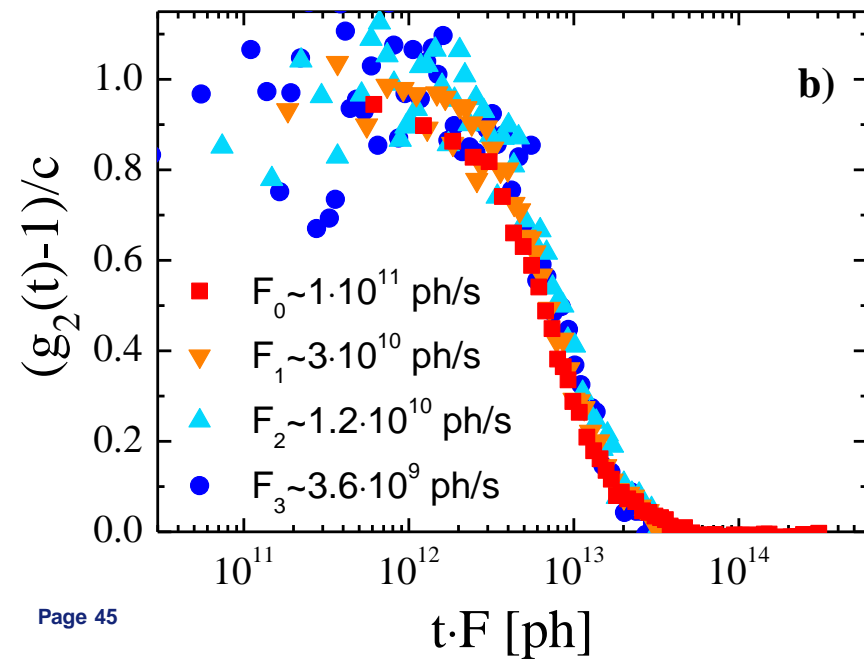
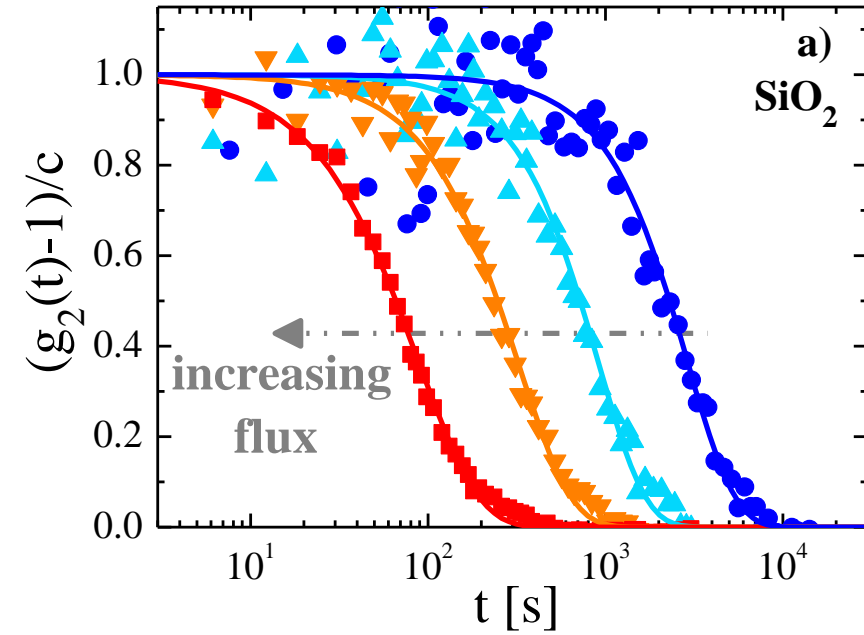
What happens in oxide glasses?

IRRADIATION-DRIVEN DYNAMICS IN SiO_2 AT LOW TEMPERATURE



IRRADIATION-DRIVEN DYNAMICS IN SiO_2 AT LOW TEMPERATURE

The atomic motion at room temperature in SiO_2 is completely induced by the incoming X-rays

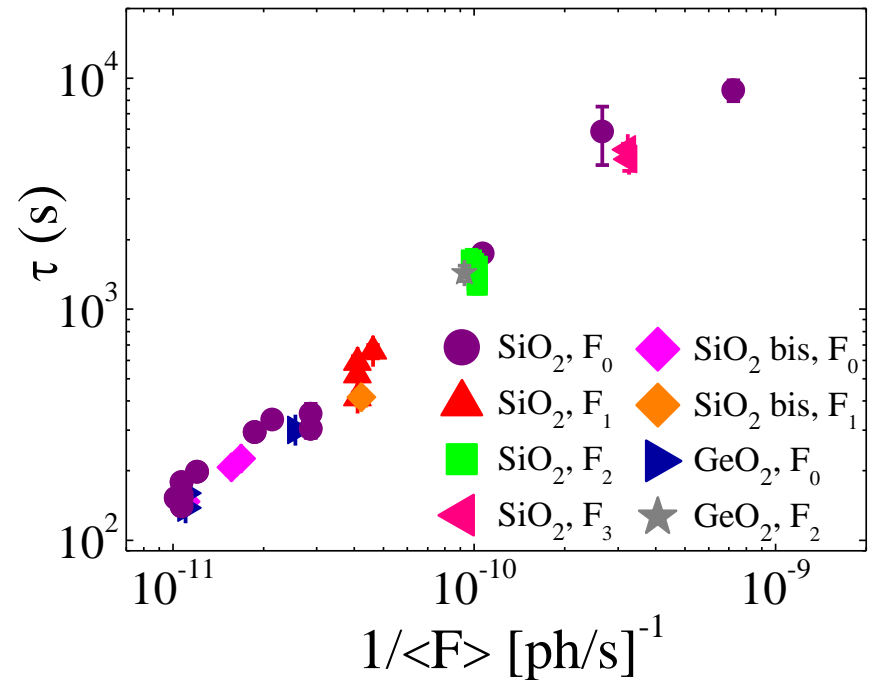
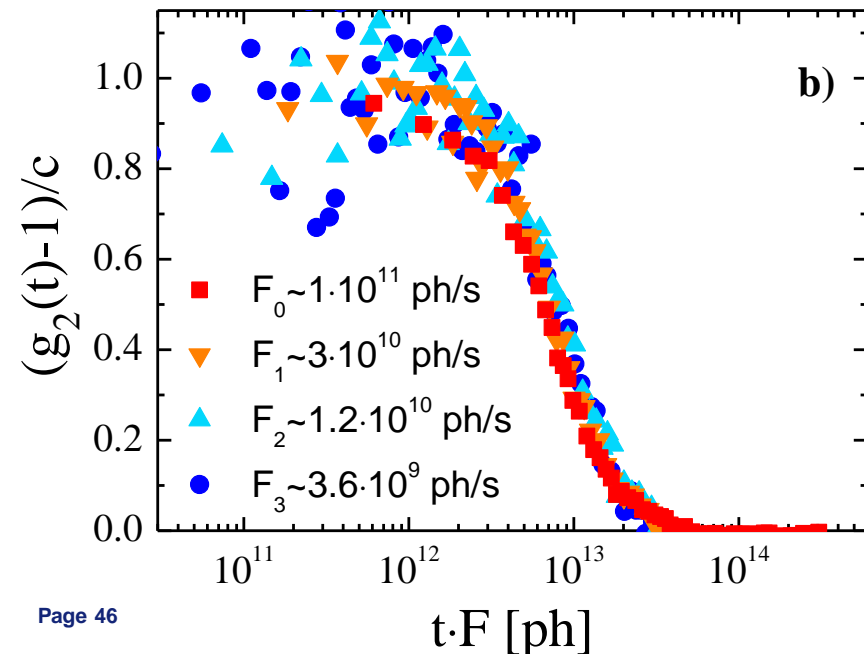
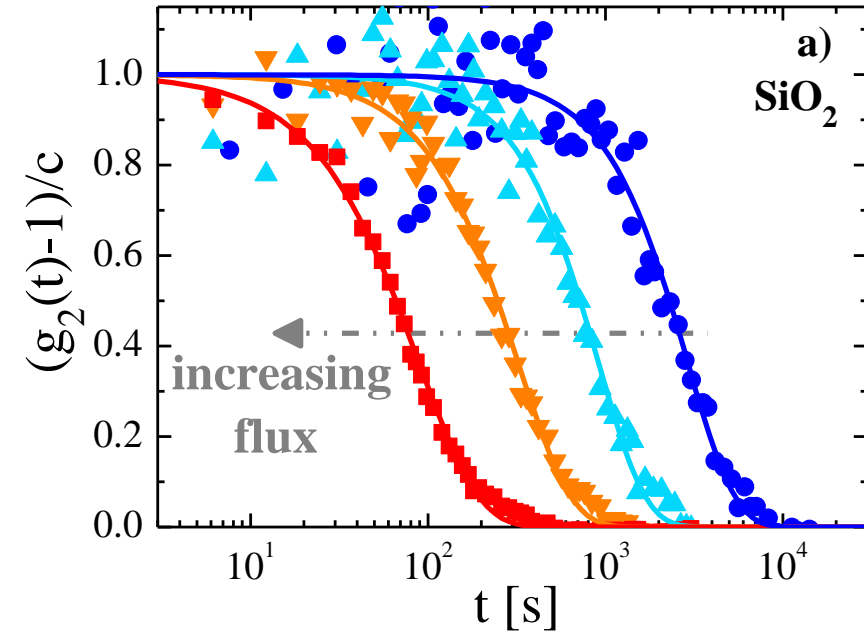


IRRADIATION-DRIVEN DYNAMICS IN SiO₂ AT LOW TEMPERATURE

The atomic motion at room temperature in SiO₂ is completely induced by the incoming X-rays

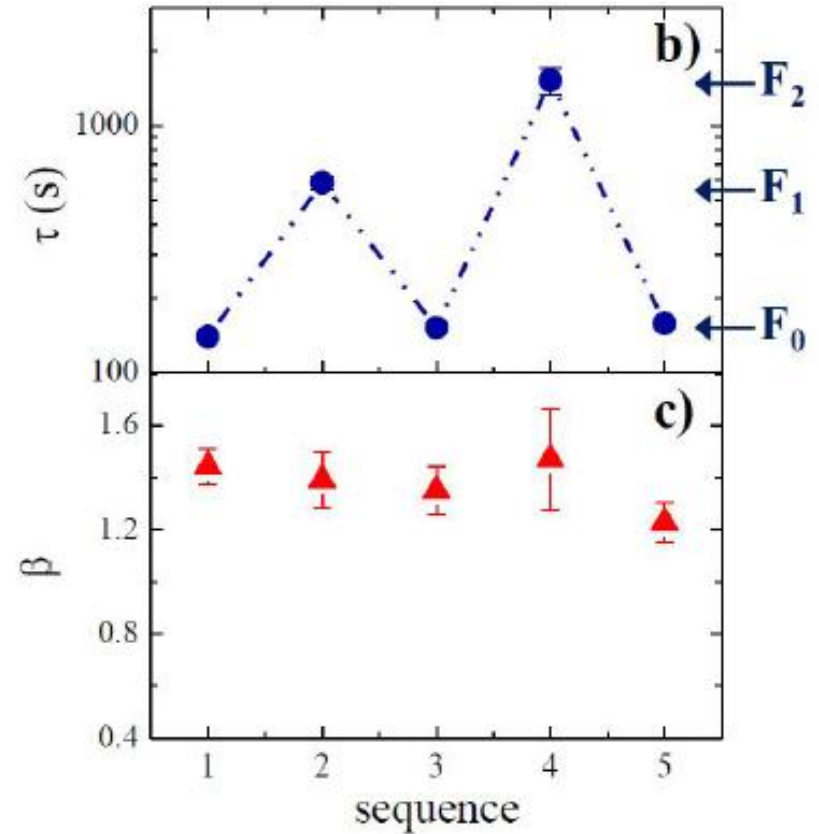
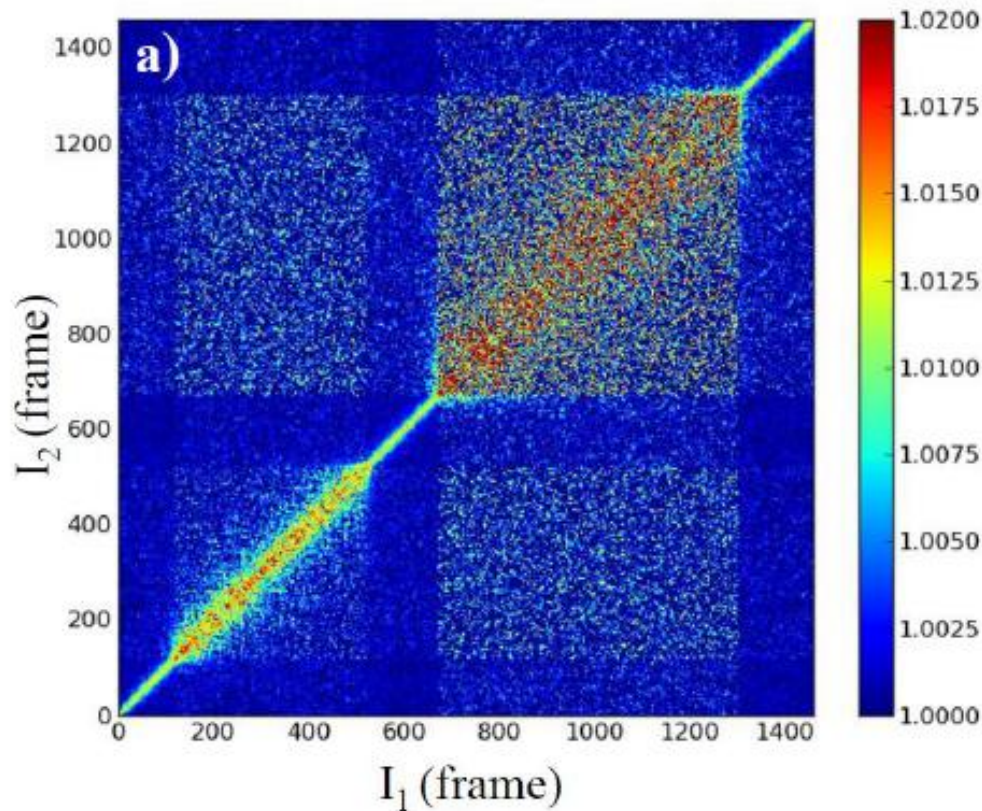


By varying appropriately the average incoming flux (flux, exposure time, delay time) is possible to tune “at will” the dynamics



NO STANDARD RADIATION DAMAGE

The effect of the X-rays is almost instantaneous and leads to a reversible and stationary atomic motion, thus independent on the global accumulated dose within the experimental time



Three main processes:

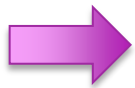
- **Knock-on events** → require high energy to break bonds
- **Electronic rearrangements** → require pre-existing defects
- **Radiolysis** → require lifetime of the excitation ~ vibrations ~1ps → **possible**

Hobbs et al. J. Nuclear Mat. (1994)

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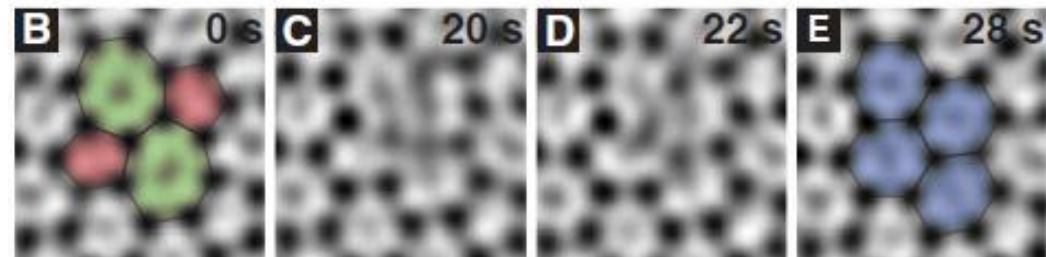
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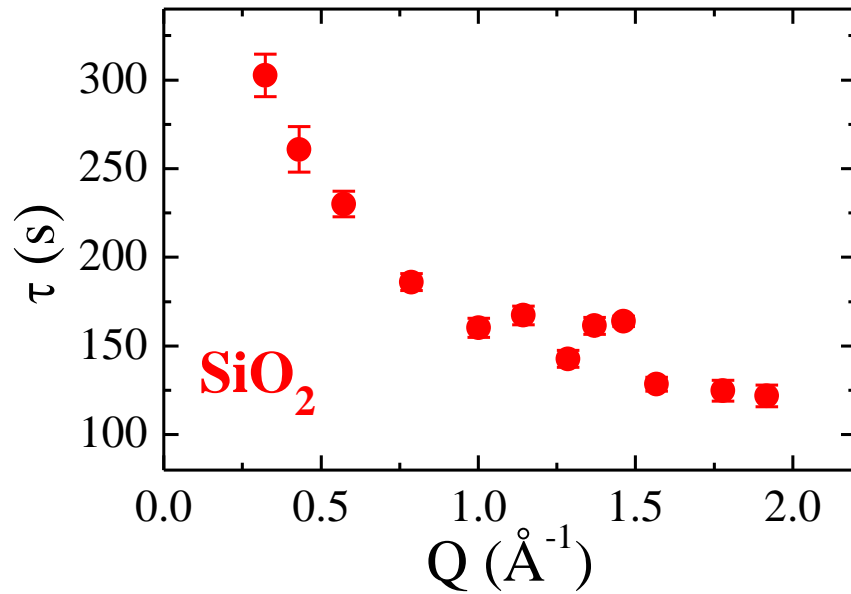
➔ **Hard X-rays as pump and probe of the dynamics**

Similar to what observed with electron transmission microscopy on bi-dimensional SiO_2



Huang et al. Science (2013)

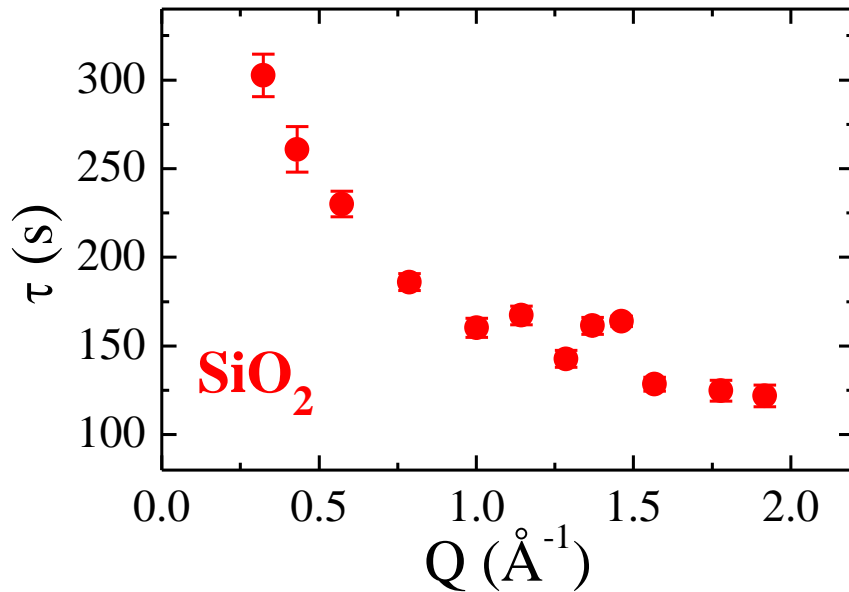
POSSIBILITY TO GET INTRINSIC PROPERTIES OF THE SYSTEM



The **wavevector dependence** of the dynamics is **independent on the flux**

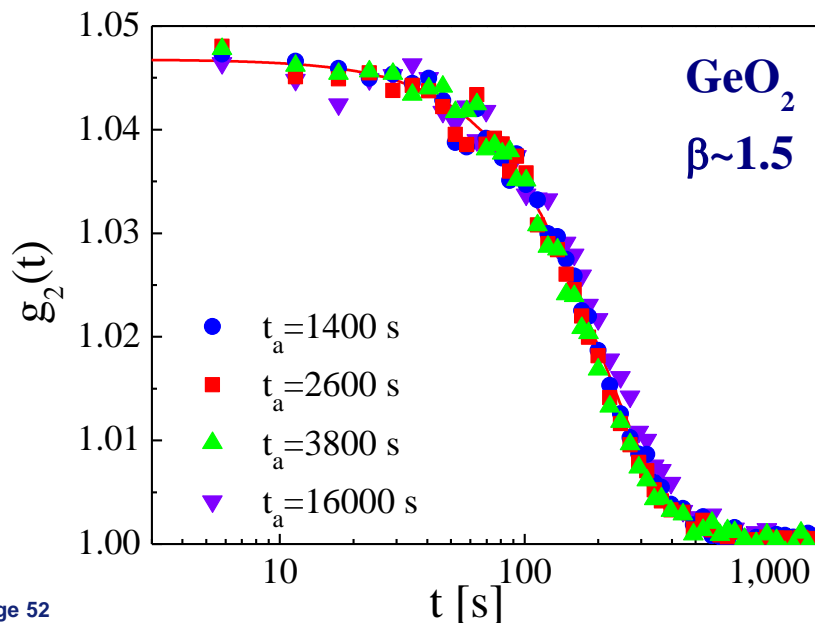
→ Typical increase as in supercooled liquids

POSSIBILITY TO GET INTRINSIC PROPERTIES OF THE SYSTEM



The **wavevector dependence** of the dynamics is **independent on the flux**

→ Typical increase as in supercooled liquids



The **shape of the curve** is also **independent on the flux**

→ Compressed in SiO_2 and GeO_2 , stretched in NS4 and lead-silicates. Stress-driven vs diffusive dynamics?

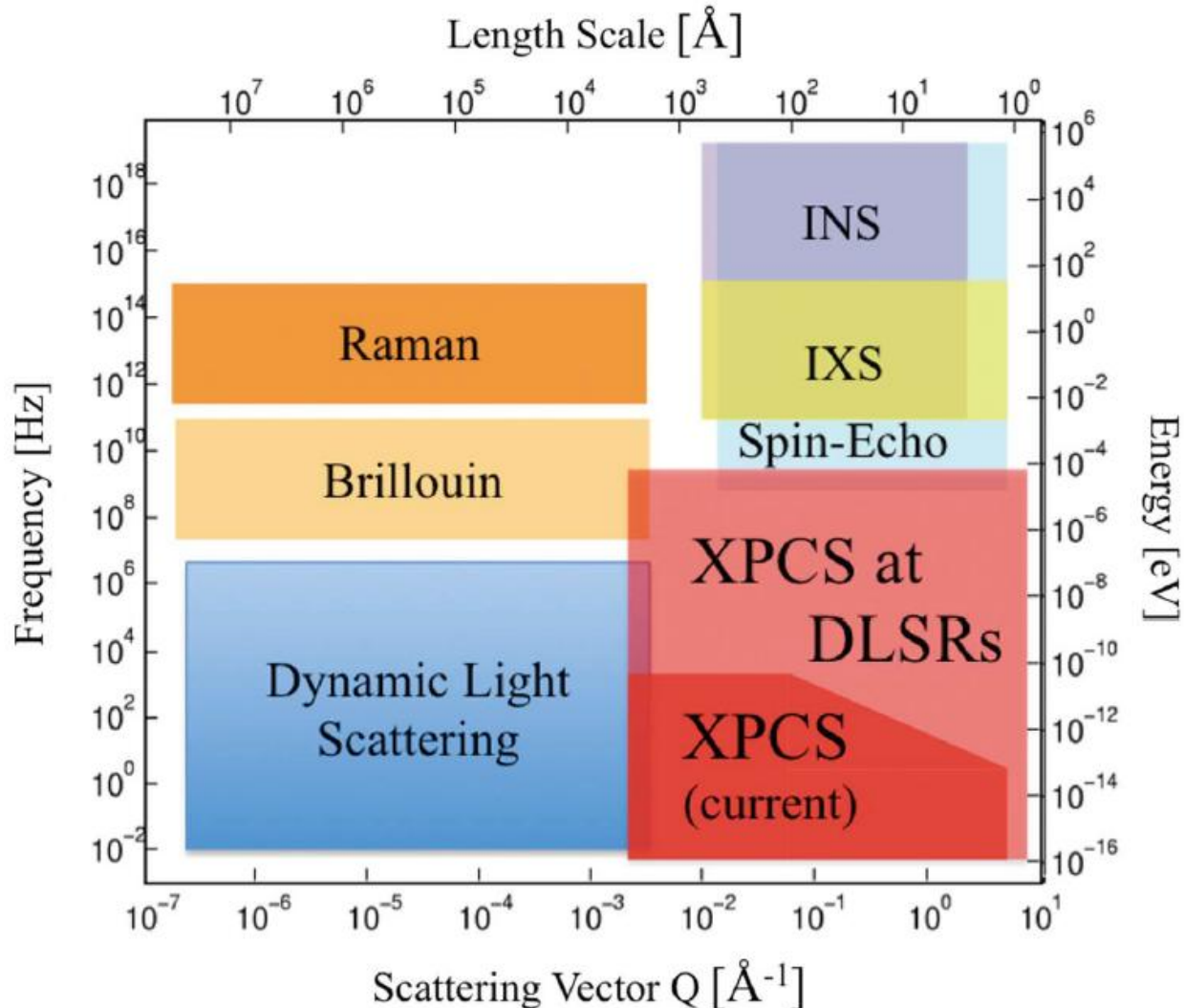
- Glassy systems
- X-ray Photon Correlation Spectroscopy
- Aging in metallic glasses
- Measurements *in operando* conditions
- Atomic dynamics in network & oxide glasses
- Conclusions and future perspectives

ESRF UPGRADE: EXTREMELY BRILLIANT SOURCE (EBS)



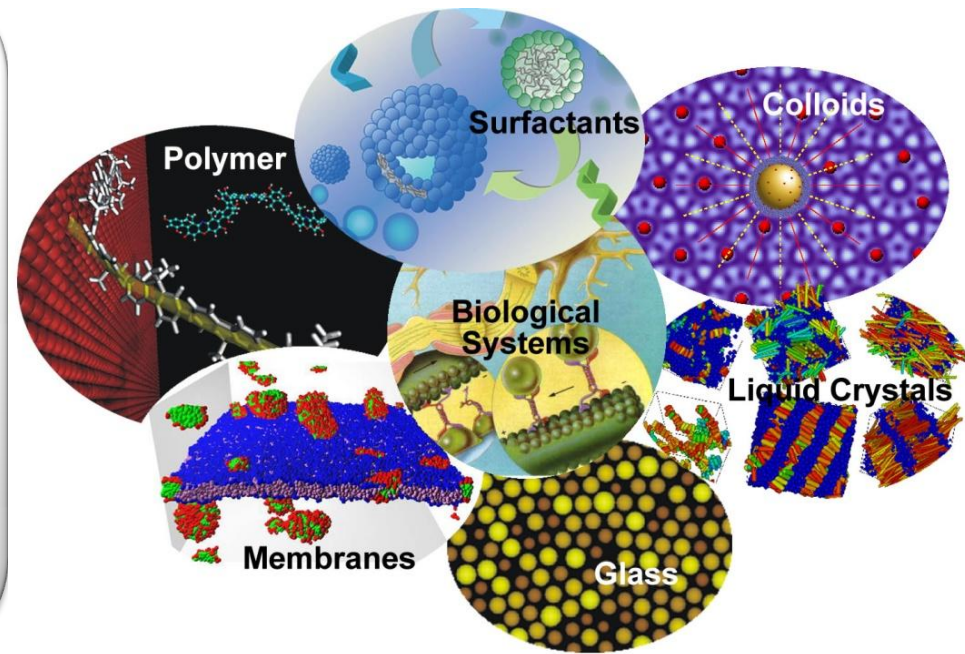
XPCS AT EBS

The EBS machine will lead to **~100X increase in coherent flux** → possibility to measure **faster time scales** (from 0.1 ms to 0.01 μ s) & to go at **higher energies** (up to 30 keV)



MANY NEW SCIENTIFIC OPPORTUNITIES

- protein dynamics in living cells
- dynamics under confinement
- dynamics of polymers, macromolecules, membranes, foams, ...
- dynamics at buried interfaces
- polyamorphism (LL/GG phase transitions)
- dynamics at extreme conditions



© Pierre Gilles De Gennes



Presentation of 8 Conceptual Design Report (CDR) for new possible beamlines

CDR1 - Beamline for coherence applications (XPCS)

CDR2 - Beamline for hard X-ray diffraction microscope

CDR3 - High throughput large field phase-contrast tomography beamline

CDR4 - Surface science beamline

CDR5 - Advanced high-flux nano-XRD beamline for science under extreme conditions

CDR6 - Facility for dynamic compression studies

CDR7 - High brilliance XAS beamline

CDR8 - Serial crystallography beamline

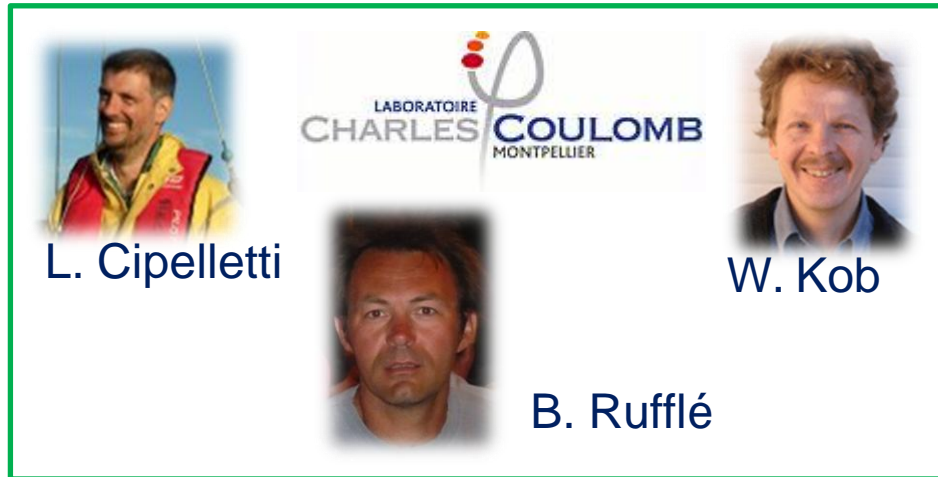
<http://www.esrf.eu/home/events/conferences/2016/ebs-science-workshop.html>

Possibility to investigate the slow dynamics in glasses at the atomic length scale

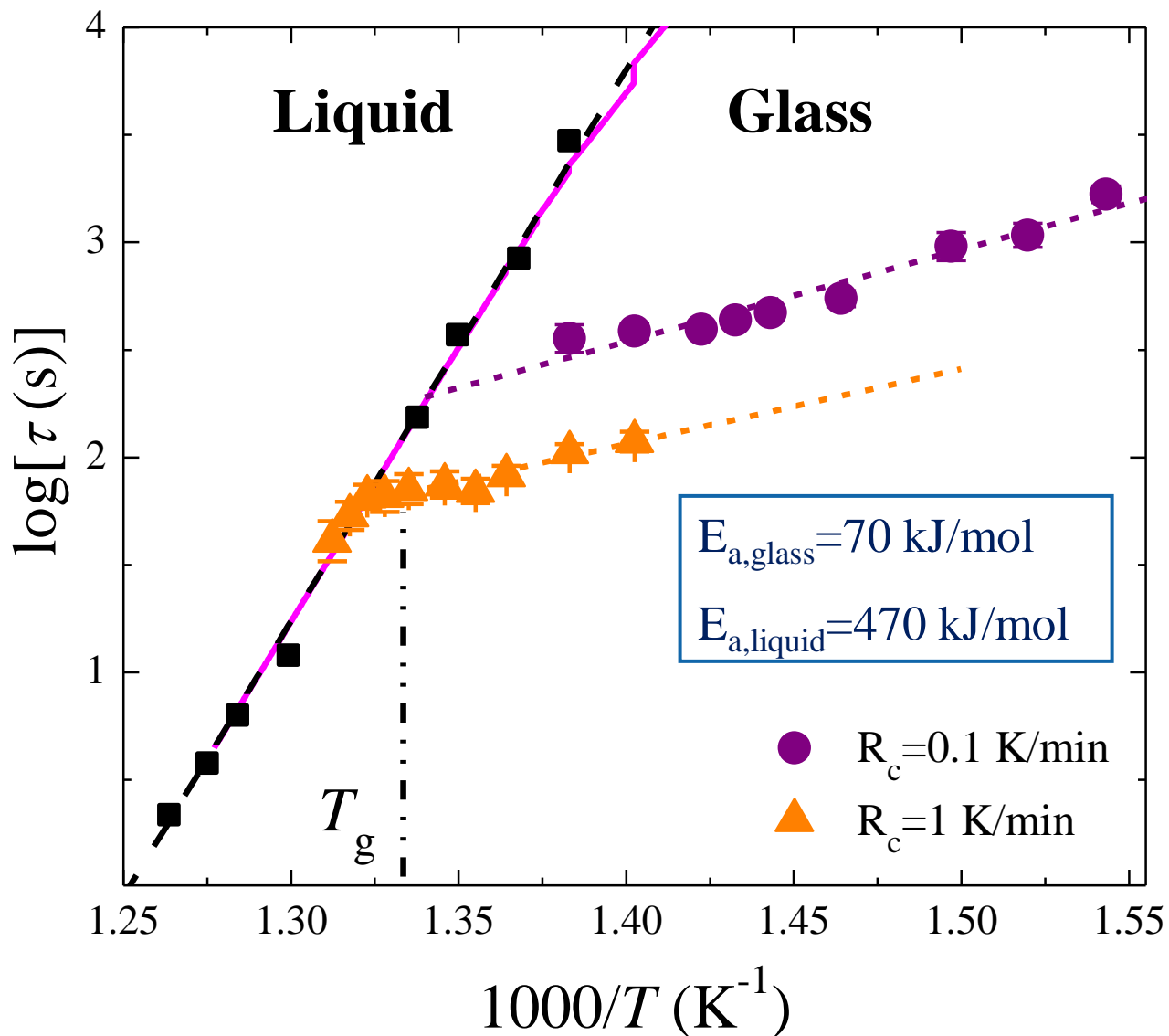
Depending on the nature of the system and the competition between intrinsic and induced dynamics is possible to move from spontaneous density fluctuations to intensity-driven atomic motion

Many new scientific opportunities with XPCS @ EBS

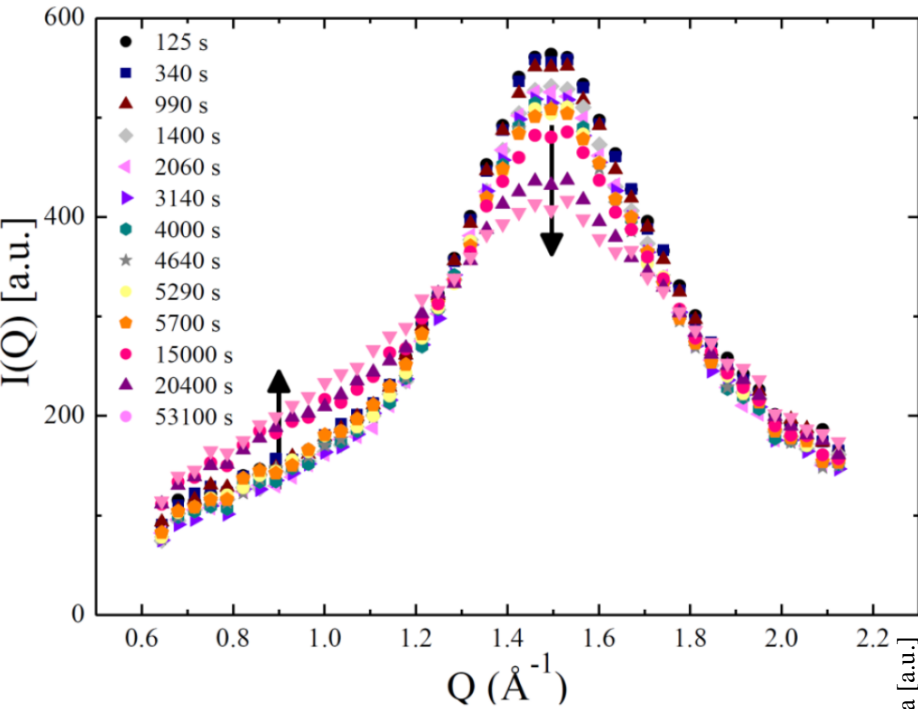
ACKNOWLEDGMENTS



Thank you for your attention!!!

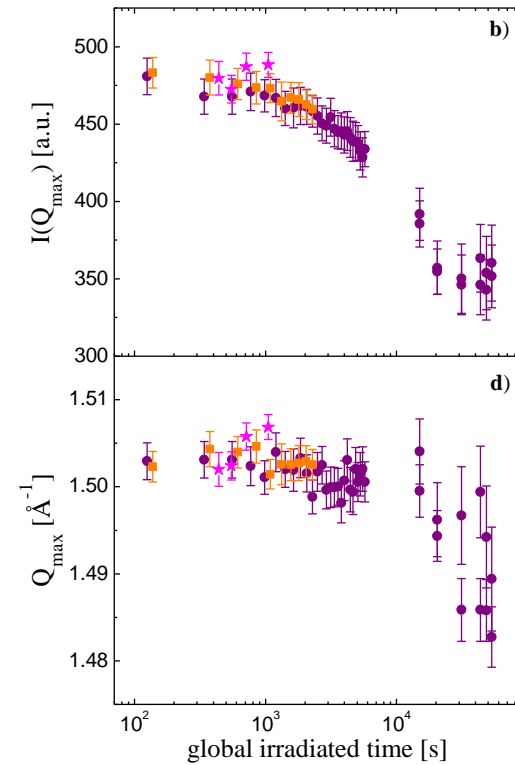
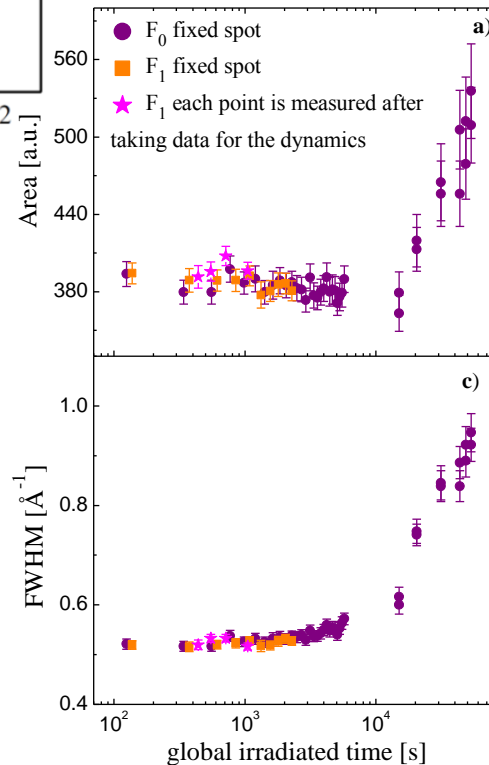


STRUCTURAL EVOLUTION

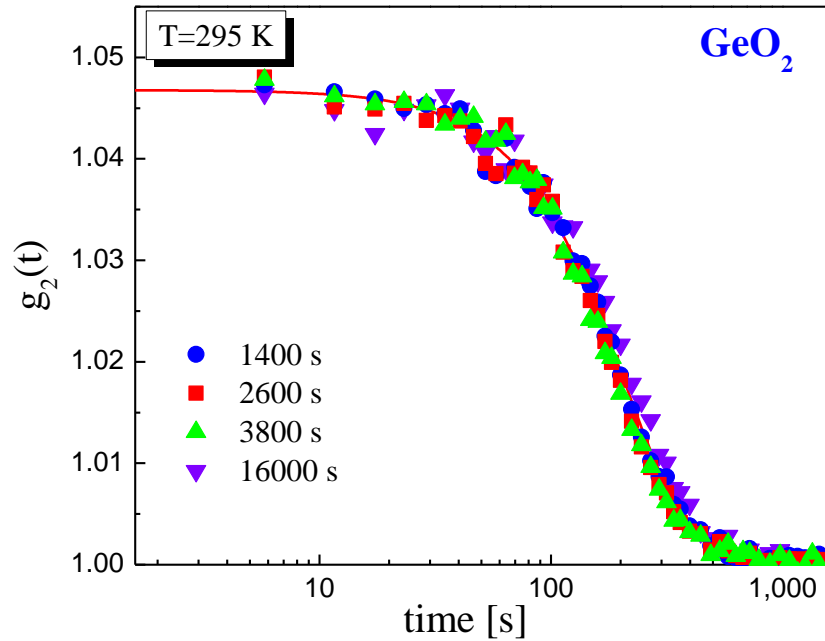


Occurrence of a real damage for longer exposure time

Threshold for structural damage at $\sim 10^4$ s of global irradiation at maximum flux

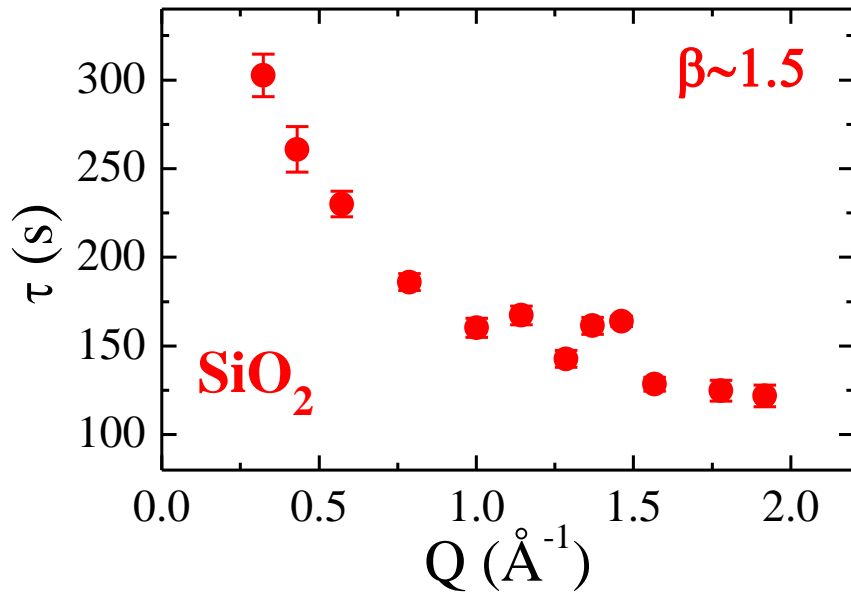


NO STANDARD RADIATION DAMAGE



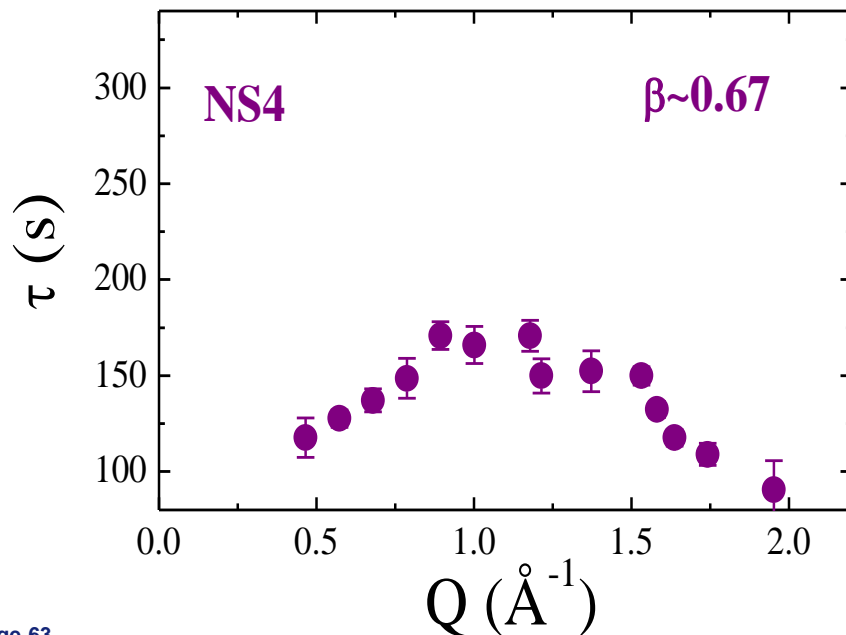
The dynamics is independent on the global accumulated dose

POSSIBILITY TO GET INTRINSIC PROPERTIES OF THE SYSTEM



Stretched vs compressed mettere una figura del confronto curve

Different Q dependence

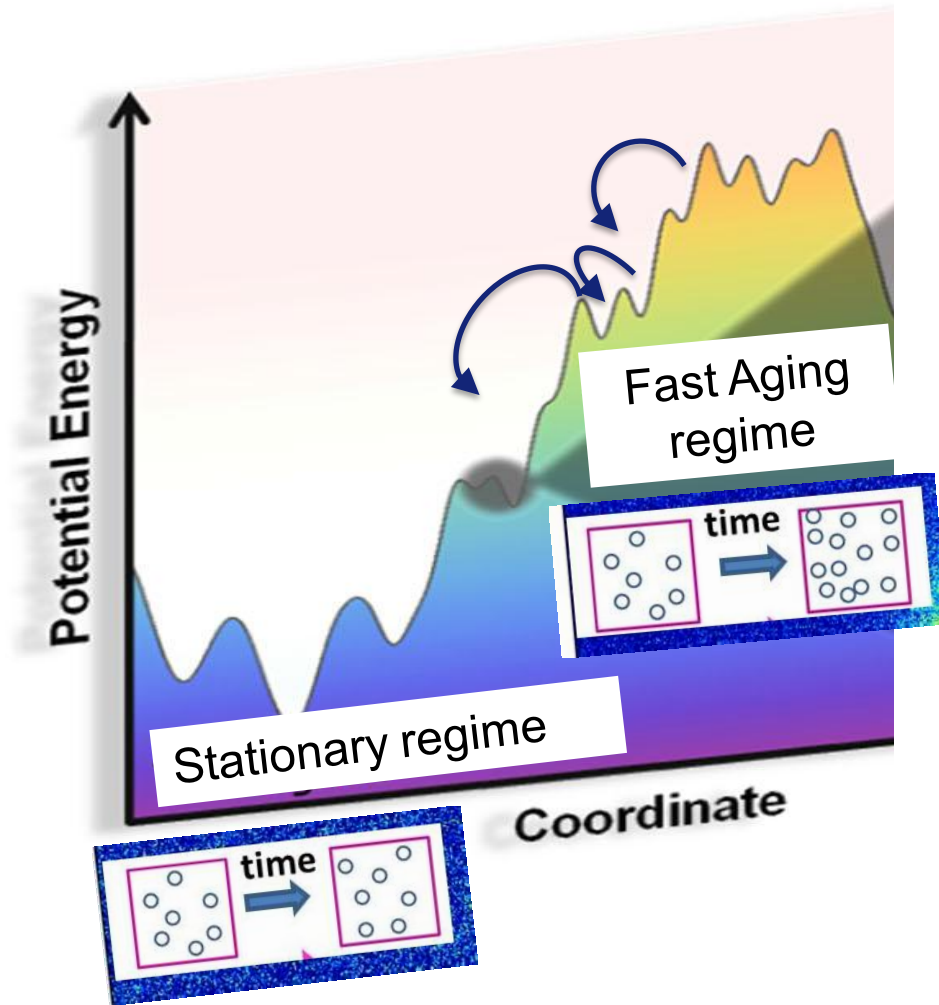


Microscopic aging:

Fast aging: thermal activation of a cascade of jumps from a high-energy minimum with irreversible atomic rearrangements changing density

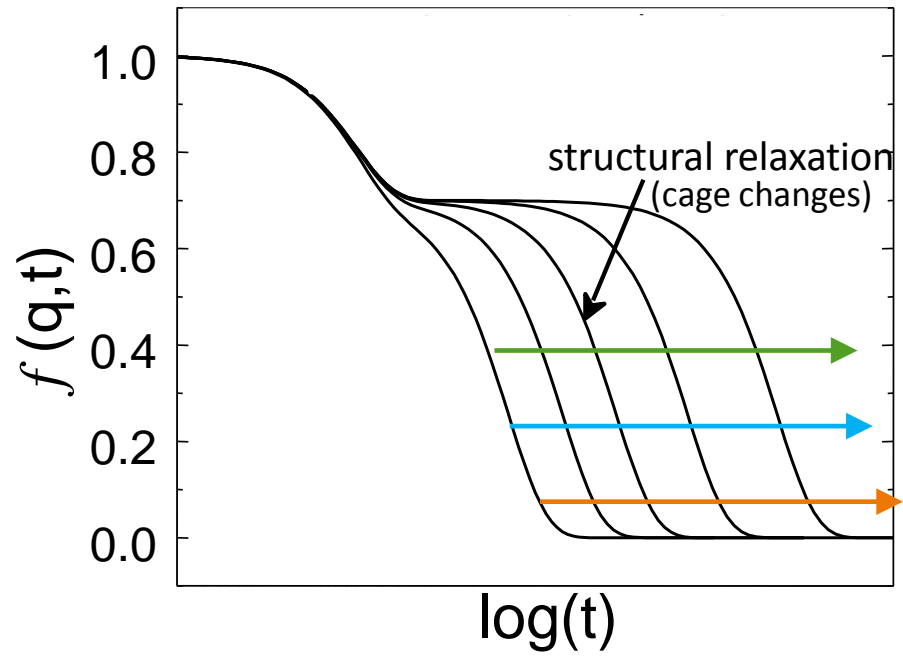
Stationary: more relaxed (low energy) minimum. Localized dynamics. no density change, MRO increase

→ the evolution between the two could be related to a **ductile to brittle transition**



RELAXATION DYNAMICS

The slow down of the dynamics toward an arrested state corresponds to a continuous shift of the decay of the density fluctuations toward longer time scales



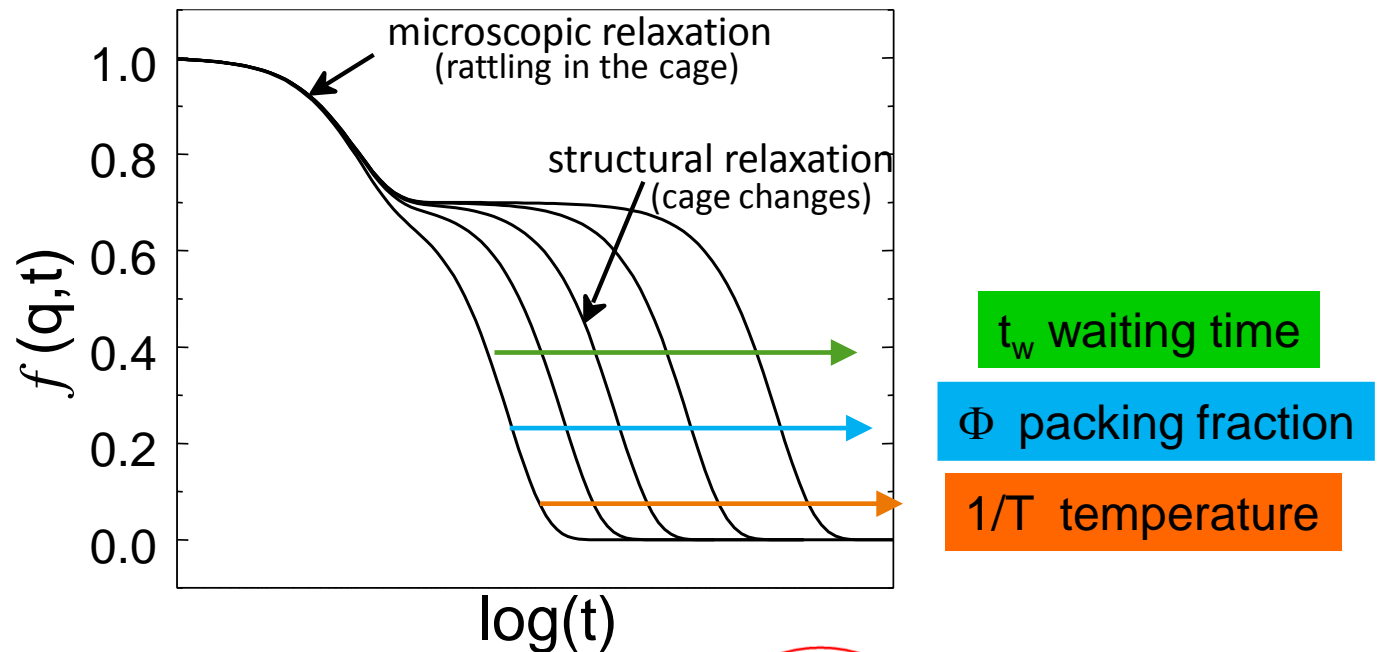
t_w waiting time

Φ packing fraction

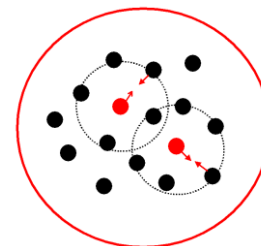
$1/T$ temperature

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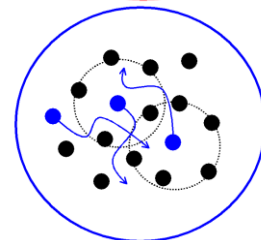


τ_1 microscopic relaxation time related to the interactions between a particle and the cage of its nearest neighbors.



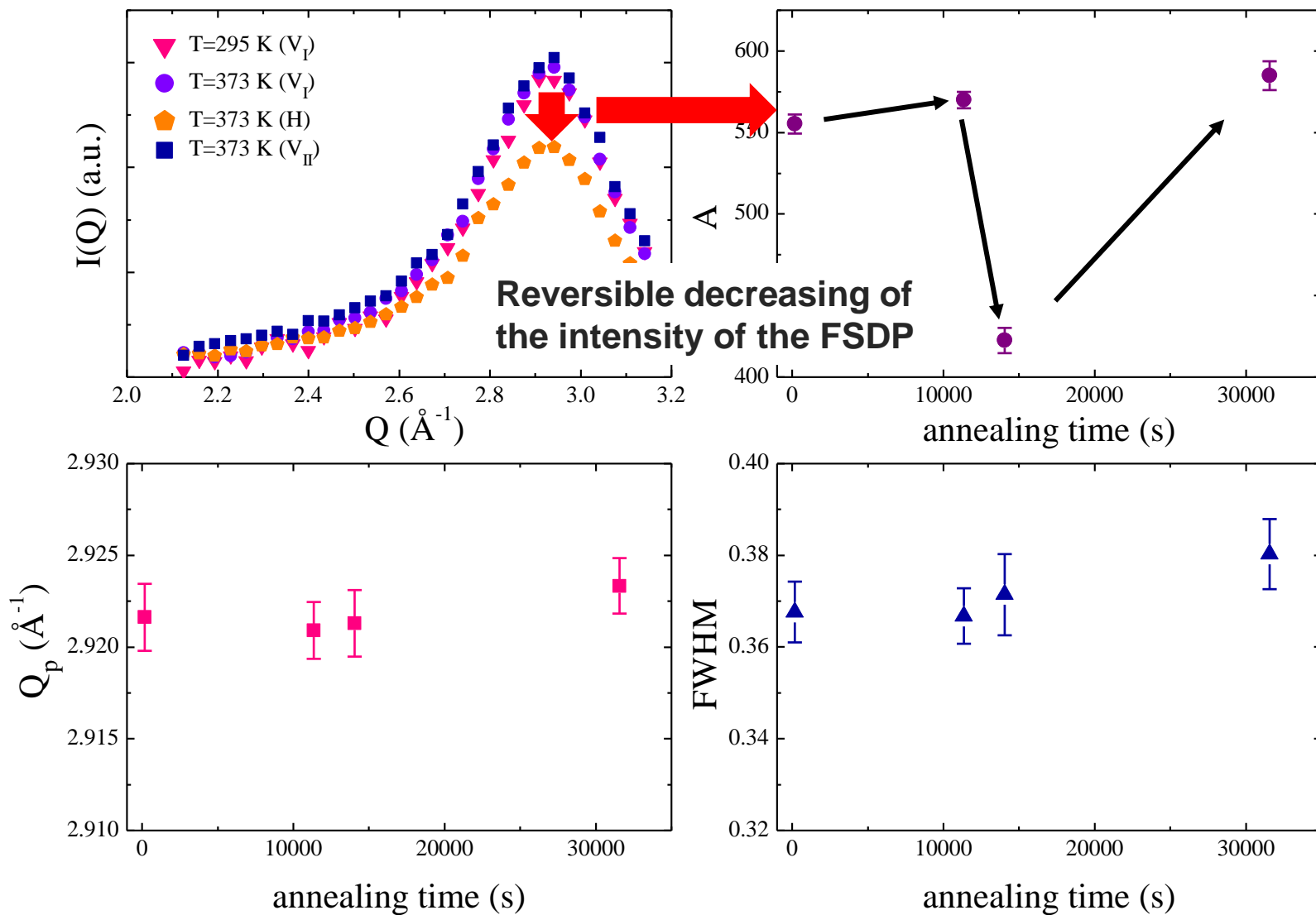
fast relaxation

τ_2 structural relaxation time related to a structural rearrangement of the particles.

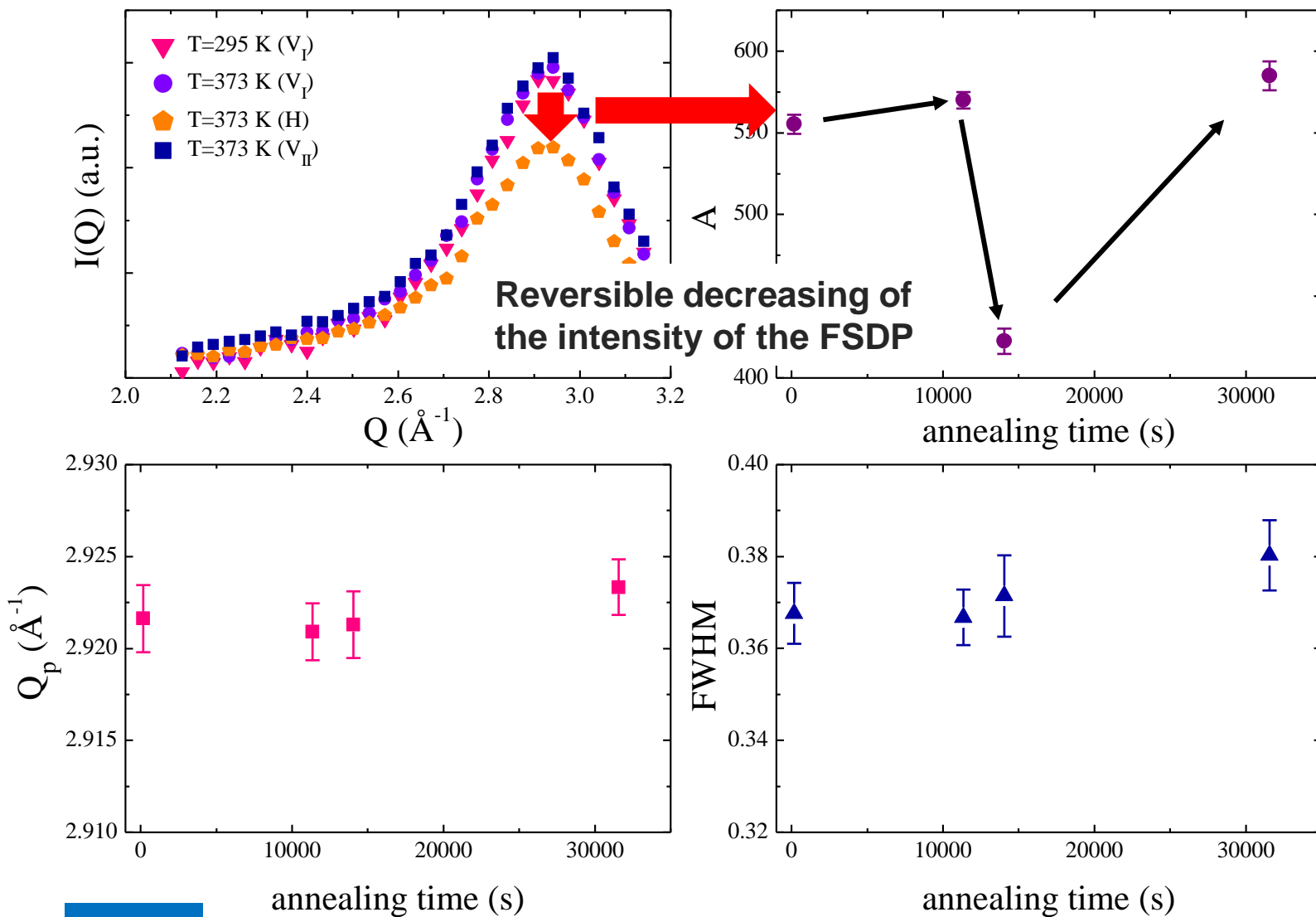


slow relaxation

EFFECT OF H₂ ON THE STRUCTURE



EFFECT OF H₂ ON THE STRUCTURE



XRD
ID10



Next step: high energy XRD & EFAPS