



# BORATE GLASSES

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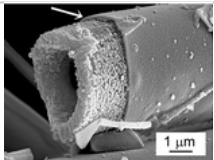
# BORATES – TECHNOLOGICAL INTEREST



*Since 1915*

## **Glass with thermal shock resistance (Pyrex)**

borosilicate glasses



*Since 60's*

## **Bioactive glasses as implant materials**

Na-K-Ca borate glasses



*Since 70's*

## **High-pressure sodium vapor lamp**

Alumino-borate glasses



*Since 70's*

## **Non-linear optical borate crystals**

Alkali-, Rare-Earth- borates

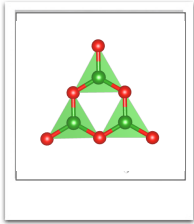


*Now*

## **Potential solid electrolytes and cathode materials**

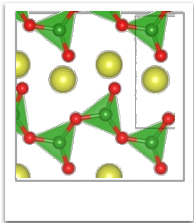
Alkali-borates with transition elements

# OUTLINE



## **B<sub>2</sub>O<sub>3</sub> glass**

Archetypal glass former

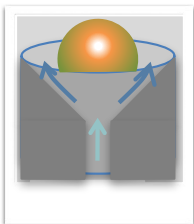


## **Alkali borate glasses (Li, Na, K, Rb, Cs)**

Vitrification domains

Alkali effects on physical properties

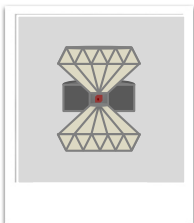
Glass structure: short- and intermediate-range order



## **From glass to melt**

Structure of alkali borate melts

Depolymerization of the borate network



## **Polyamorphism**

## B<sub>2</sub>O<sub>3</sub> VS SiO<sub>2</sub>

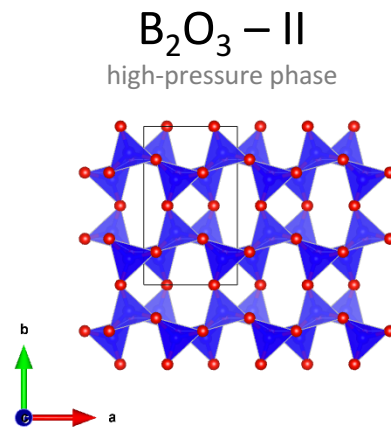
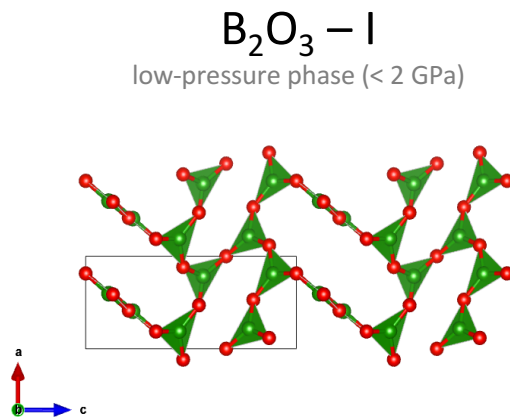
- An important advantage of borate glasses over silica glasses is their significantly lower melting temperature

	B <sub>2</sub> O <sub>3</sub>	SiO <sub>2</sub>
Average network connectivity	3	4
Mass Density	1.844 g.cm <sup>-3</sup>	2.202 g.cm <sup>-3</sup>
T <sub>g</sub>	260°C	1100°C
T <sub>m</sub>	450°C	1728°C
Liquid viscosity (log $\eta$ at 1200°C)	1.66 P	12.6 P
chemical durability	low	high
Thermal expansion ( $\alpha$ )	161.6 × 10 <sup>-7</sup> K <sup>-1</sup>	5.35 × 10 <sup>-7</sup> K <sup>-1</sup>

- Used in combination with other oxides such as Al<sub>2</sub>O<sub>3</sub> or SiO<sub>2</sub>, this leads to:
  - improve chemical durability,
  - decrease melting temperatures.

## PECULIARITIES OF $B_2O_3$

- $B_2O_3$  does not crystallize at ambient pressure : « *ideal glass former* »
- But two polymorphs of  $B_2O_3$  can be formed when pressure is applied

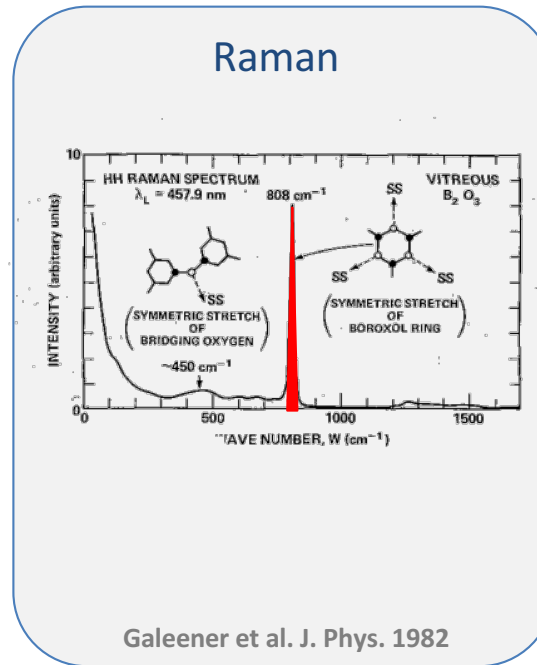
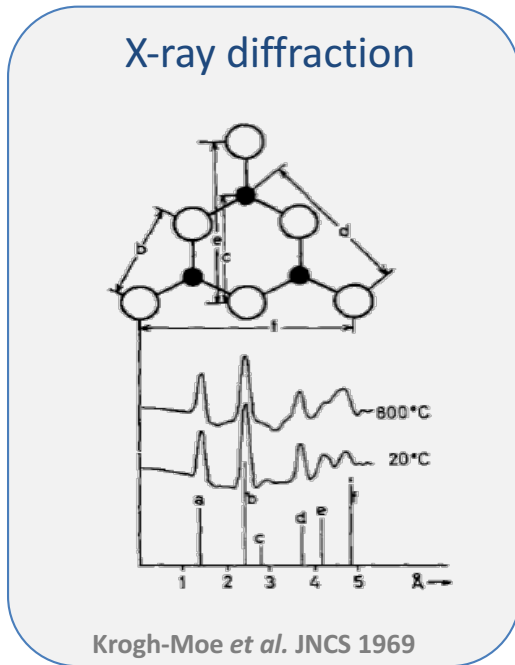
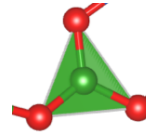


$$d(v-B_2O_3) = 1.84 \text{ g.cm}^{-3} \ll d(B_2O_3 - I) = 2.55 \text{ g.cm}^{-3}$$

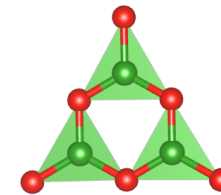
→ Why the glass structure is less compact ?

# PECULIARITIES OF $B_2O_3$

- Glassy  $B_2O_3$  is solely constituted by  $BO_3$  units



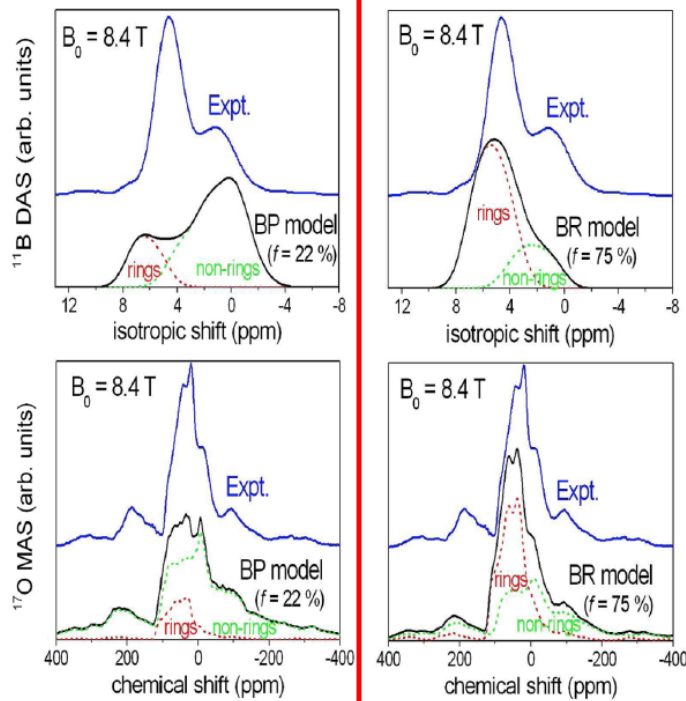
Boroxol rings  
( $B_3O_6$ )



→ Existence of an intermediate range order in glass ...

# PROPORTION OF BOROXOLS IN V-B<sub>2</sub>O<sub>3</sub> ?

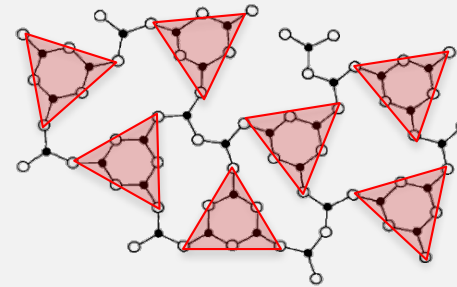
[<sup>11</sup>B and <sup>17</sup>O] NMR



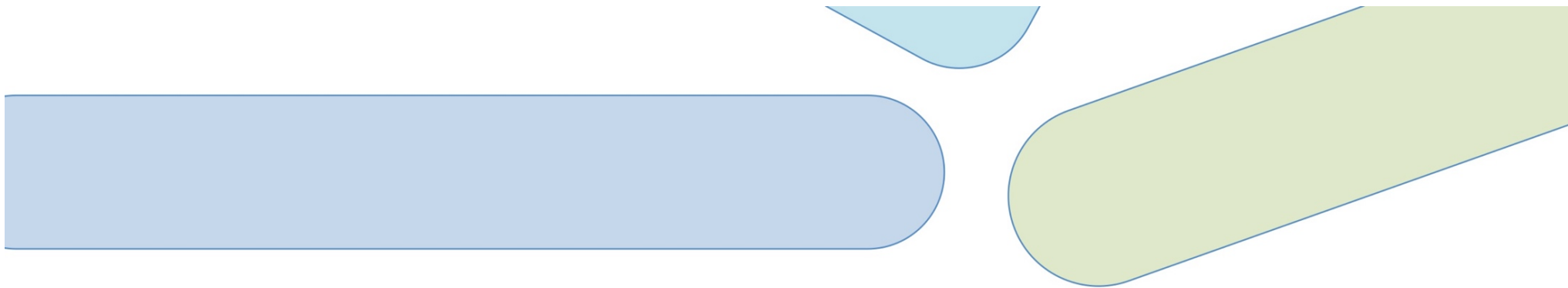
22%  
boroxol rings

75%  
boroxol rings

~75% of boroxol rings in B<sub>2</sub>O<sub>3</sub> glass



Krogh-Moe JNCS 1969 / Ferlat PRL 2008



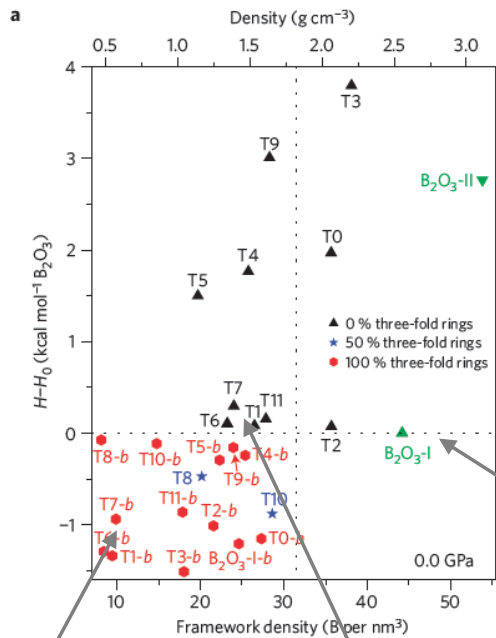
Why B<sub>2</sub>O<sub>3</sub> does not crystallize at ambient pressure ?



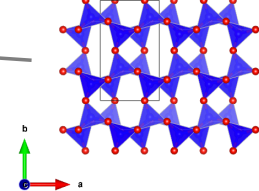
# PECULIARITIES OF $B_2O_3$

- Predictions of crystalline  $B_2O_3$  forms

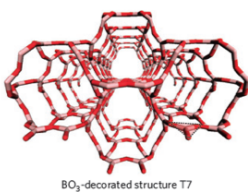
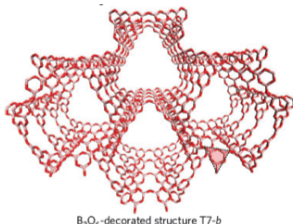
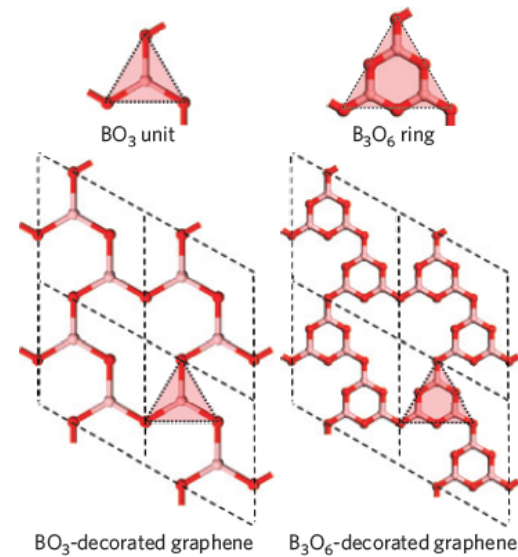
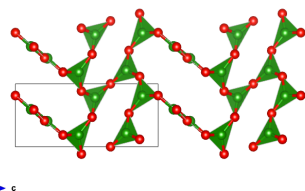
Ambient pressure



$B_2O_3$  - II  
high-pressure phase



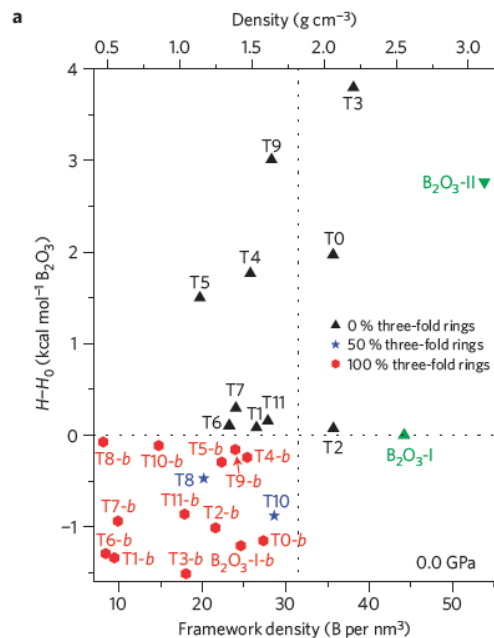
$B_2O_3$  - I  
low-pressure phase (< 2 GPa)



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- Predictions of crystalline  $B_2O_3$  forms

Ambient pressure

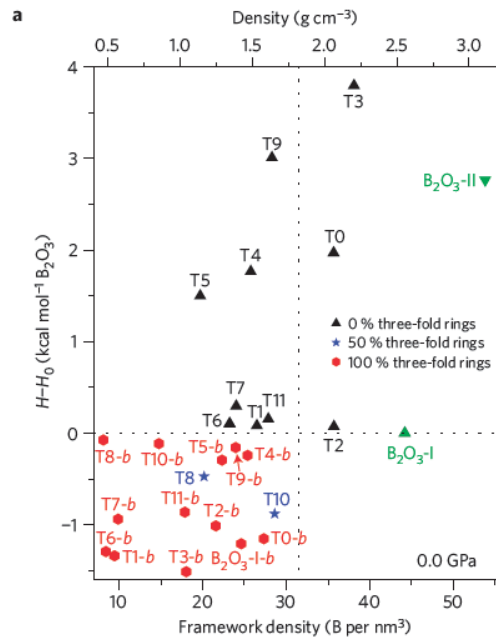


→ At ambient pressure, the crystallization is avoided as a result of the existence of several competing phases that eventually induces the system amorphization.

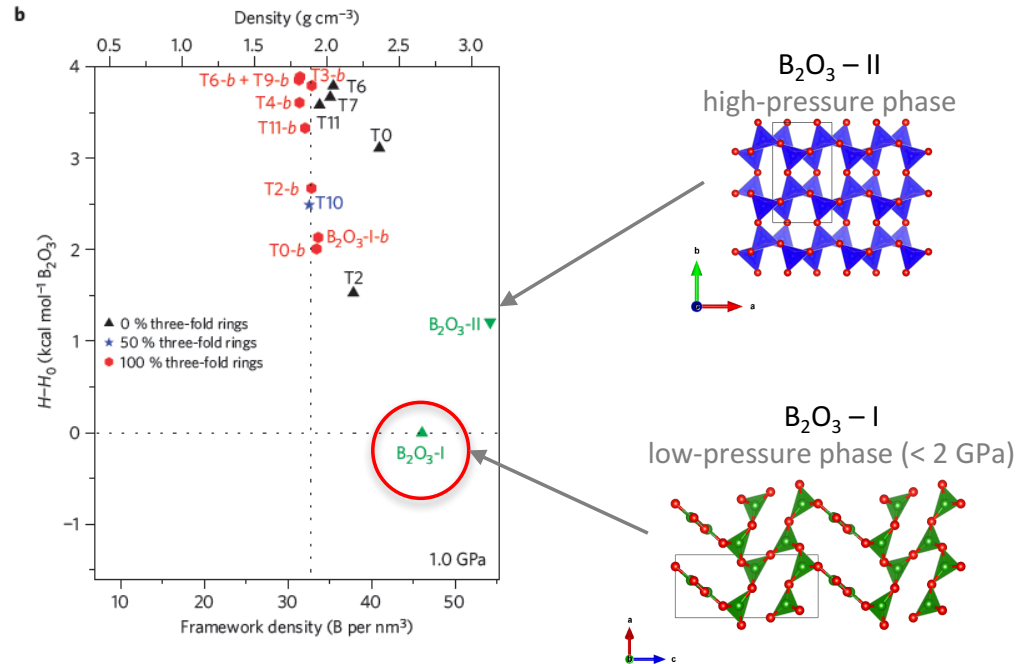
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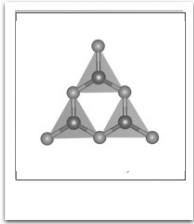
Ambient pressure



1 GPa

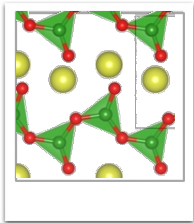


# OUTLINE



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Archetypal glass former

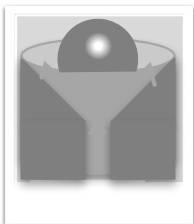


## **Alkali borate glasses (Li, Na, K, Rb, Cs)**

Vitrification domains

Alkali effects on physical properties

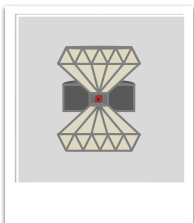
Glass structure: short- and intermediate-range order



## **From glass to melt**

Structure of alkali borate melts

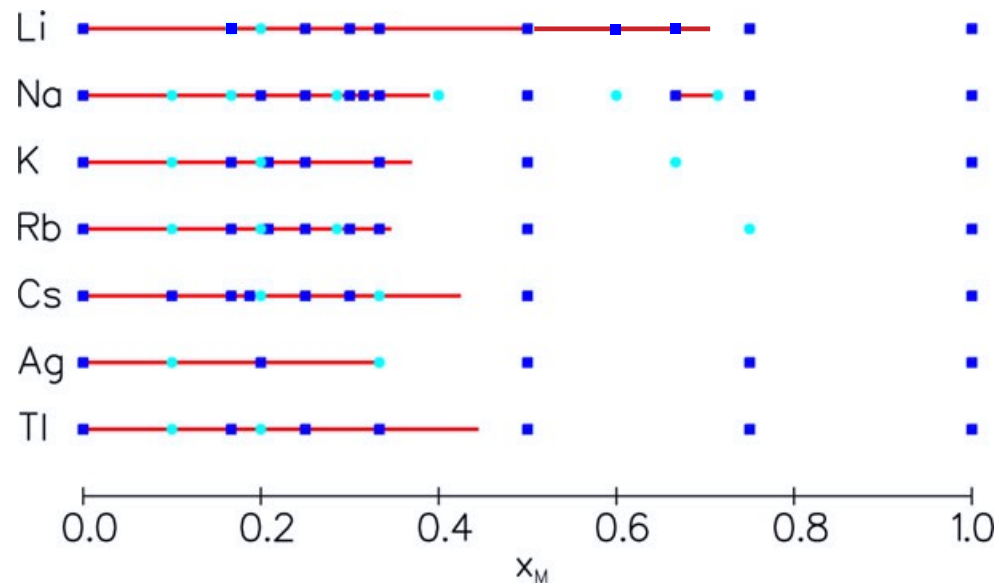
Depolymerization of the borate network



## **Polyamorphism**

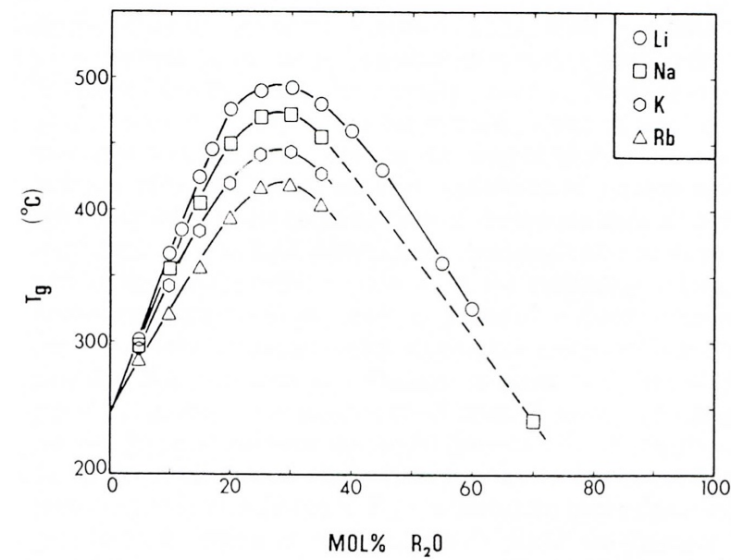
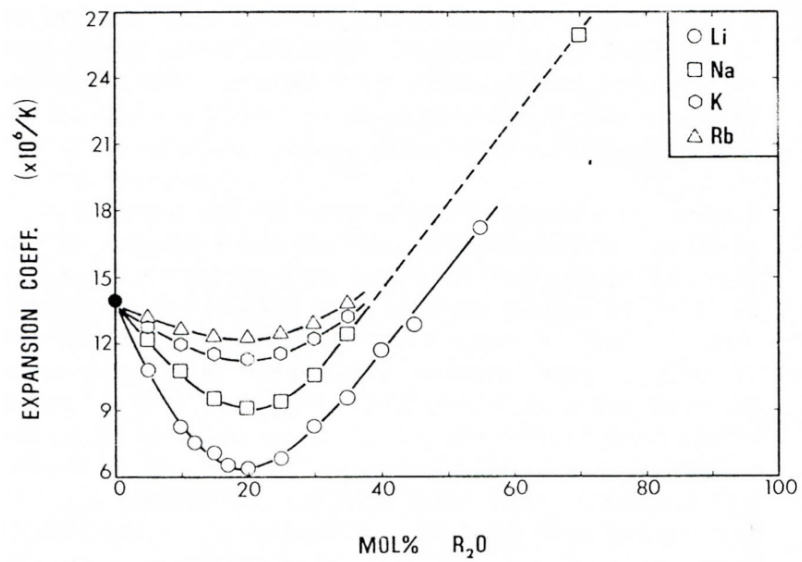
# ALKALI BORATE: VITRIFICATION DOMAINS

- A large composition range:



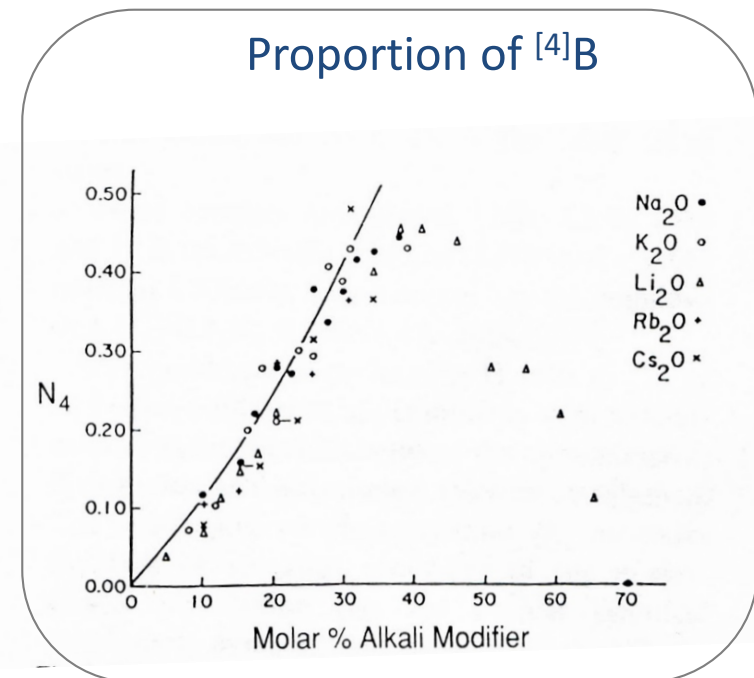
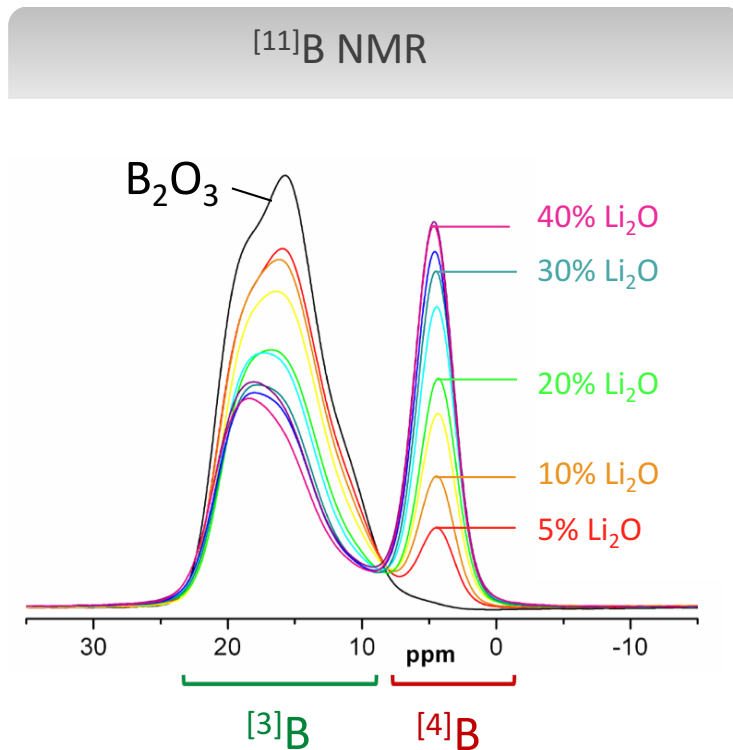
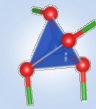
- crystalline phase with unknown structures
- crystalline phase with known structures
- glass forming region

# BORATE ANOMALIES

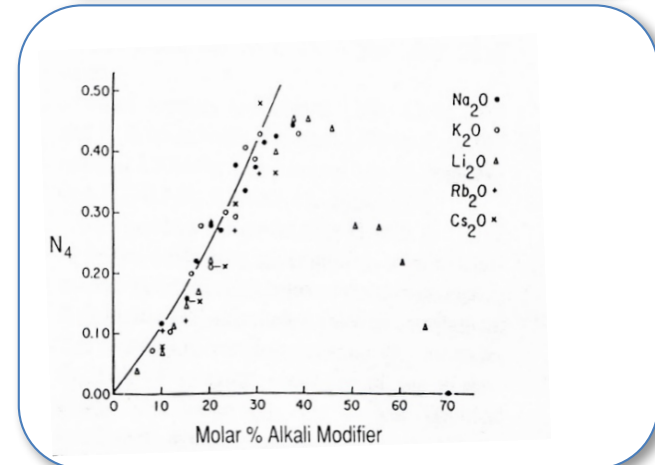
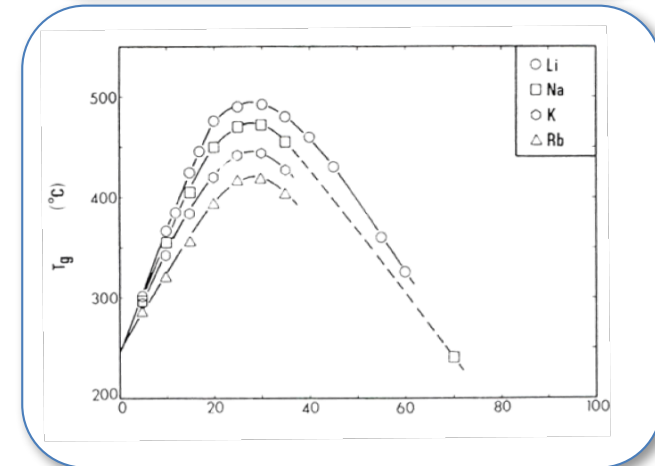
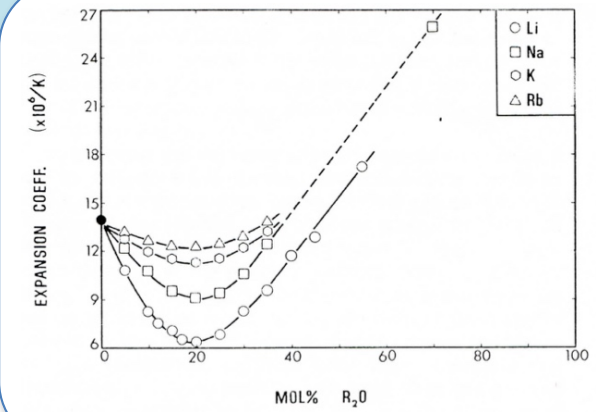


# WHAT HAPPENS WHEN WE ADD ALKALIS ?

- By adding alkali oxide into  $B_2O_3$  :  $[^3]B \rightarrow [^4]B$  conversion



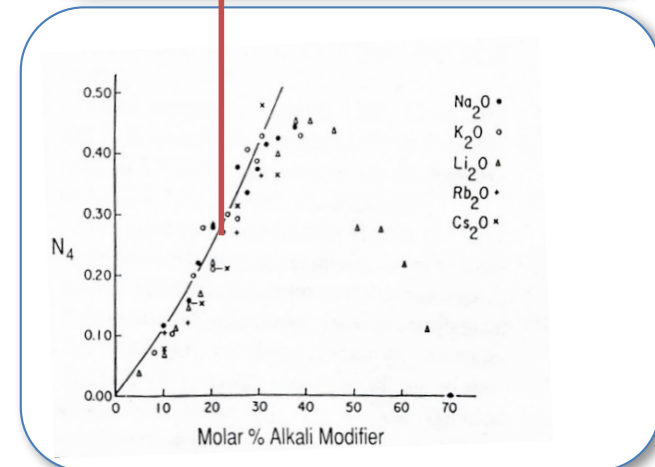
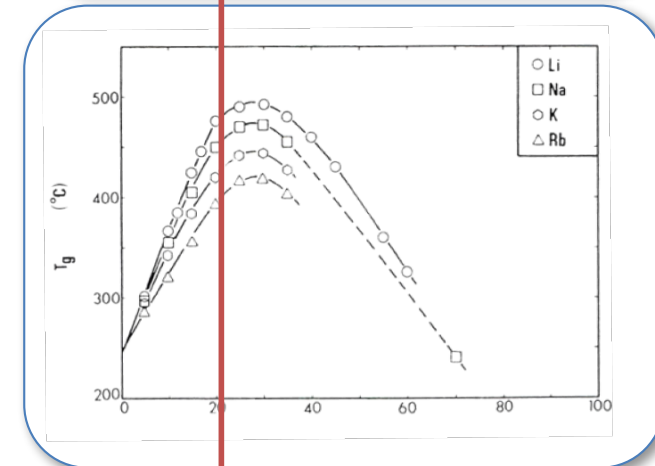
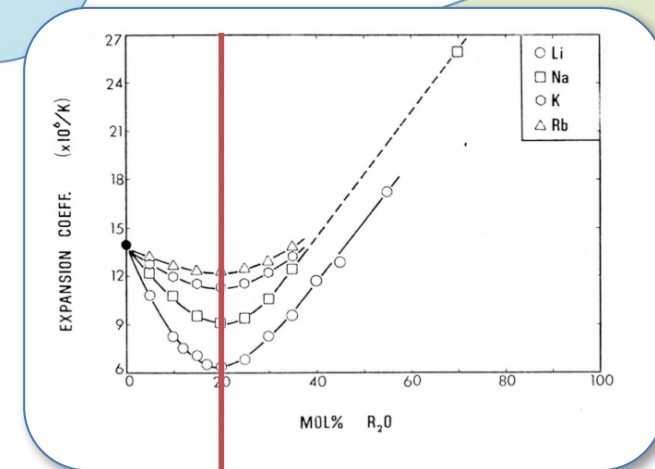
# RELATION $N_4$ AND THE ANOMALIES ?





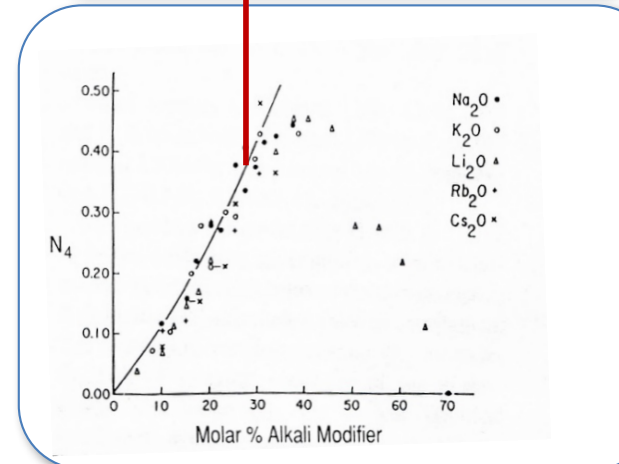
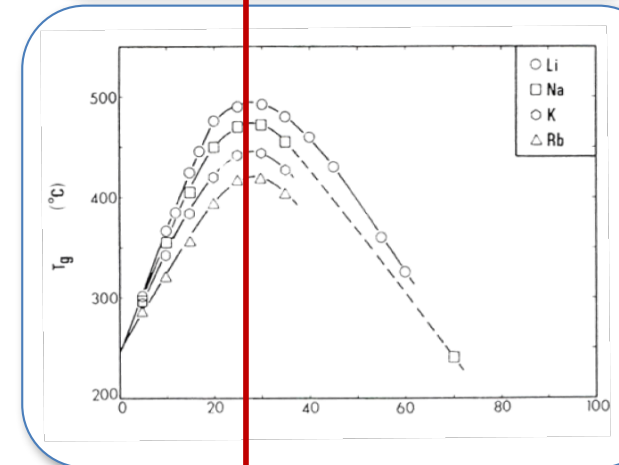
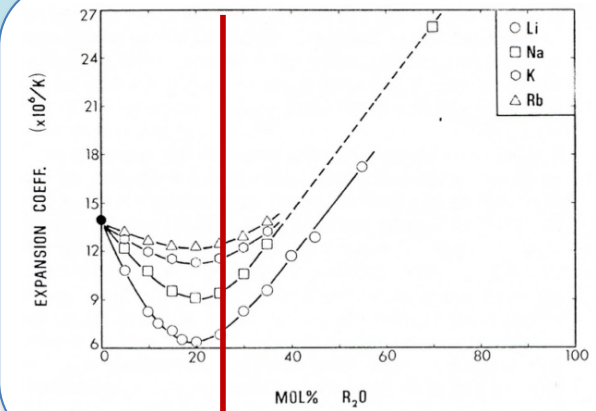
## RELATION $N_4$ AND THE ANOMALIES ?

- Expansion coefficient minimum at 20mol%



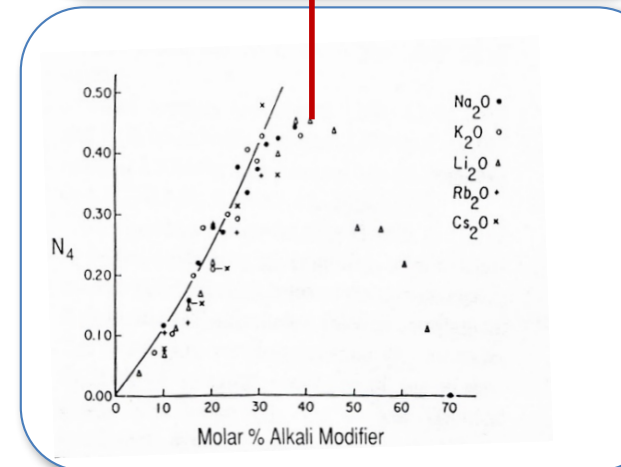
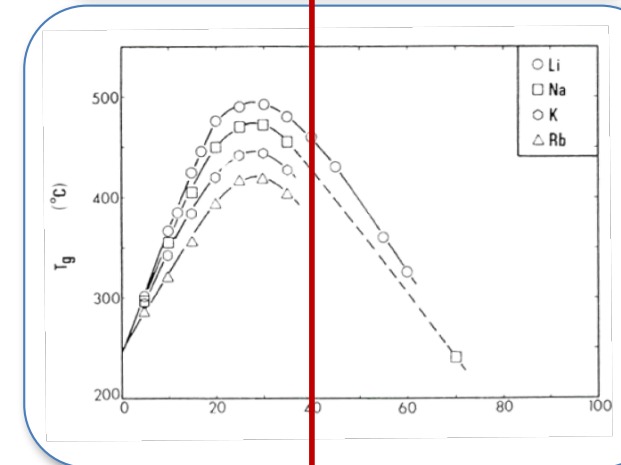
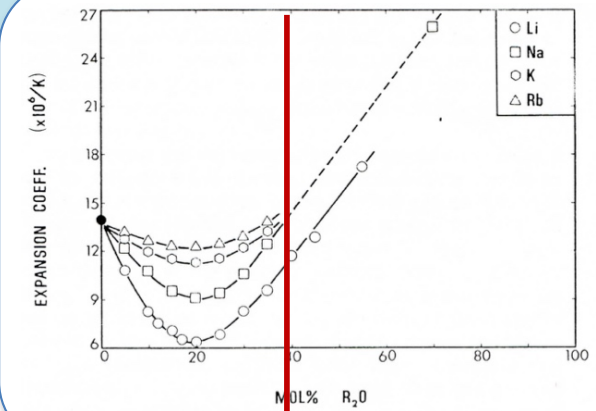
## RELATION $N_4$ AND THE ANOMALIES ?

- Expansion coefficient minimum at 20mol%
- $T_g$  maximum at 27mol%



## RELATION $N_4$ AND THE ANOMALIES ?

- Expansion coefficient minimum at 20mol%
- $T_g$  maximum at 27mol%
- $N_4$  maximum around 45mol%

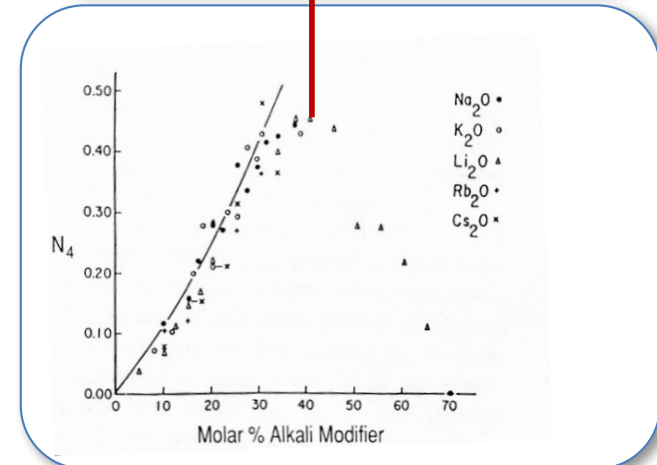
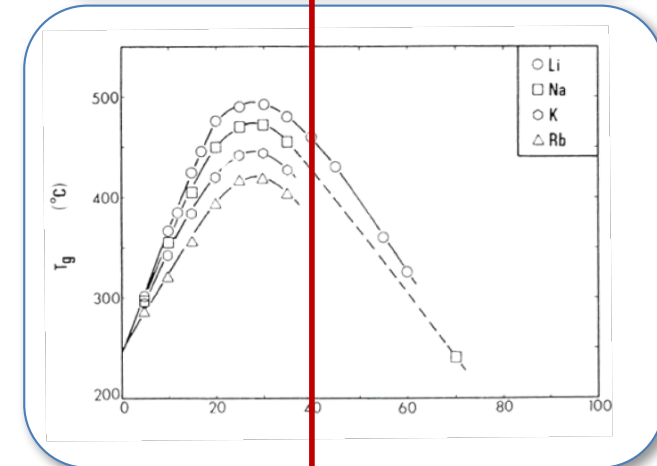
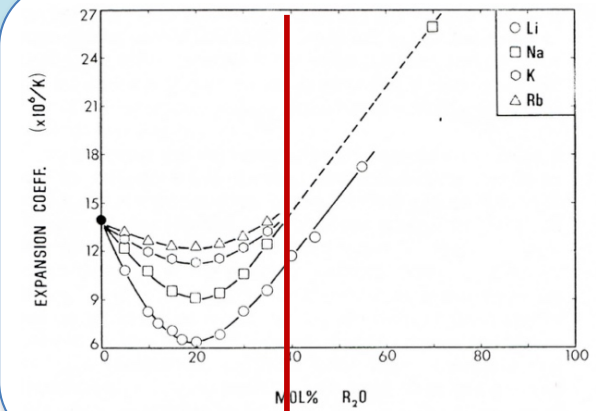


## RELATION $N_4$ AND THE ANOMALIES ?

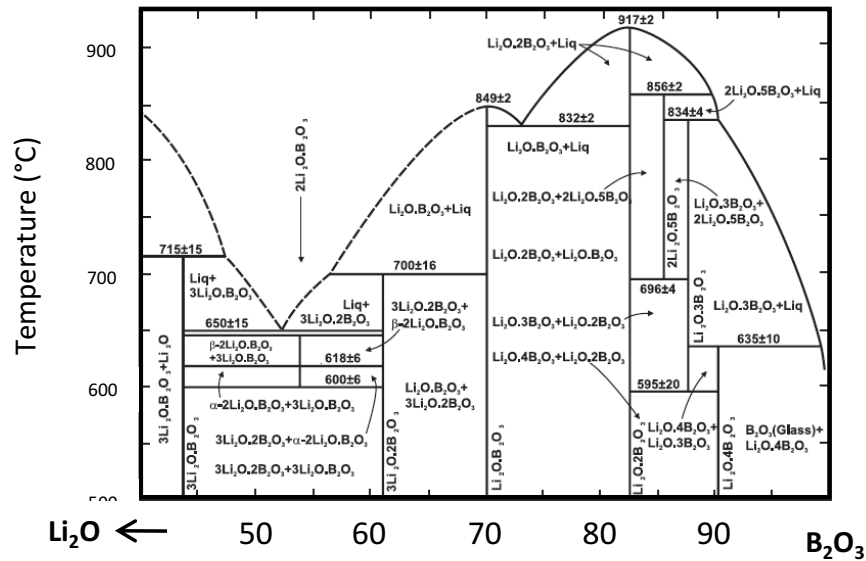
- Expansion coefficient minimum at 20mol%
- $T_g$  maximum at 27mol%
- $N_4$  maximum around 45mol%

→ *The origin of these anomalies is not fully ascribable to the presence of  $[4]B...$*

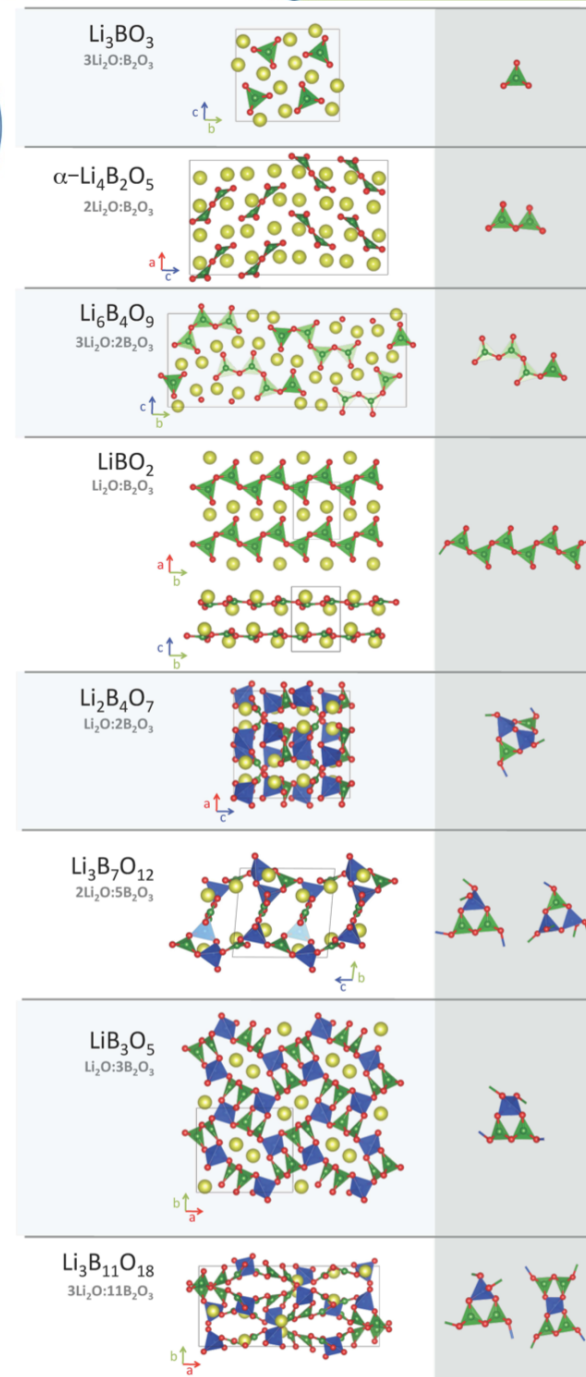
→ *Let's have a look at the crystalline samples...*



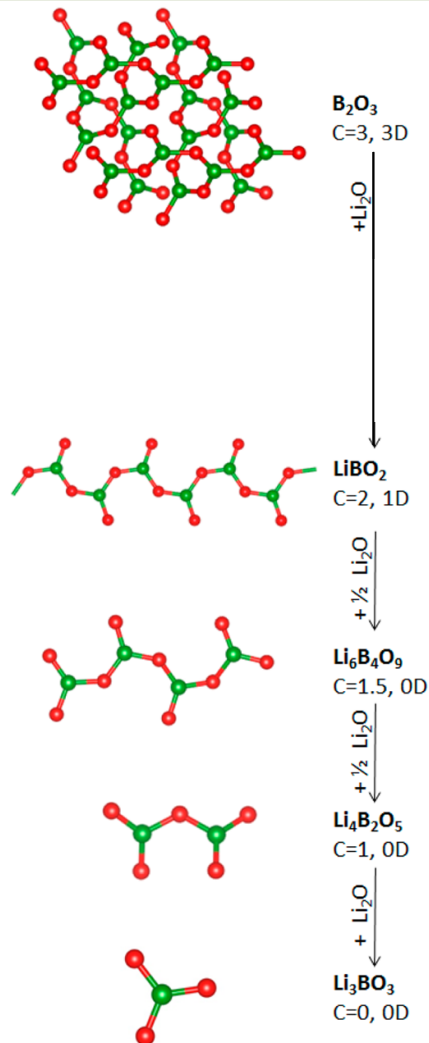
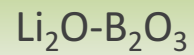
# Li<sub>2</sub>O-B<sub>2</sub>O<sub>3</sub> SYSTEM



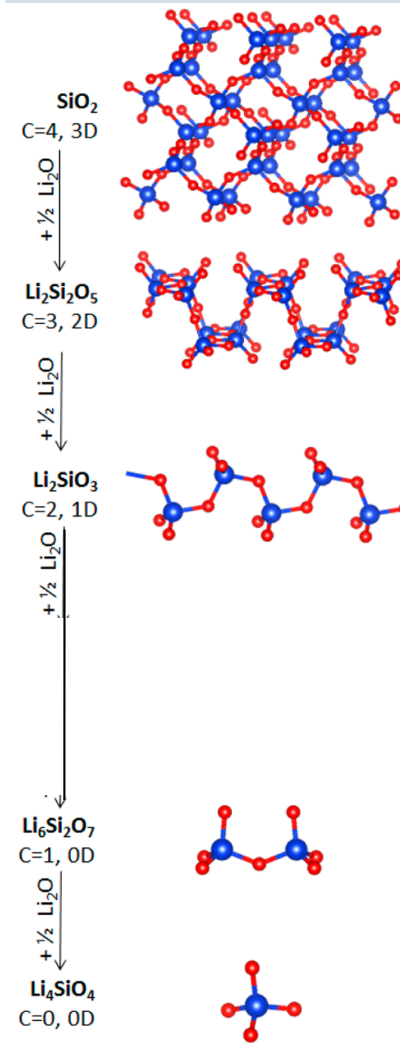
→ Presence of complex structures called *superstructural units*



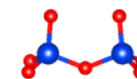
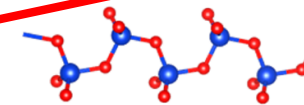
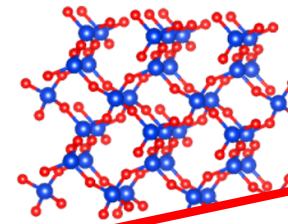
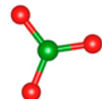
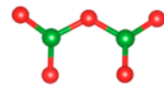
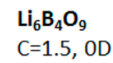
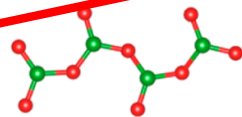
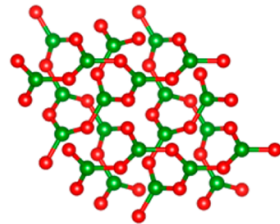
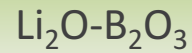
# DEPOLYMERIZATION MECHANISMS



Reduction of dimensionality



# DEPOLYMERIZATION MECHANISMS



Almost perfect parallelism between  $\text{B}_2\text{O}_3$  and  $\text{SiO}_2$

## Li<sub>2</sub>O-B<sub>2</sub>O<sub>3</sub>

Stoichiometry	Crystal cell	Constituting units
Li <sub>3</sub> BO <sub>3</sub> 3Li <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
α-Li <sub>4</sub> B <sub>2</sub> O <sub>5</sub> 2Li <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
Li <sub>6</sub> B <sub>4</sub> O <sub>9</sub> 3Li <sub>2</sub> O:2B <sub>2</sub> O <sub>3</sub>		
LiBO <sub>2</sub> Li <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
Li <sub>2</sub> B <sub>4</sub> O <sub>7</sub> Li <sub>2</sub> O:2B <sub>2</sub> O <sub>3</sub>		
Li <sub>3</sub> B <sub>7</sub> O <sub>11</sub> 2Li <sub>2</sub> O:5B <sub>2</sub> O <sub>3</sub>		
LiB <sub>3</sub> O <sub>5</sub> Li <sub>2</sub> O:3B <sub>2</sub> O <sub>3</sub>		
Li <sub>3</sub> B <sub>11</sub> O <sub>18</sub> 3Li <sub>2</sub> O:11B <sub>2</sub> O <sub>3</sub>		

## Na<sub>2</sub>O-B<sub>2</sub>O<sub>3</sub>

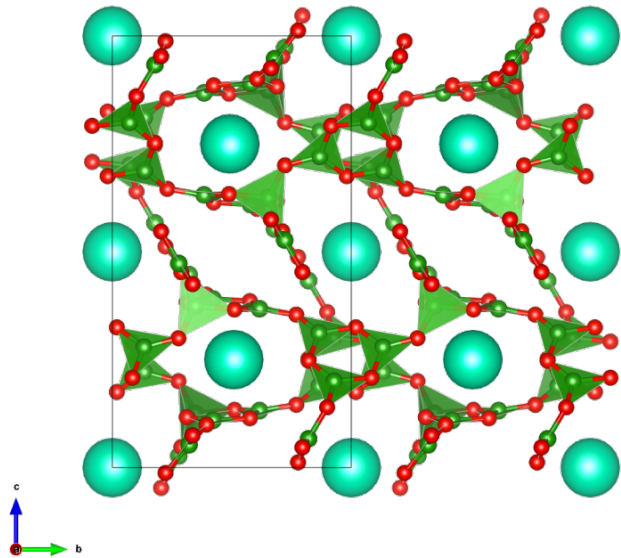
Stoichiometry	Crystal cell	Constituting units
Na <sub>3</sub> BO <sub>3</sub> 3Na <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
Na <sub>4</sub> B <sub>2</sub> O <sub>5</sub> 2Na <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
Na <sub>6</sub> B <sub>4</sub> O <sub>9</sub> 3Na <sub>2</sub> O:2B <sub>2</sub> O <sub>3</sub>	?	
NaBO <sub>2</sub> Na <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
Na <sub>2</sub> B <sub>4</sub> O <sub>7</sub> Na <sub>2</sub> O:2B <sub>2</sub> O <sub>3</sub>		
Na <sub>3</sub> B <sub>7</sub> O <sub>12</sub> 3Na <sub>2</sub> O:7B <sub>2</sub> O <sub>3</sub>		
NaB <sub>3</sub> O <sub>5</sub> Na <sub>2</sub> O:3B <sub>2</sub> O <sub>3</sub>		
Na <sub>2</sub> B <sub>8</sub> O <sub>13</sub> Na <sub>2</sub> O:4B <sub>2</sub> O <sub>3</sub>		

## K<sub>2</sub>O-B<sub>2</sub>O<sub>3</sub>

Stoichiometry	Crystal cell	Constituting units
KBO <sub>2</sub> K <sub>2</sub> O:B <sub>2</sub> O <sub>3</sub>		
K <sub>2</sub> B <sub>4</sub> O <sub>7</sub> K <sub>2</sub> O:2B <sub>2</sub> O <sub>3</sub>		
KB <sub>3</sub> O <sub>5</sub> K <sub>2</sub> O:3B <sub>2</sub> O <sub>3</sub>		
KB <sub>5</sub> O <sub>8</sub> K <sub>2</sub> O:5B <sub>2</sub> O <sub>3</sub>		

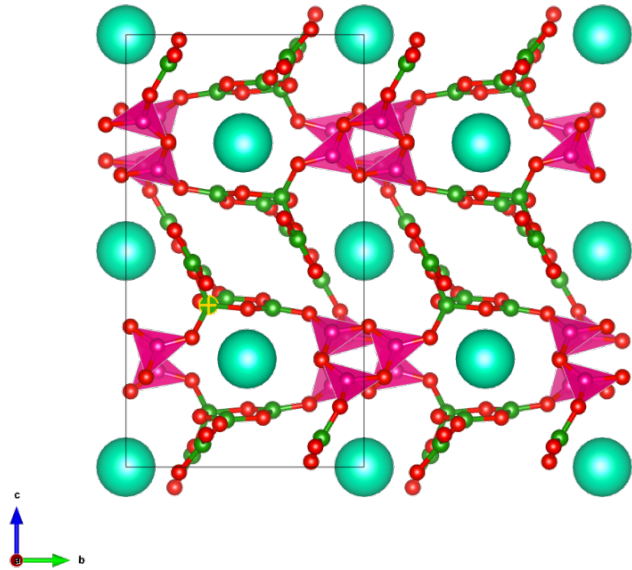


# CAESIUM ENEBORATE CASE



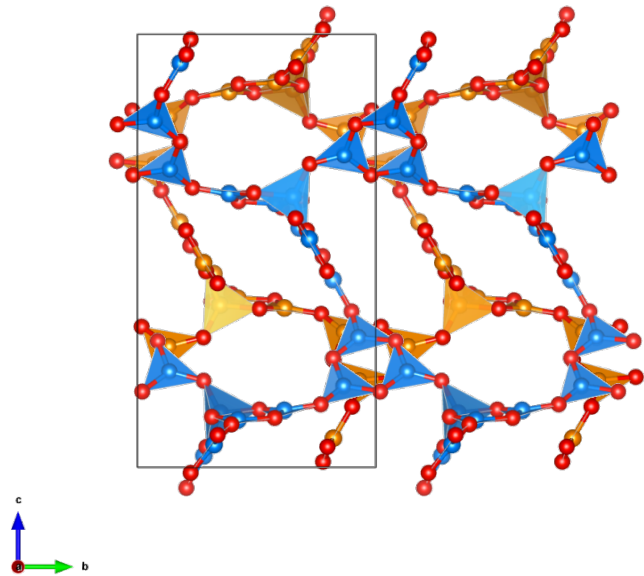
□ 90%  $\text{BO}_3$  units

## CAESIUM ENEBORATE CASE



- 90%  $\text{BO}_3$  units
- Unique crystal containing **boroxol rings**
- 30% of  $\text{BO}_3$  units involved in boroxol rings

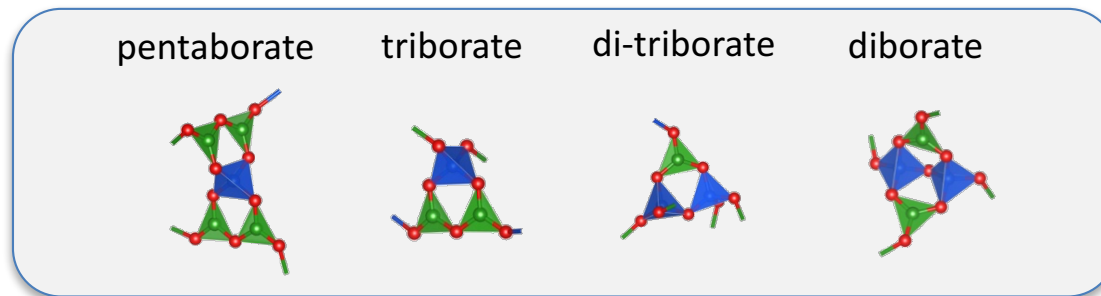
## CAESIUM ENEBORATE CASE



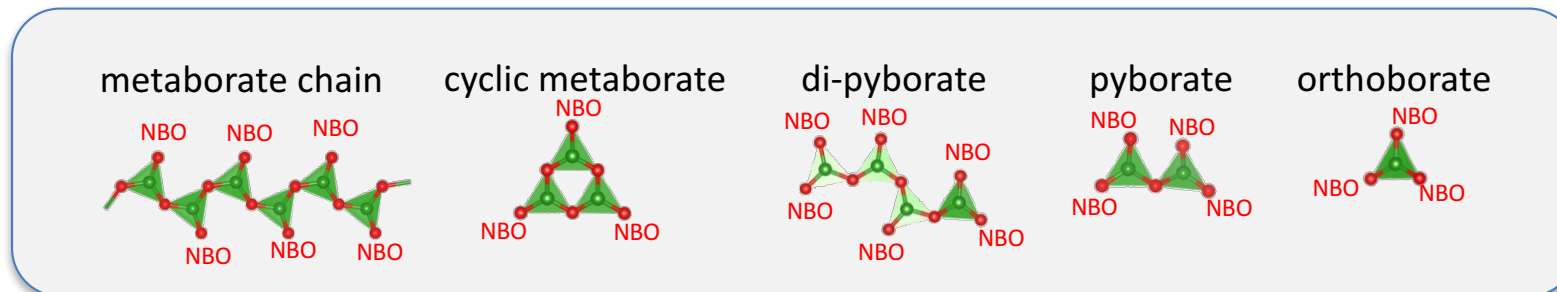
- 90%  $\text{BO}_3$  units
- Unique crystal containing **boroxol rings**
- 30% of  $\text{BO}_3$  units involved in boroxol rings
- 2 independent sub-networks

# ZOOLOGY OF THE SUPERSTRUCTURAL UNITS

- Examples of units containing  $[^3]\text{B}$  and  $[^4]\text{B}$



- Examples of units containing only  $[^3]\text{B}$



→ Existence of an intermediate range order in glasses ???

# BORATE GLASSES – STRUCTURE

## NMR

$^{11}\text{B}$  (NA: 80%) is the most sensitive ++

$^{17}\text{O}$  (NA: 0.038%) is the only NMR active nucleus → Isotopic enrichment mandatory +-

## Raman IR

Well suited ++

## ND neutron

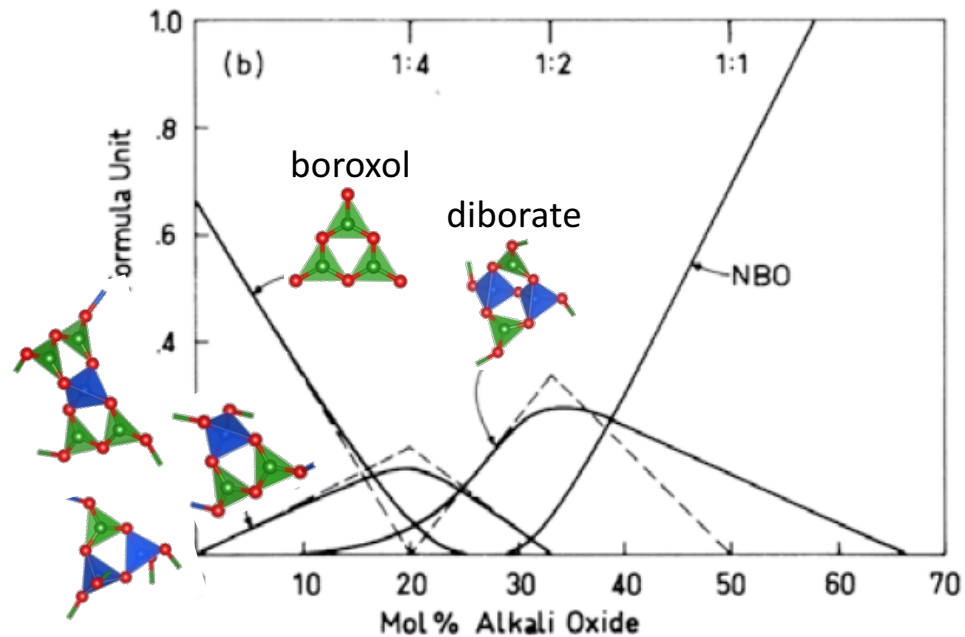
$^{11}\text{B}$  is highly absorbing → Isotopic enrichment mandatory +-

## XAS X-rays

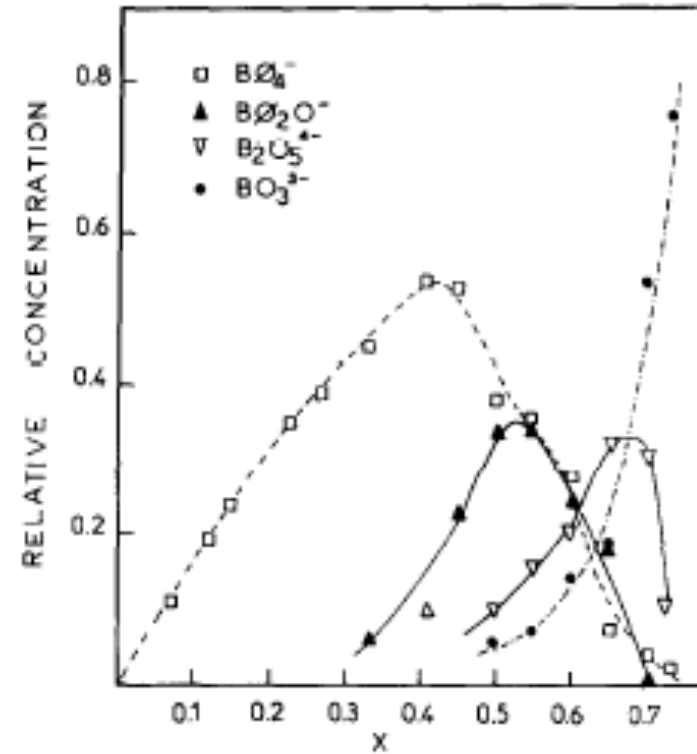
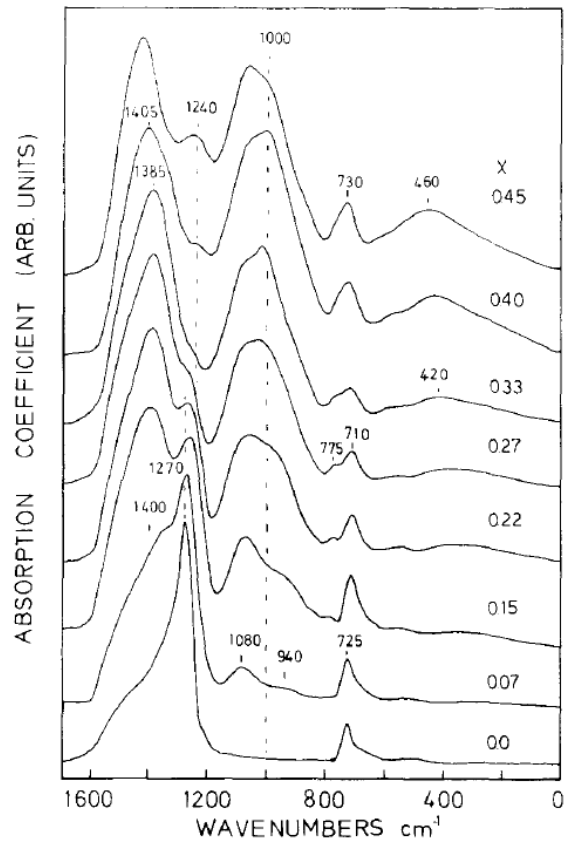
Low-Z elements → K-edges are surface sensitive – high-vacuum mandatory +-

## KROGH-MOE – GRISCOM MODEL

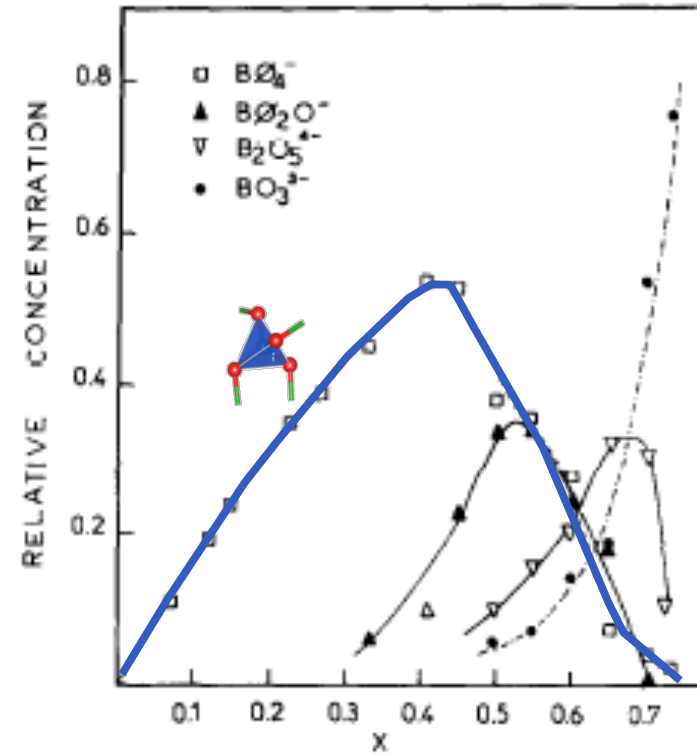
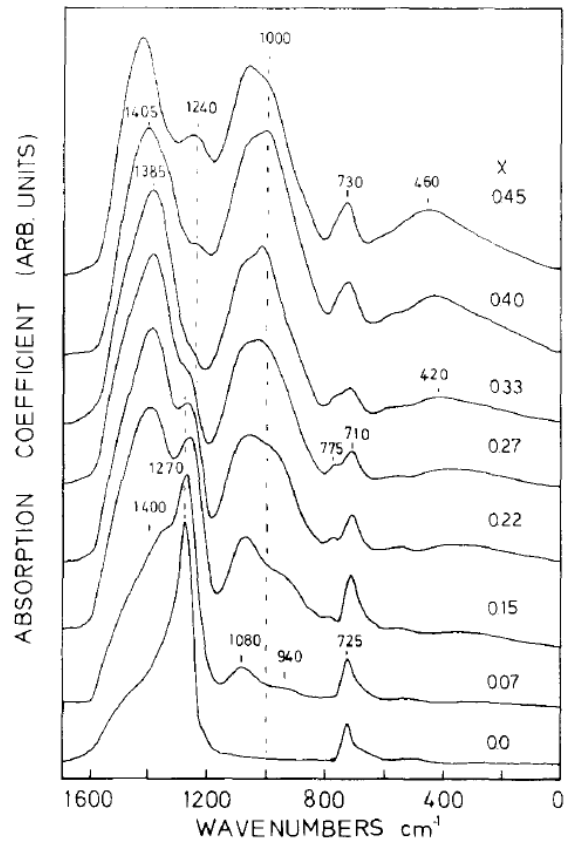
- Krogh-Moe in 1962 predicted the distribution of superstructural units in sodium borate glasses as a function of  $\text{Na}_2\text{O}$  content.



# SUPERSTRUCTURAL UNITS BY IR AND RAMAN SPECTROSCOPIES

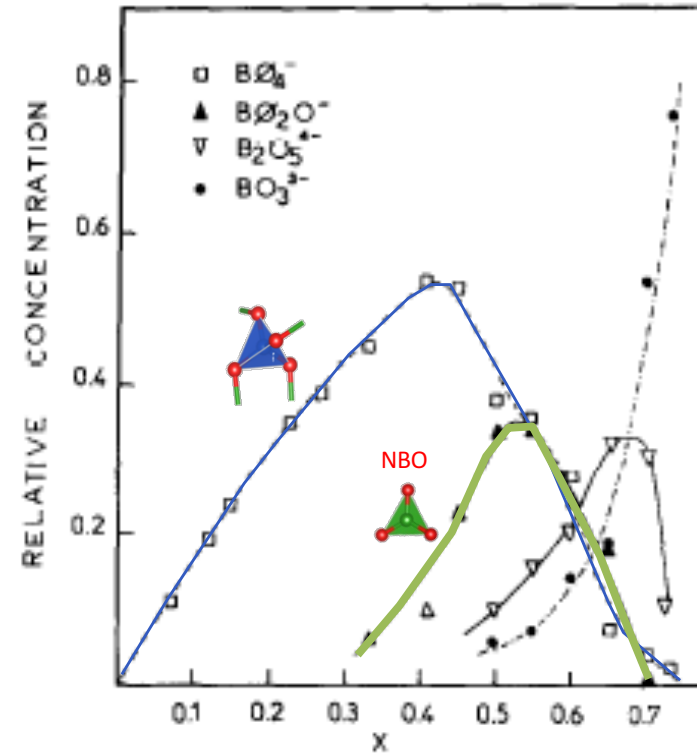
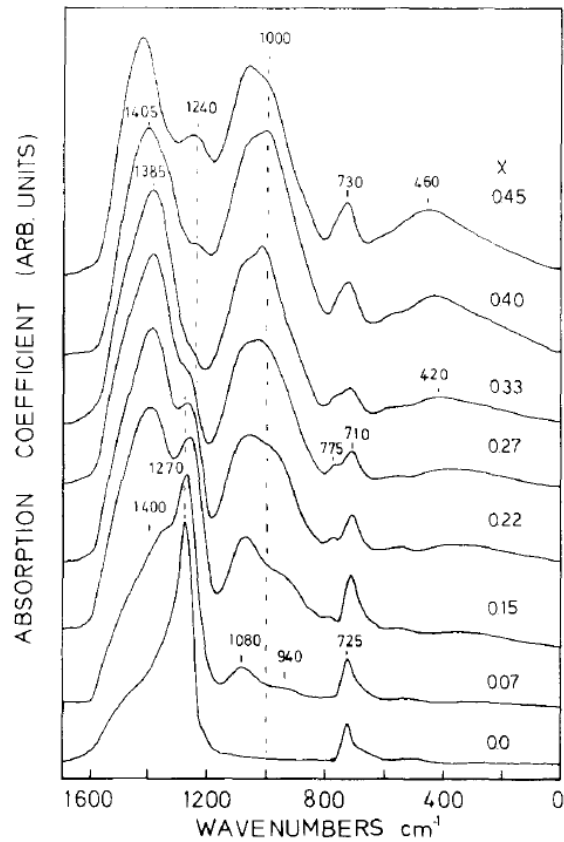


# SUPERSTRUCTURAL UNITS BY IR AND RAMAN SPECTROSCOPIES

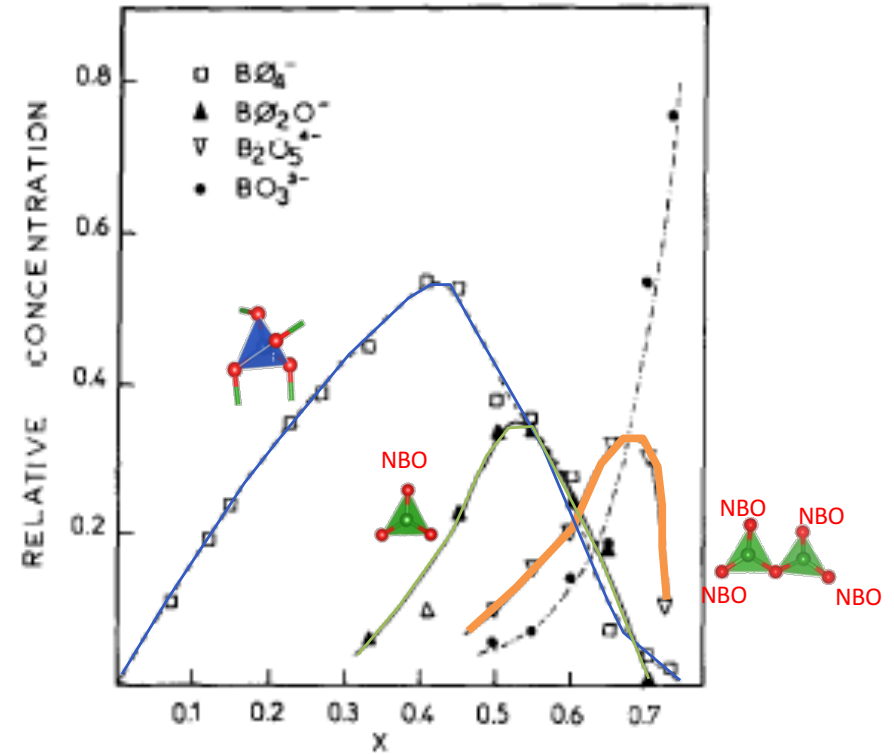
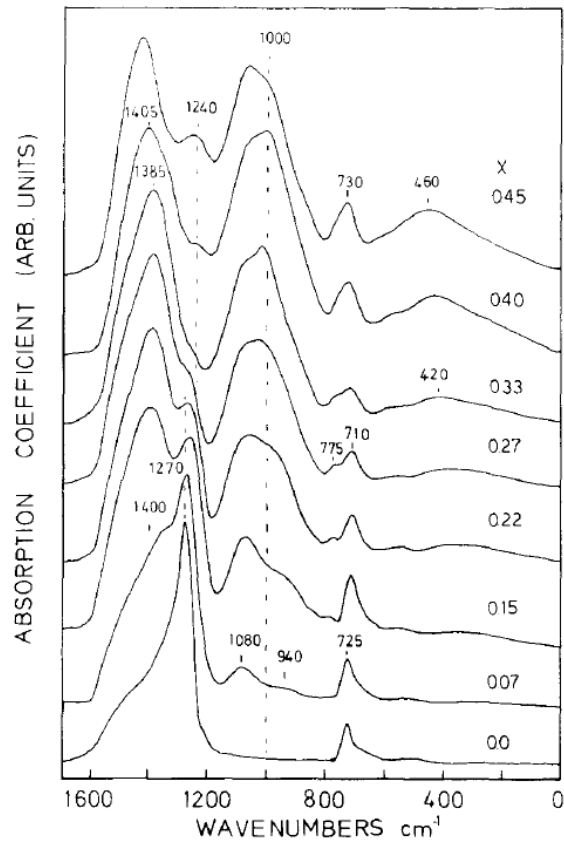




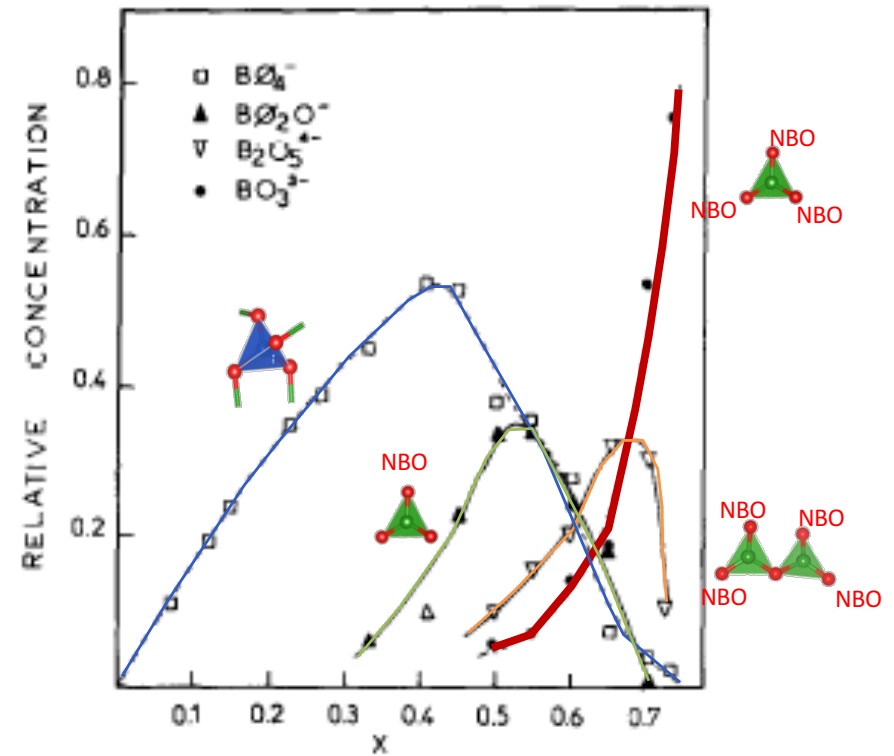
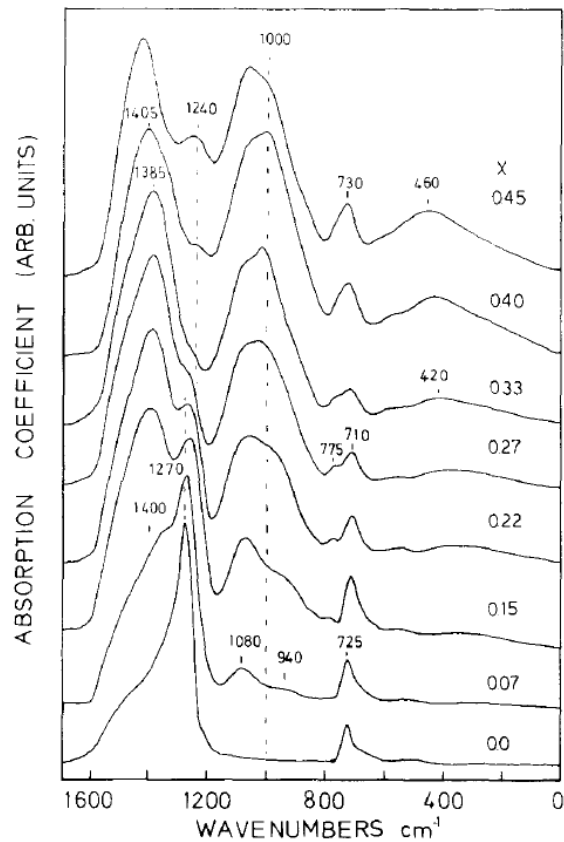
# SUPERSTRUCTURAL UNITS BY IR AND RAMAN SPECTROSCOPIES



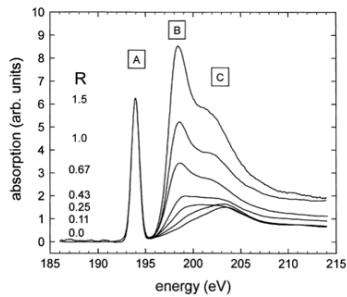
# SUPERSTRUCTURAL UNITS BY IR AND RAMAN SPECTROSCOPIES



# SUPERSTRUCTURAL UNITS BY IR AND RAMAN SPECTROSCOPIES

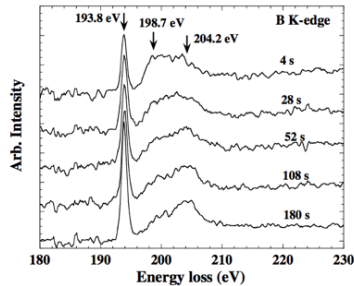


# PROBING LOCAL ORDER AND ELECTRONIC STRUCTURE



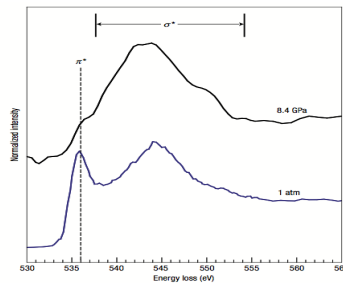
## X-ray absorption spectroscopy (XAS)

- ⇒ High vacuum – Surface sensitive - High resolution (<0.2 eV)
- ⇒ No complex sample environments



## Energy Electron Loss Spectroscopy (EELS)

- ⇒ Vacuum – Beam damage – Low resolution ( $\sim 0.7$  eV)
- ⇒ Access to edges at very low energies (Li), spatial resolution
- ⇒ No complex sample environments

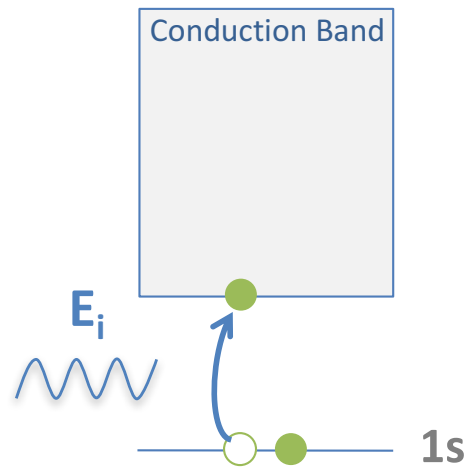


## Non-resonant Inelastic X-ray scattering (NRIXS)

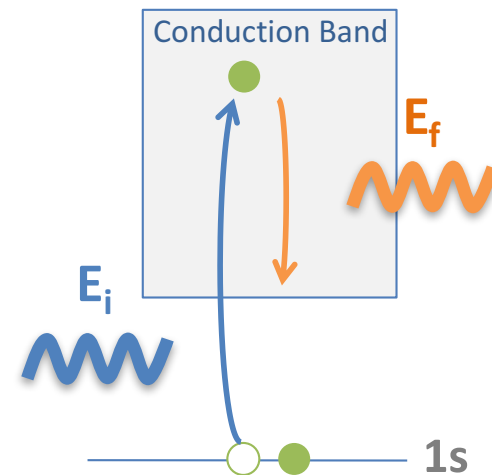
- ⇒ Low resolution ( $\sim 0.7$  eV) – Long experiments
- ⇒ Access to edges at very low energies (Li  $\Rightarrow$  Ne)
- ⇒ Various complex sample environments (high-pressure, high-temperature...)

# NRIXS: A SUBSTITUTE FOR SOFT X-RAY XAS

XAS



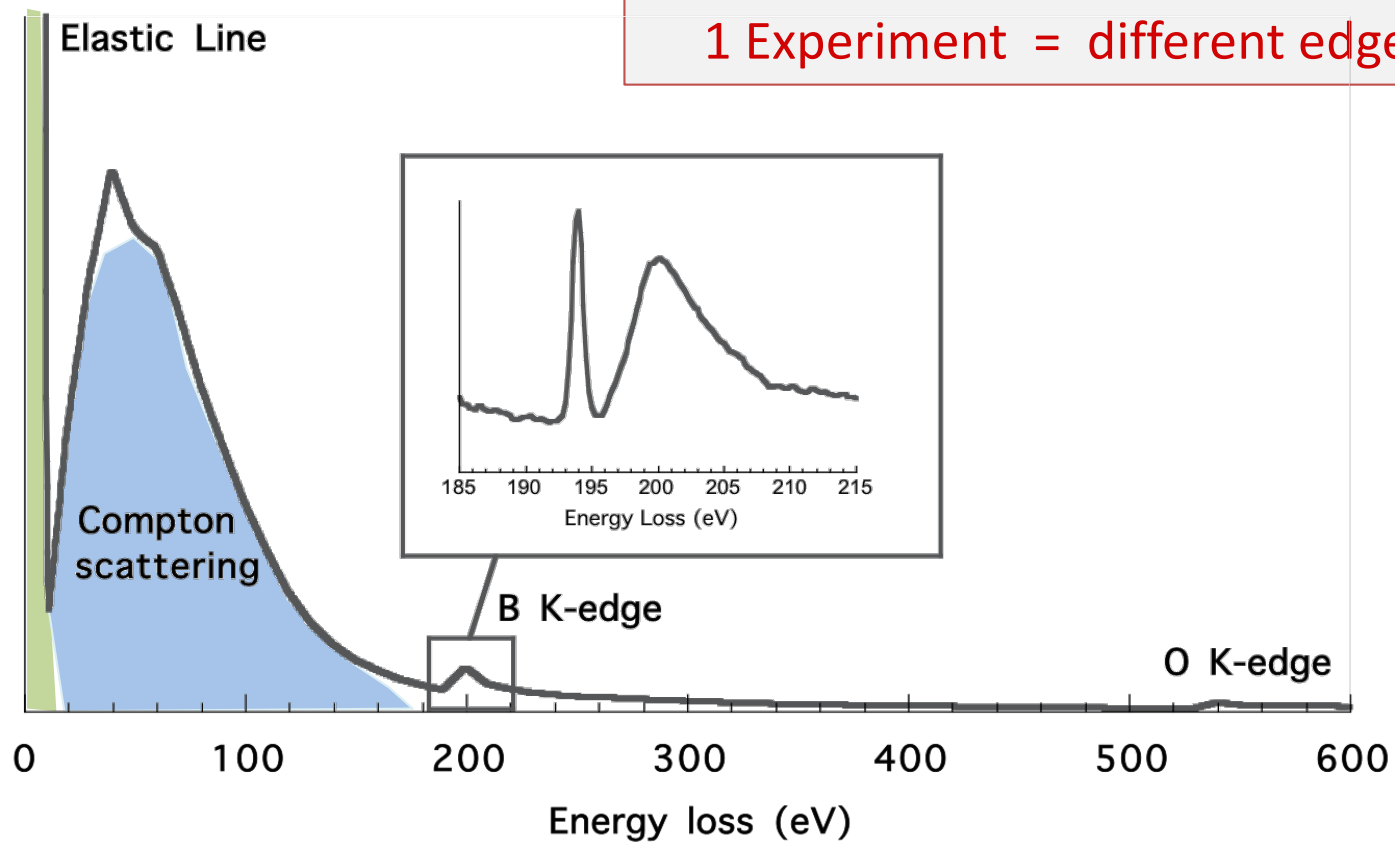
NRIXS



For low-Z elements: soft x-rays  
surface sensitive  
vacuum condition

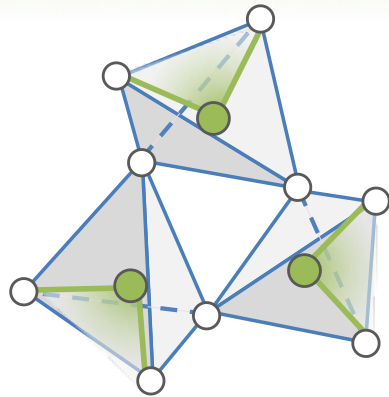
hard x-rays  
bulk sensitive  
**atmospheric condition**  
**complex sample environments**

# NRIXS: A SUBSTITUTE FOR SOFT X-RAY XAS



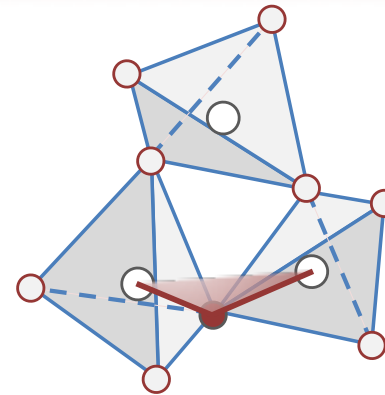
# STRUCTURE OF OXIDE GLASSES USING X-RAYS

VIEW THROUGH THE  
NETWORK FORMER CATIONS



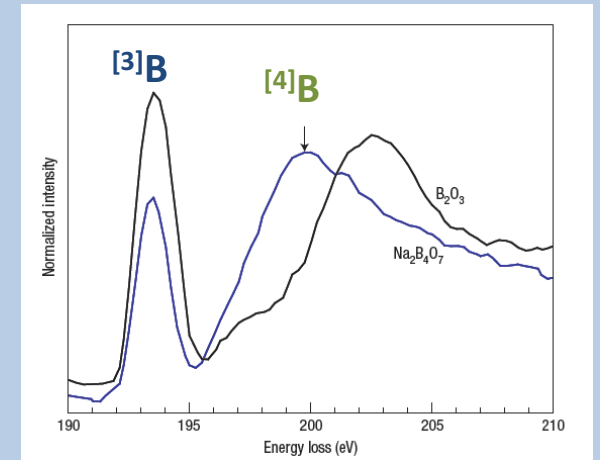
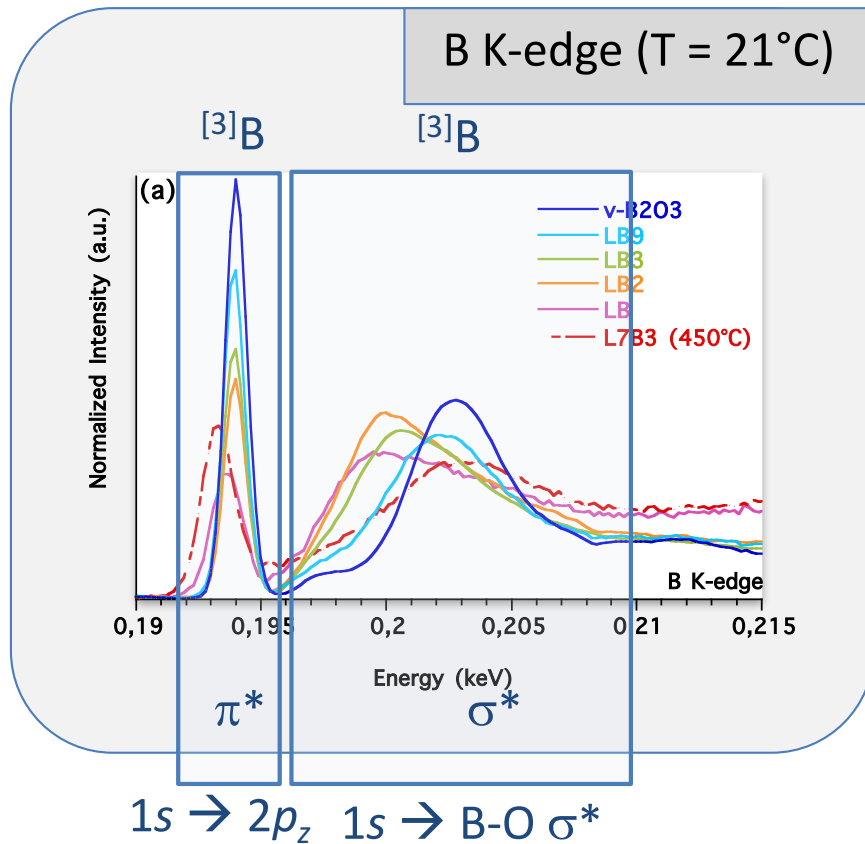
B K-edge

VIEW THROUGH THE  
LIGANDS



O K-edge

# B K-EDGE IN LITHIUM BORATE GLASSES

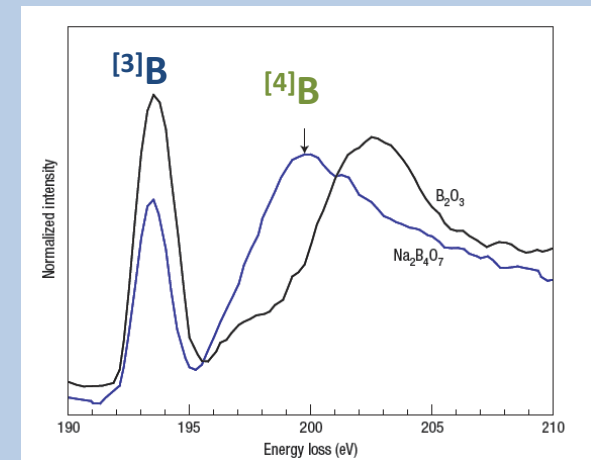
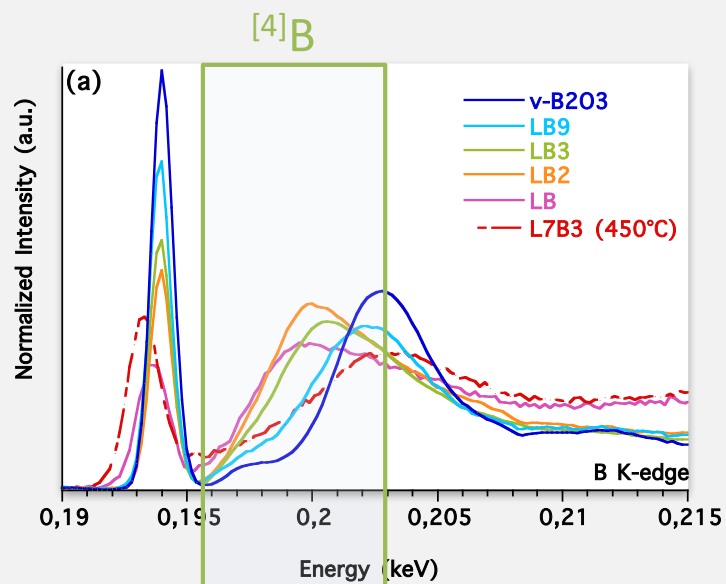


Lee *et al.*, Nature materials (2005) 4, 851-854.



# B K-EDGE IN LITHIUM BORATE GLASSES

B K-edge (T = 21°C)

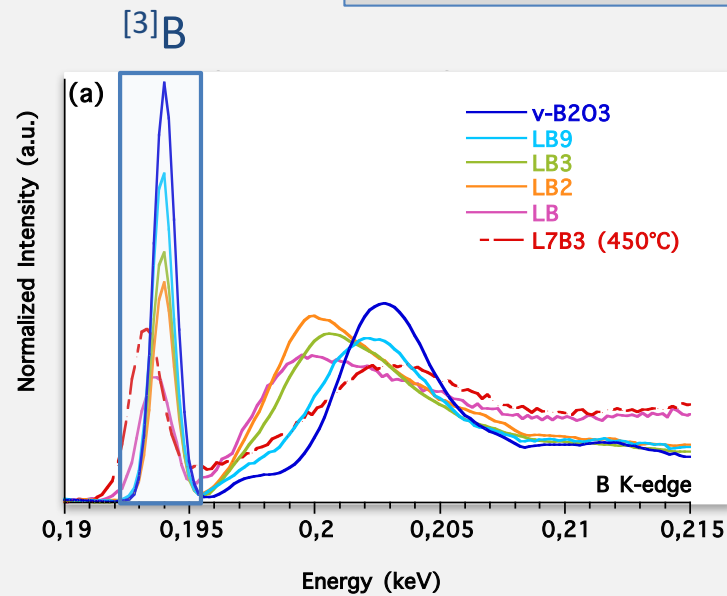


Lee *et al.*, Nature materials (2005) 4, 851-854.

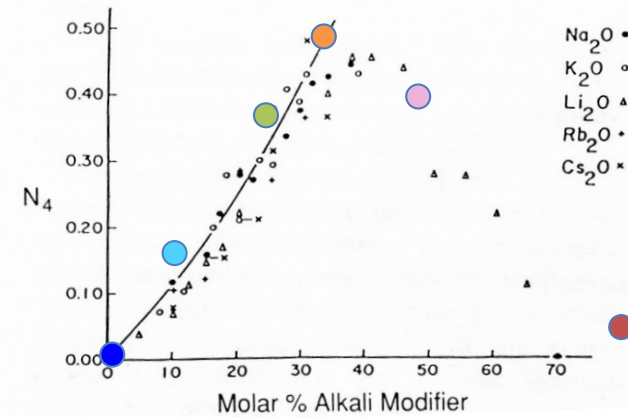
Modifications of the B K-edge due to the  $BO_3 / BO_4$  ratio

# EFFECT OF LI CONTENT ON THE $^{[3]}B/^{[4]}B$

B K-edge (T = 21°C)

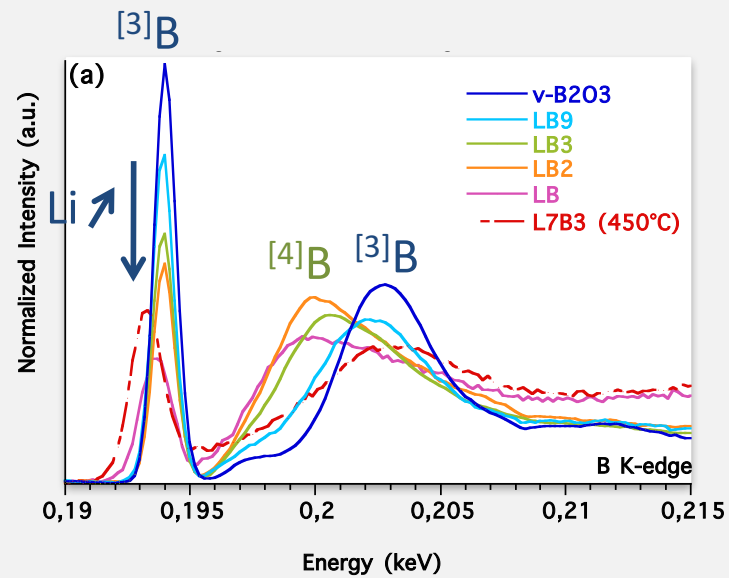


Zhong JNCS 1989, 111, 67.



# WHAT CAN WE LEARN FROM THE O K-EDGE ?

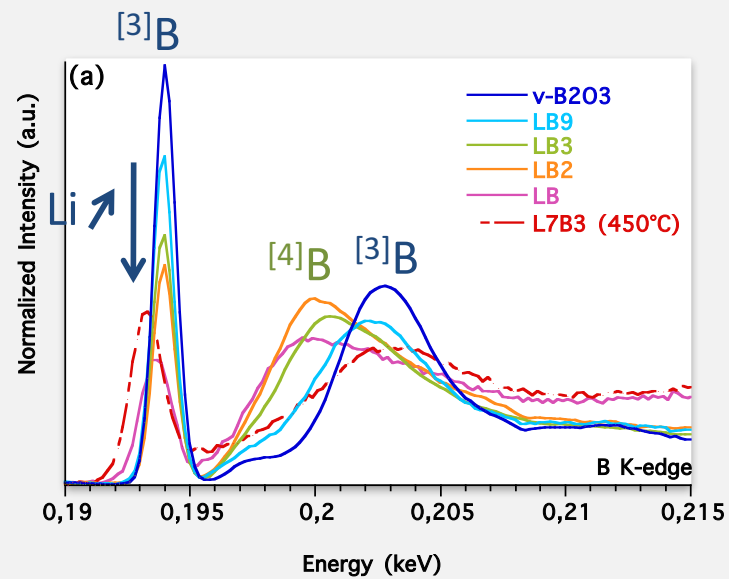
B K-edge (T = 21°C)



Coordination change for B

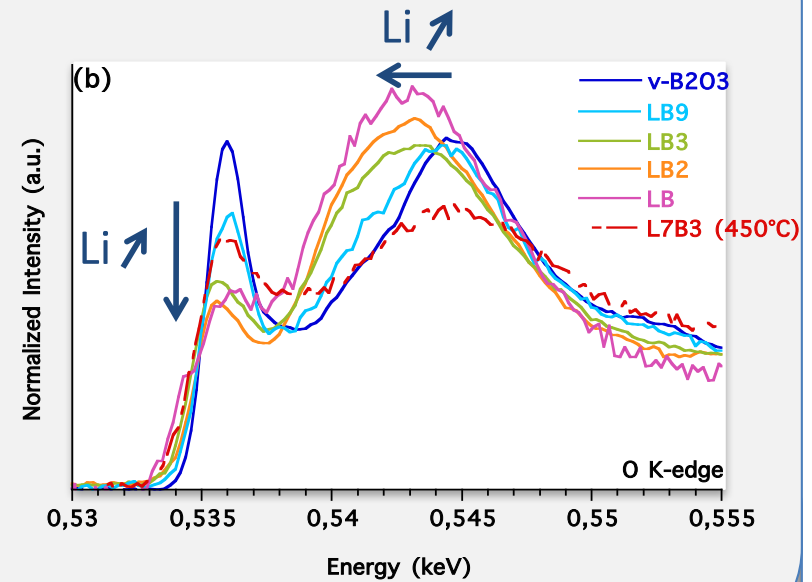
# WHAT CAN WE LEARN FROM THE O K-EDGE ?

B K-edge (T = 21°C)



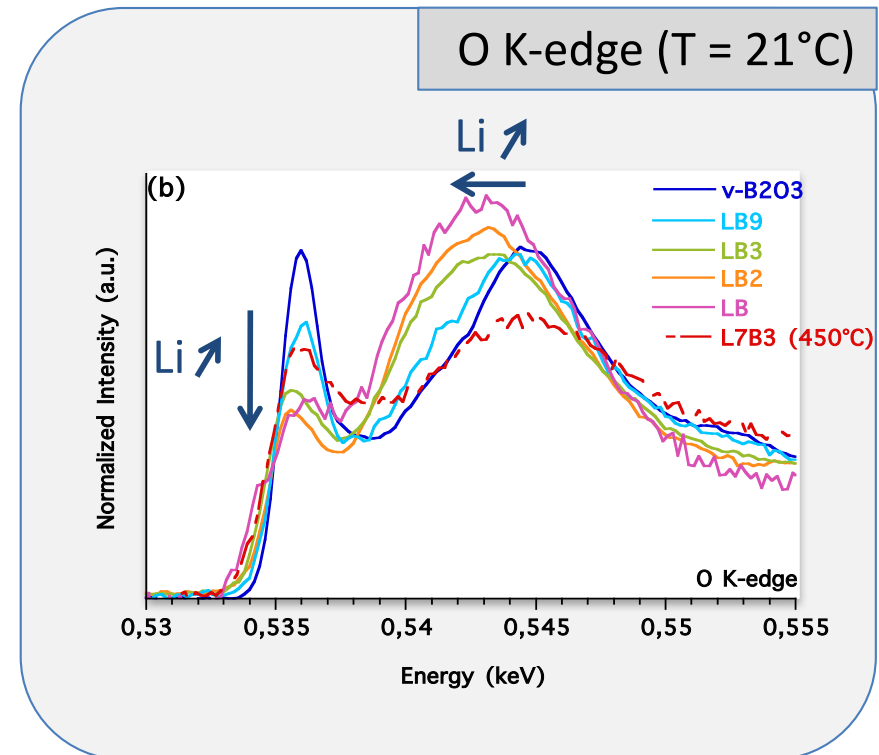
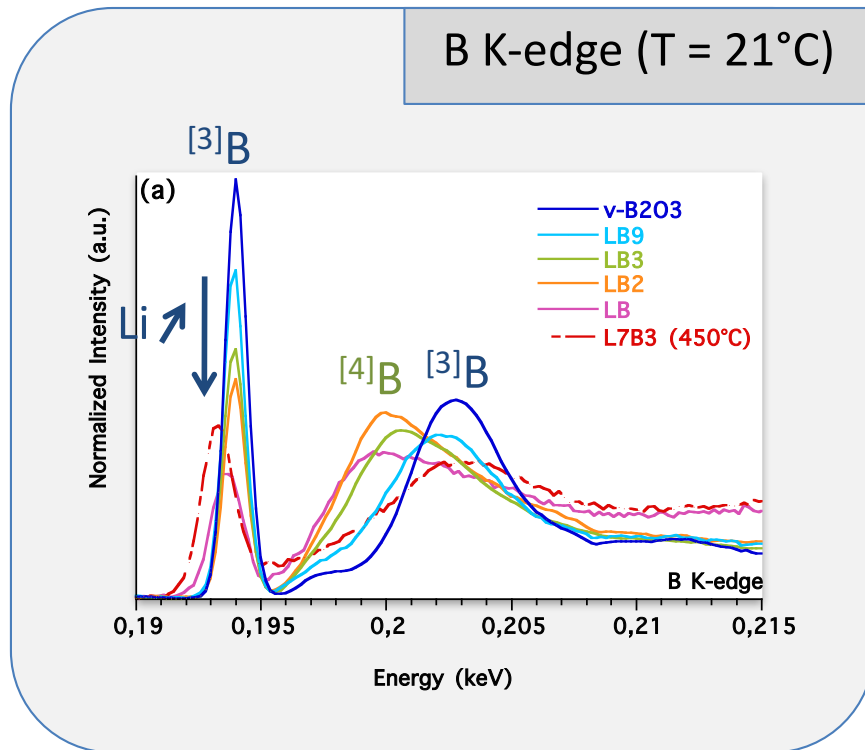
Coordination change for B

O K-edge (T = 21°C)



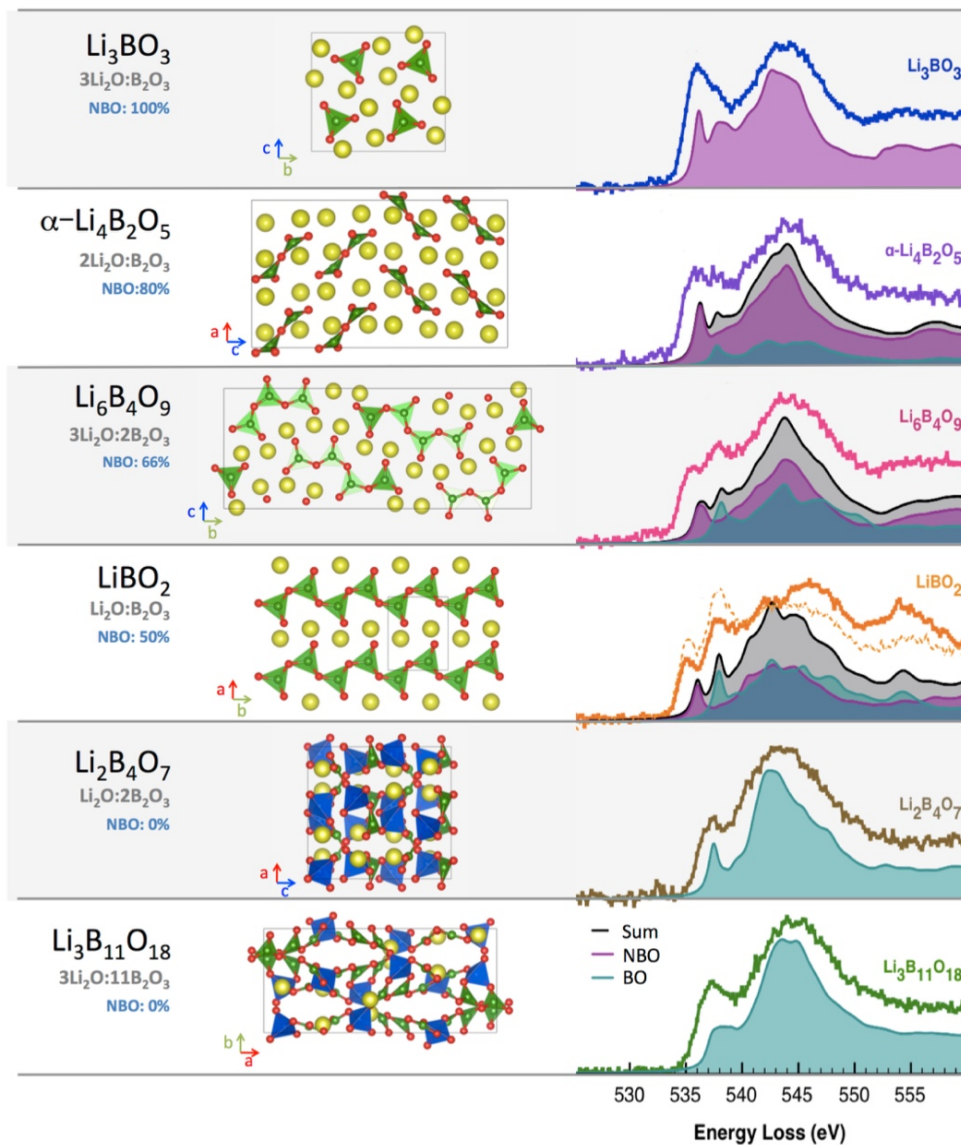
No coordination change for O

# WHAT CAN WE LEARN FROM THE O K-EDGE ?



→ We need crystalline samples in order to better understand the O K-edge

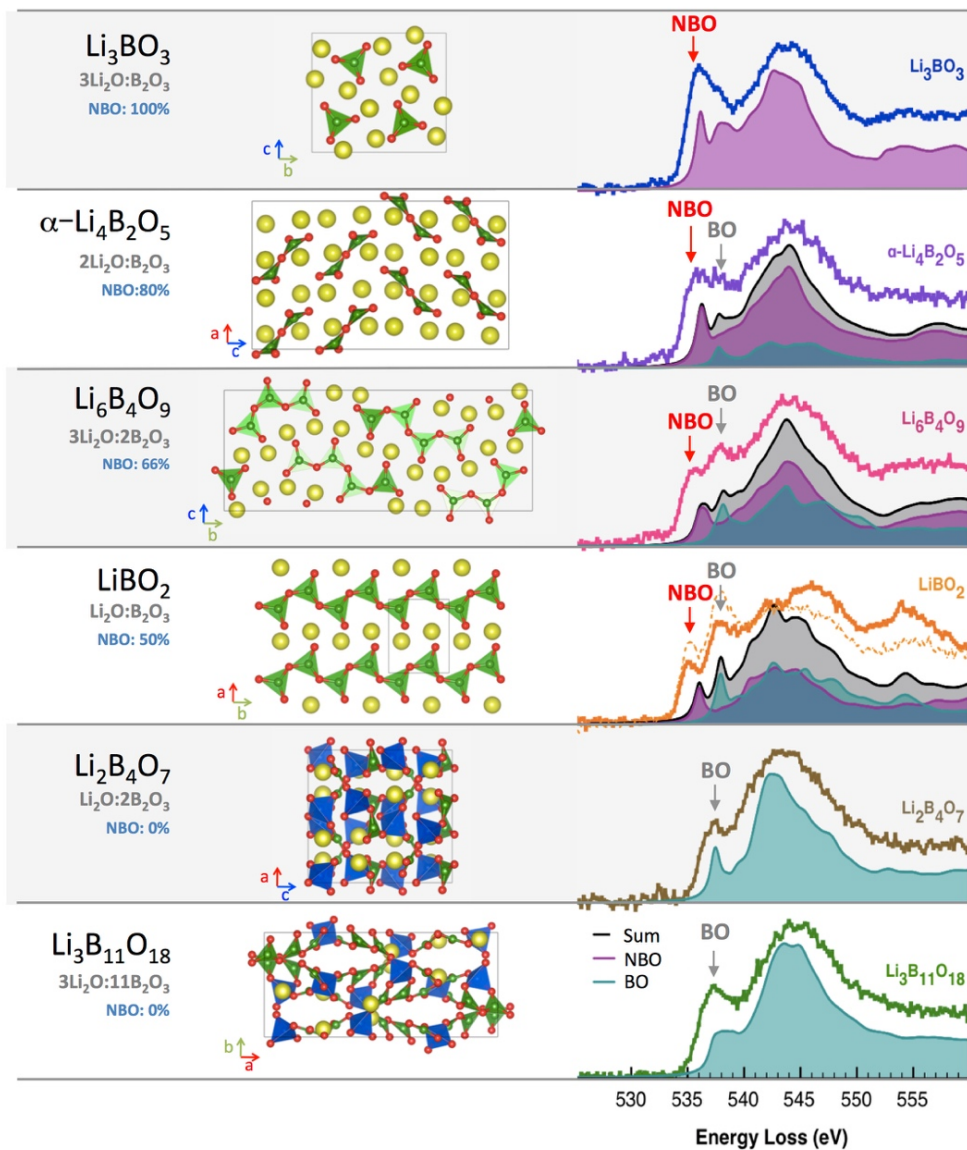
# O K-EDGE ON LI BORATE CRYSTALS



## DFT Calculations

- Contribution from NBOs
- Contribution from BOs

# O K-EDGE ON LI BORATE CRYSTALS



## DFT Calculations

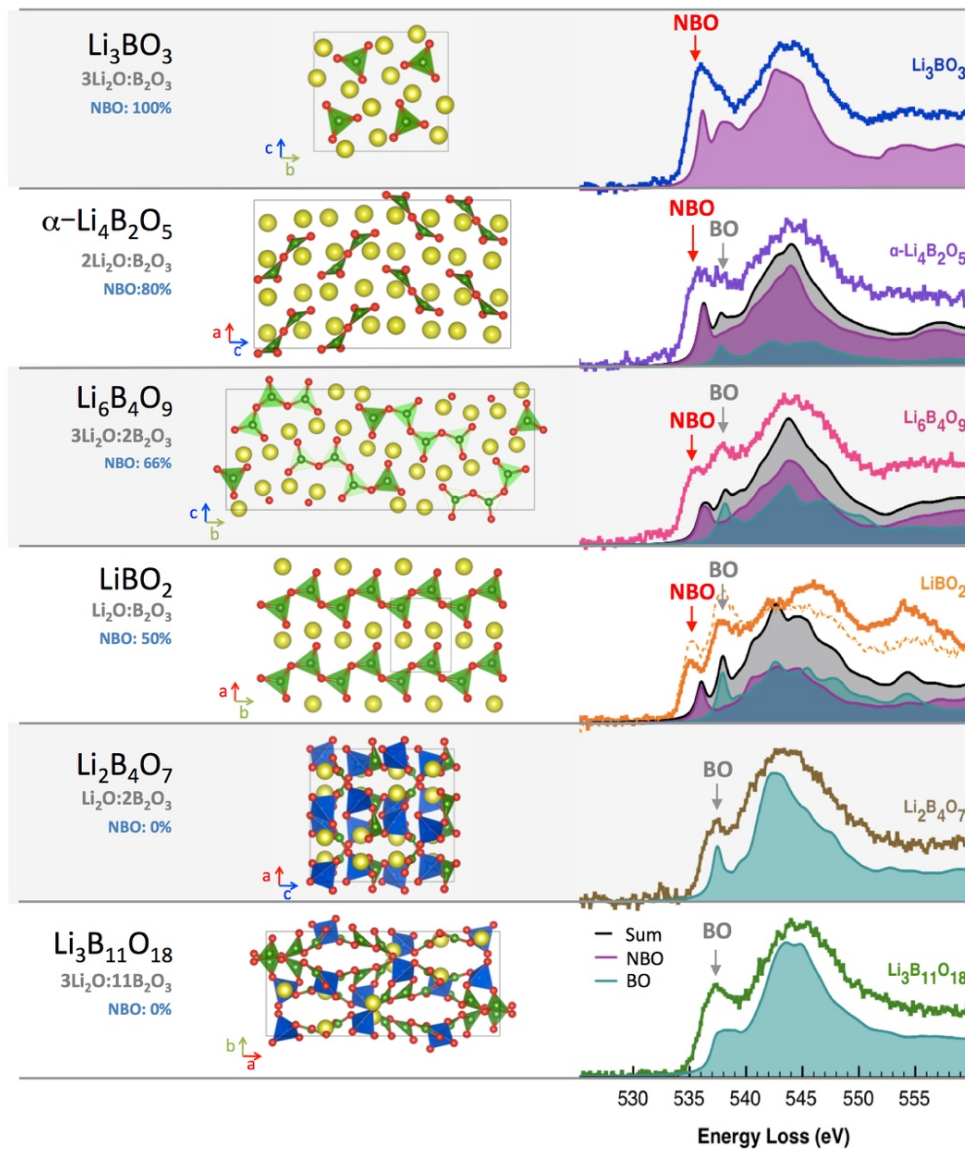
- Contribution from NBOs
- Contribution from BOs

DETERMINATION OF A PROBE OF NBOs



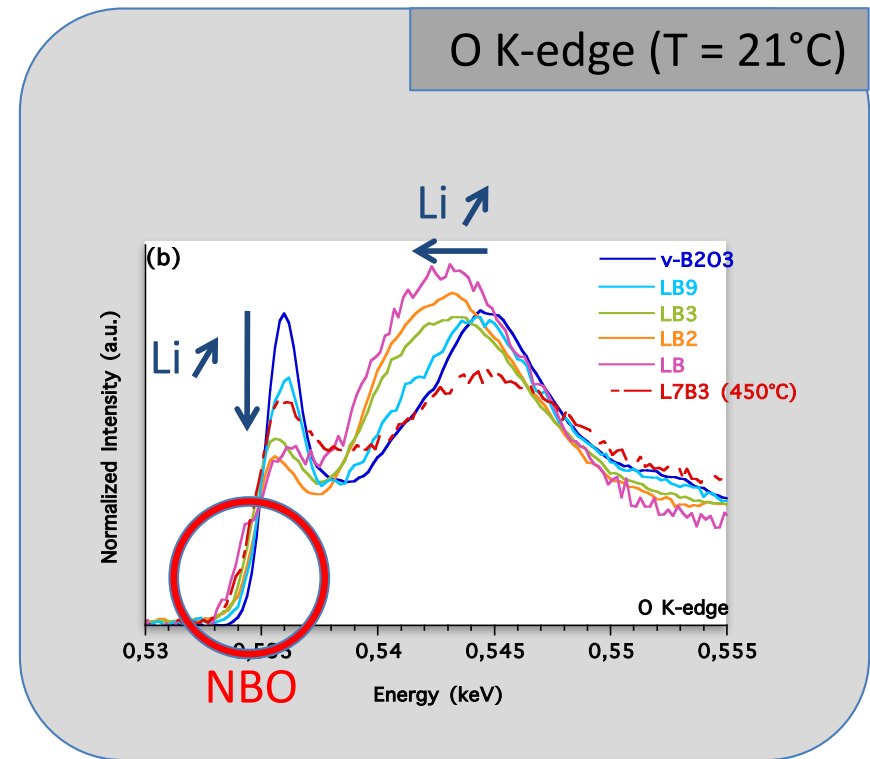
PROBE OF THE DEPOLYMERIZATION OF  
THE BORATE NETWORK

# O K-EDGE ON LI BORATE CRYSTALS



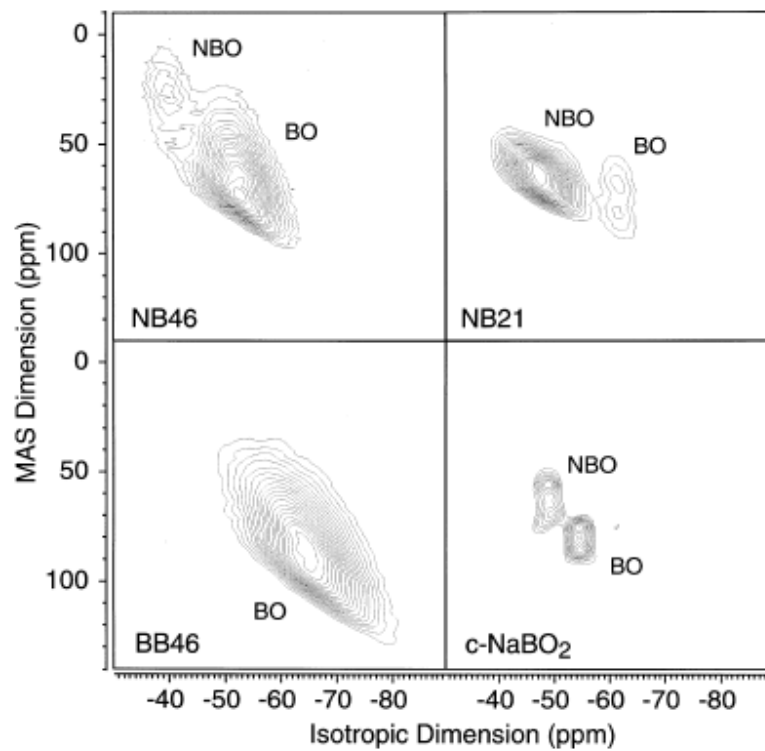
## DFT Calculations

- Contribution from NBOs
- Contribution from BOs

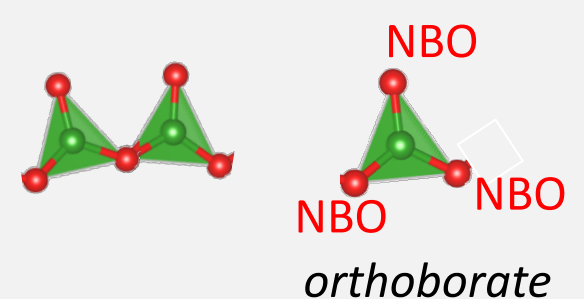
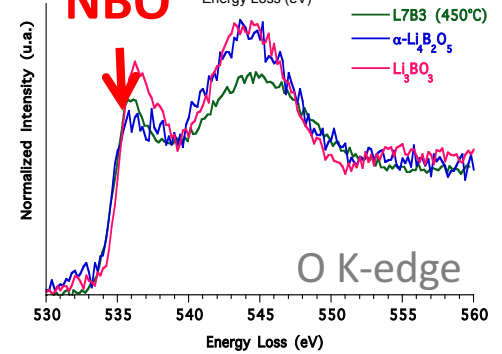
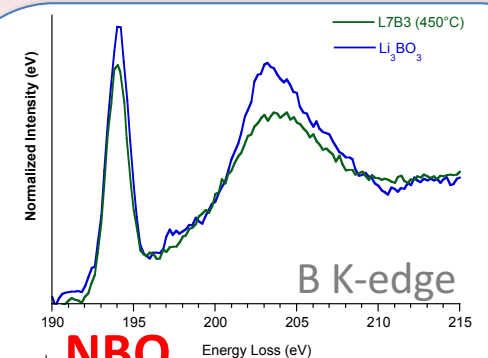
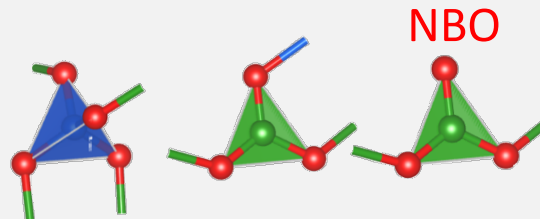
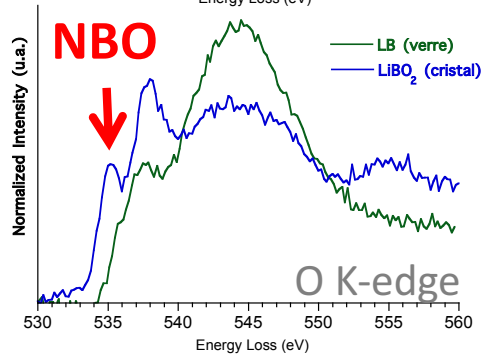
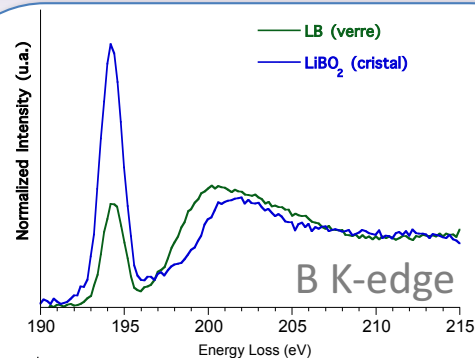
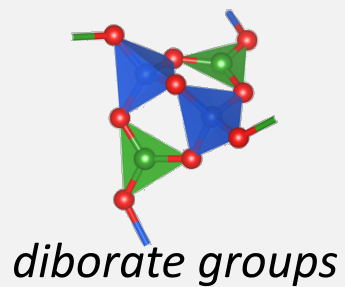
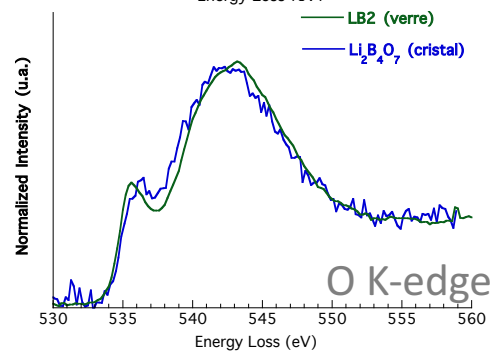
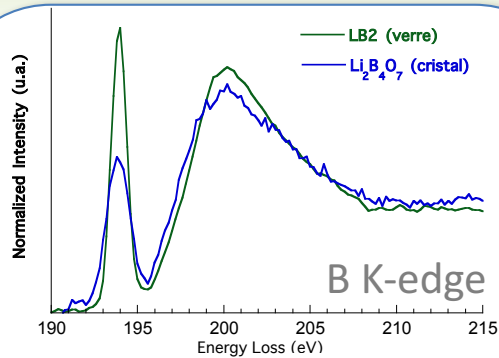
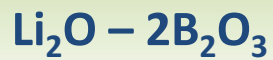




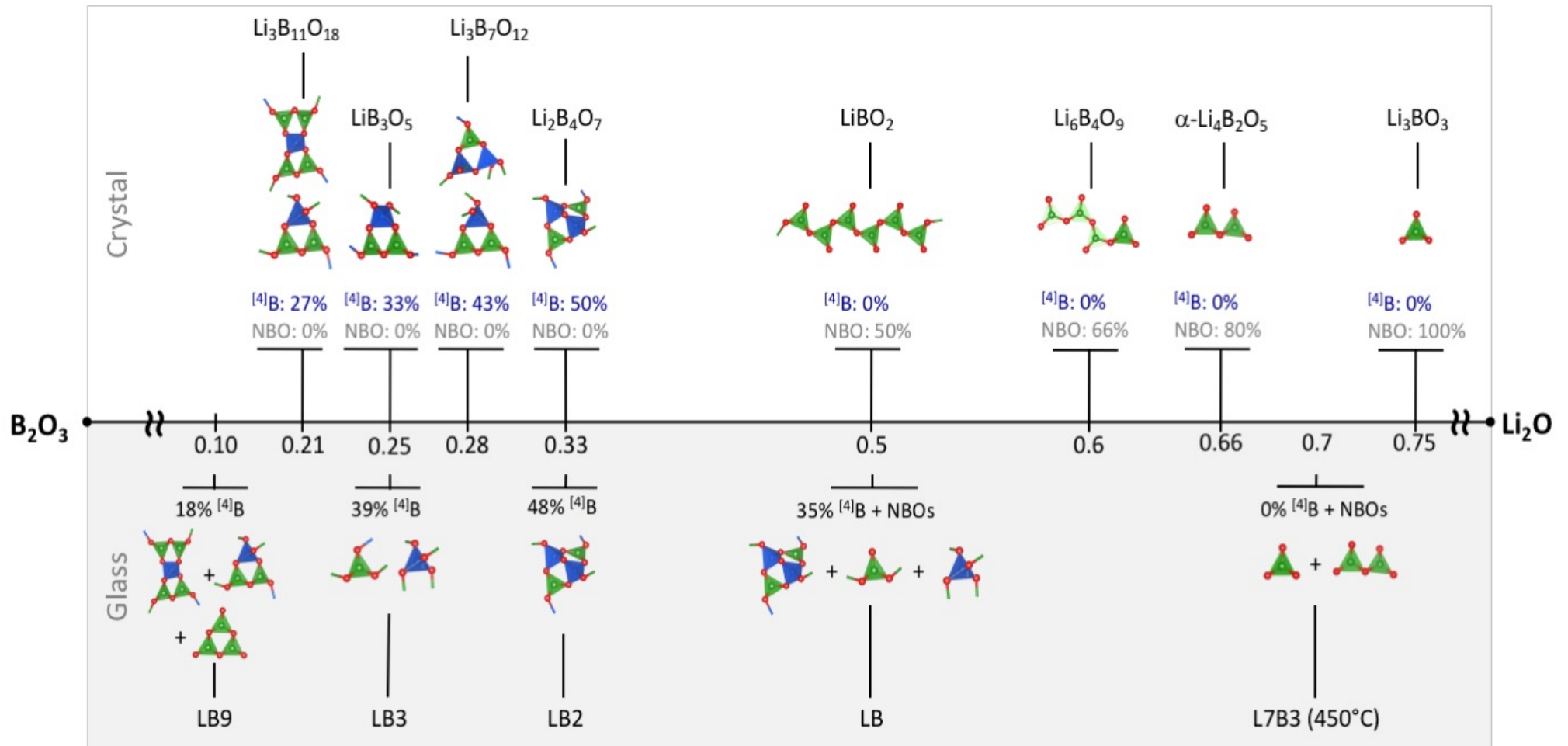
# NBO BY $^{17}\text{O}$ NMR 3QMAS



# GLASSES VS CRYSTALS

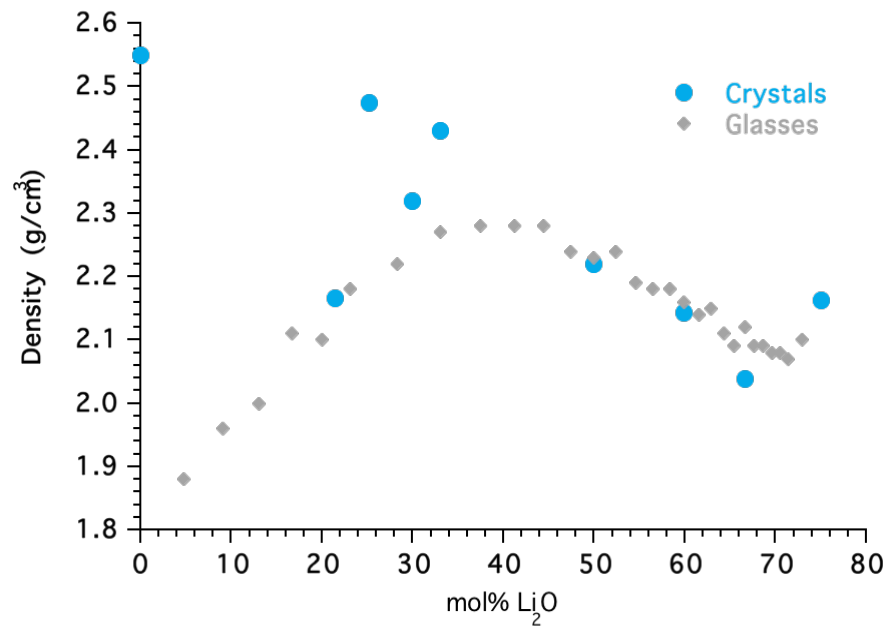


# LITHIUM BORATES: GLASS VS. CRYSTAL

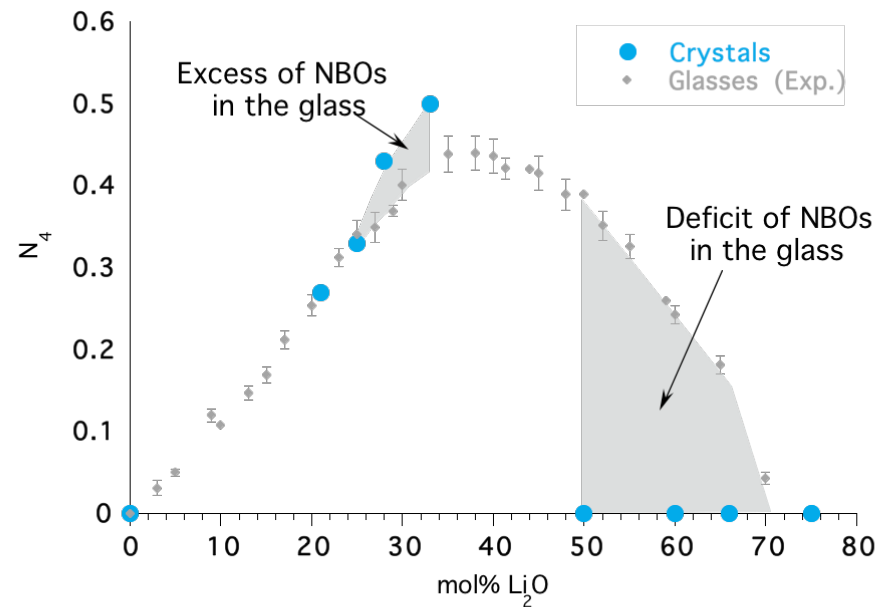


# LITHIUM BORATES: GLASS VS. CRYSTAL

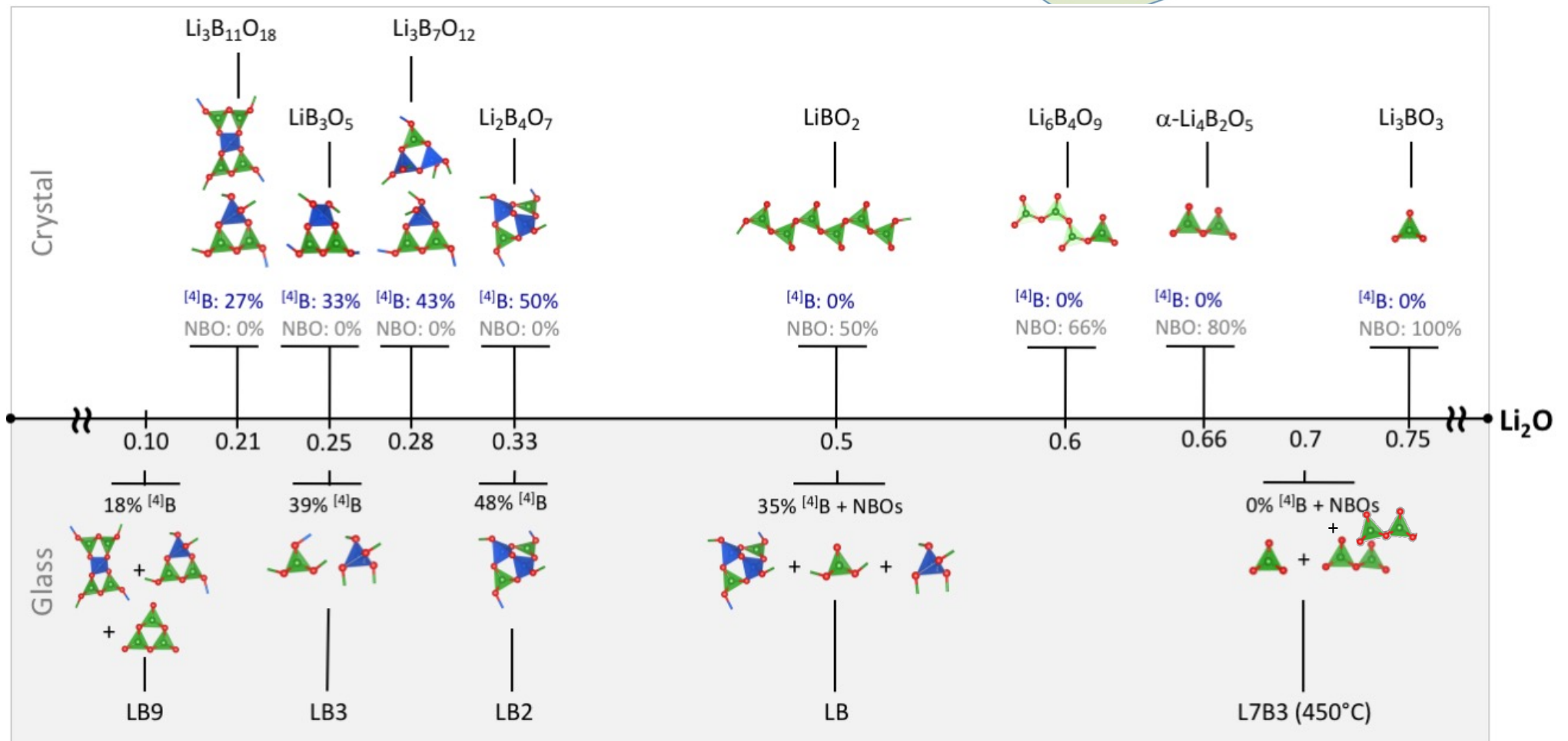
## DENSITY



## PROPORTION OF [4]B



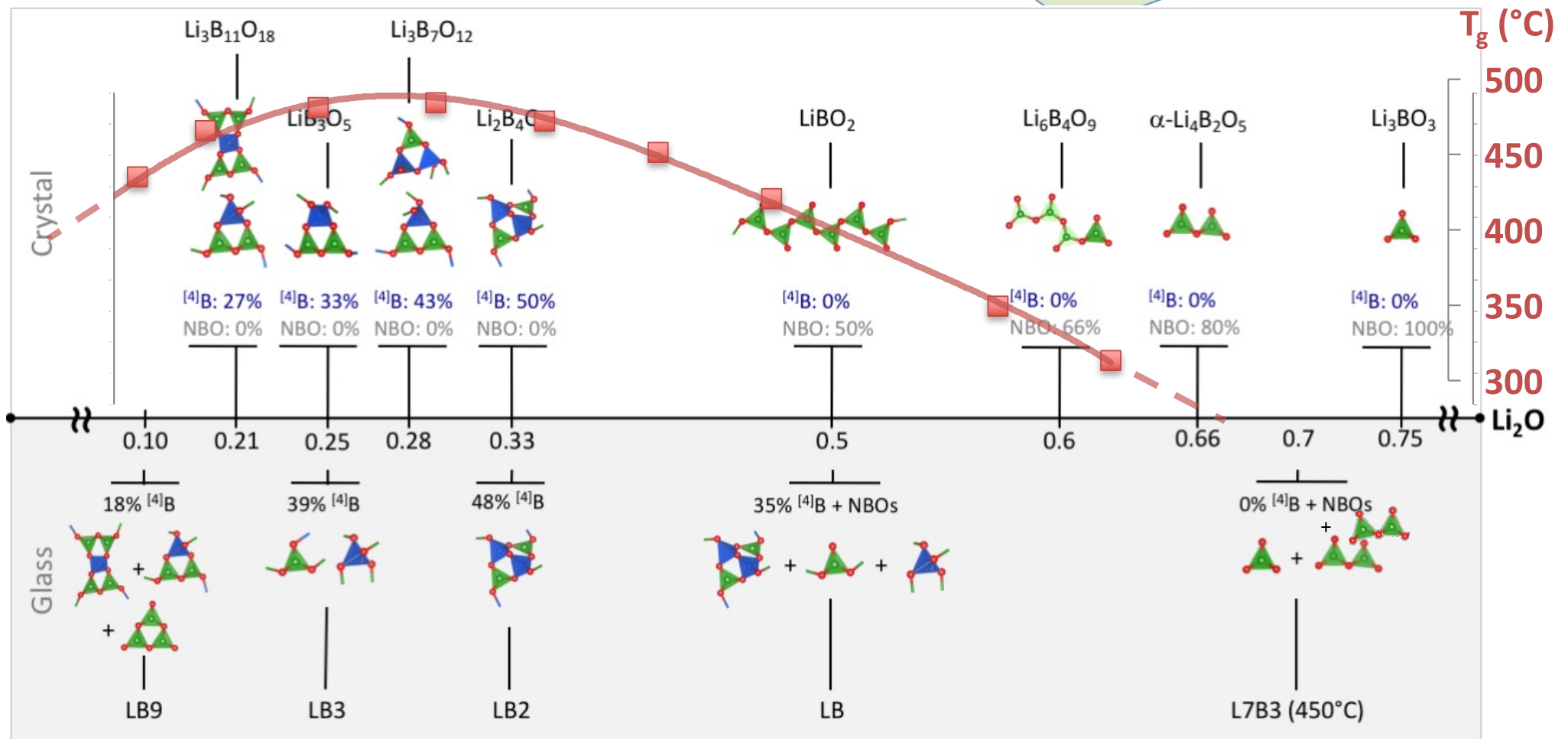
# BORATE ANOMALIES ...



Similarities glass ↔ cristal  
at the local scale

Discrepancies in the local  
structure of glass & cristal

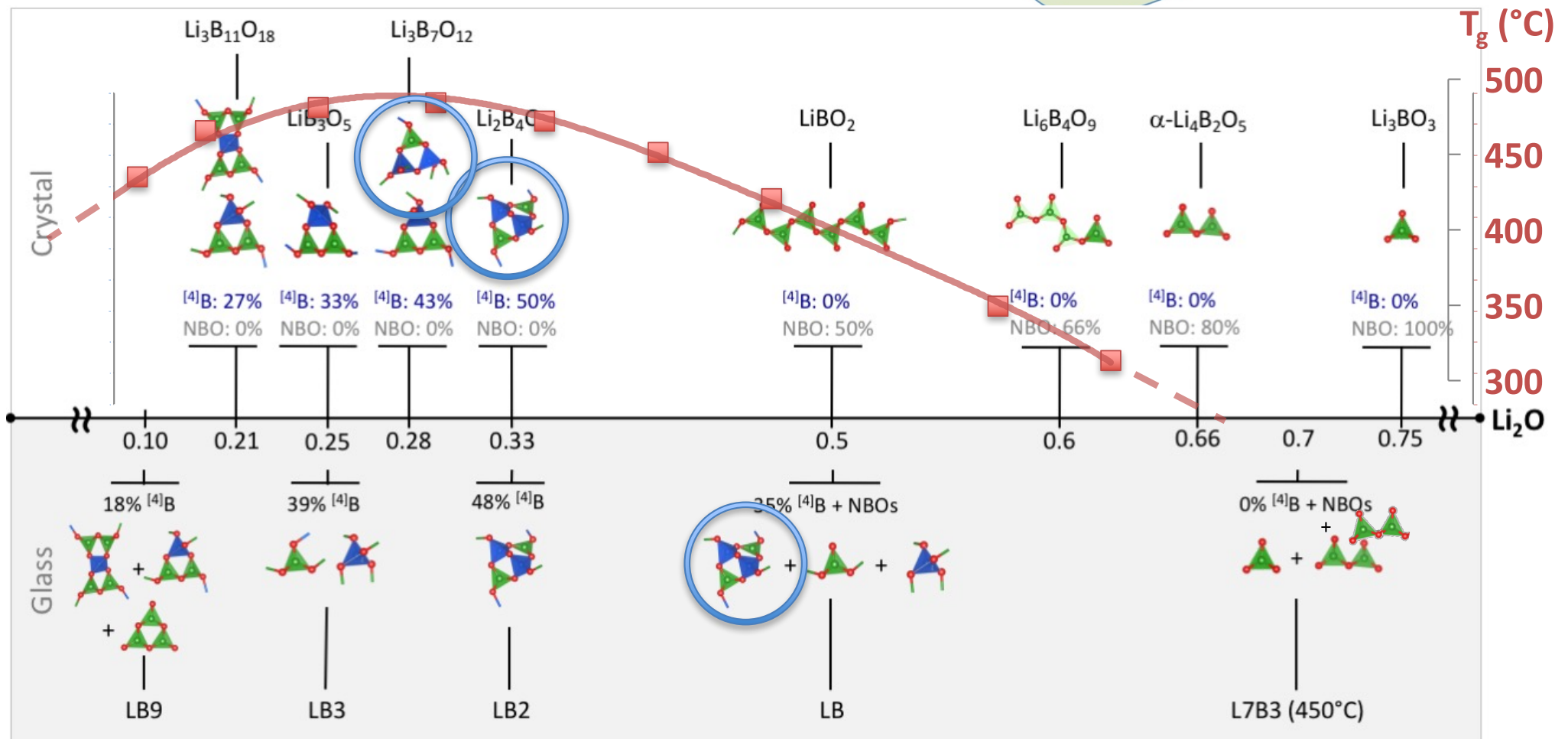
# BORATE ANOMALIES ...



Similarities glass ↔ cristal  
at the local scale

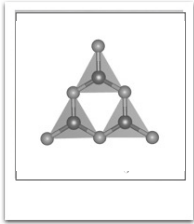
Discrepancies in the local  
structure of glass & cristal

# BORATE ANOMALIES ...



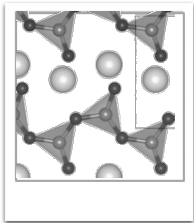
Important role of the superstructural units (medium range order)

# OUTLINE



## **B<sub>2</sub>O<sub>3</sub> glass**

Archetypal glass former

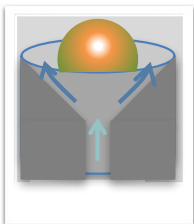


## **Alkali borate glasses (Li, Na, K, Rb, Cs)**

Vitrification domains

Alkali effects on physical properties

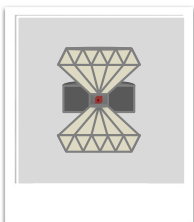
Glass structure: short- and intermediate-range order



## **From glass to melt**

Structure of alkali borate melts

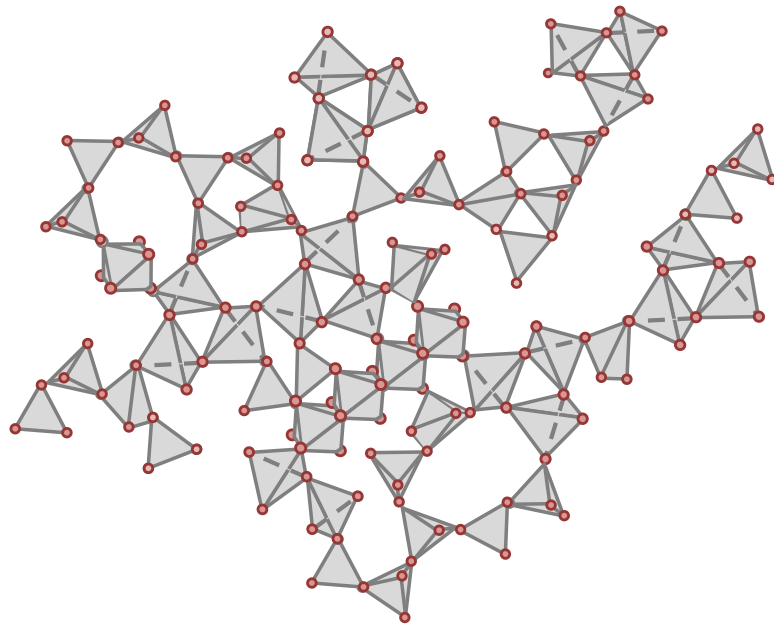
Depolymerization of the borate network



## **Polyamorphism**



# BORATE GLASSES - LOW Z ELEMENTS



## PERIODIC TABLE OF ELEMENTS

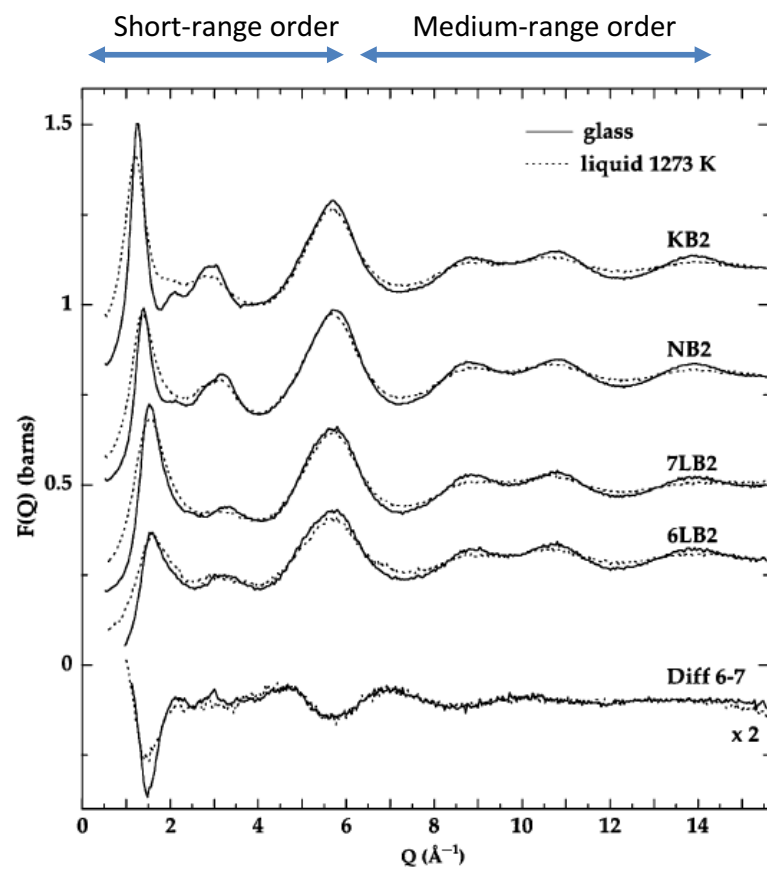
H						He	
Li	Be	B	C	N	O	F	Ne
Na	Mg	Al	Si	P	S	Cl	Ar

⇒ Tractable when working at ambient conditions  
(<sup>17</sup>O NMR, Soft X-ray XANES, ND, EELS, ...)

⇒ Much more complex problem under extreme  
conditions (HP/HT)

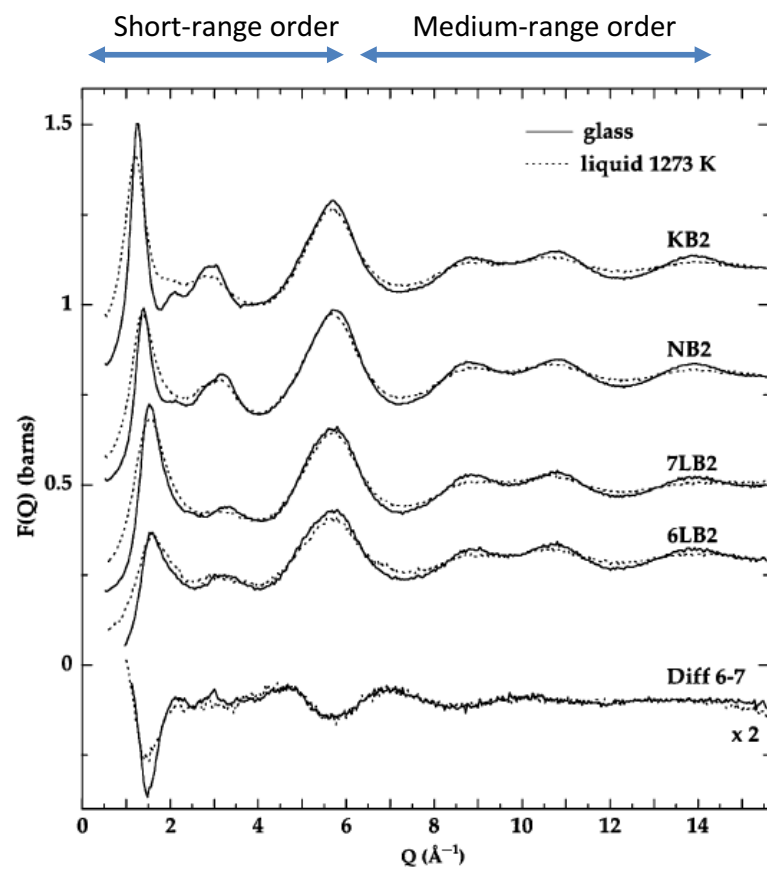
# STRUCTURE OF BORATES VS. TEMPERATURE

## NEUTRON DIFFRACTION

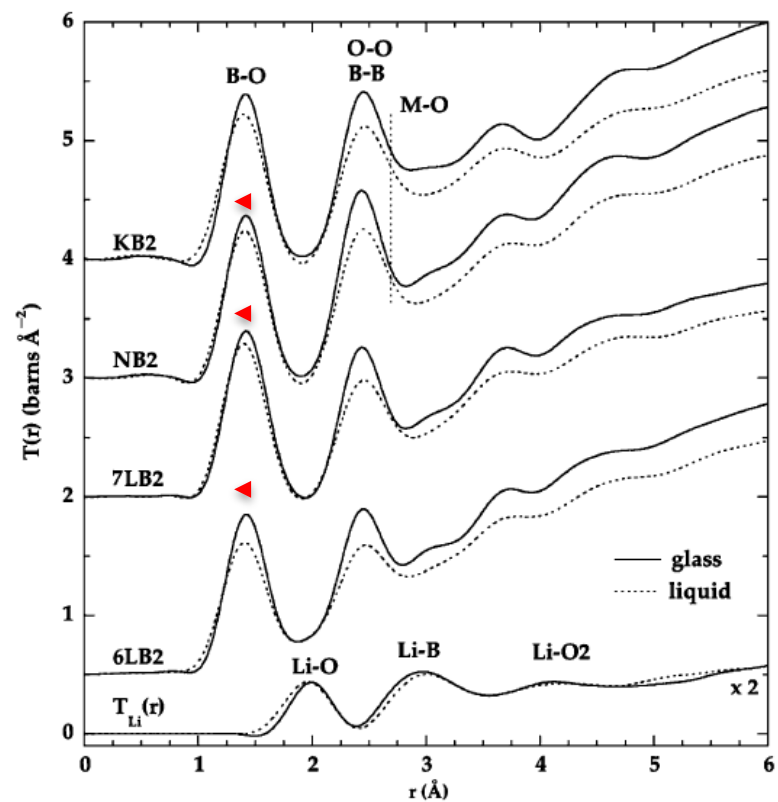


# STRUCTURE OF BORATES VS. TEMPERATURE

## NEUTRON DIFFRACTION



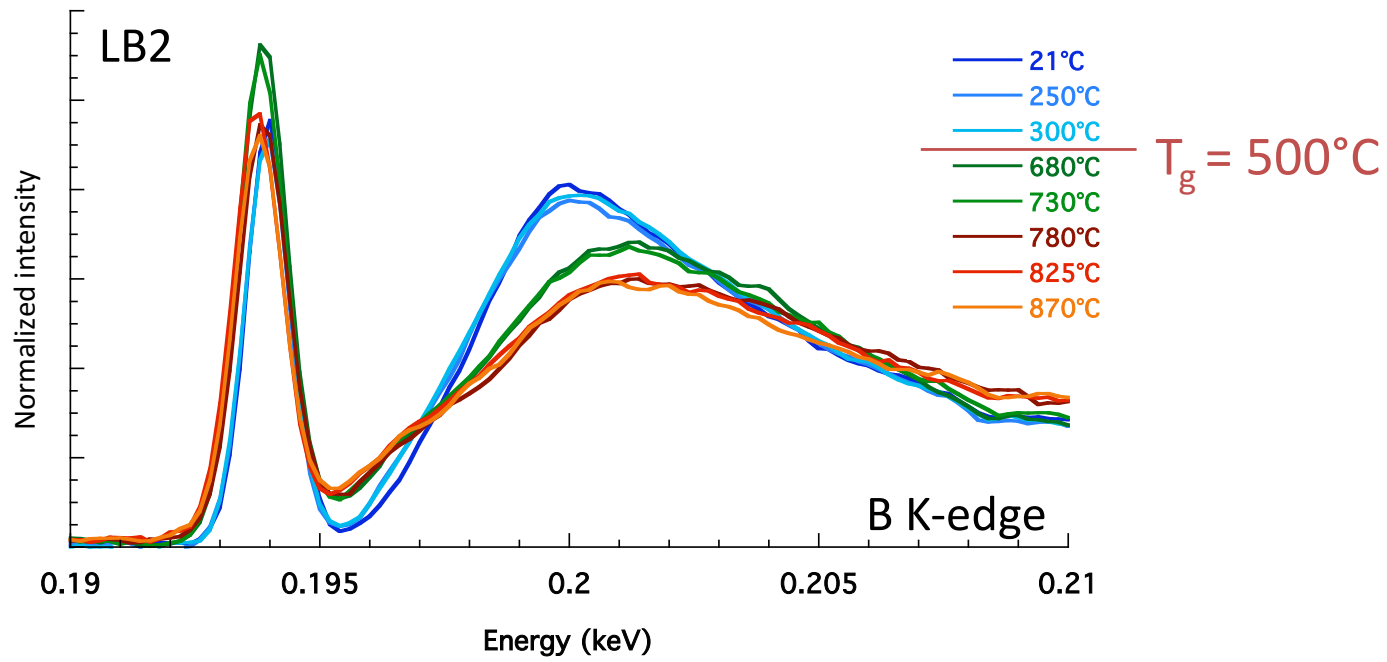
Total Structure Factor



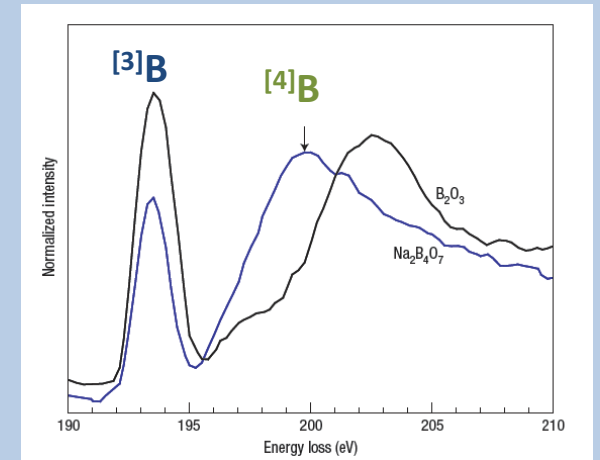
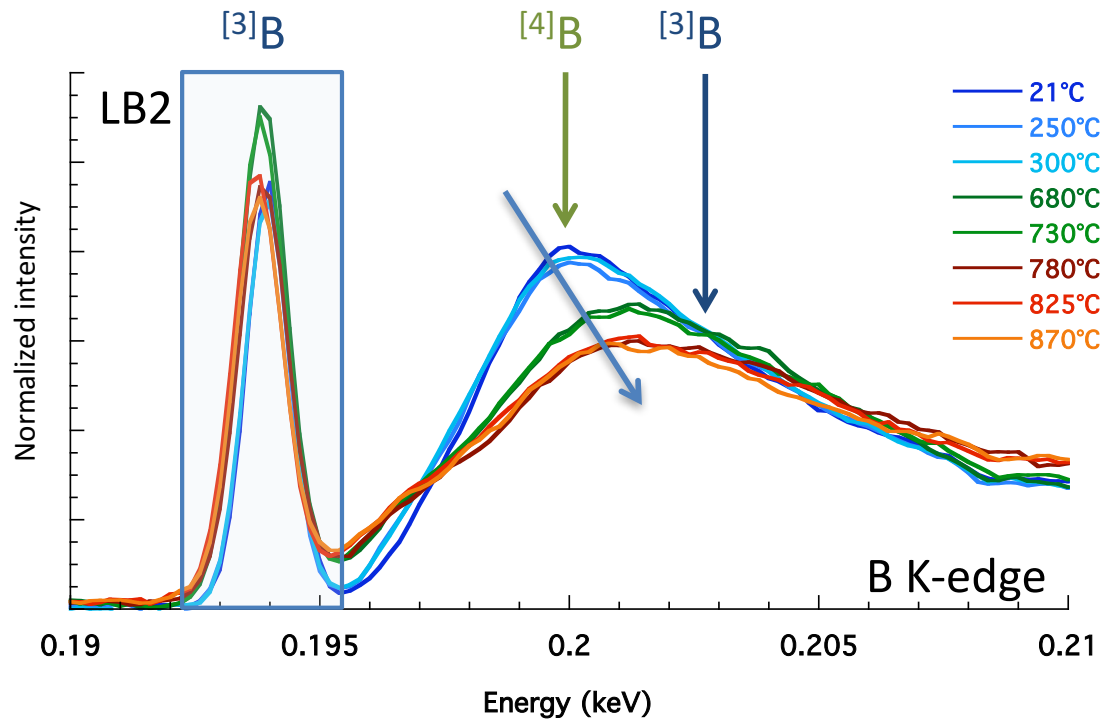
Total Correlation Function

# LITHIUM BORATES VS TEMPERATURE

NRIXS  $\approx$  XAS at B K-edge



# LITHIUM BORATES VS TEMPERATURE



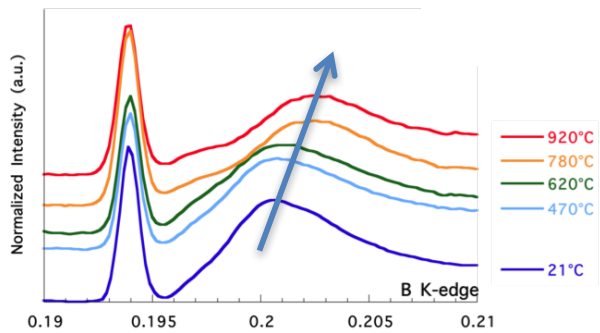
Lee *et al.*, Nature materials (2005) 4, 851-854.

There is a conversion  $[4]B \rightarrow [3]B$  during the heating

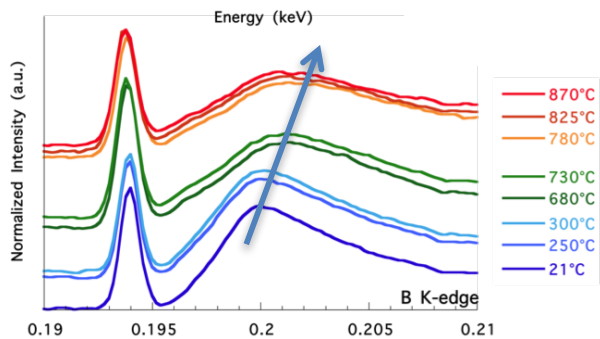
# O K-EDGE OF LI BORATES VS TEMPERATURE

## B K-edge

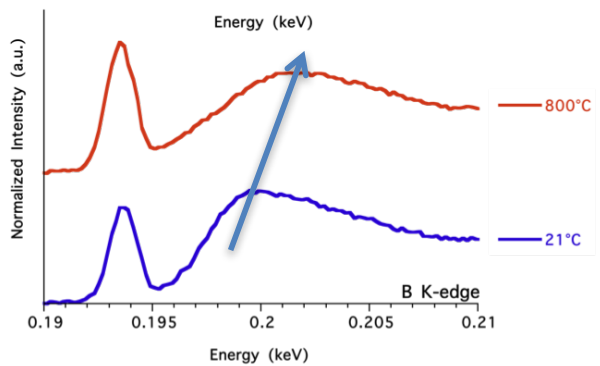
LB3



LB2



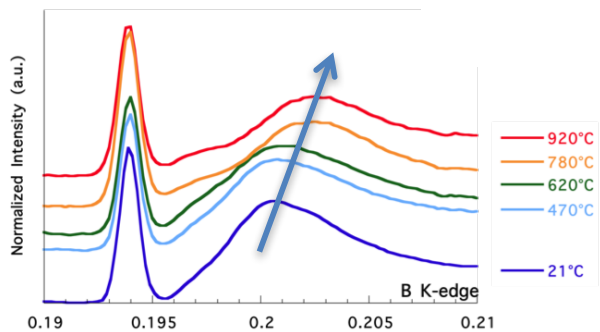
LB



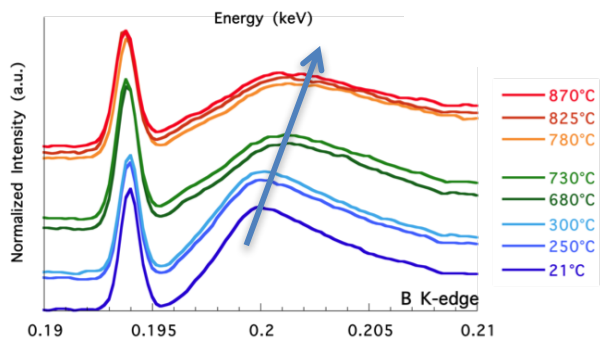
# O K-EDGE NRIXS OF LI BORATES VS TEMPERATURE

## B K-edge

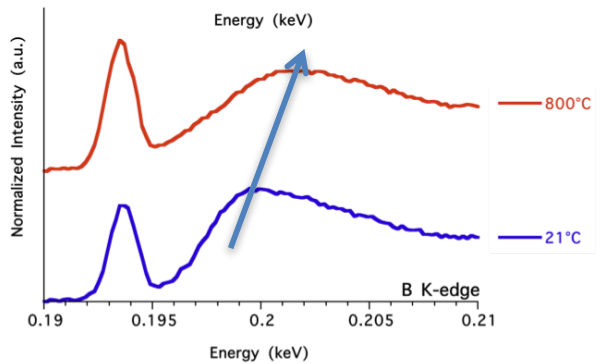
LB3



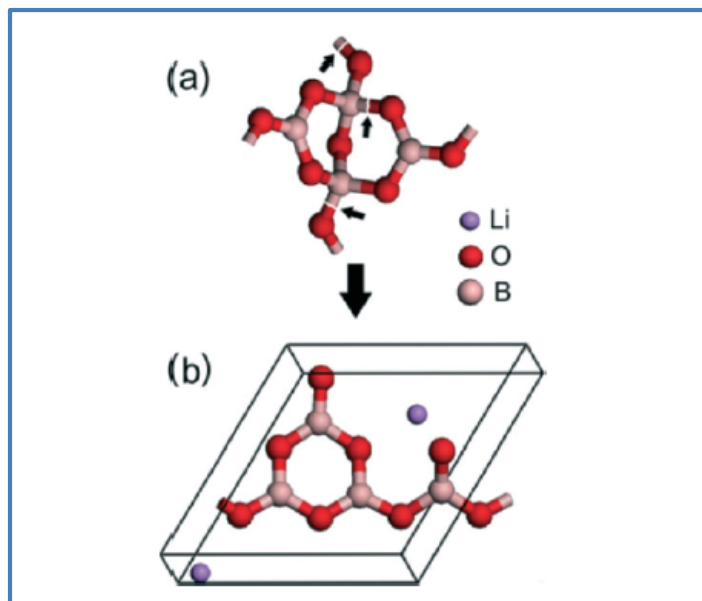
LB2



LB



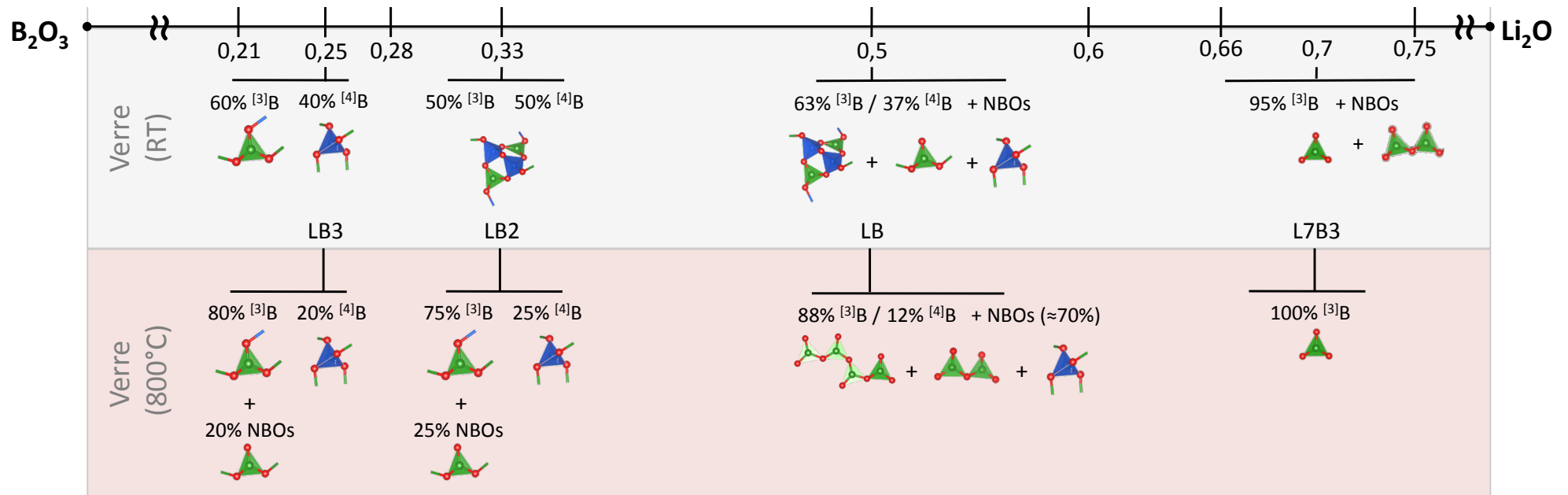
Wan et al. Cryst. Eng. Comm. 2014



# GLASS VS MELT....

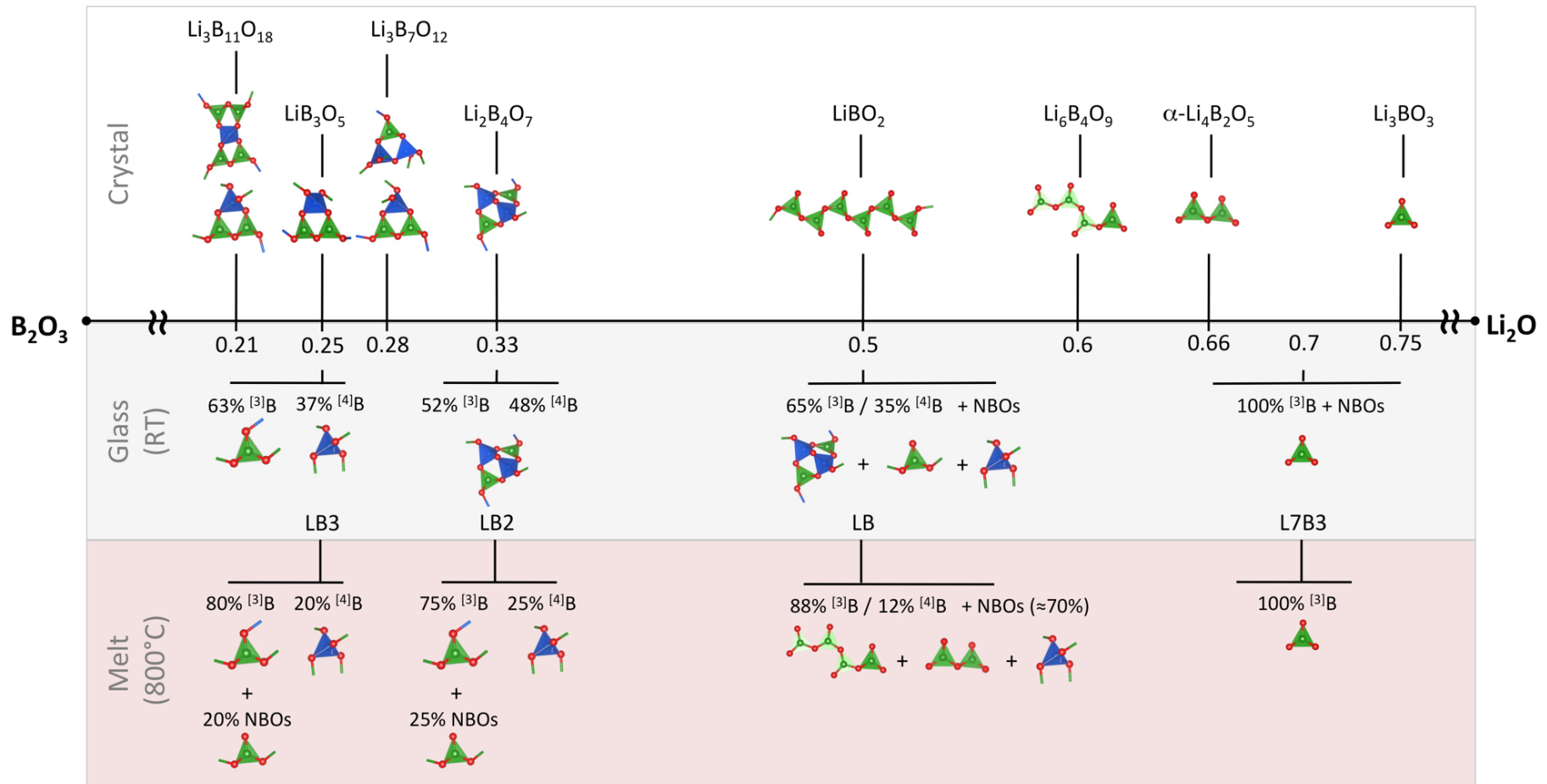
□ Increasing temperature has a large impact on :

- $N_4$
- number of NBOs
- diffusivity of alkali ion
- density
- configurational entropy

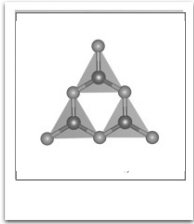




# GLASS vs CRYSTAL vs MELT

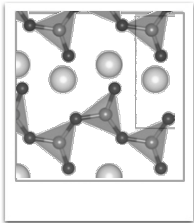


# OUTLINE



## **B<sub>2</sub>O<sub>3</sub> glass**

Archetypal glass former

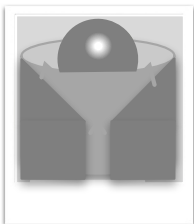


## **Alkali borate glasses (Li, Na, K, Rb, Cs)**

Vitrification domains

Alkali effects on physical properties

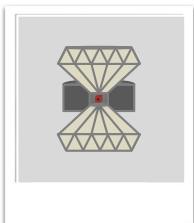
Glass structure: short- and intermediate-range order



## **From glass to melt**

Structure of alkali borate melts

Depolymerization of the borate network



## **Polyamorphism**

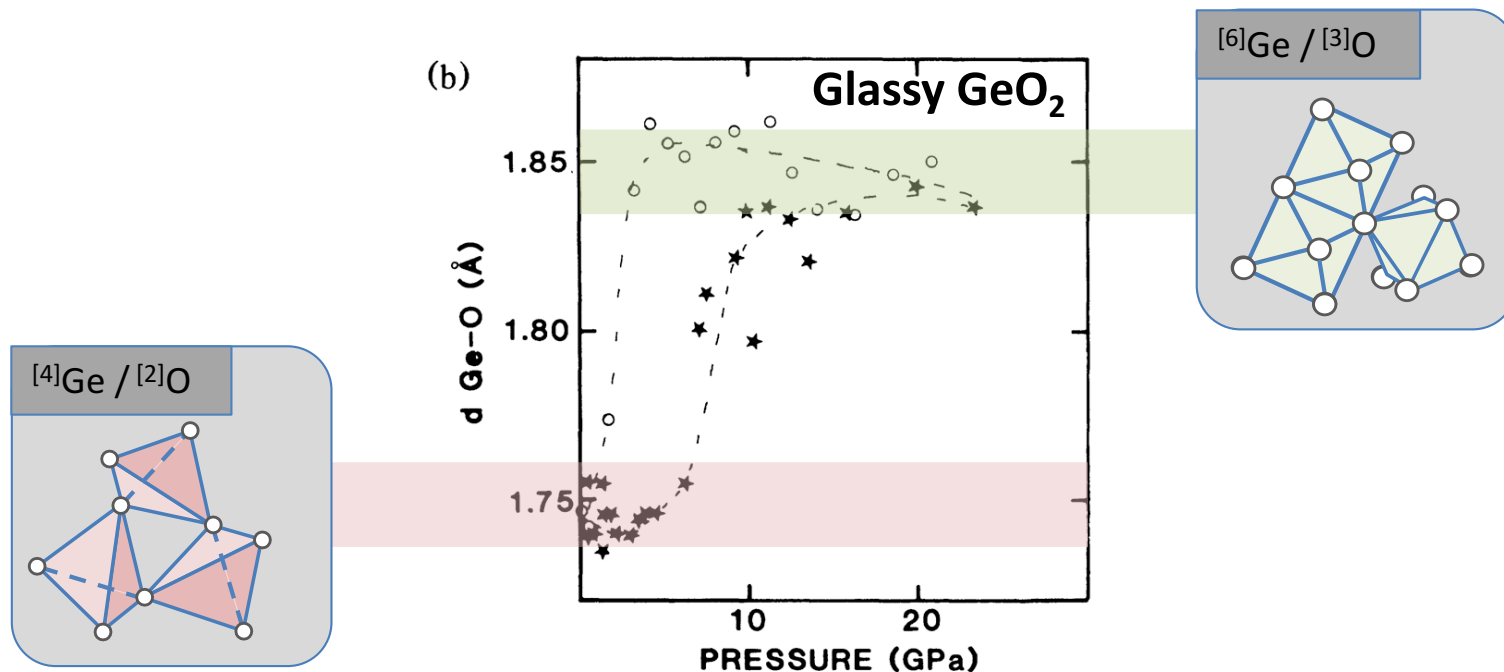


# POLYAMORPHISM

**POLYAMORPHISM:** ABILITY FOR A SYSTEM TO FORM SEVERAL DISTINCT AMORPHOUS STRUCTURES OF IDENTICAL COMPOSITION.

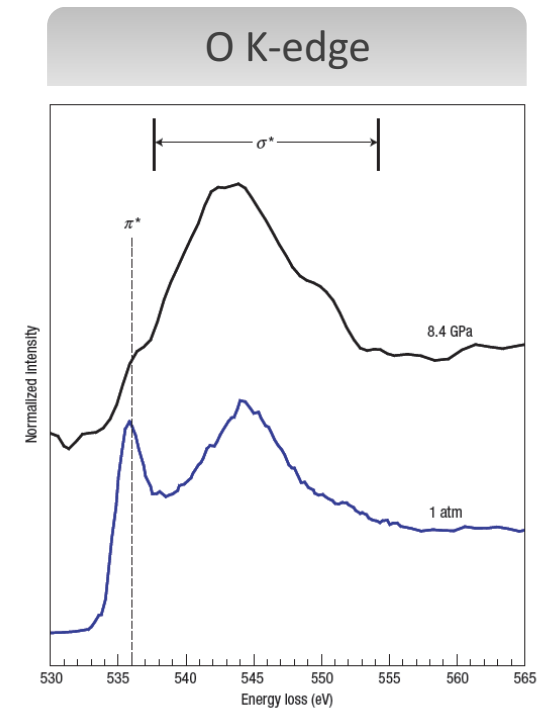
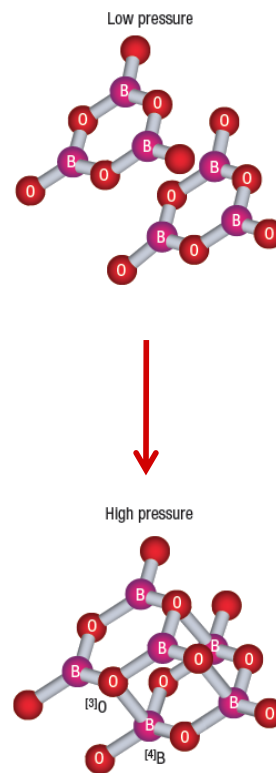
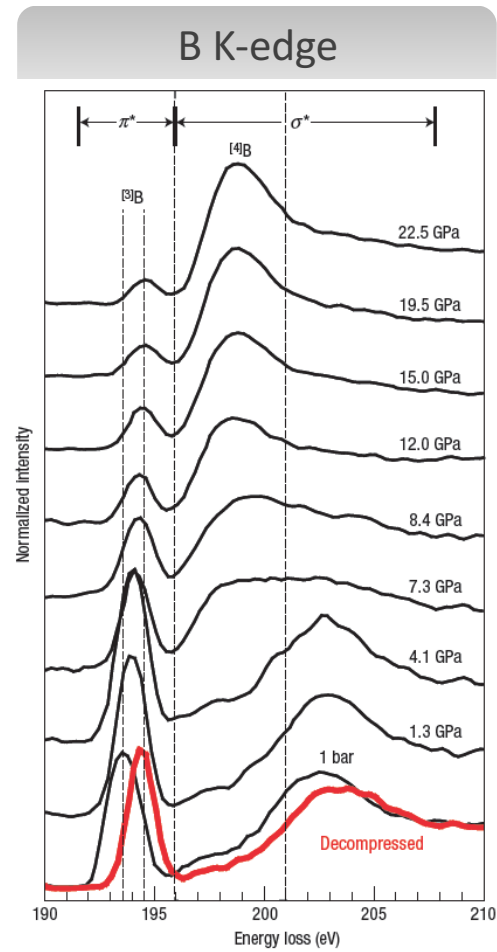
# POLYAMORPHISM

**POLYAMORPHISM:** ABILITY FOR A SYSTEM TO FORM SEVERAL DISTINCT AMORPHOUS STRUCTURES OF IDENTICAL COMPOSITION.

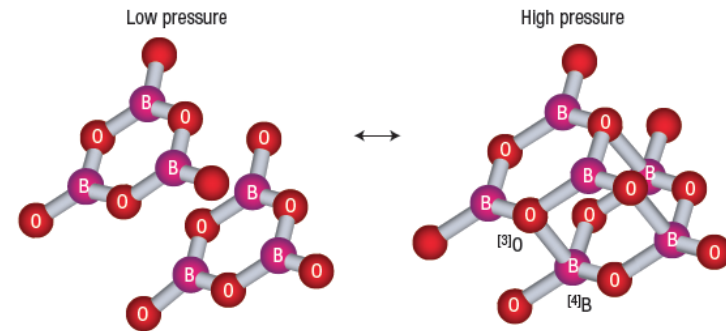
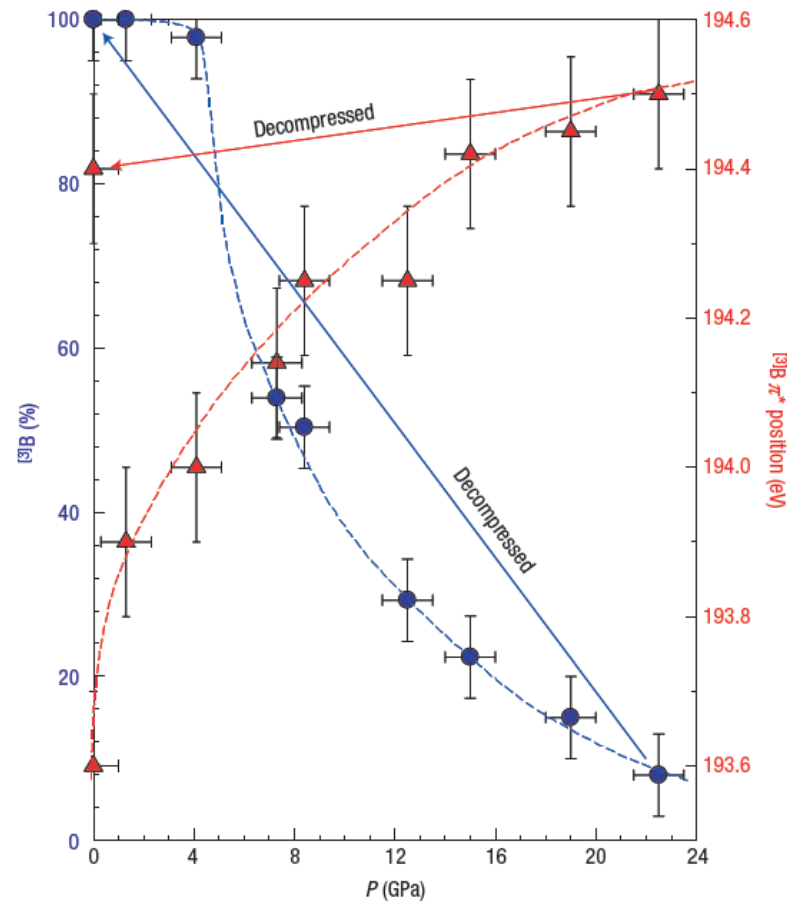


→ Four-fold to six-fold coordinated Ge atom transition

# POLYAMORPHISM IN GLASSY $B_2O_3$



# POLYAMORPHISM IN GLASSY $B_2O_3$

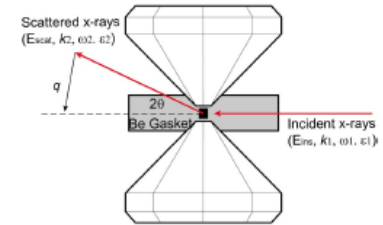
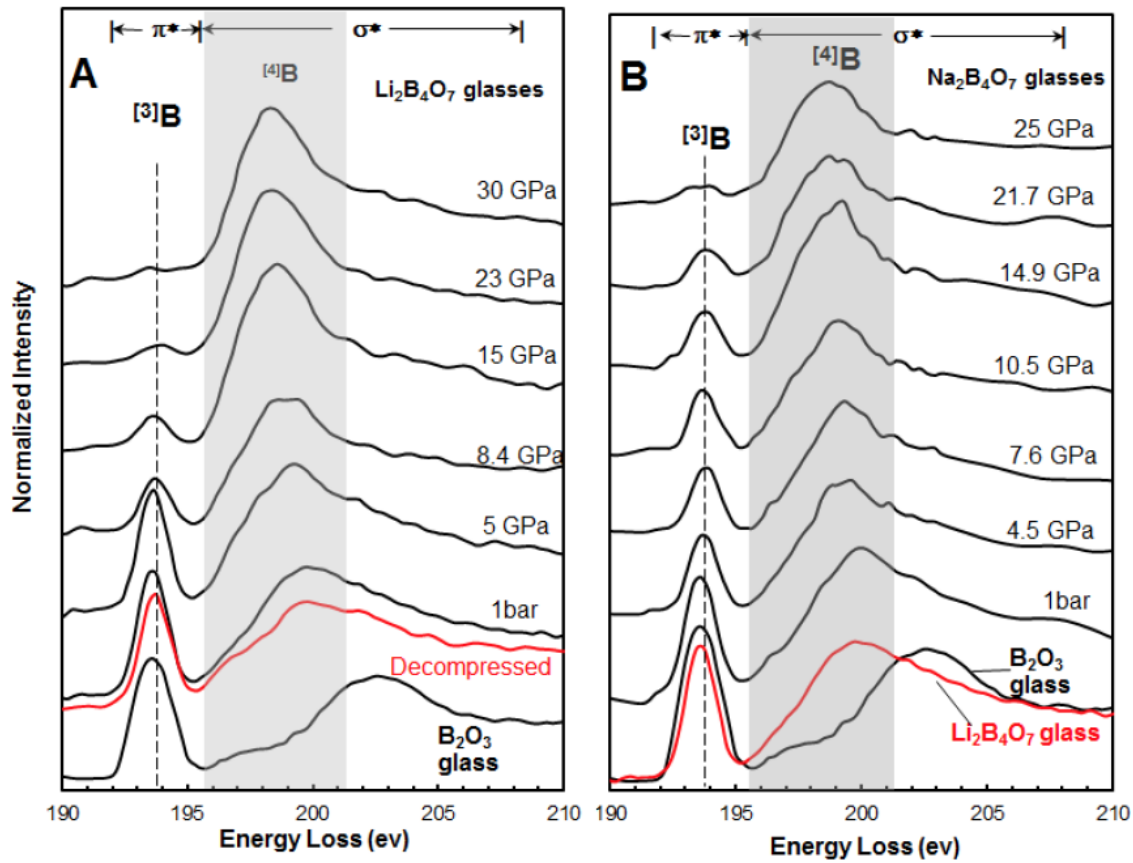


With compression:

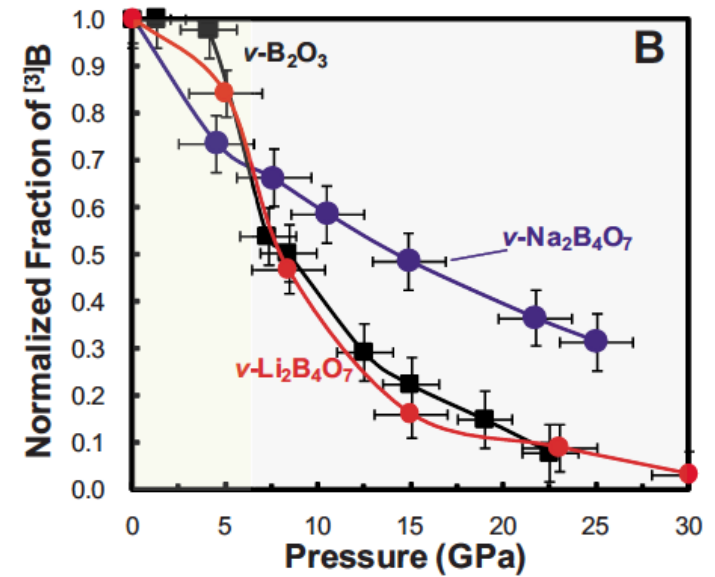
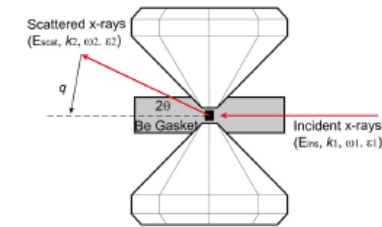
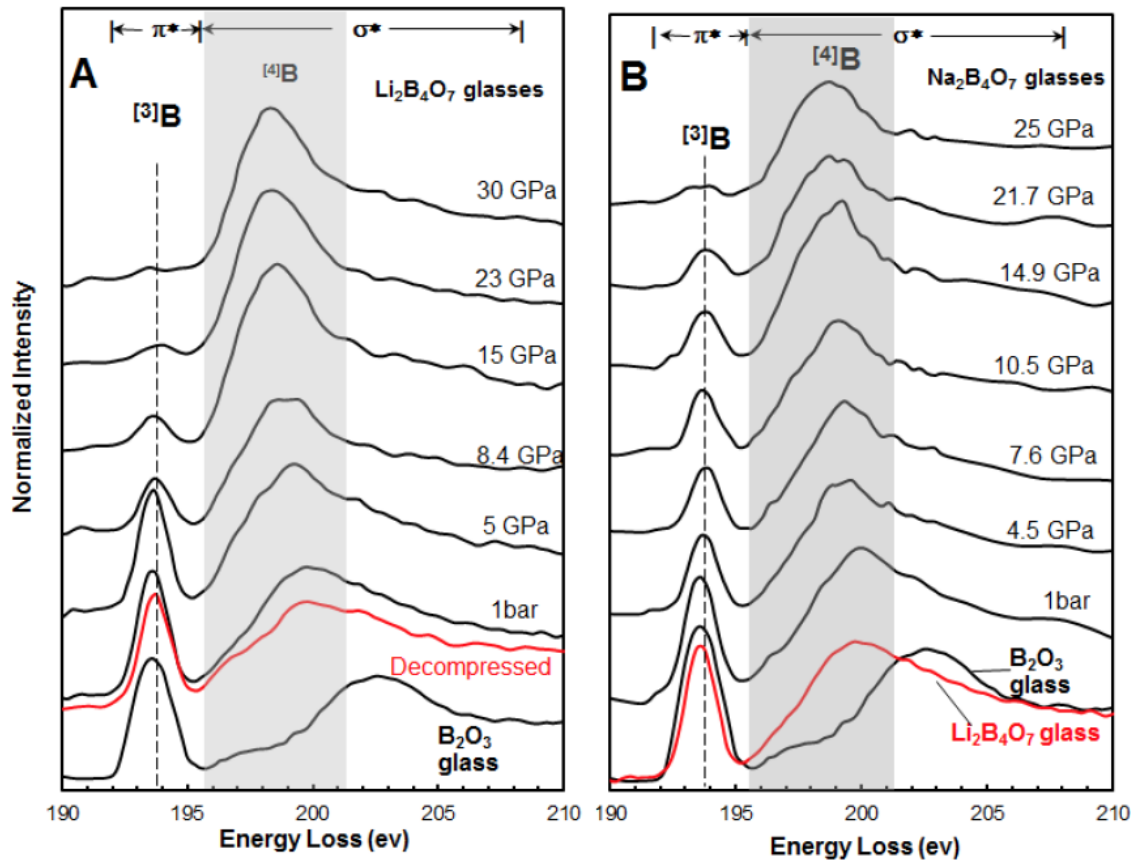
$[3]B \rightarrow [4]B$  conversion

$[2]O \rightarrow [3]O$  conversion

# POLYAMORPHISM IN ALKALI BORATES

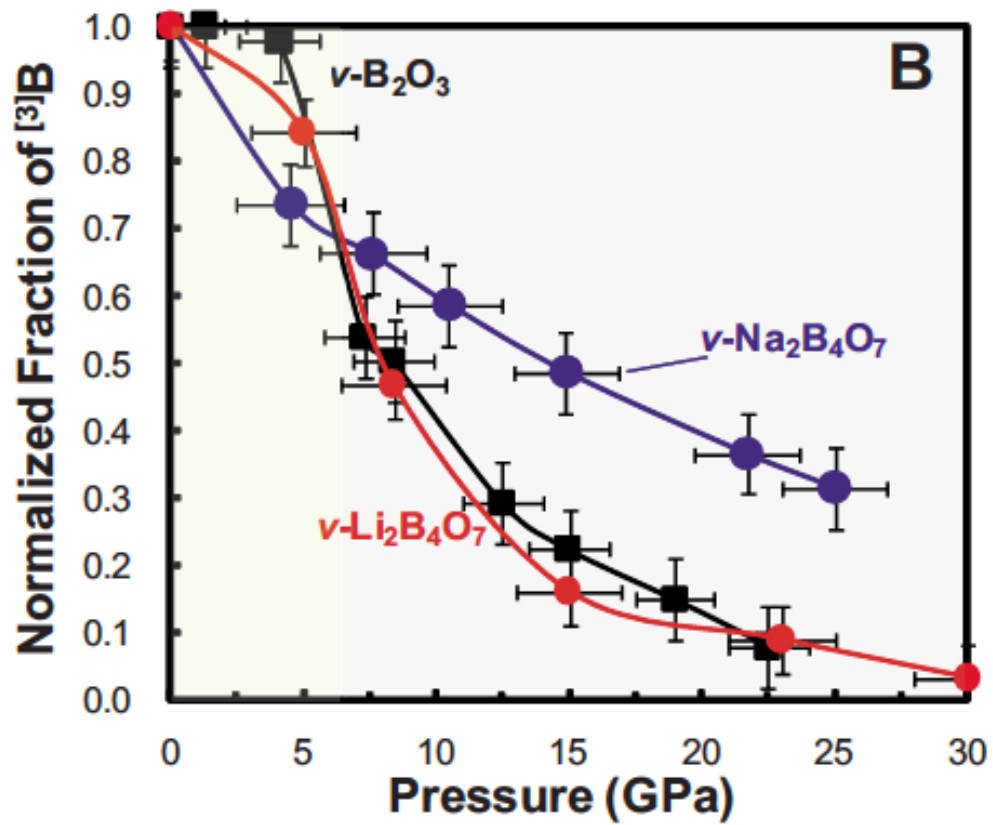


# POLYAMORPHISM IN ALKALI BORATES



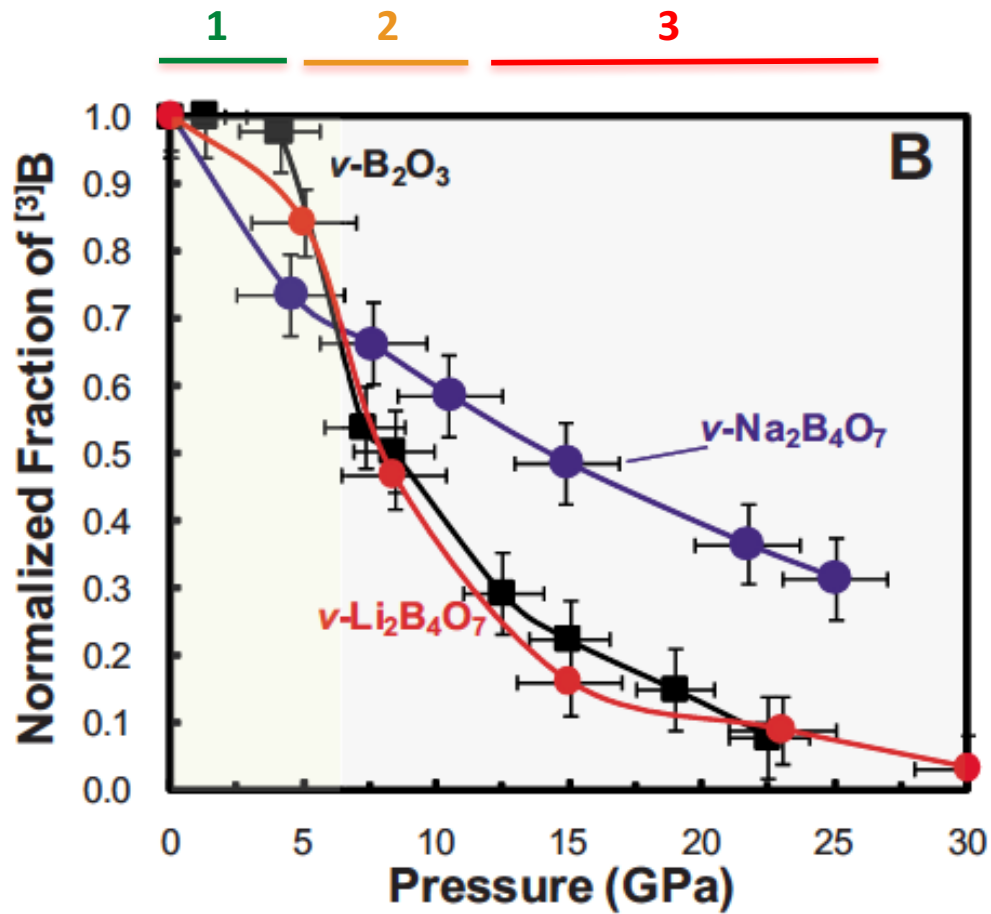


# POLYAMORPHISM IN ALKALI BORATES



# POLYAMORPHISM IN ALKALI BORATES

## MULTI-STEP DENSIFICATION MECHANISM



**1 – Topological variations**  
(no coordination change)

**2 – Coordination change**  
(maximum speed of conversion)

**3 – Larger energy cost of the conversion**

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