Diffusion issues related to silicate glass corrosion stephane.gin@cea.fr



C g z



Various types of glasses in various environments













VERRE & DIFFUSION

Ecole thématique du CNRS sur la diffusion chimique dans les phases vitreuses et liquides

3 - 8 Octobre 2021 Fréjus - Villa Clythia











Reactions and Transport

Reactive Behaviors

- Selective dissolution of glass network
- Restructuring of glass to form gel (dissolution reprecipitation under some conditions)
- Evolution of gel structure
- Dissolution of gel
- Precipitation of 2nd phases



- These layers control both the solution chemistry and the glass dissolution kinetics
- Features and properties of alteration layers depend on many intrinsic and extrinsic parameters
- There is no predictive model available to date





- 1. Diffusion in pristine glass = interdiffusion = ion exchange
- 2. Diffusion in gel
- 3. Diffusion in secondary phases





Contrary to crystals, glass structure is more open, allowing small species (Li, He, H^+ , $H_2O...$) to diffuse in the bulk material AND the glass surface in contact with water offers a large diversity of chemical bonds (distribution of bond lenghts and angles)





Topological analysis of interstials in crystal and equivalent glass

Density: crystal = 2.62 g/cm³ and glass = 2.34 g/cm³
 Dissolution: r_{glass} ~50 r_{crystal}

Perez et al. Geochim. Cosmochim. Acta 254 (2019)

Water diffusion in glass

1





Bunker, J.Non-Cryst.Sol. (1994)

Water diffusivity in glass is strongly tied to its topology (the degree of disorder depends on the glass composition and the





Ref	Glass	H/Na
Lanford et al. (1979)	SiO ₂ , Na ₂ O, CaO	2.9 ± 0.3
Houser et al. (1980)	SiO ₂ , Na ₂ O	1.75
Tsong et al. (1981)	SiO ₂ , Na ₂ O	2.0 ± 0.3
		or 3.2 ± 0.4 depending % Na
Dran et al (1989)	SiO ₂ , Na ₂ O, CaO	\sim 2 near surface, \sim 1 near diss front
Ferrand et al. (2006)	SON68	2.6 ± 0.3



Hydration/Interdiffusion

Glass	$\mathbf{D} \mathbf{m}^2 \cdot \mathbf{s}^{-1}$	Temp °C	pН	Ref
Si-Na	$10^{-17} - 10^{-15}$ 10^{-15}	60 - 100 30	Pure water 4.9	Rana et Douglas (1961b) Houser et al. (1980)
Si-Na-Ca	$\frac{10^{-18}}{10^{-20}} - 10^{-17}$	90 60-100	Pure water	Lanford et al. (1979) Rana et Douglas (1961b)
Medieval K-stained glass window	10 ⁻¹⁷ - 10 ⁻¹⁸	T° ambiante		Starponiah at Liboural
Medieval Na- stained glass window	10 ⁻²⁰	T° ambiante		(2006)
Si-Na-Al	$10^{-18} - 10^{-19}$ 10^{-16}	70 25	Pure water	Smets et Lommen (1982) Boksay et al. (1975) Smit et Stein (1979)
Borosilicates	10^{-17} - 10^{-18}	90	9,5	McGrail et al. (1984)
SON68	$10^{-21} - 10^{-22} \\ 10^{-21} - 10^{-24}$	90 30 - 90	7 - 10 7 - 10	Ferrand et al. (2006) Chave et al. (2007)
Obsidian	10 ⁻²²	22	Pure water	Friedman et al. (1966)

Ea = 30-45 kJ/mol





Jabraoui, J.Phys. Chem. C, 125 (2021)

- MD and ab initio calculations show that IX is a complex process that cannot be dissociated with water dissociation, SiOH, BOH formation, hydrolysis...
- The gap between the atomistic level and the macroscopic observations is not bridged.

Gin et al., *npj. Mater. Degrad.* 5, 42 (2021)

2 - Diffusion through the gel layer



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Complementary Analytical & Numerical Techniques applied to a reference 6 oxide borosilicate glass



Chemistry

ToF-SIMS low lateral resolution 0 SEM, TEM qualitative, Z>6 0 XPS only 5-10 nm 0 APT time consuming 0

Collin, npj-Mat. Degrad. 3 (2019) Gin, Geochim. Cosmochim. Acta 202 (2017) Guo, Nat. Mater 19 (2020) Perea, npj-Mat. Deg. 4 (2020)

Structure

IR, Raman 0 bulk technique NMR \cap MD potentials dependent 0

Porosity

 \cap

0

0

- SAXS.SANS 0
 - $\Phi_{\rm pore}$ > 1-2 nm model dependent **BET/BJH**
- SE 0
- **Cryo-APT** 0
- TEM 0
 - MD, KMC rules under development

Kaya, J. Non-cryst. Sol., 556 (2021) Collin, npj-Mat. Degrad. 2 (2018) Collin, npj-Mat. Degrad. 2 (2018)

Cailleteau, Nat. Mater. 7 (2015) Ngo, npj-Mat. Degrad. 2 (2018) Ngo, npj-Mat. Degrad. 2 (2018) Schreiber, Ultramic. 194 (2018) Mir, npj-Mat. Degrad. 4 (2020) Rimsza, npj-Mat. Degrad. 2 (2018); Kerisit, J. Non-cryst. Sol., 556 (2021)

Transport

- **ToF-SIMS** 0
 - qualitative MD, KMC for simple systems

Gin. Nat. Comms. 9 (2018) Collin, npj-Mat. Degrad. 2 (2018)

















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WATER DYNAMICS IN THE GEL (1)



- ¹⁸O[·] enrichment in the gel (up to the reaction front) is noticed even after 3 min of tracing
- Profiles are flat
- ¹⁸O[.] enrichment in the gel slowly increases with time
- Quantification of isotopic exchange is possible

Experimental data were implemented in a mathematical model to calculate diffusion coefficients:



Schematic representation of the mathematical model considering two diffusion coefficient D_c and D_d .

 D_d values calculated from experimental data (Fraction of oxygen exchanged calculated from ToF-SIMS data).

 $D_c > 10^{-11} \text{ m}^2 \cdot \text{s}^{-1}, D_d = 10^{-21} - 10^{-23} \text{ m}^2 \cdot \text{s}^{-1}$

This difference cannot be explained by nanoconfinement effect only, as MD simulation yields mean diffusion coefficient of nanoconfined water ($\Phi = 1$ nm) roughly one to two orders of magnitude lower than that of bulk water \rightarrow non-accessible porosity due to network reorganization.

Effect of alkali on glass alteration was also studied (~20 mmol·L⁻¹ of XCl with X = Li, Na, K and Cs):



ET(B) well correlated with square root of time in no-alkali, Li and Na case \rightarrow diffusion coefficient can be calculated and are low (9 order of magnitude lower than that of bulk water).

	10			A 11 11			, · · · · ·			97	
	8		Without With Li With Na With K	Alkalı			Ţ	• •	-	92	
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	-	.6		00	0	0	0	0		07	
	0 🤇))	4		8	12		16		0	
					Time ^{1/2}	² (d)					
			San	nple	D	(10 -18	m ²	s ⁻¹)			
			Jun		+	(• /	_		
					1	<u> </u>)				

Sample	D (10 ⁻¹⁸ m ² .s ⁻¹)		
ISG – PW	3		
ISG – Li	4		
ISG – Na	4		

Huge effect of exogenous species on glass alteration and passivation.



Knowing **O repartition** (%):

O _{BO}	O _{NBO}	O of H ₂ O mol
71.6	15.5	12.9

In the following mass balance calculations Mobile species in red

Let us play again with O isotopes





>60% of O exchanged after 4 or 10 days of tracing for baby gel (3 days of prealteration), <24% of O exchanged after 4 month of tracing for adult gel (1625 days of pre-alteration).

Gel reactivity decreases with aging



Spectroscopic Ellipsometry experimental data were collected on pristine glass, newly-formed gel and mature gel:



Maturation of gel lead to a bigger-sized porosity, but less accessible \rightarrow **network reorganization** leads to a rate drop.

ARTICLES

materials

17

Laser

beam

laser

focus

Solution

inlet

Thermo couple

* v

Real-time in situ observations of reaction and transport phenomena during silicate glass corrosion by fluid-cell Raman spectroscopy

Thorsten Geisler 1*, Lars Dohmen^{1,2}, Christoph Lenting' and Moritz B.K. Fritzsche¹

Glass 60 SiO₂, 20 Na₂O 20 B₂O₃ Altered at 85°C in NaHCO₃ and D₂O





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ARTICLE OPEN Insights into the mechanisms controlling the residual corrosion rate of borosilicate glasses

Stephane Gin 12¹²², Xiaolei Guo², Jean-Marc Delaye¹, Frédéric Angeli¹, Kamalesh Damodaran¹, Véronique Testud³, Jincheng Du⁴, Sebastien Kerisit 13¹² and Seong H. Kim 13¹⁶









Interestingly, the continuous and monotonic trend displayed by $r_m(B)$ over a broad pH range may suggest that the same reaction could control B dissolution between pH 3 and 9. According to first principles simulations, this reaction implies the nucleophilic attack of H⁺ on B and the hydrolysis of B–O–Si by H₂O⁴⁷. The apparent activation energy for B release at pH 3 is 59 ± 5 kJ mol⁻¹ (Fig. 5b), a value close but slightly lower than that for Si dissolution (64 ± 3 kJ mol⁻¹ from pH 3 to 5.6)³⁵. From the parabolic trend following the initial release of B, an apparent diffusion coefficient of B was calculated at 90 °C (Fig. 5d):

$$D_B(m^2 s^{-1}) = 10^{-13.65 - 0.51pH}$$
⁽²⁾



Etudes récentes sur le verre ISG à 90°C pH 7, 9, 5, 3

- 1. Les liaisons Si-O-B s'hydrolysent d'autant plus vite que le pH est bas. L'Ea associée est légèrement plus faible que celle associée à la dissolution des Si-O-Si. Il est proposé qu'une seule réaction impliquant les protons contrôle l'hydrolyse des liaisons Si-O-B.
- 2. Na et Ca ne sont dissous que si les laisons Si-O-B sont hydrolysées.
- 3. Certaines liaisons Si-O-Si sont hydrolysées mais le Si ne passe pas en solution.
- 4. Le départ du B permet la recondensation des Si-OH, étape clé de la restructuration du gel
- 5. Le départ du B est responsible de plus de 90% de la porosité du gel
- 6. Le B hydrolysé diffusent lentement dans le gel. Si l'hydrolyse est plus rapide que la diffusion, il s'accumule dans le gel et ralentit la poursuite de l'hydrolyse.

Modèle de formation du gel par réorganisation in situ sans précipitation de phases secondaires





Questions préalables à l'établissement d'un modèle pour la Vr

(2)

- 1. Où se fait le blocage dans le gel $(H_2O \text{ et } B_{aq})$?
- 2. Le B dissous et piégé dans le gel a-t-il un rôle sur la diffusion de l'eau?
- 3. Quelle est la spéciation et la diffusivité du B dissous dans le gel ?
- 4. Quelle est l'activité de l'eau et des ions en milieu confiné ?
- 5. Comment est répartie la porosité dans le gel (ouverte vs fermée) en lien avec la taille des espèces diffusantes ?

Cea 3. Diffusion in alteration products

Verney-Carron et al., *Geochim. Cosmochim. Acta* 2008 ; 2010 Verney-Carron et al., *J. Nucl. Mat.*, 2010



Quantitative validation of a mechanistic model over 1800 years – Applicable to other glasses by the analogy of the mechanisms involved



NANO-SIMS FOR PROBING ATMOSPHERIC CORROSION OF STAINED GLASS WINDOW

Altered stained-glass window (14th century), re-corroded in laboratory under 90% RH of $D_2^{18}O$

Sessegolo et al., npj-Mater. Degrad. 2 (2018)



Alteration resumed – Interdiffusion is the dominant mechanism
 The alteration layer is not protective
 An interdiffusion coefficient of 5 x 10⁻²⁰ m²/s is derived



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