

Structure "and properties" of glasses and melts

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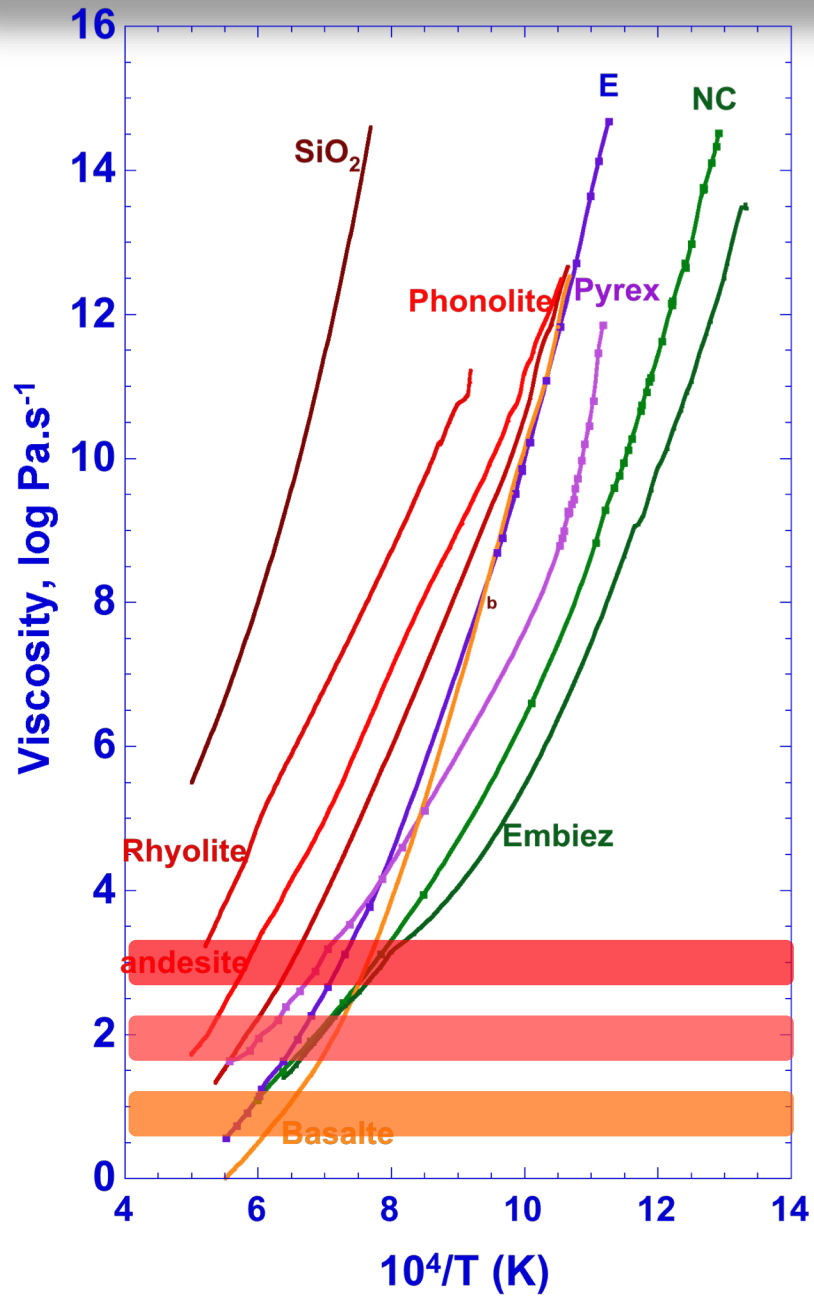
Thanks :

C. Le Losq, IPGP

P. Florian, L. Hennet, D. Massiot -CEMHTI

L. Cormier - IMPMC

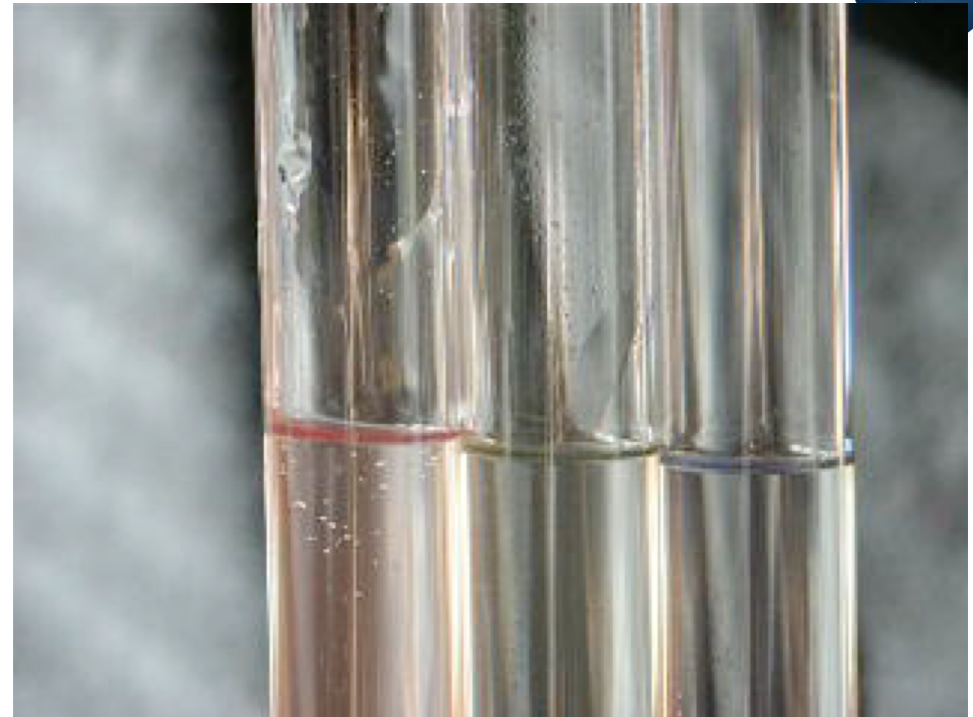
N. Trcera - SOLEIL

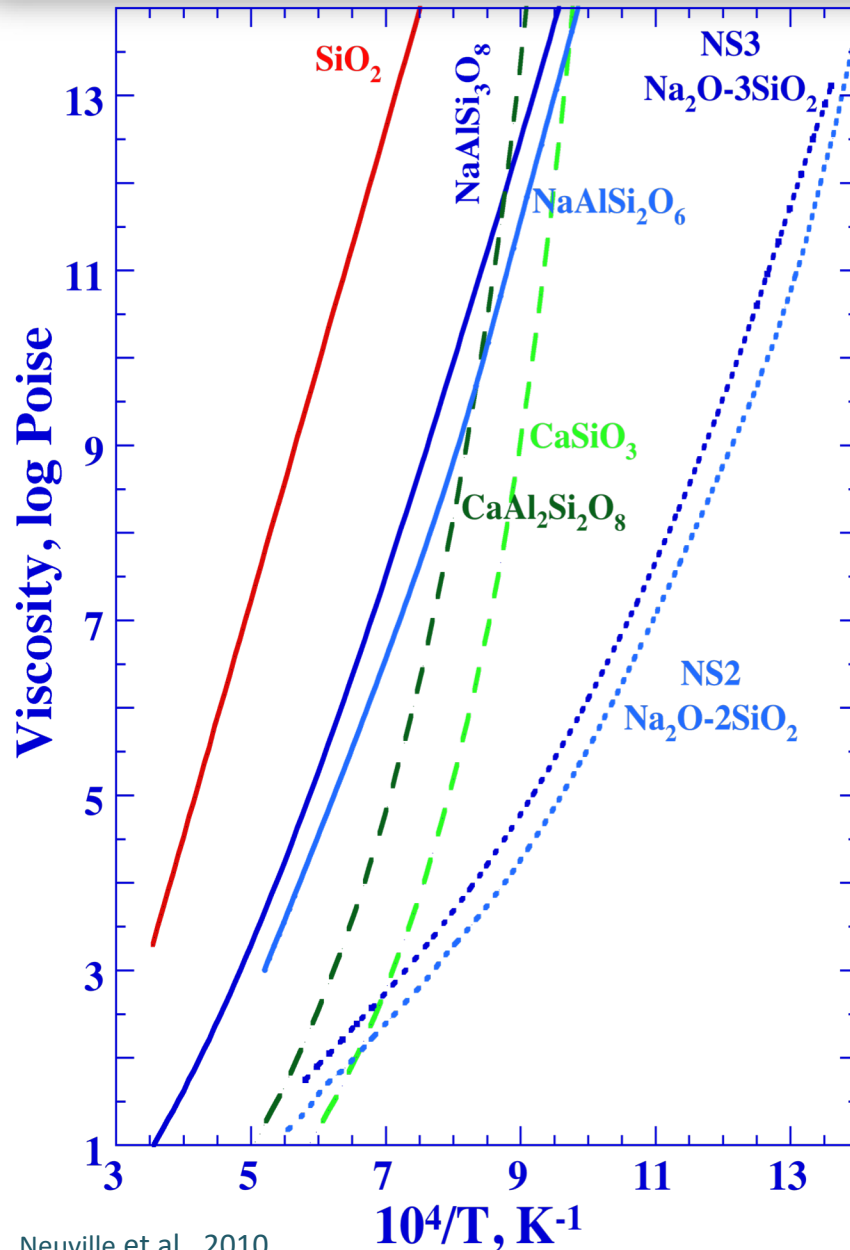


1

2

3





Arrhenius :

$$\eta(T) = A \cdot \exp(E/RT)$$

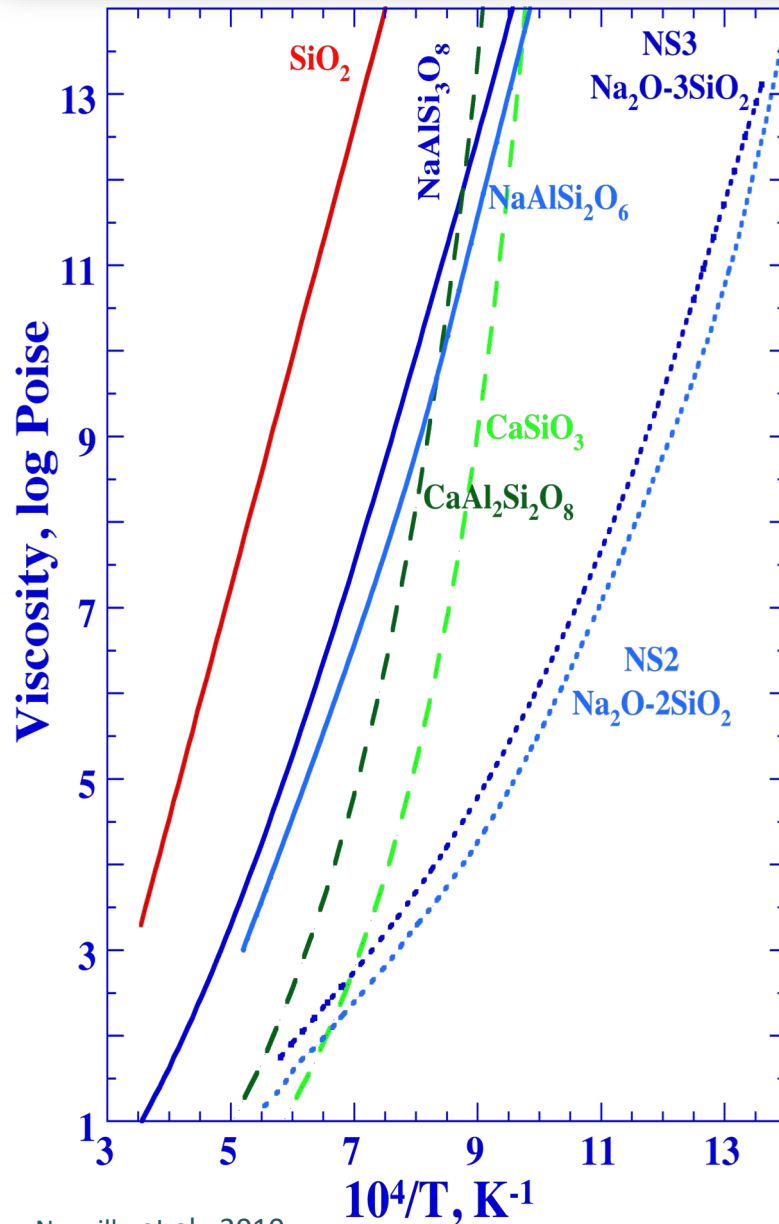
$$\Leftrightarrow \log \eta = A + B/T$$

Yes but only for SiO₂, GeO₂, NaAlSiO₈, KAlSiO₈ because activation energy change from 2000kJ/mol at 1000K up down 300kJ/mol at 1800K for NS3.

Need TVF equation

$$\log \eta = A_1 + B_1/(T-T_1)$$

But, just a fit



$$\eta(T) = A_e \cdot \exp[B_e / TS^{\text{conf}}(T)]$$

Proposed by Adam and Gibbs 1965

First used to silicate melts by Urbain 1972,

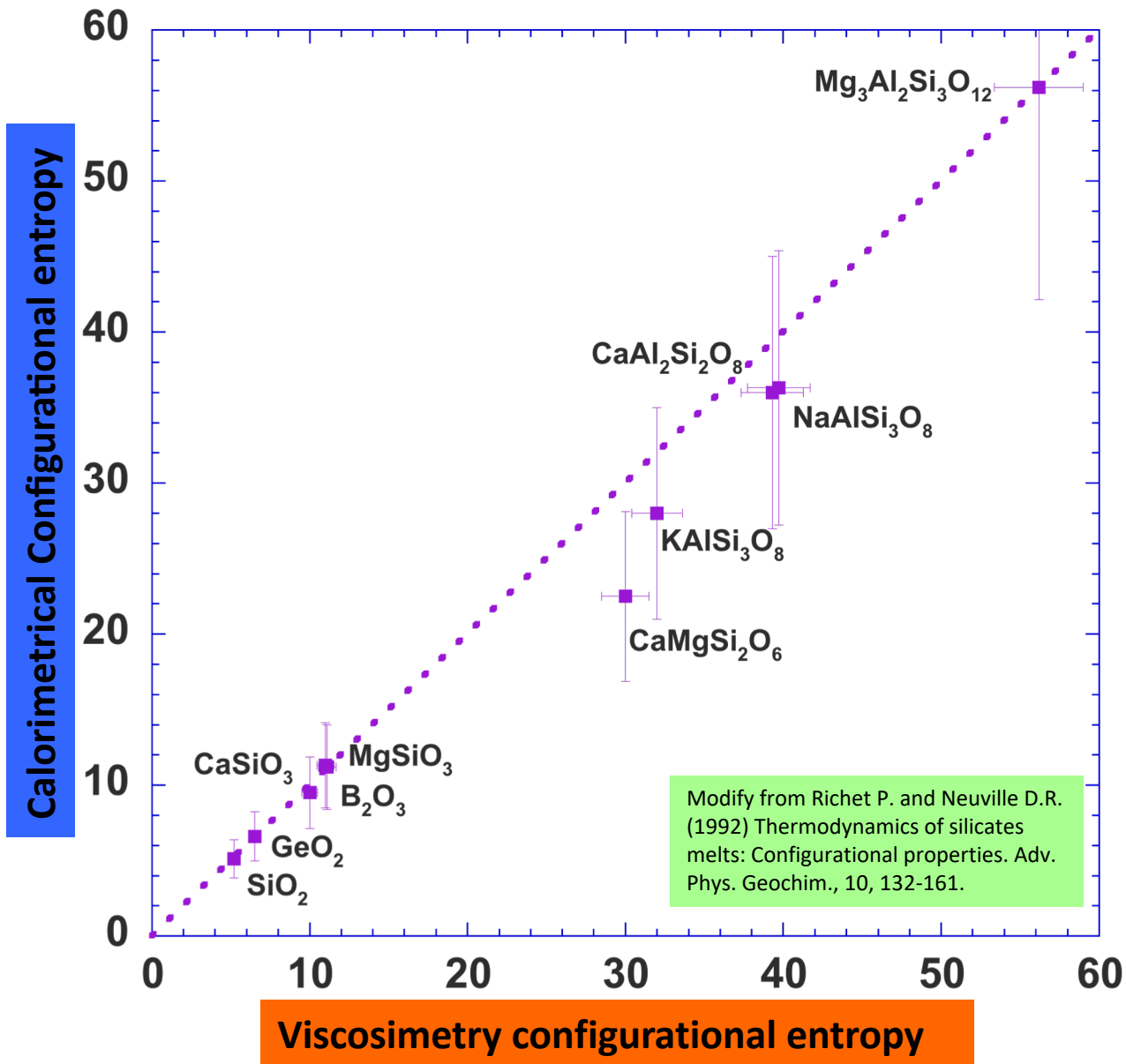
Wong and Angell 1976,

Scherer 1984, Richet 1984, Neuville and Richet 1991....

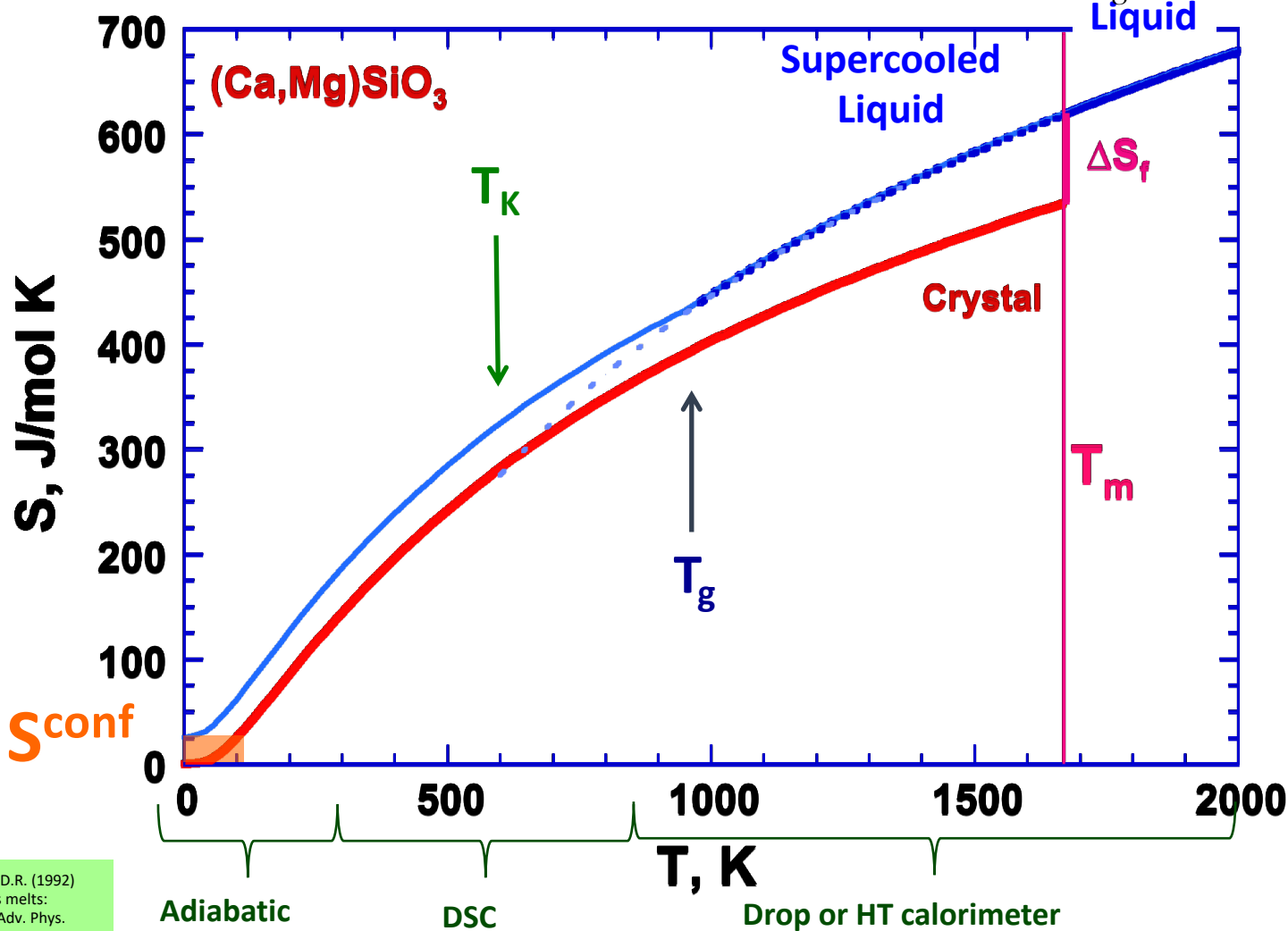
$$S^{\text{conf}}(T) = S^{\text{conf}}(T_g) + \int_{T_g}^T C_p^{\text{conf}} / T dt$$

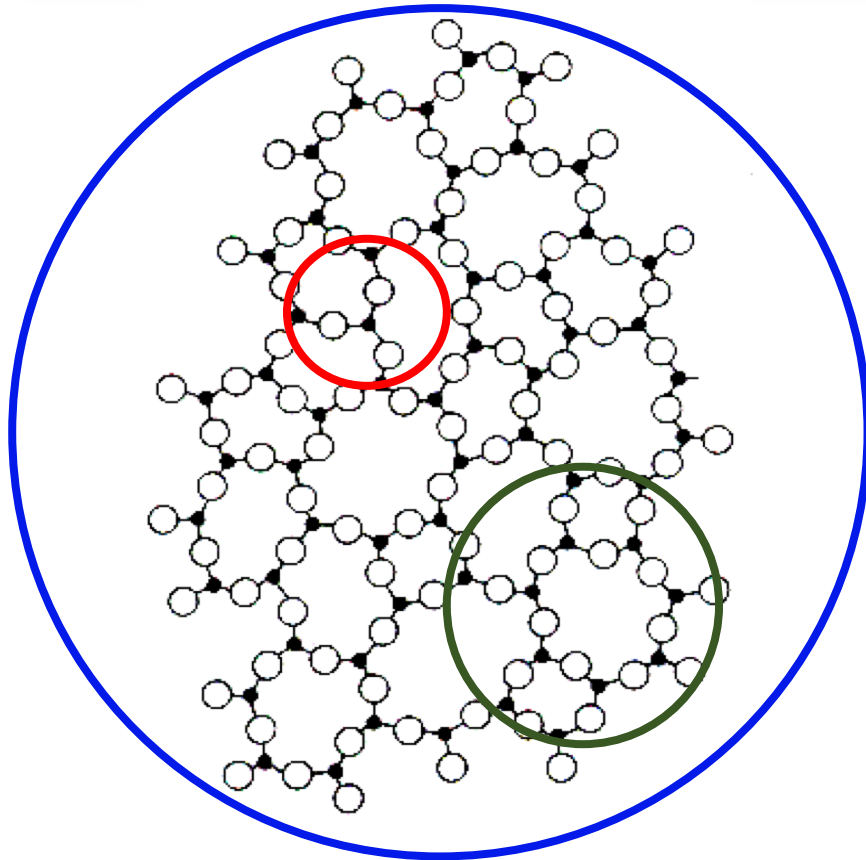
$$C_p^{\text{conf}}(T) = C_{pg}(T_g) - C_{pl}(T)$$

Calorimetry measurements
=> Easy



$$S^{conf}(T_g) = \int_0^{T_m} \frac{C_p^{Crystal}}{T} \cdot dT + \Delta S_f + \int_{T_m}^{T_g} \frac{C_p^{liquid}}{T} \cdot dT + \int_{T_g}^0 \frac{C_p^{glass}}{T} \cdot dT$$





Short range order $< 3 \text{ \AA}$:

- Coordination, bond length, bond angle
- homopolar (-Se-Se, -C-C, -As-As) versus heteropolar (Si-O, B-O, Ge-S)

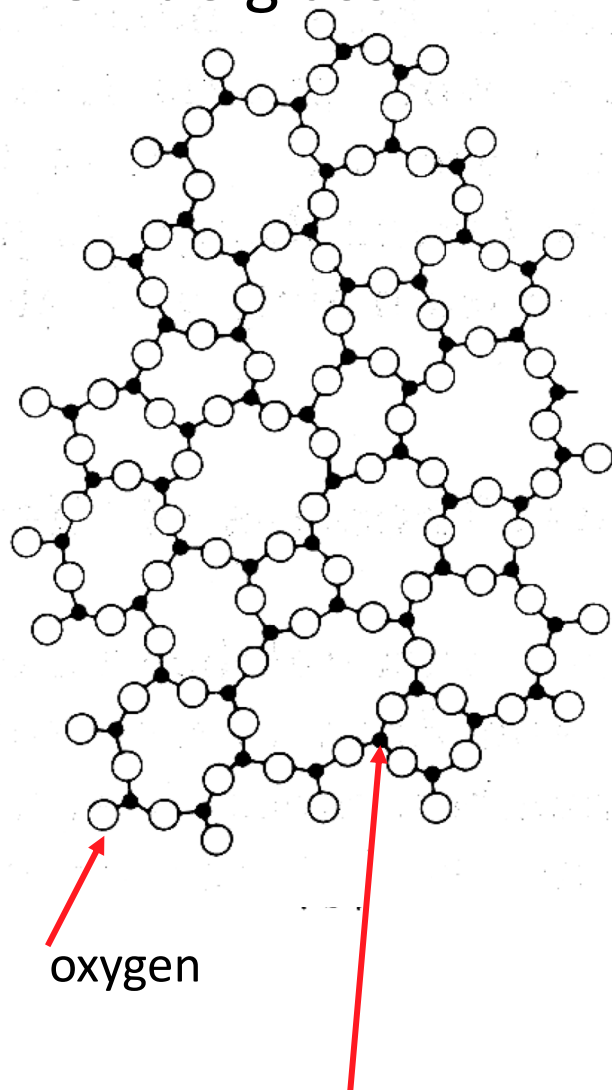
Medium (intermediate) range order ($\sim 3\text{-}10\text{\AA}$):

- angles between structural units
- connectivity between structural units (linkage by corner, edge, face)
- dimensionnality, rings

Almost no long order (no periodicity !):

- phase separation
- inhomogeneities

Oxide glass

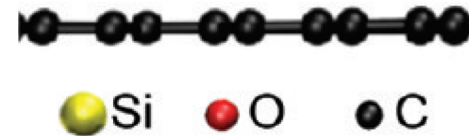
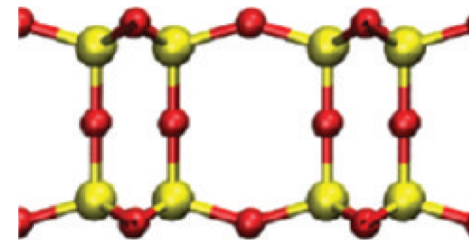
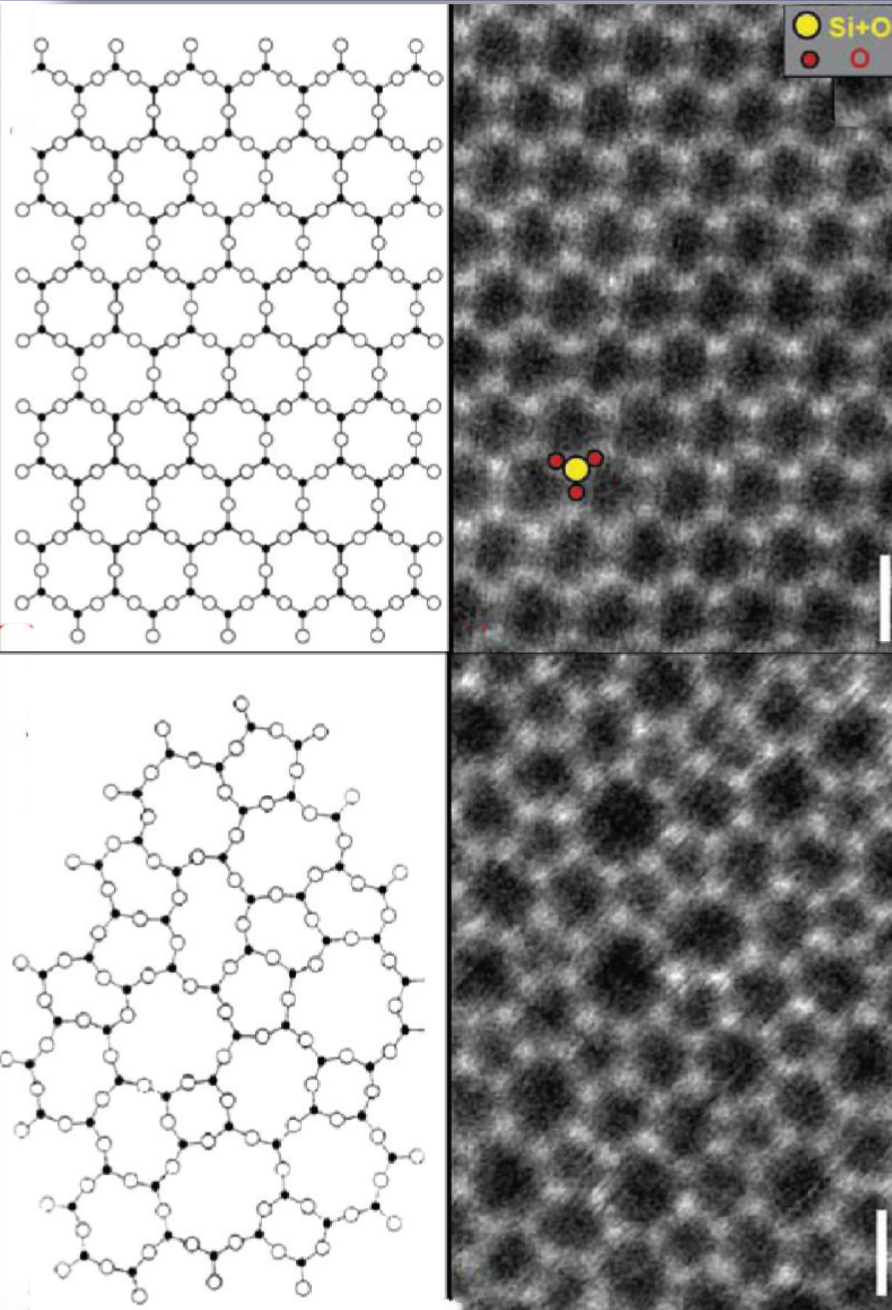


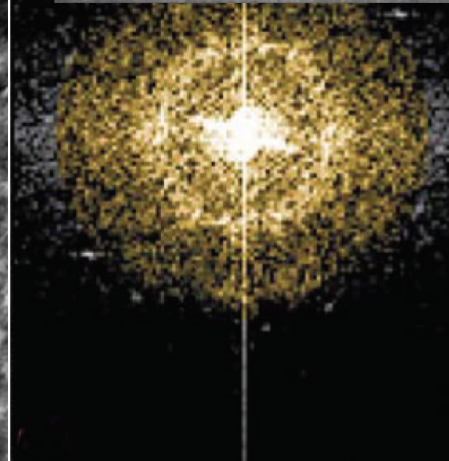
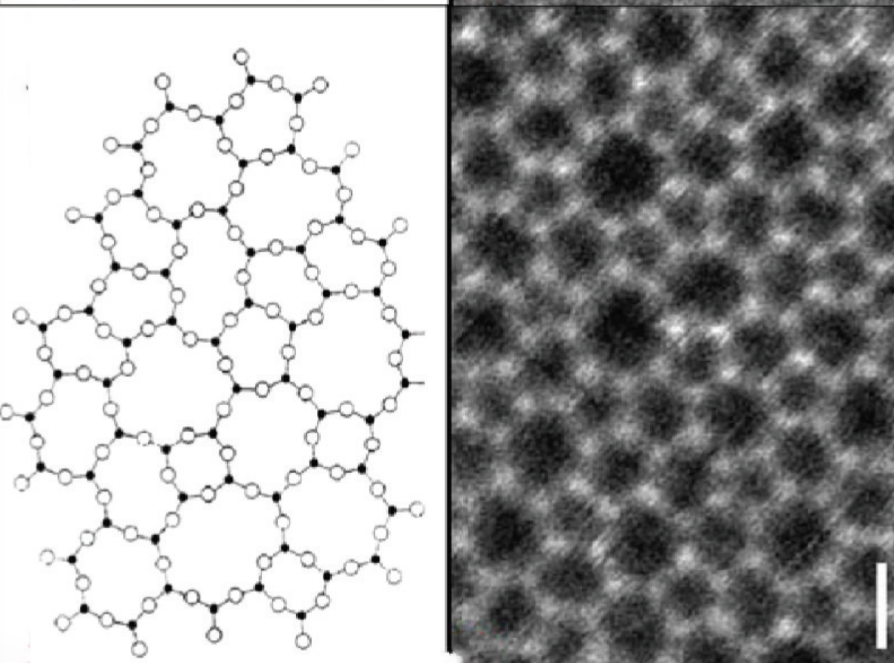
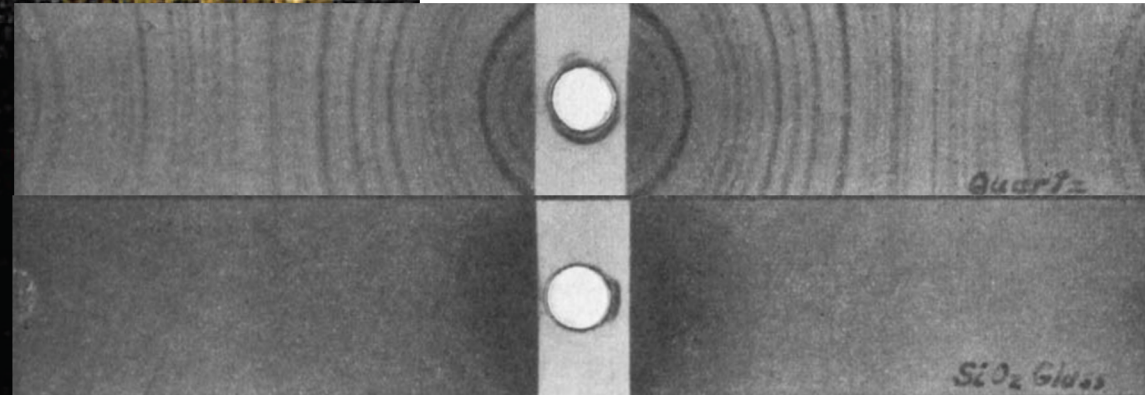
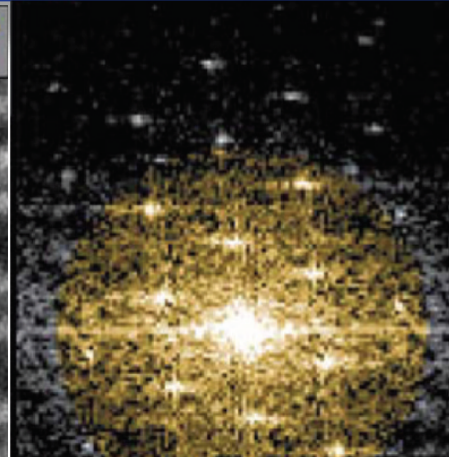
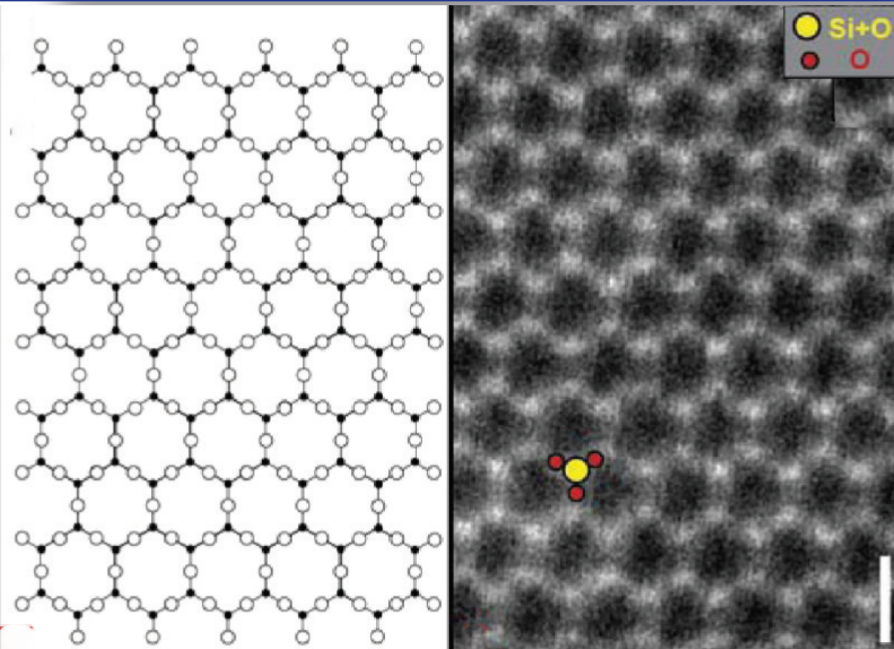
oxygen

Network former (Si, Al, Fe, Ti, B...)

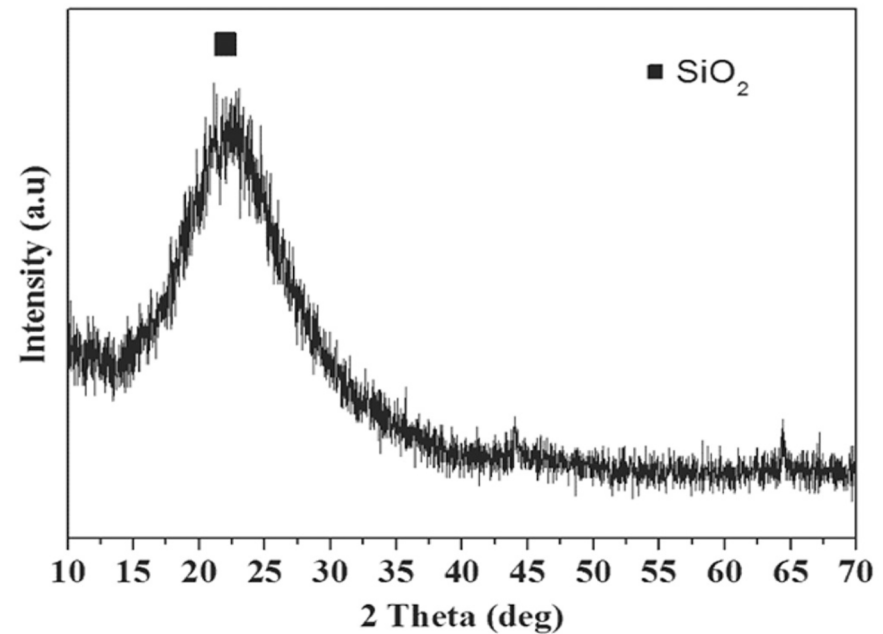
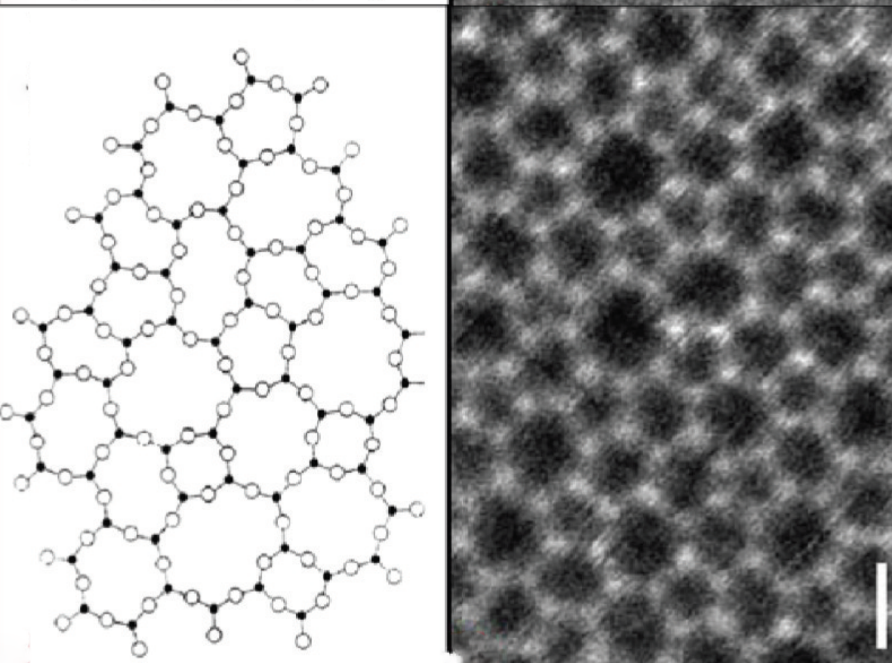
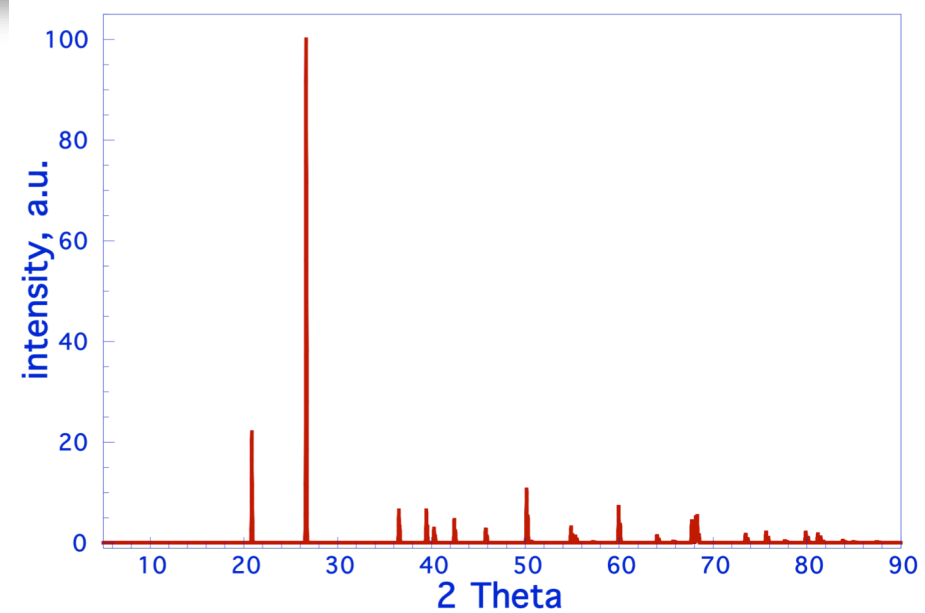
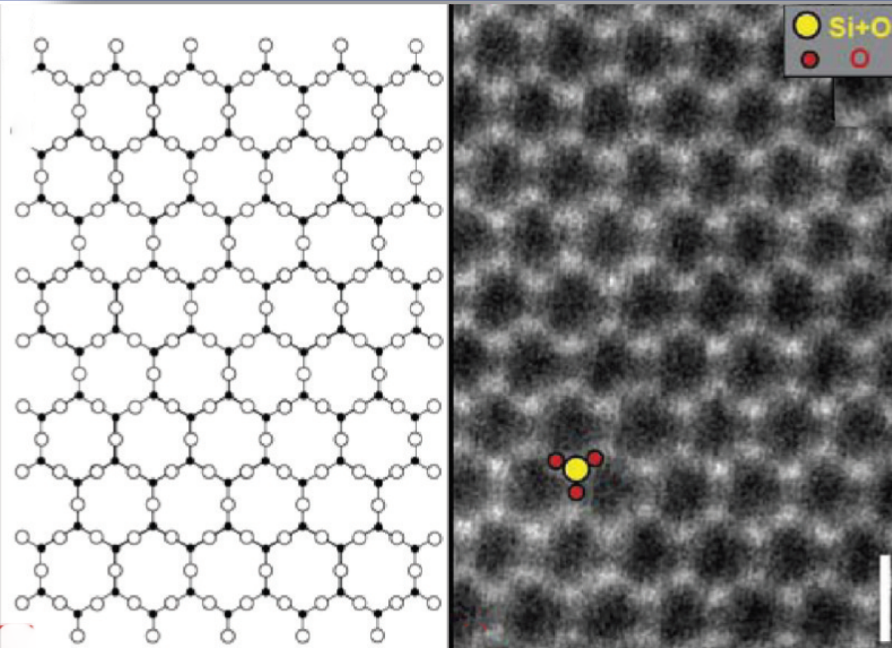
Zachariasen model (1932)

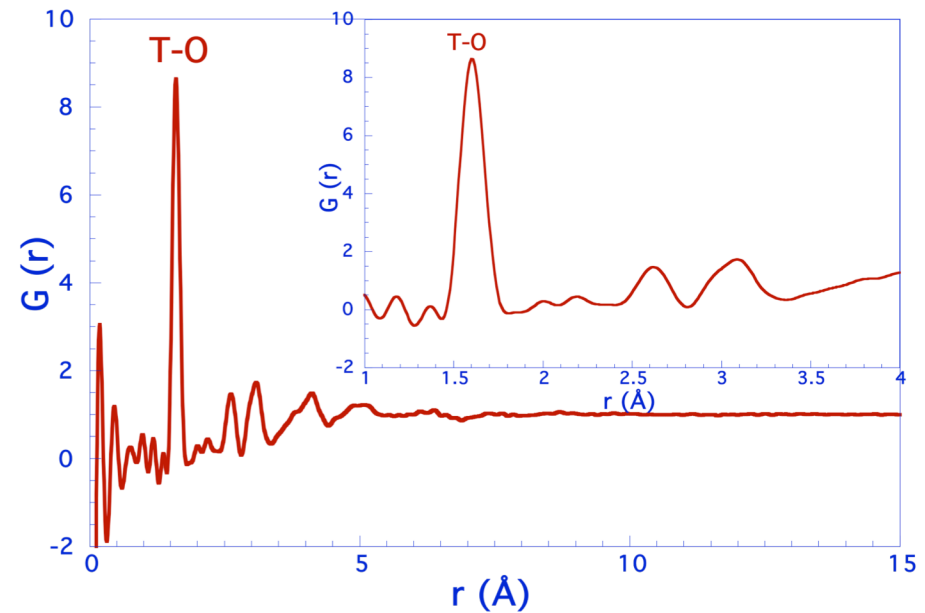
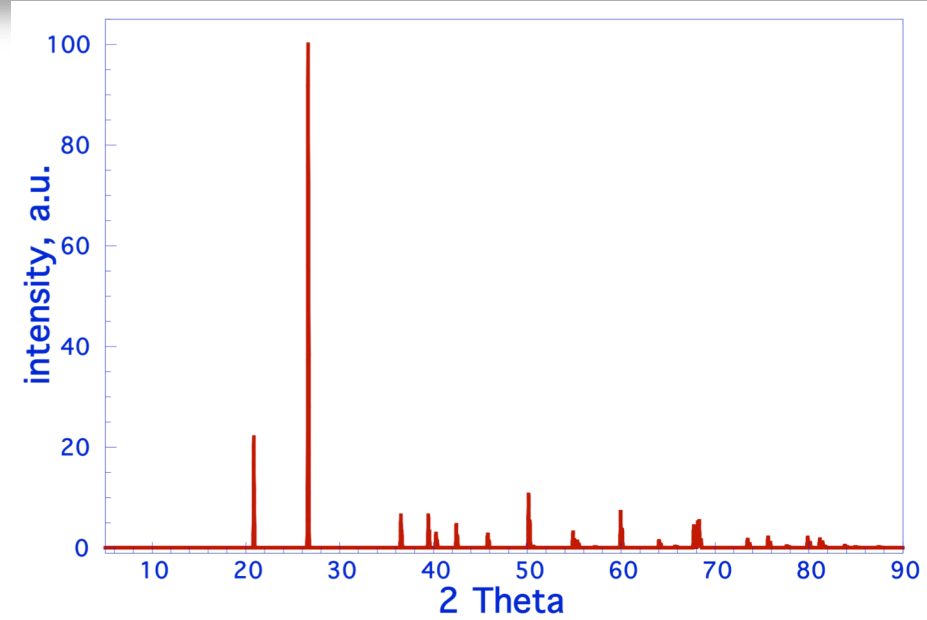
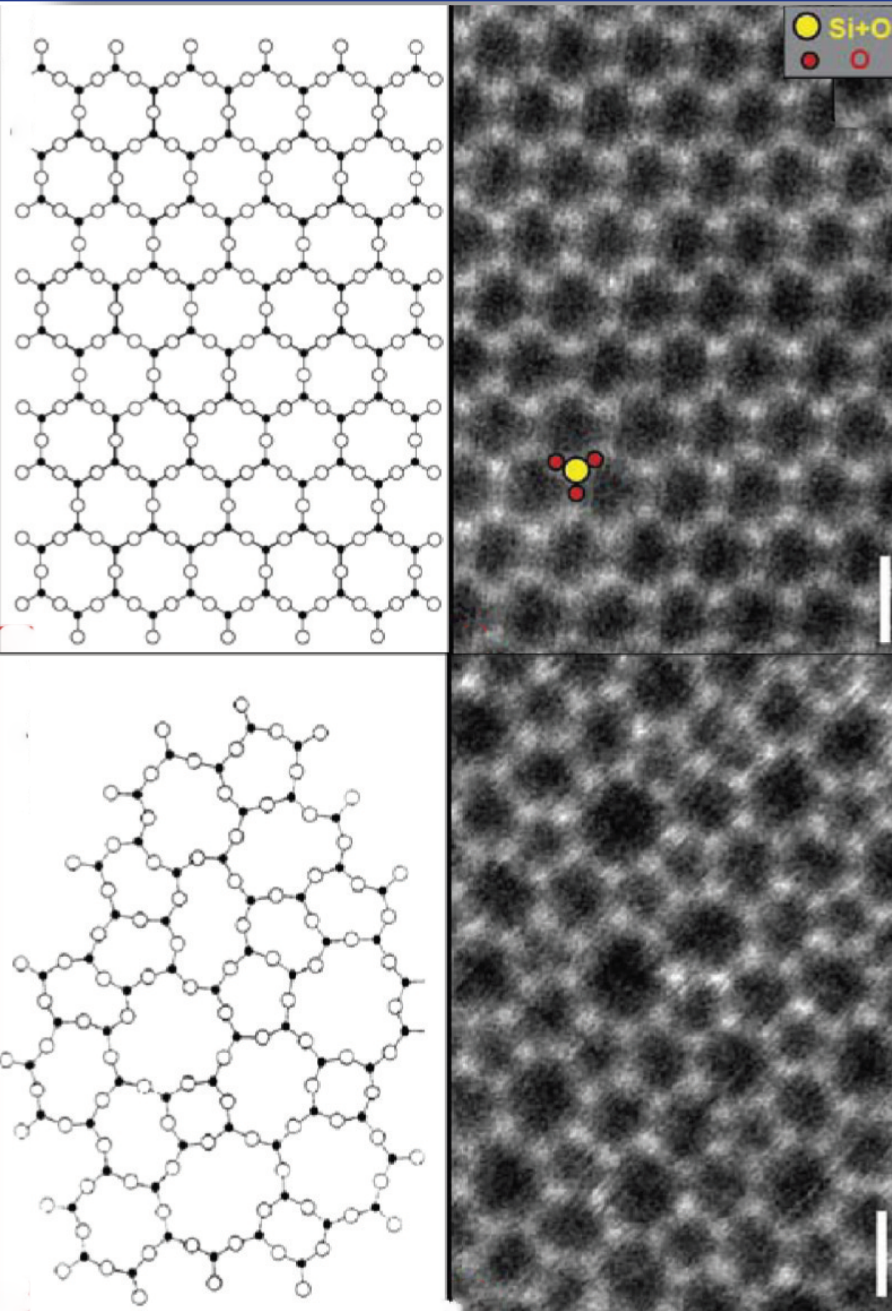
1. Each oxygen atom linked (bonded) to no more than two glass-forming cations (e.g. Si^{4+}).
2. Oxygen coordination number (CN) around glass-forming cation is small: 3 or 4.
3. Cation polyhedra share corners, not edges or faces.
4. The polyhedral structural units form a **3-D continuous random network** in which every polyhedron shares at least 3 corners with its neighbors.

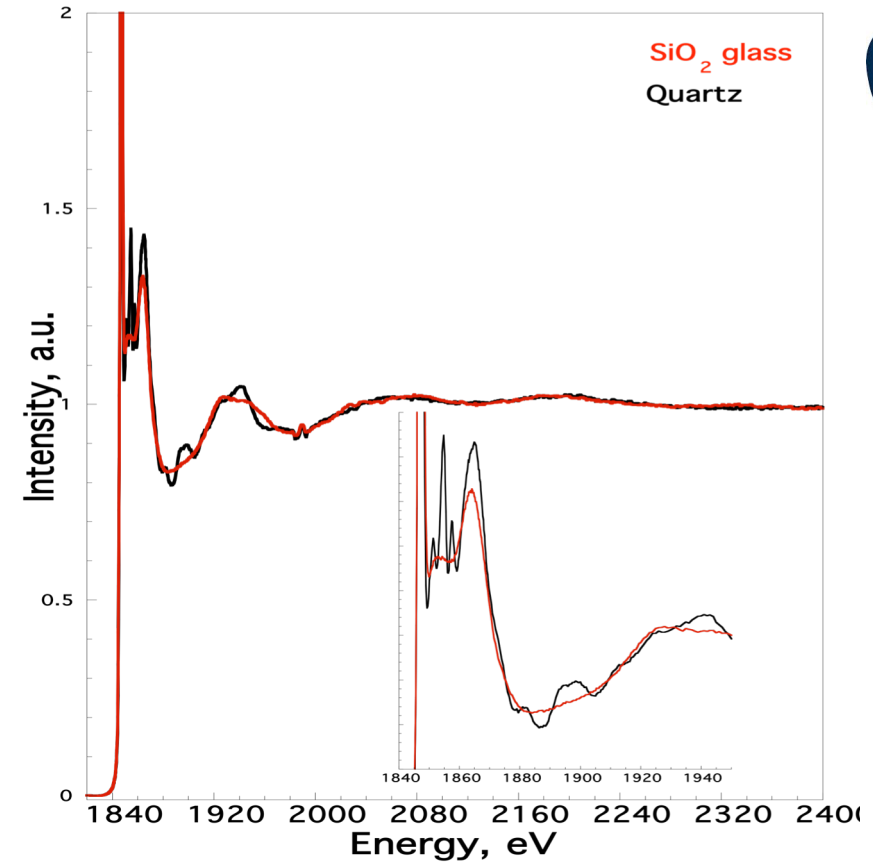
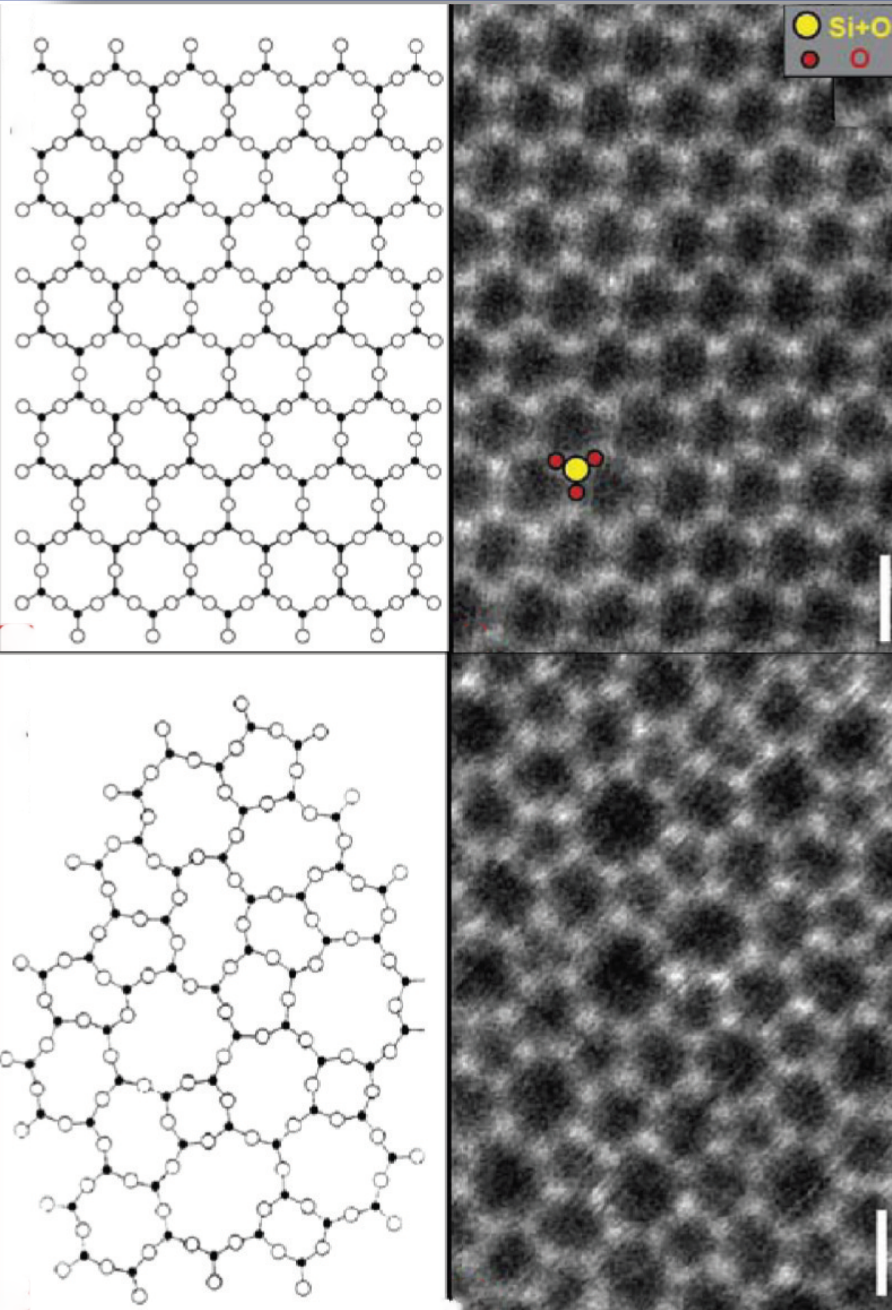




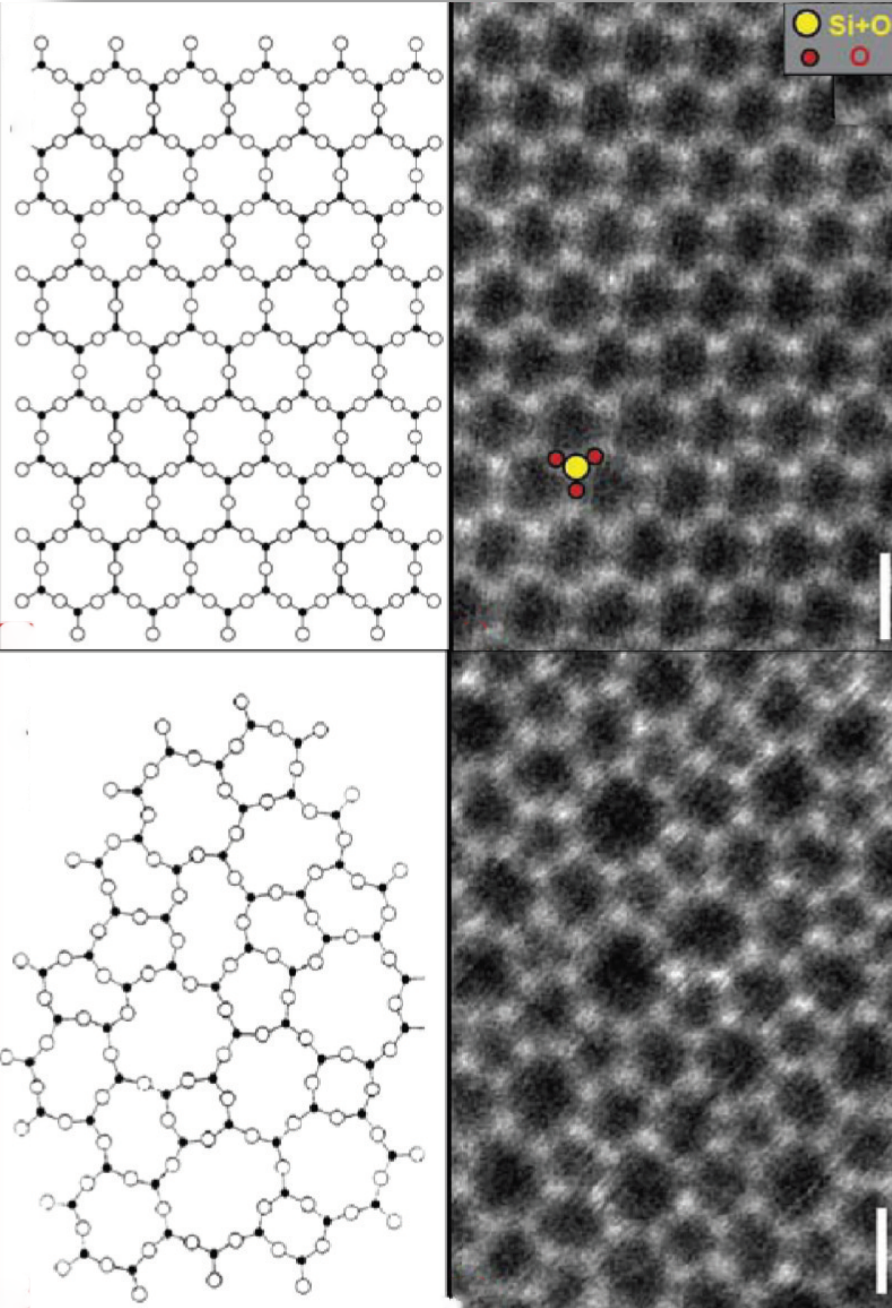
Warren, 1934





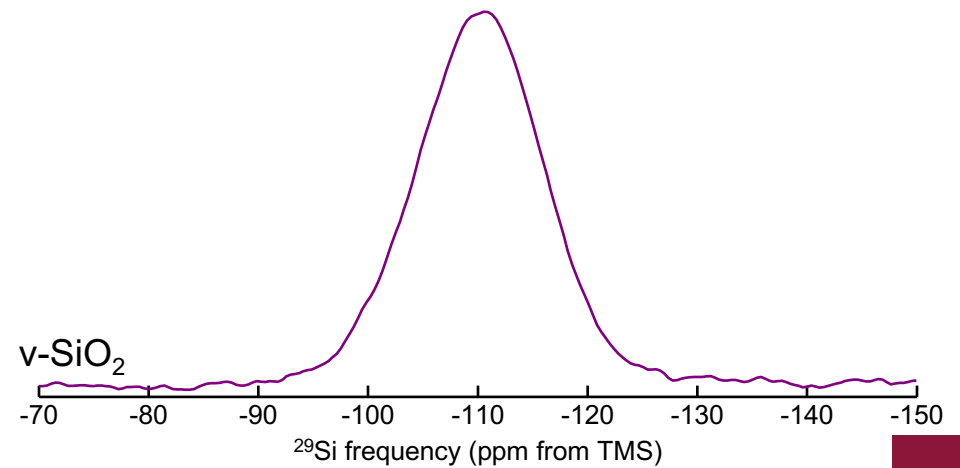


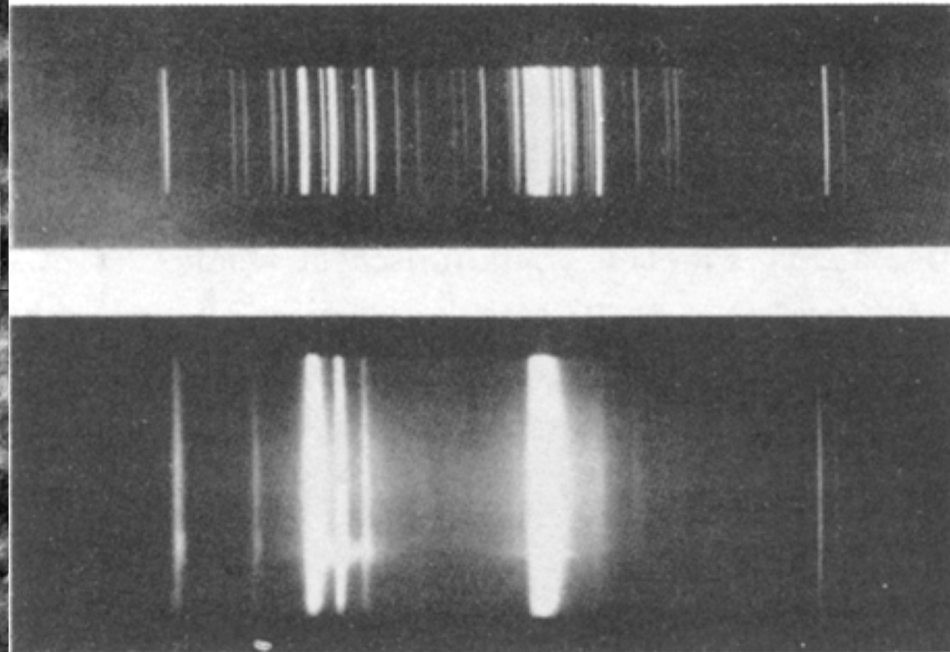
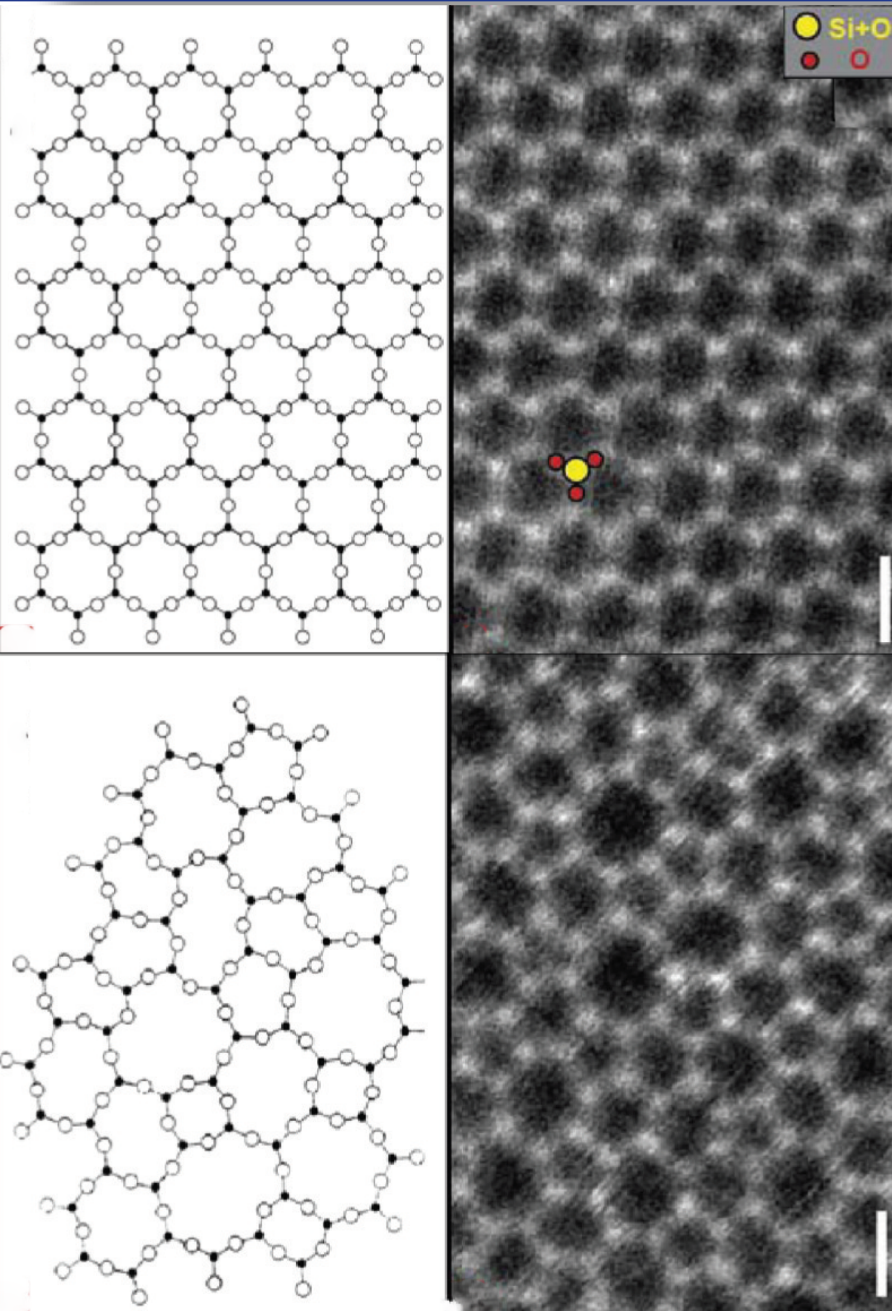
XANES at the Si K-edge



low-cristobalite

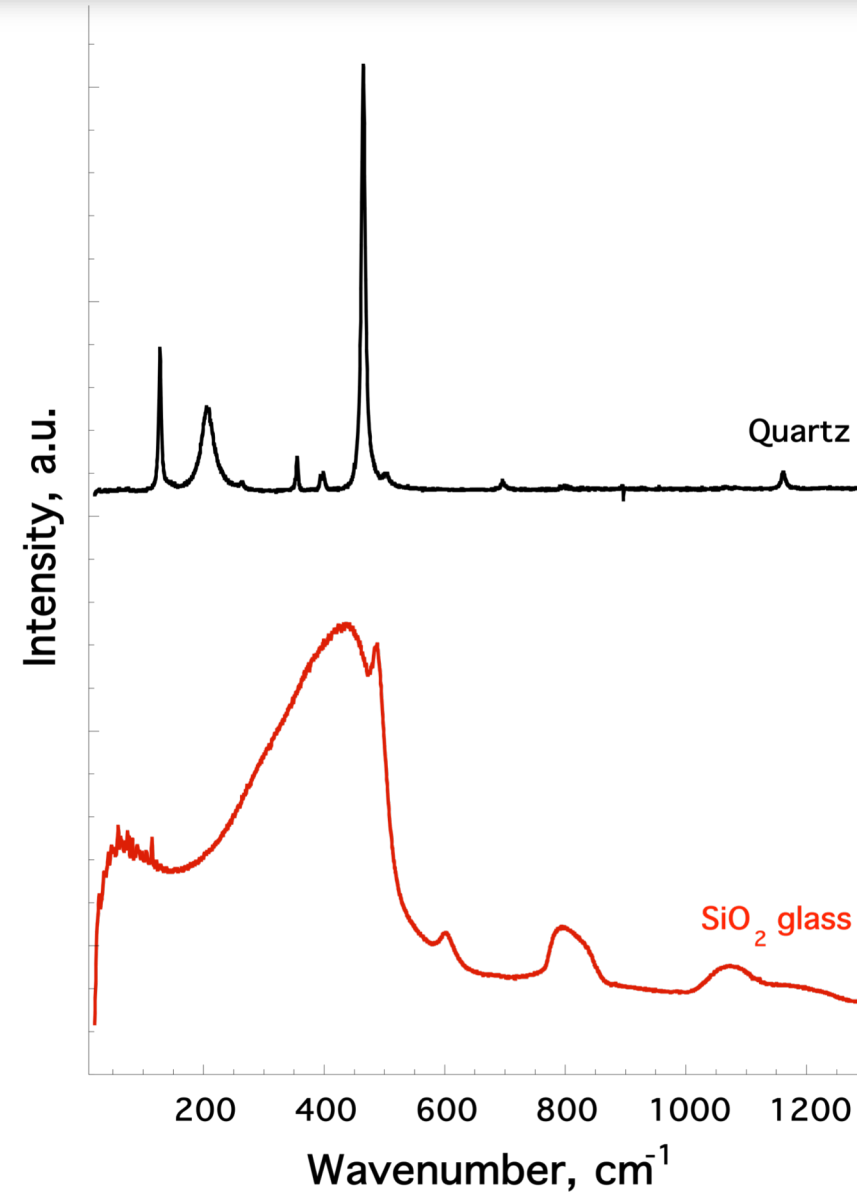
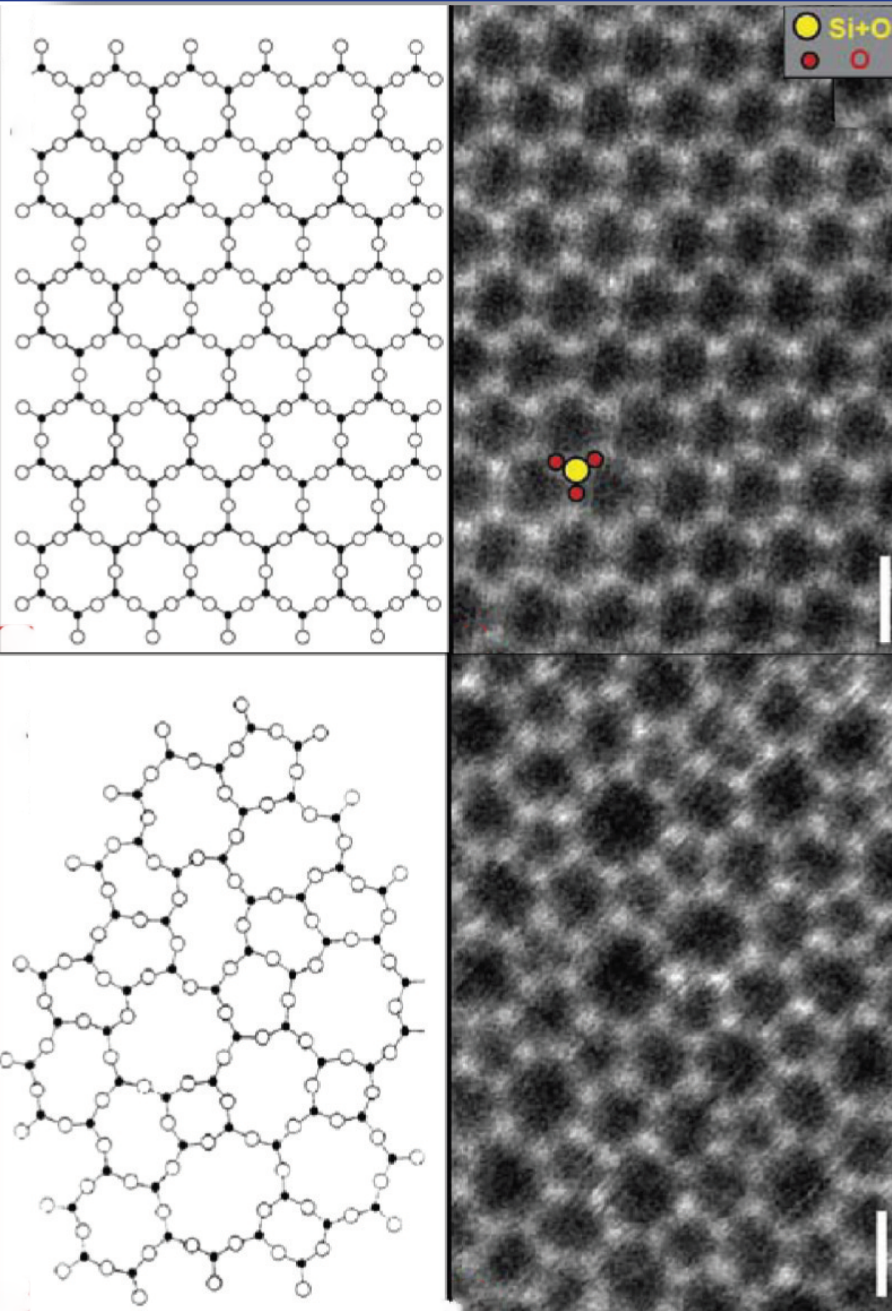
²⁹Si NMR

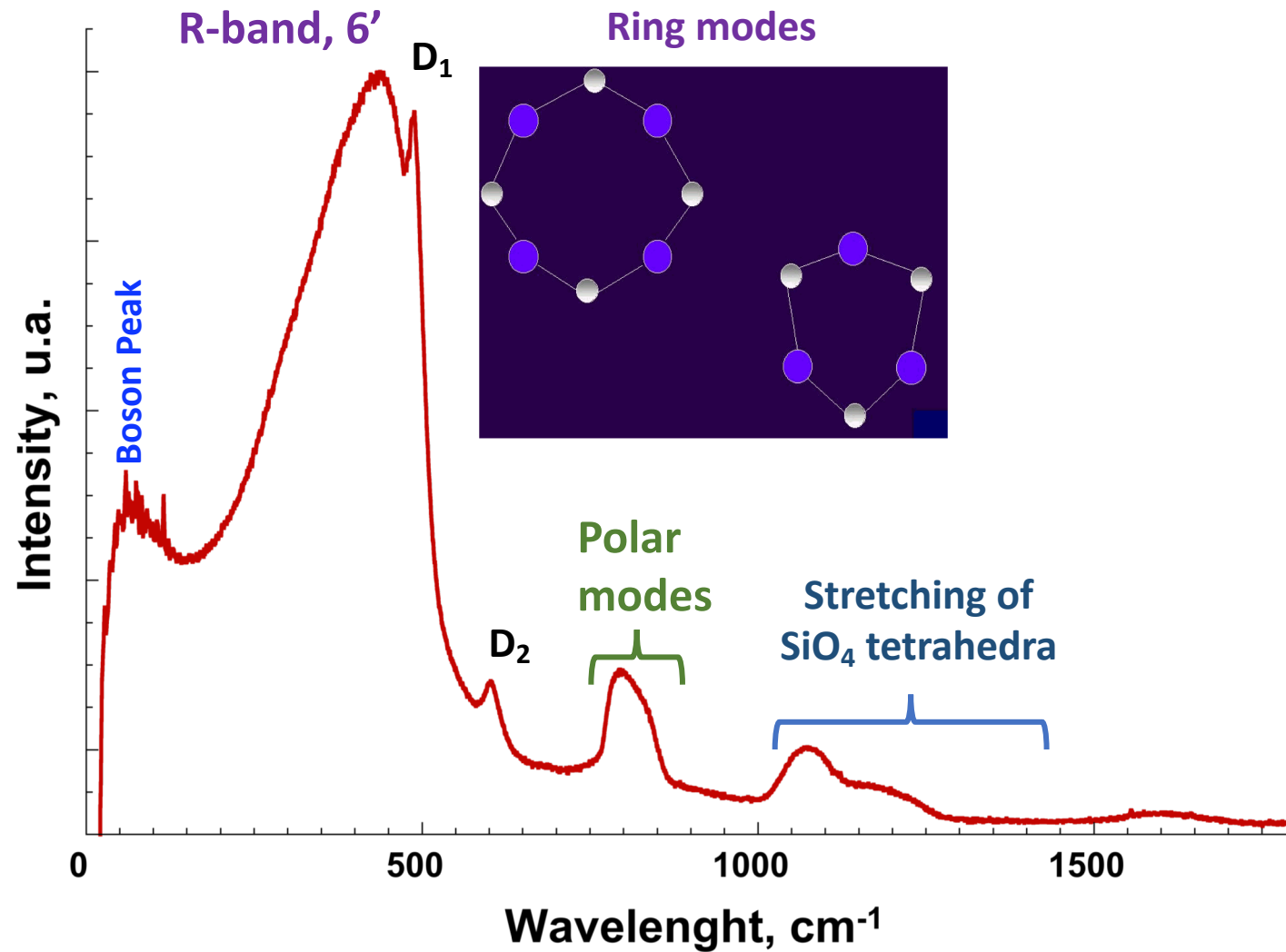


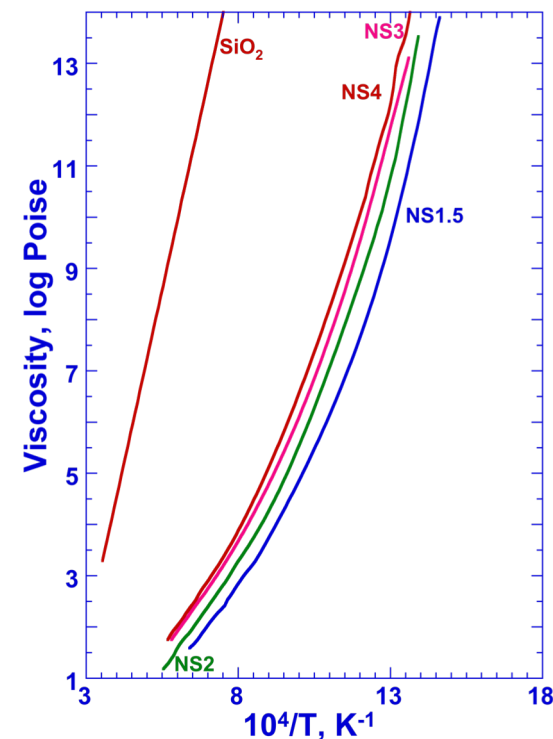
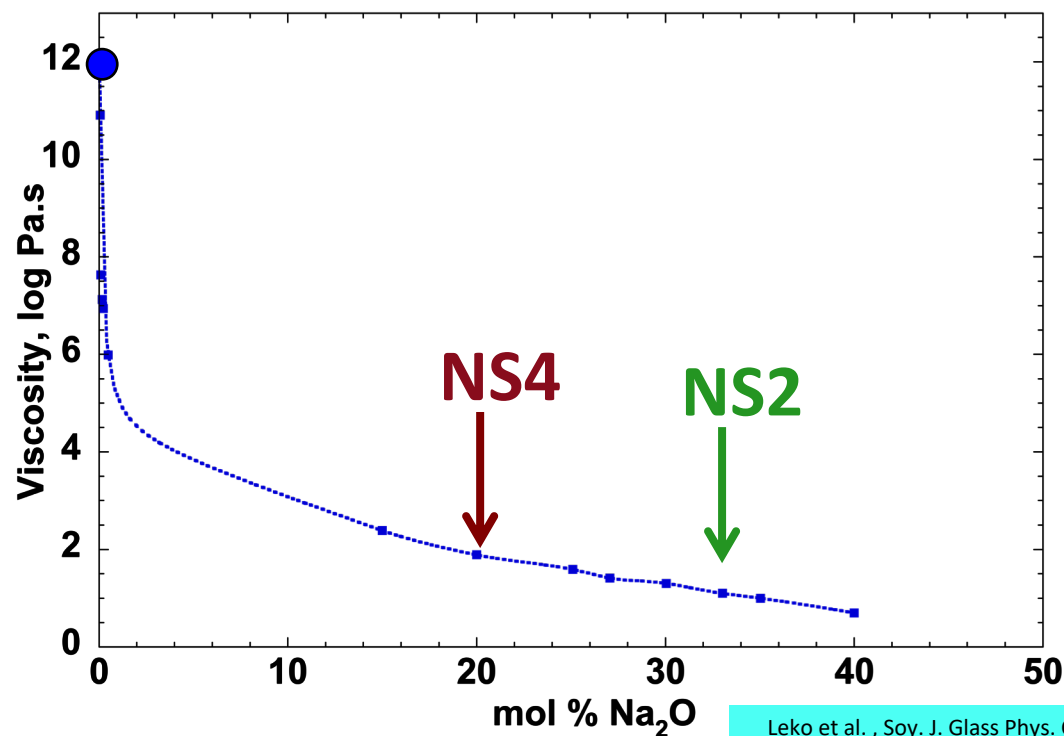
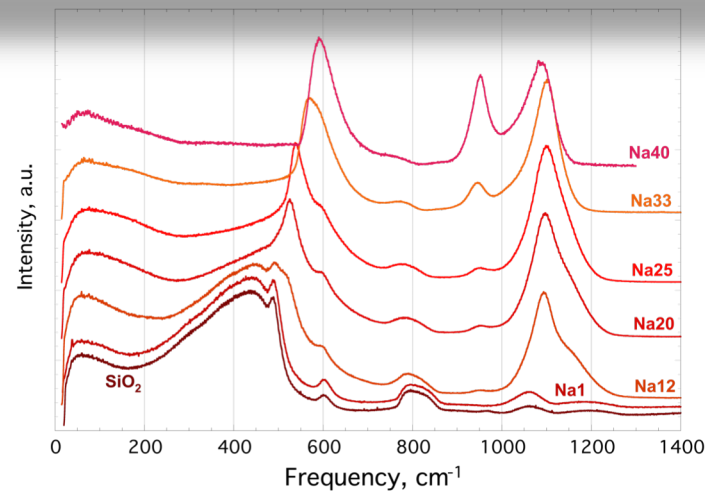
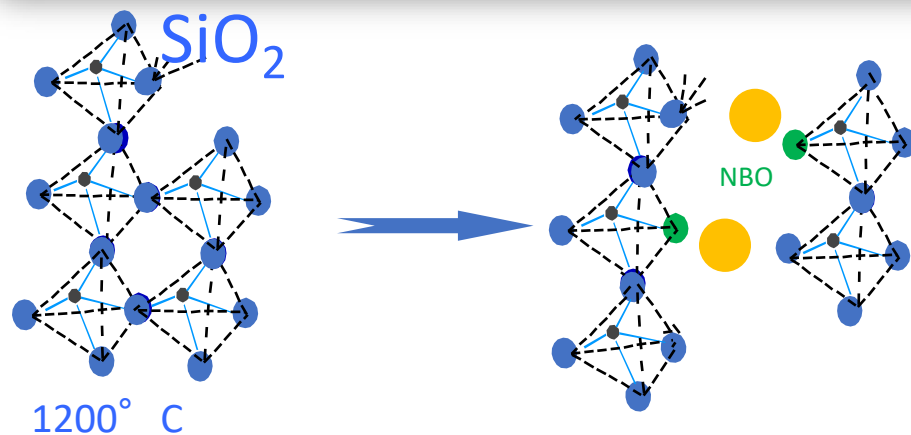


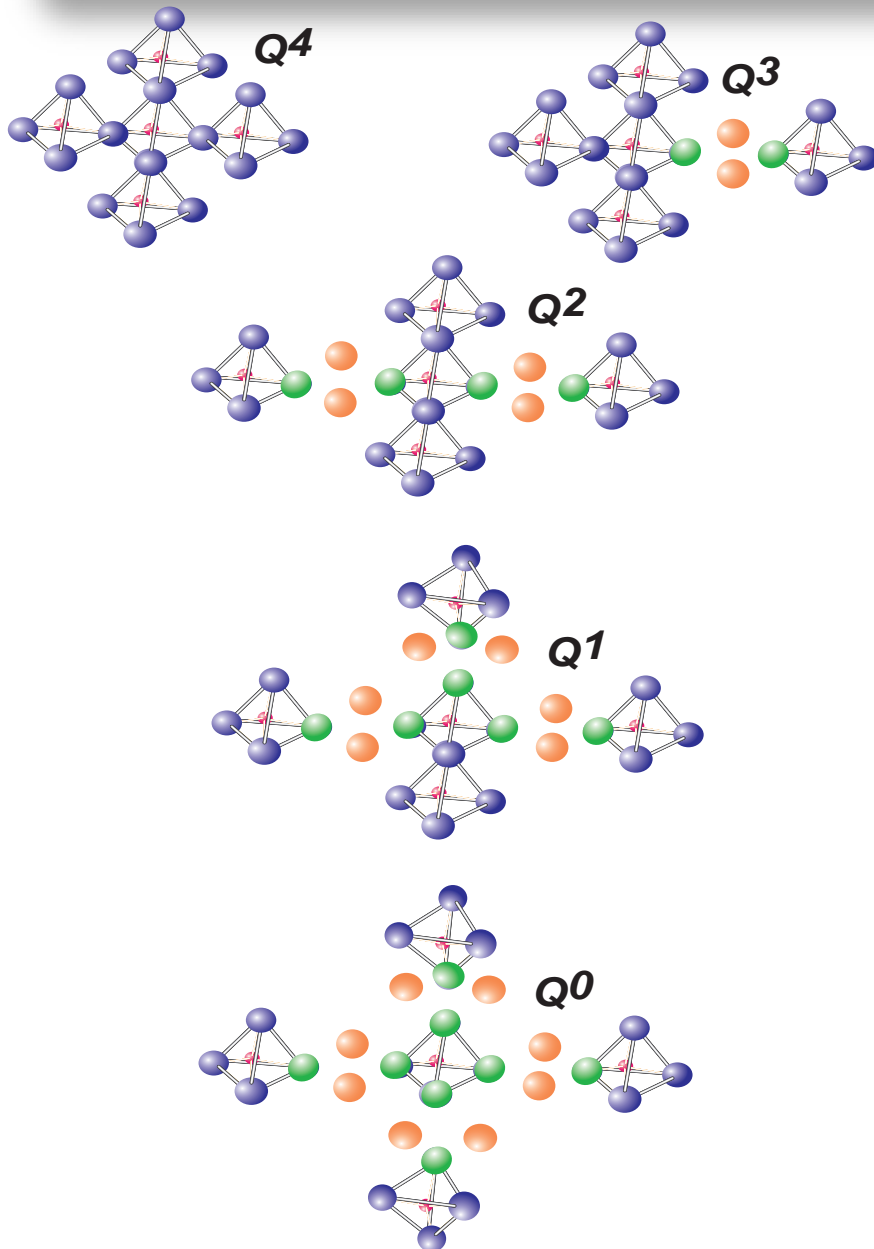
Kristallinischer
Quarz

Amorpher
Quarz







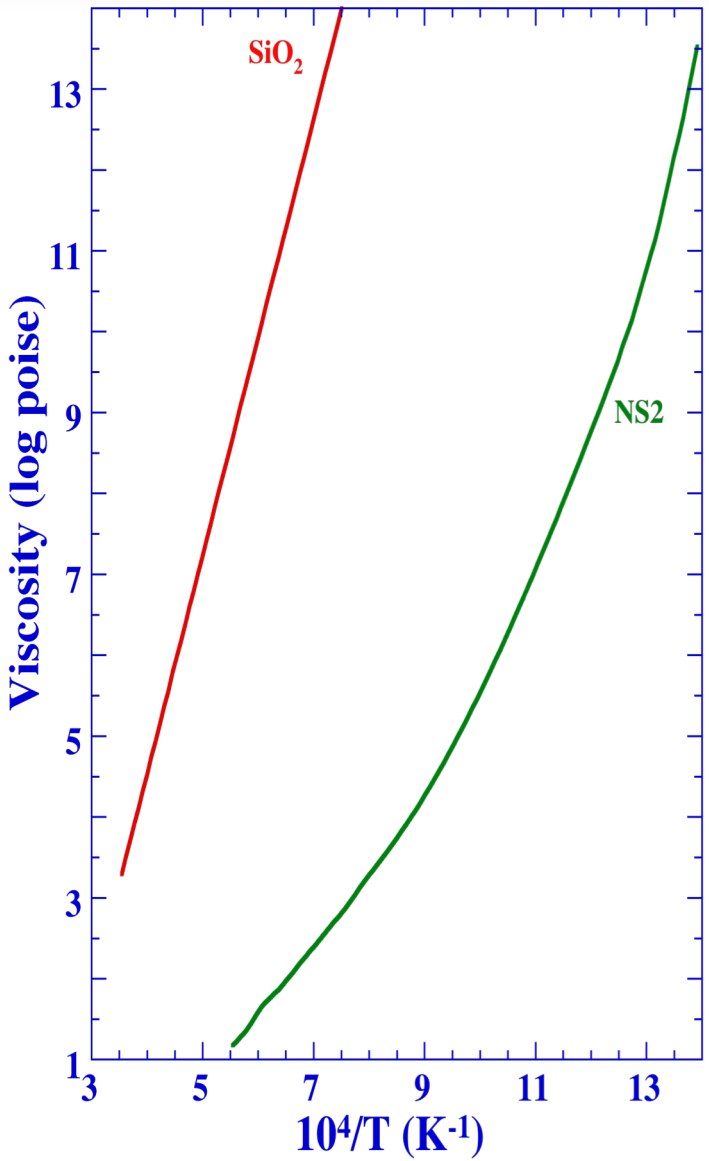
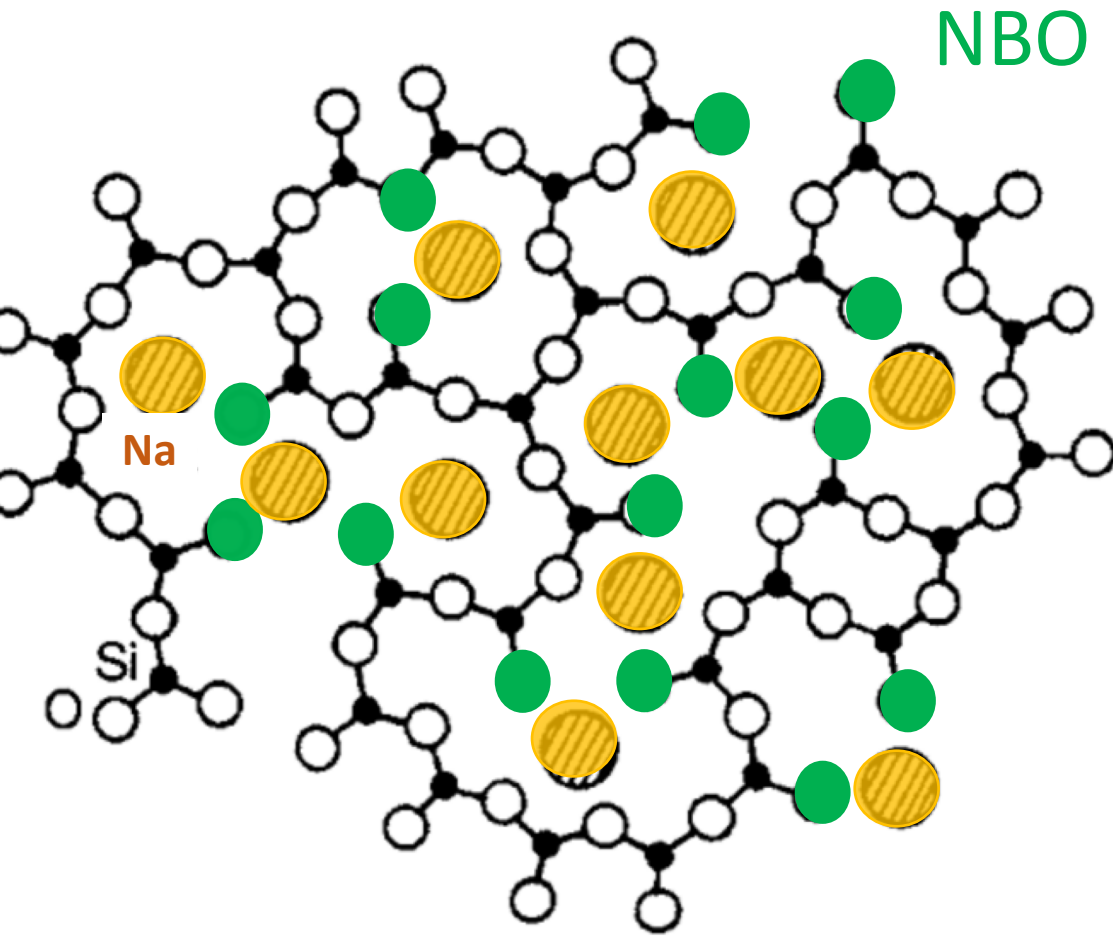


Non-network formers (alkali, alkaline-earth, transition elements) decrease the network connectivity, T_g , η by forming **non-bridging oxygens (NBO)** ($\neq$ bridging oxygens BO)

=>Network modifier

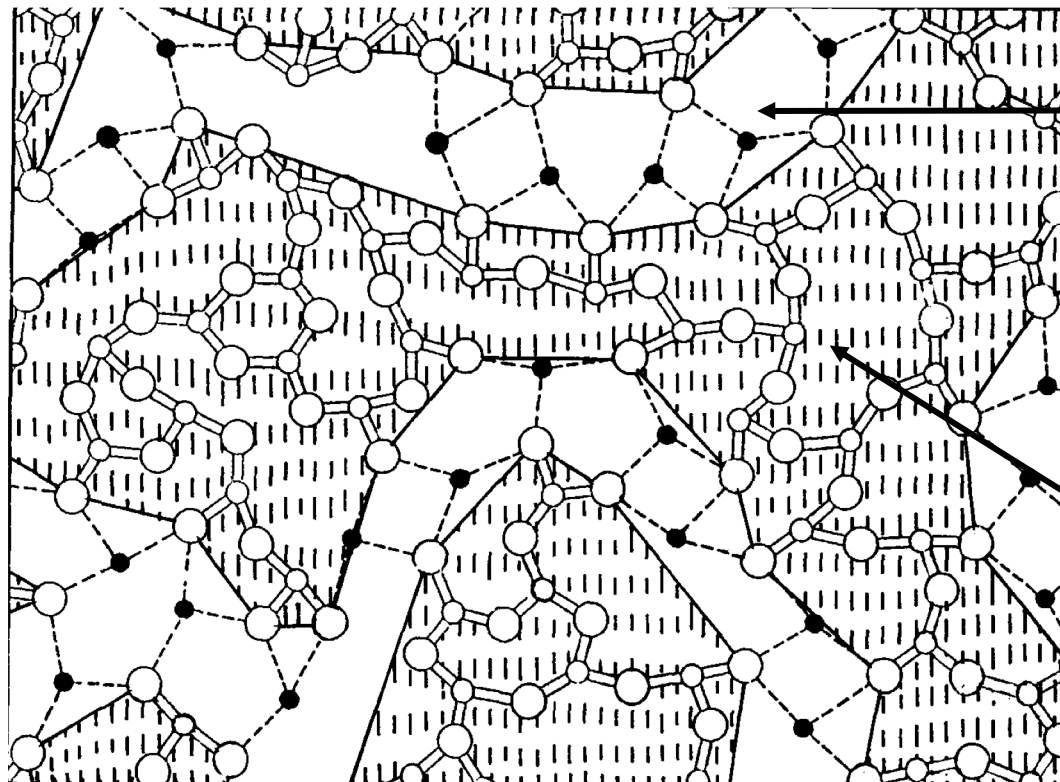
Qⁿ species $n =$
number of bridging
oxygens by tetrahedra

Zachariasen's rules do not consider at all modified oxides or multicomponent systems, or even non-oxide glasses



Zachariasen–Warren network theory

Modified random network - MRN (Greaves, 1985)



Zones rich with network
modifiers

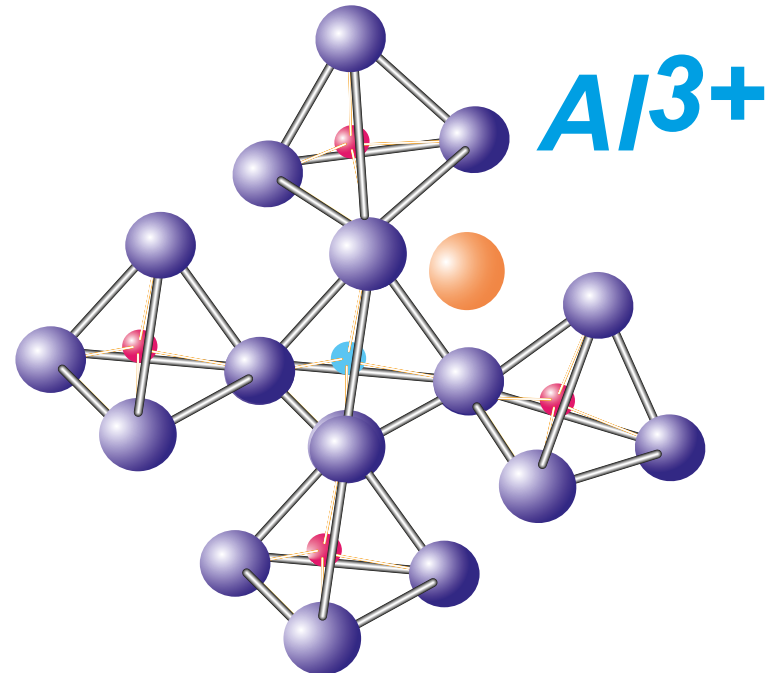
Zones rich with
network
formers

Relationships with conductivity, alteration etc

Al substitute to Si in tetrahedral position

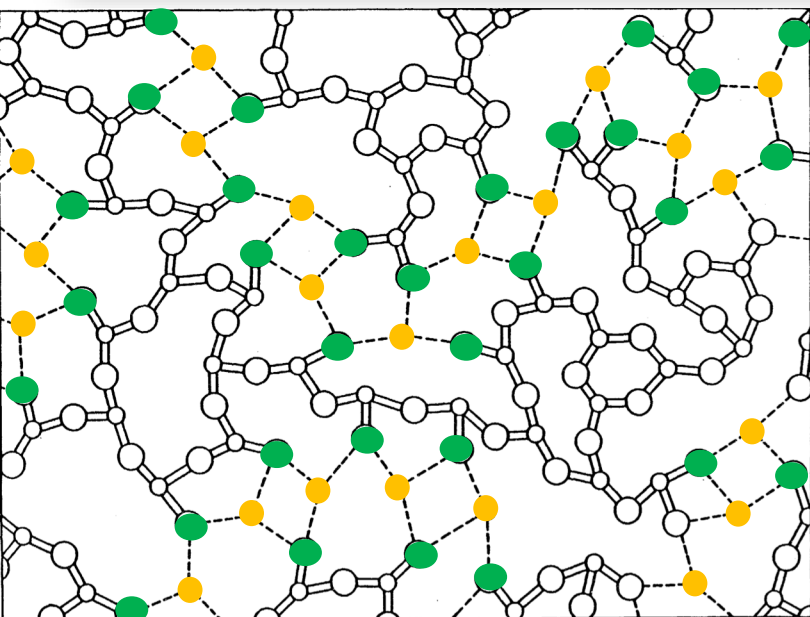
Al : (Ne)3s²3p¹ : 3 valence electrons => ions Al³⁺

(AlO₄)⁻ charge
electroneutrality
ensured by the
presence of alkali or
alkaline earth

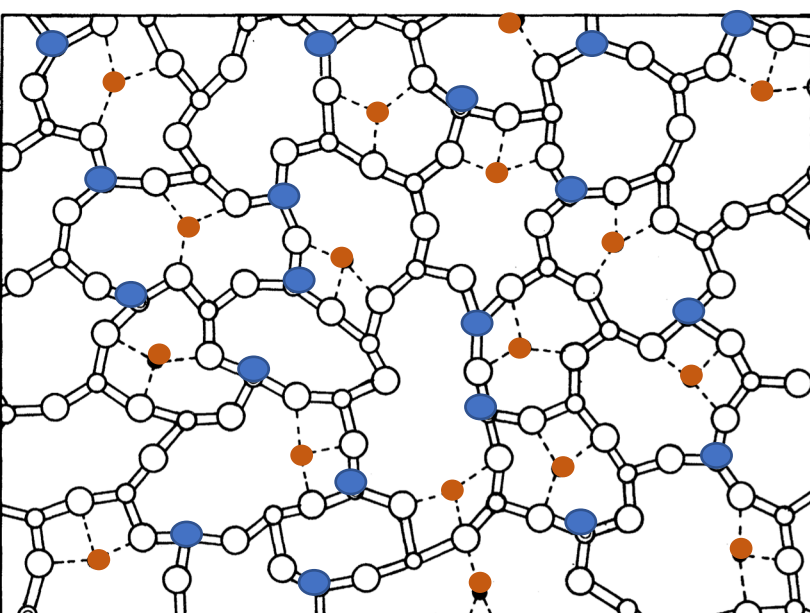


Similar for (BO₄)⁻

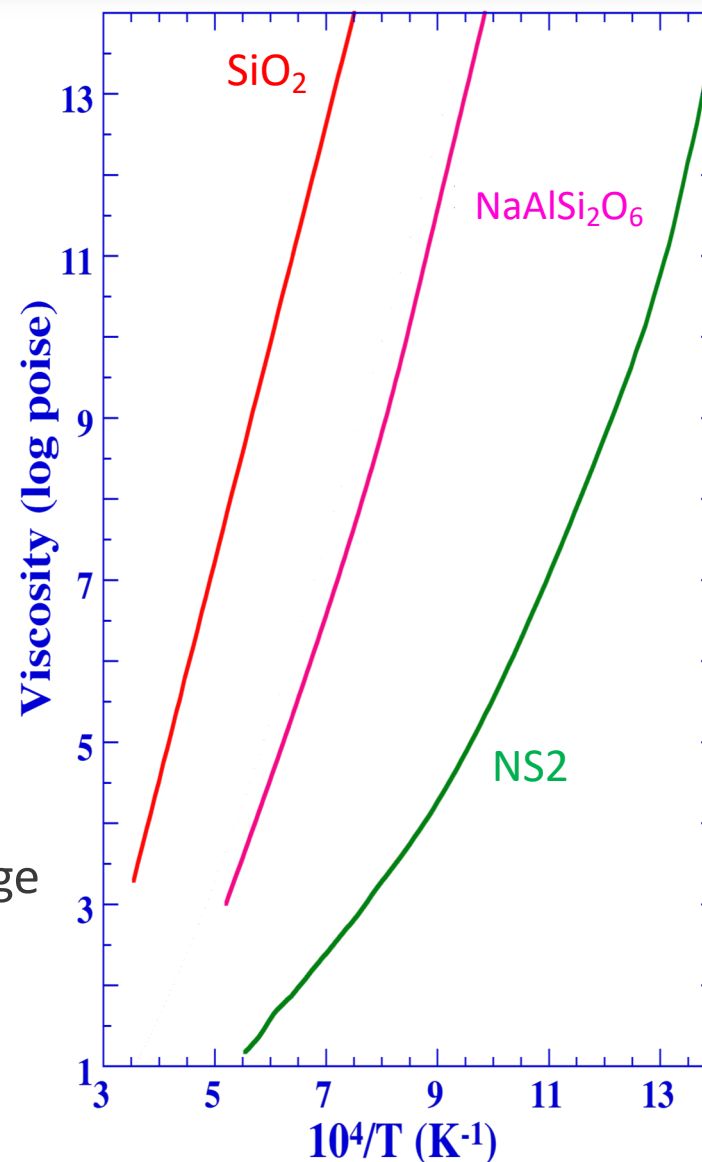
MRN - Modified random network versus CCRN Compensated Continuous Random Network

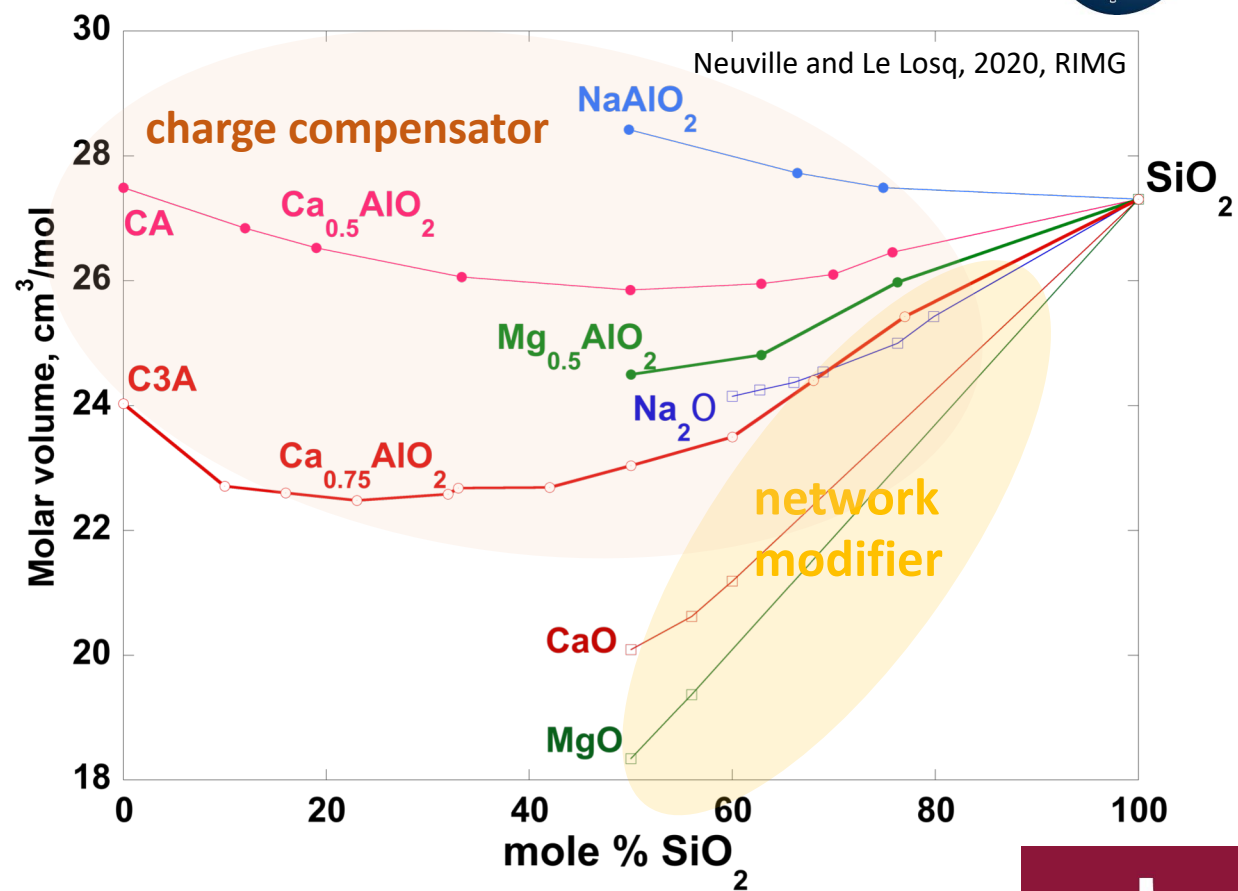
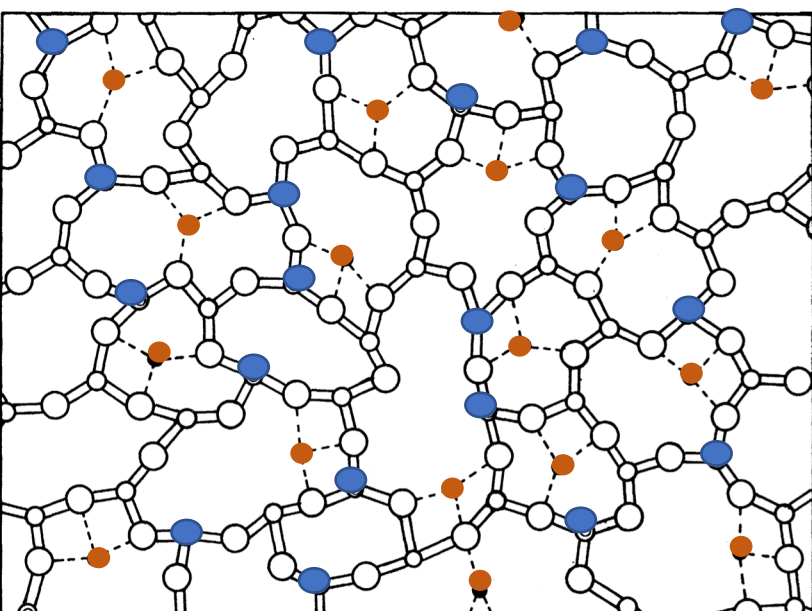
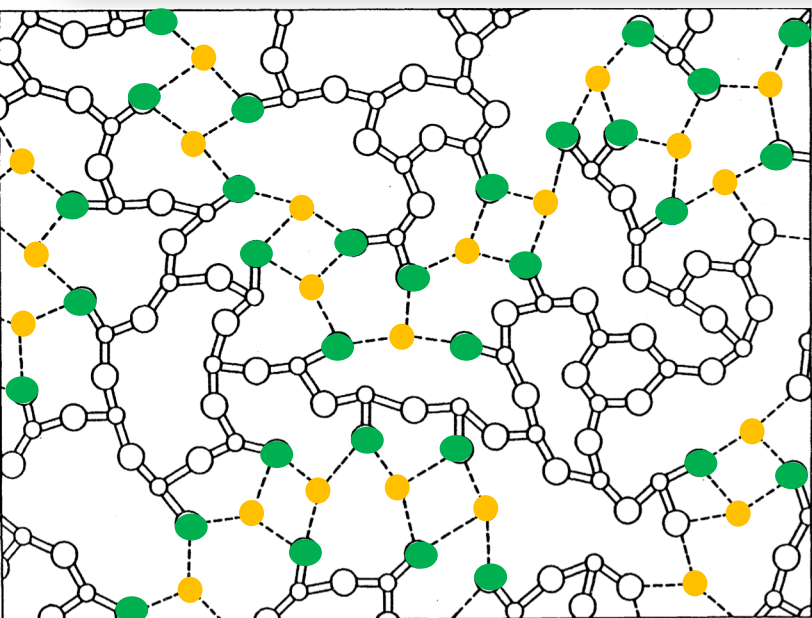


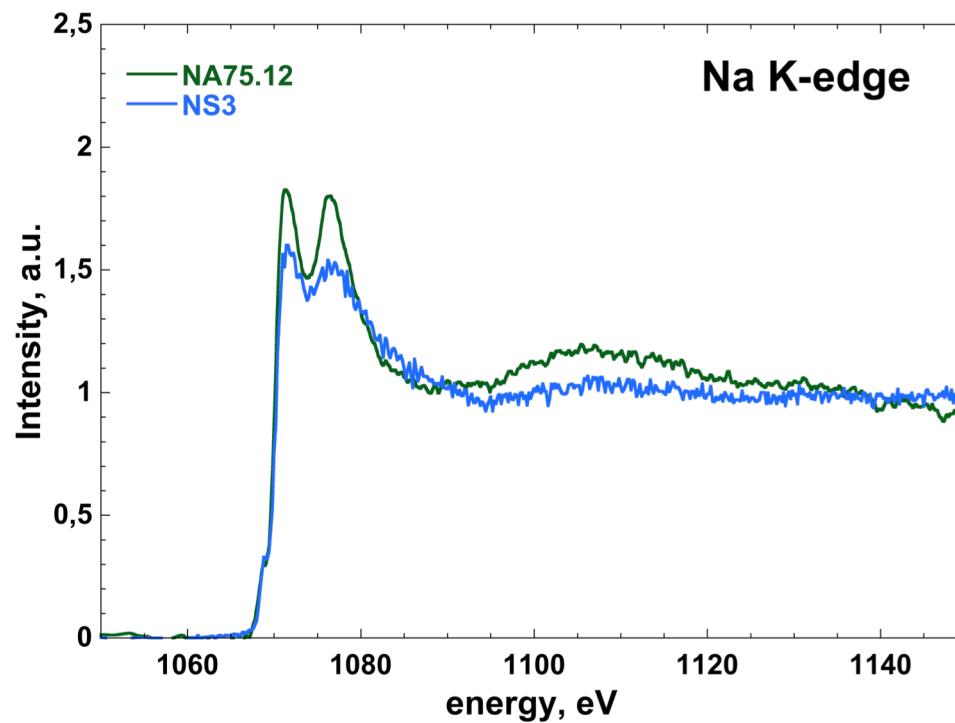
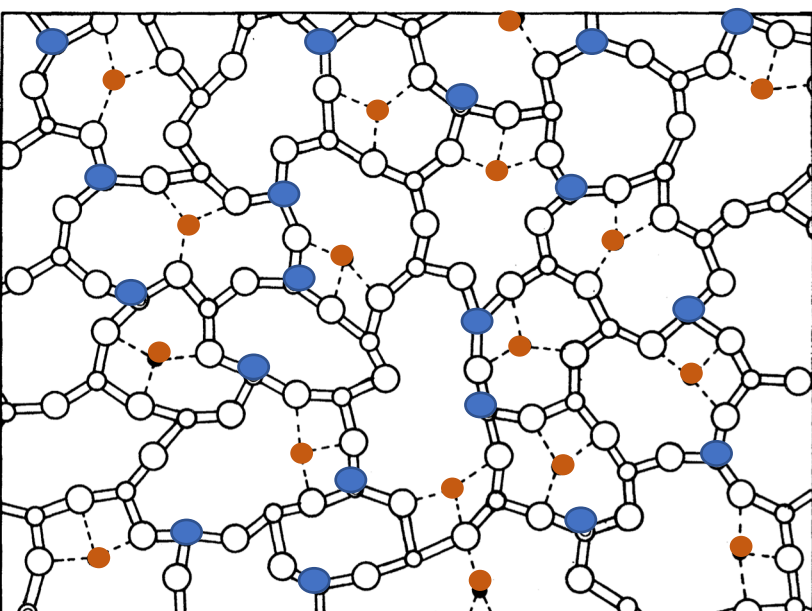
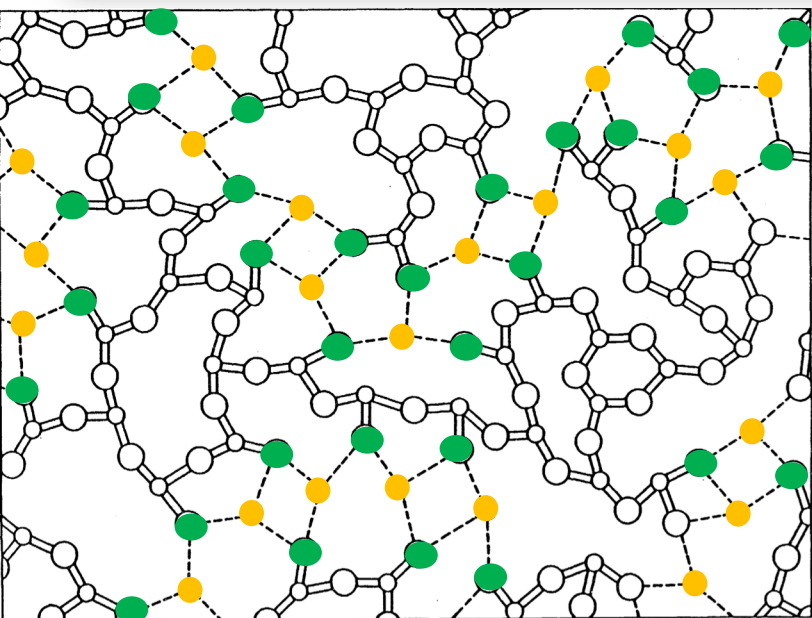
Cations connected to **NBO** associated to the negative charge of O^- => **network modifier**



Cations connected to **BO** and acting as charge compensator near $(AlO_4)^-$, $(BO_4)^-$ => **charge compensator**

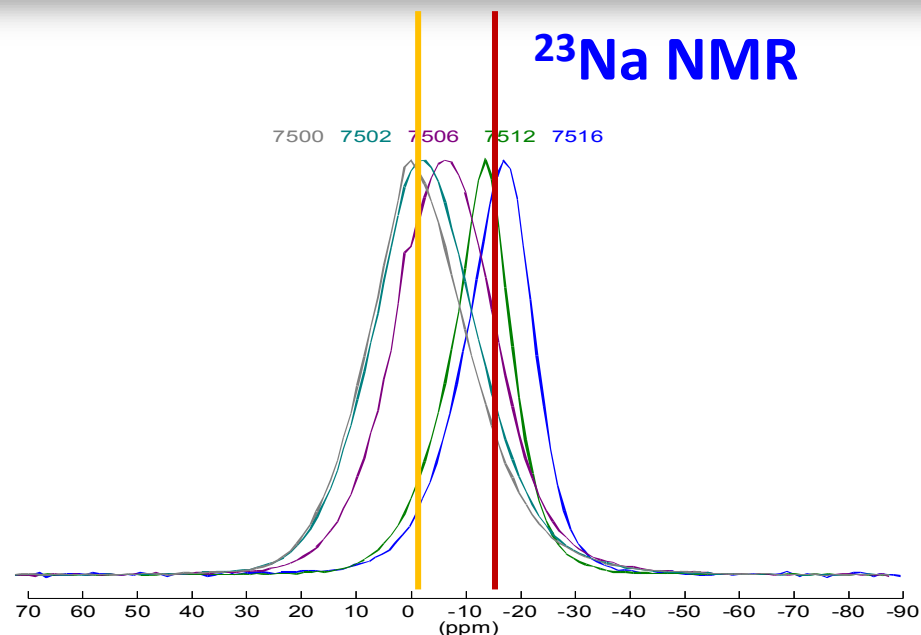
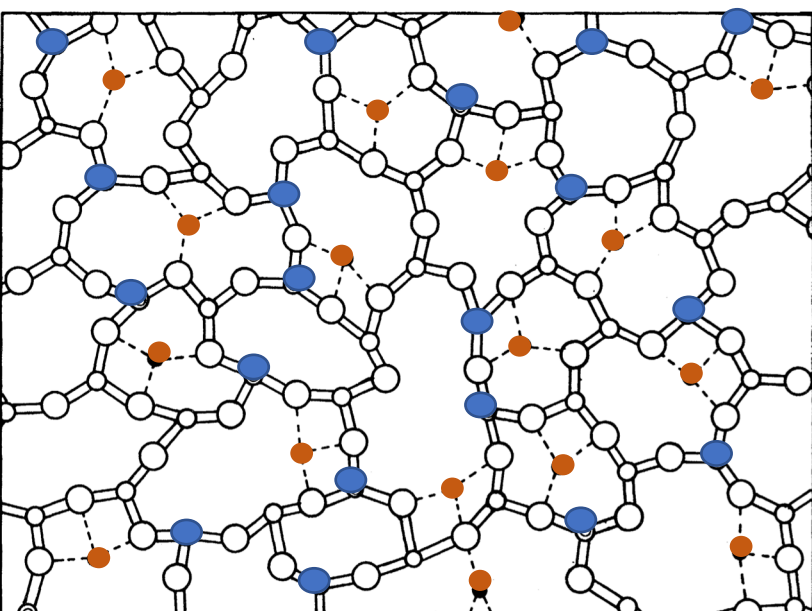
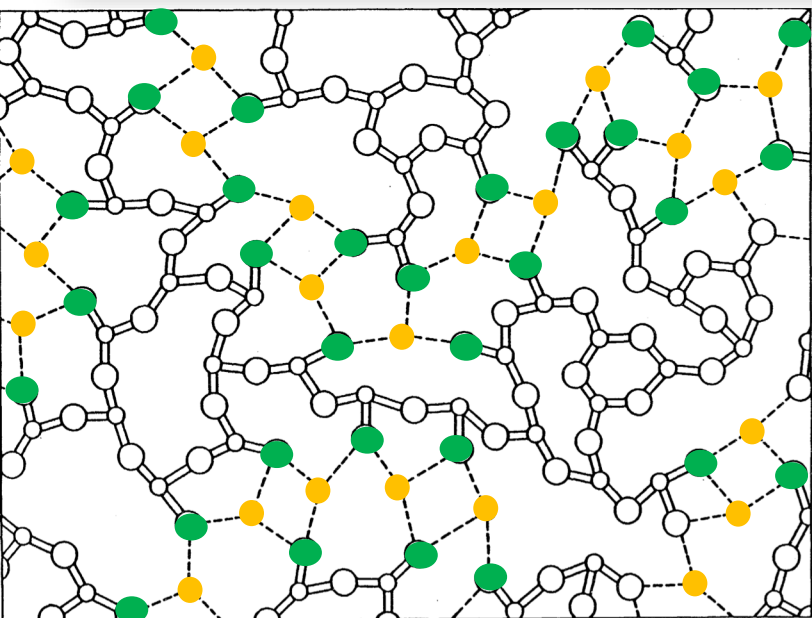






NS3 => $^{[6]}\text{Na}$ network modifier

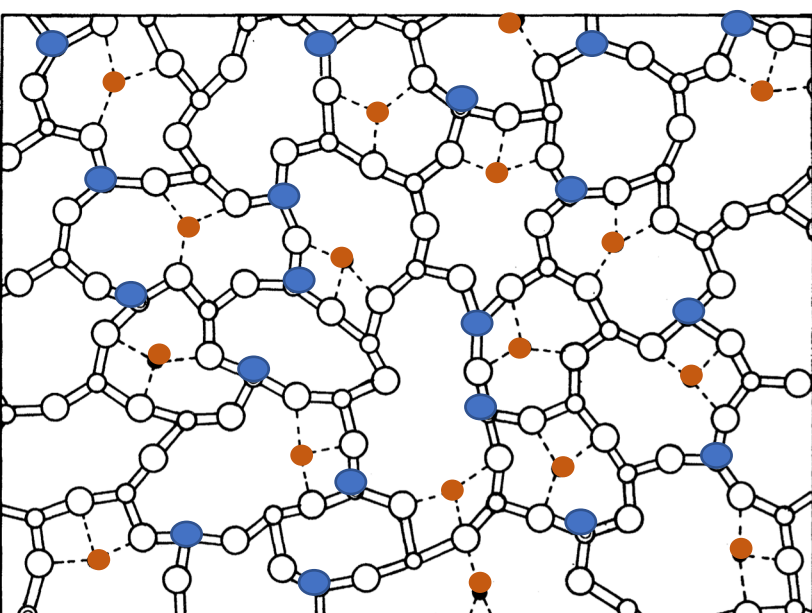
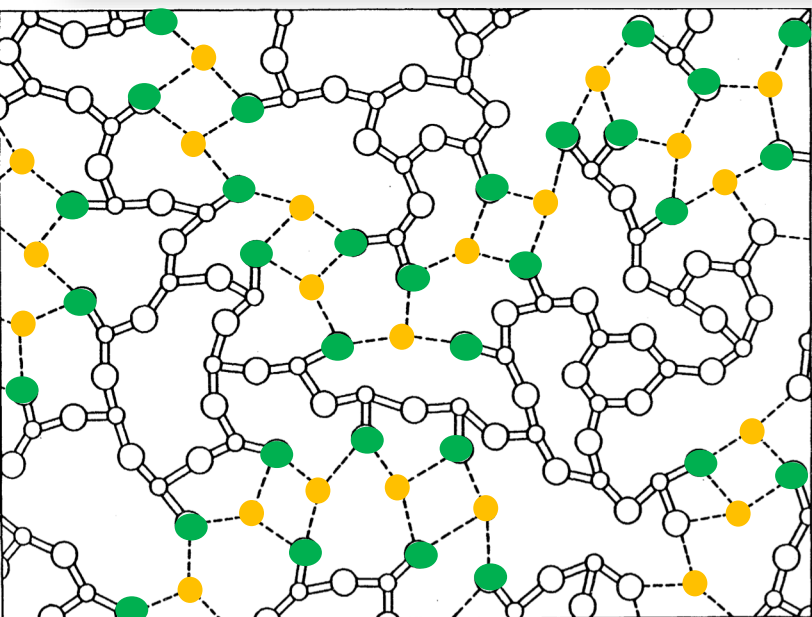
NA75.12 => $^{[9]}\text{Na}$ charge compensator



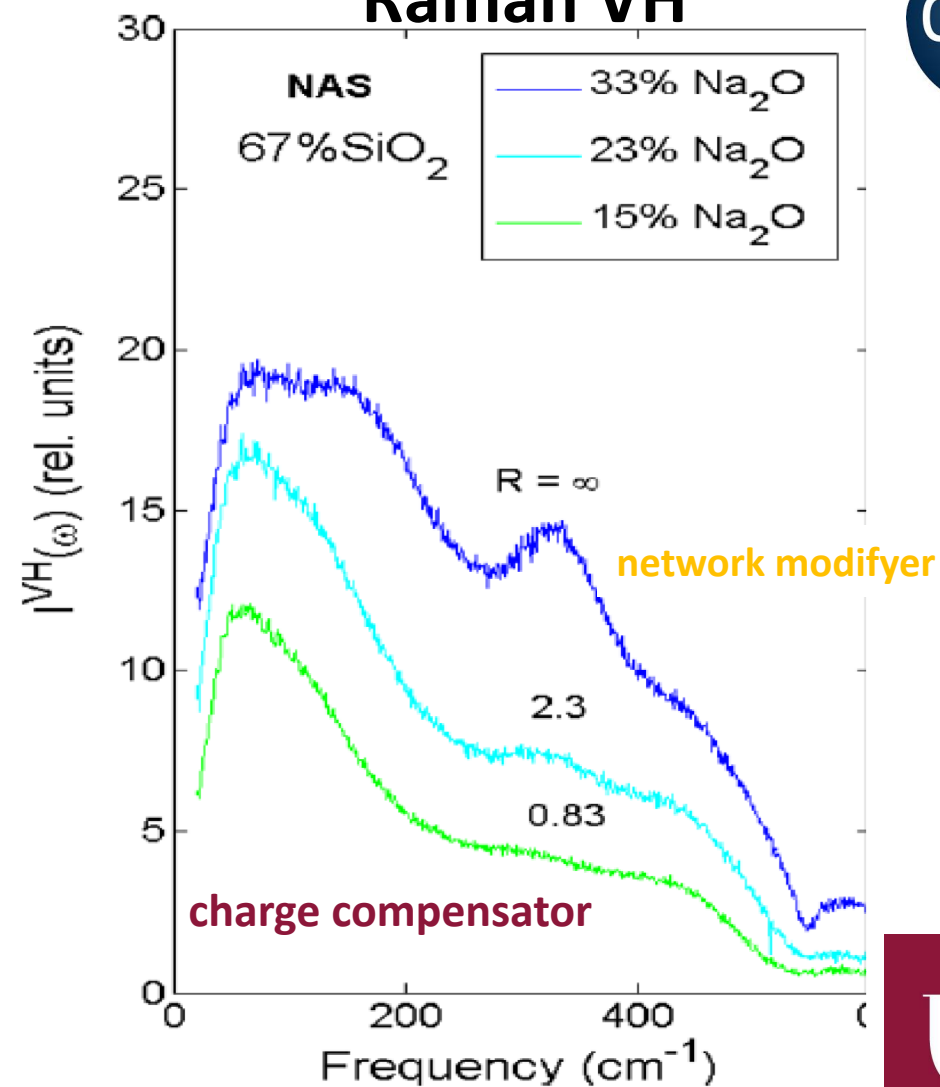
⇒ Chemical shift of ^{23}Na ,
from **network modifier**
to **charge compensator**

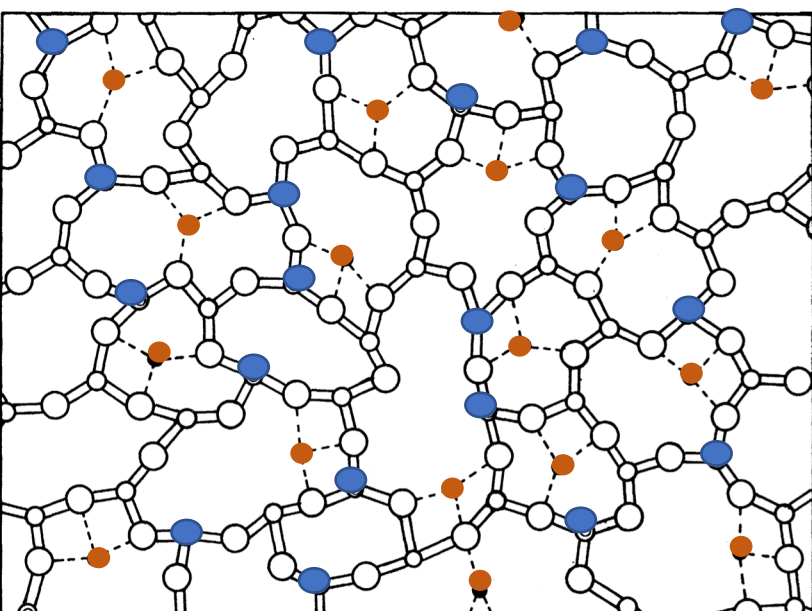
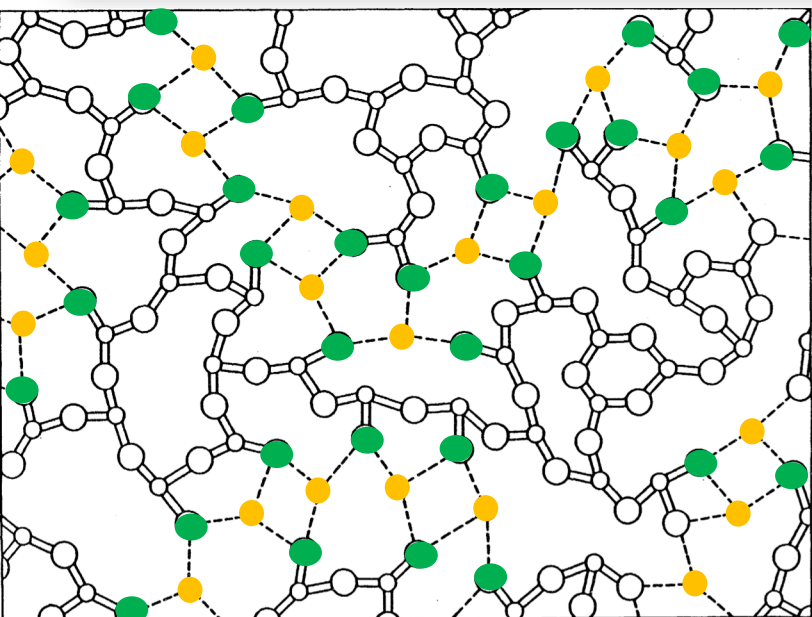
Le Losq Ch., Neuville D.R., Florian P., G.S. Henderson and Massiot D.
(2014) Role of Al^{3+} on rheology and nano-structural changes of sodium
silicate and aluminosilicate glasses and melts. *Geochimica Cosmochimica
Acta*, 126, 495-517.

Hehlen B. and Neuville D.R. (2015) Raman response of network modifier cations in aluminosilicate glasses. The Journal of Physical Chemistry B. 119, 4093–4098.

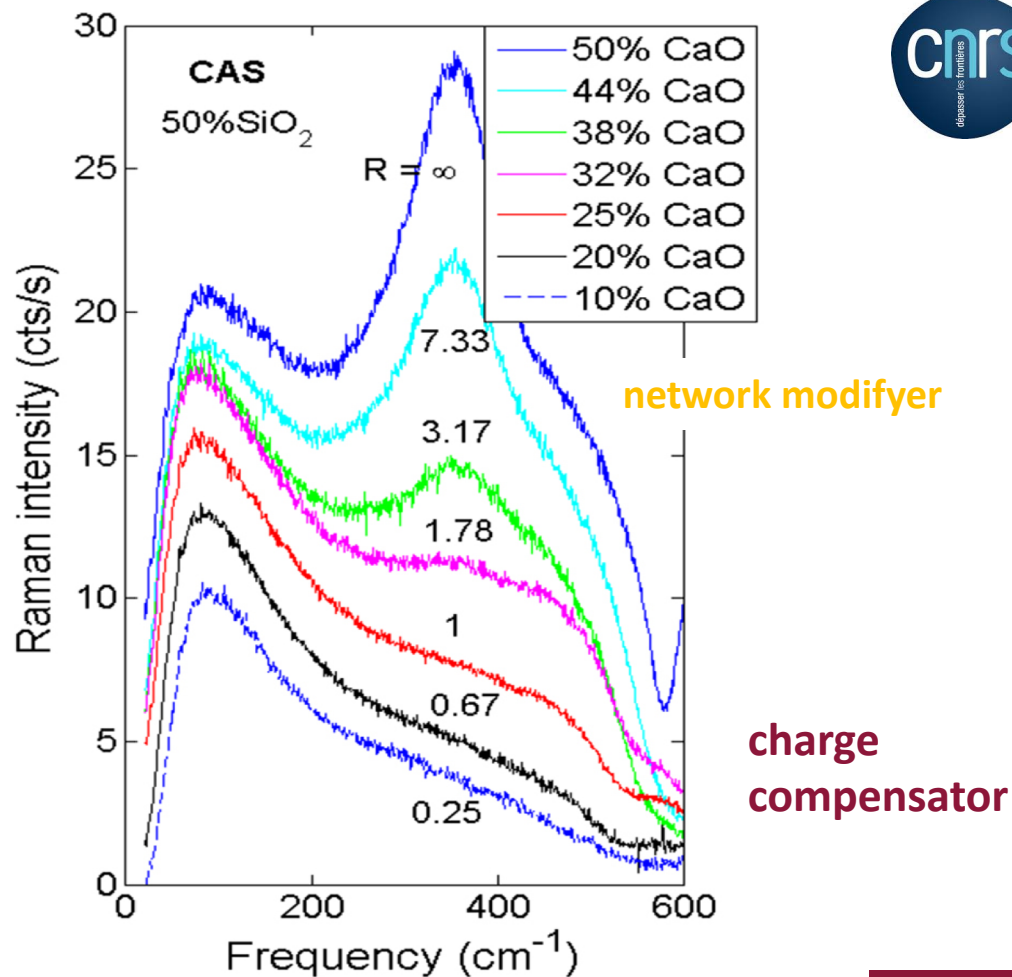


Raman VH

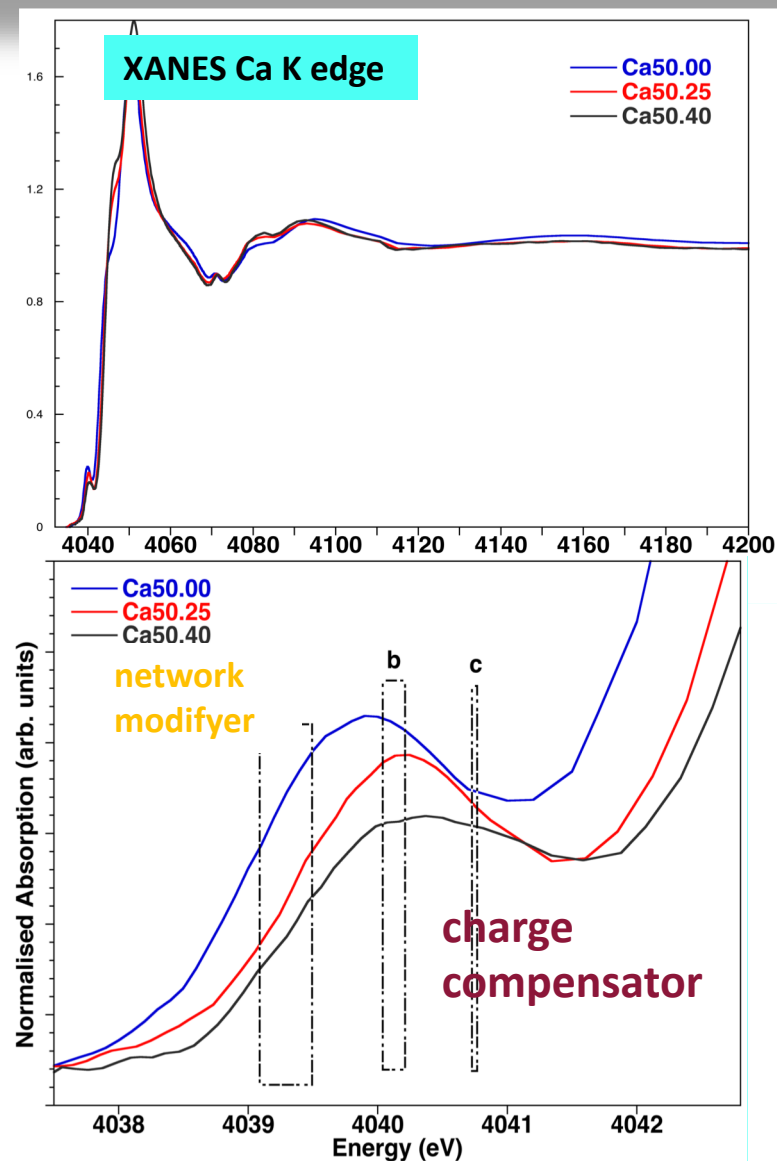
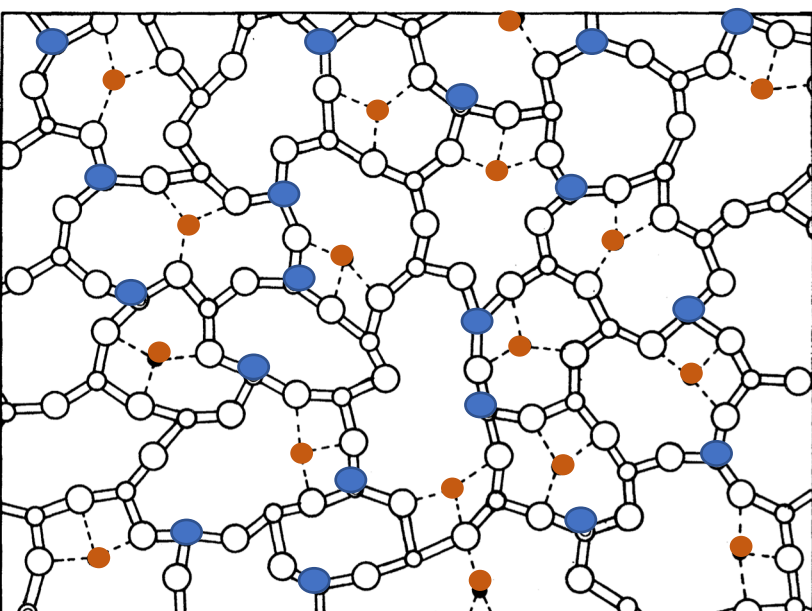
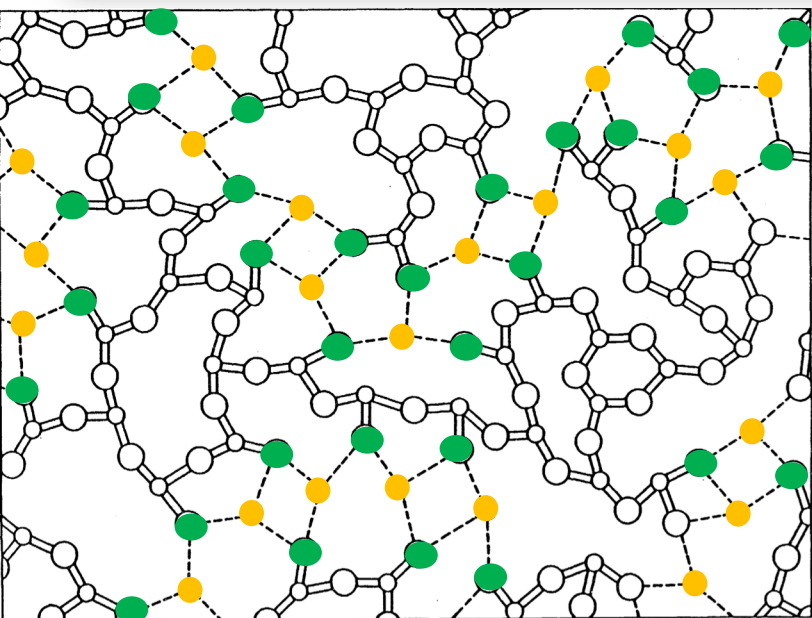




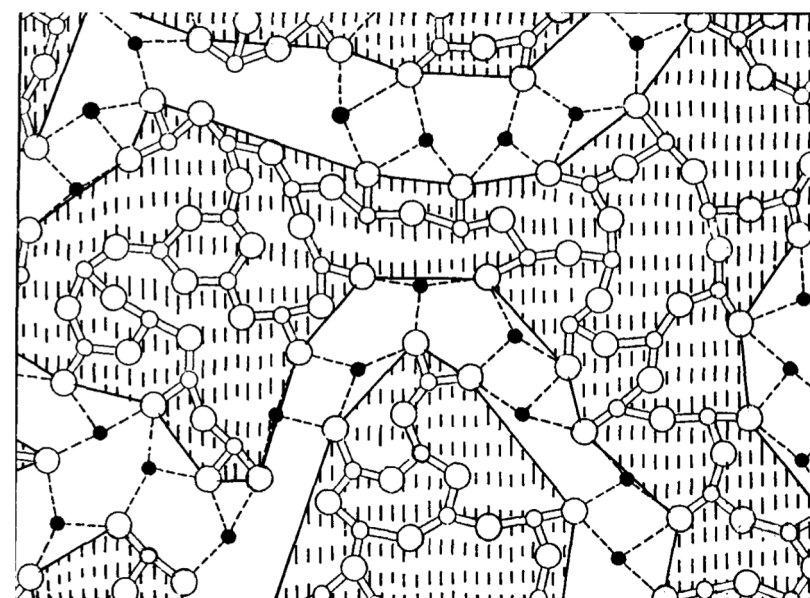
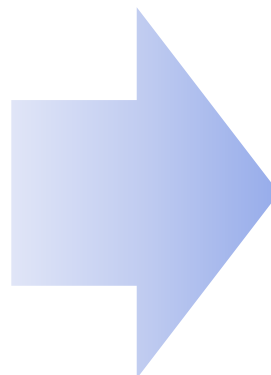
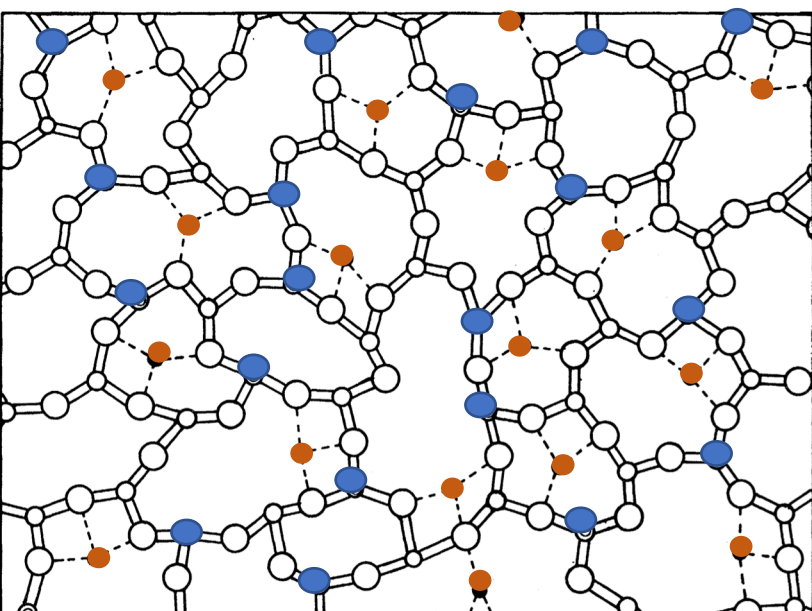
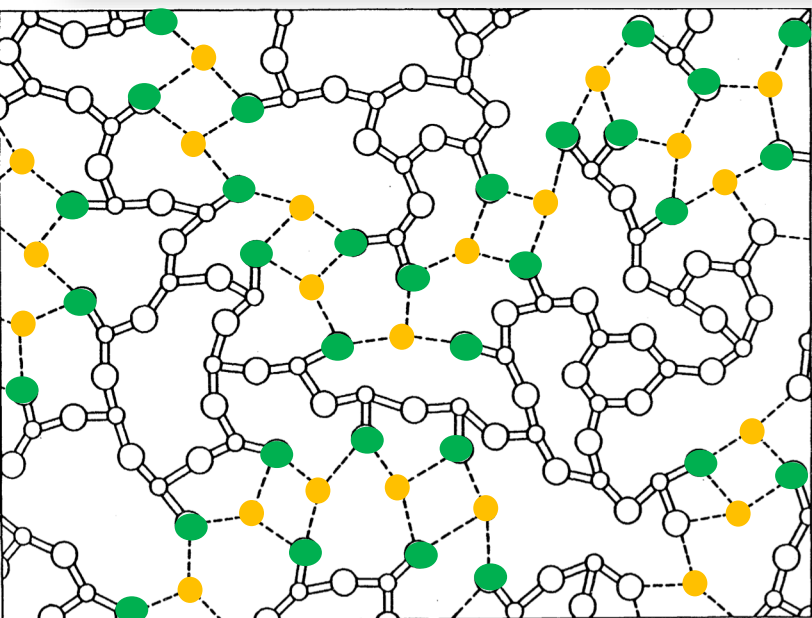
Raman VH



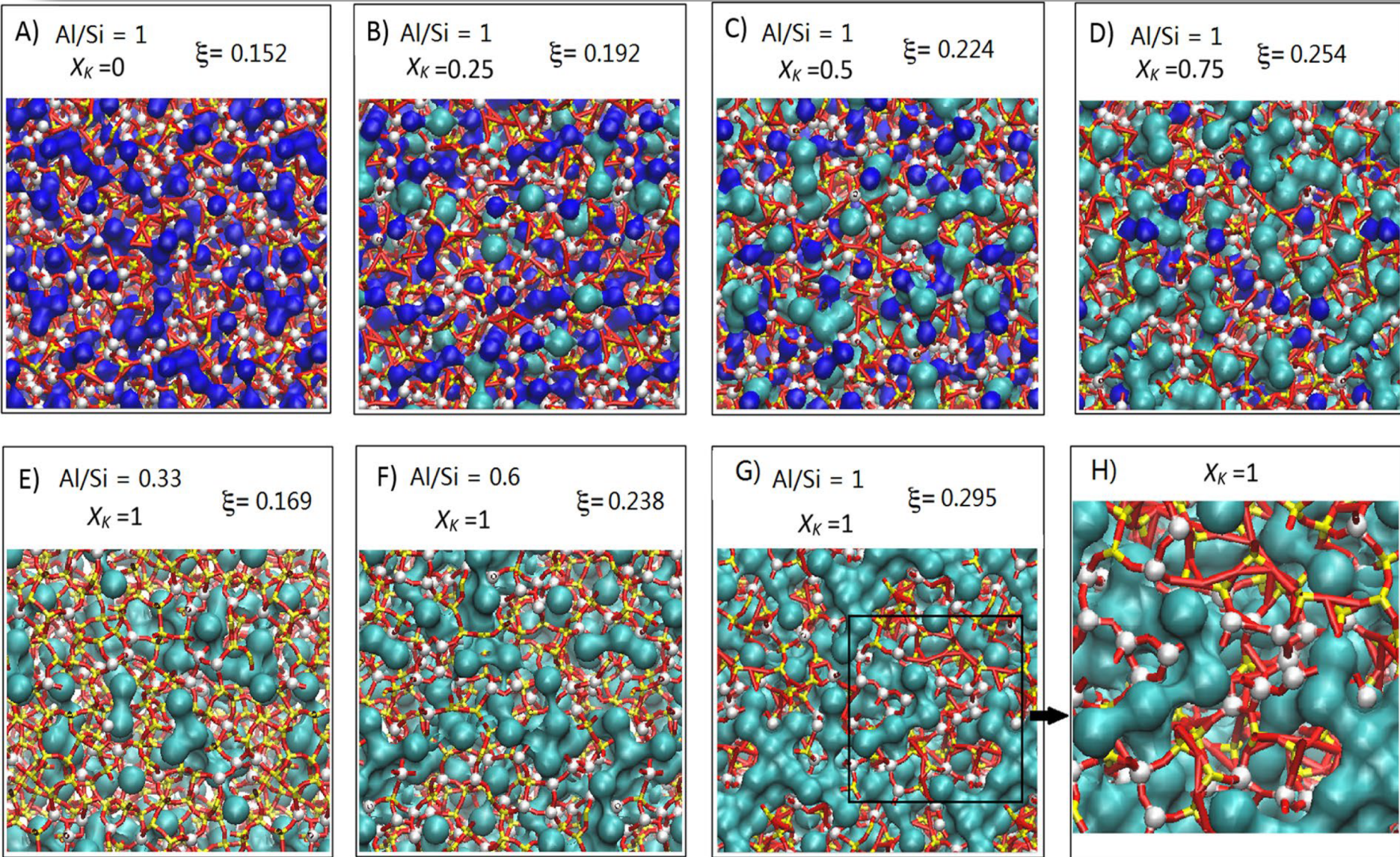
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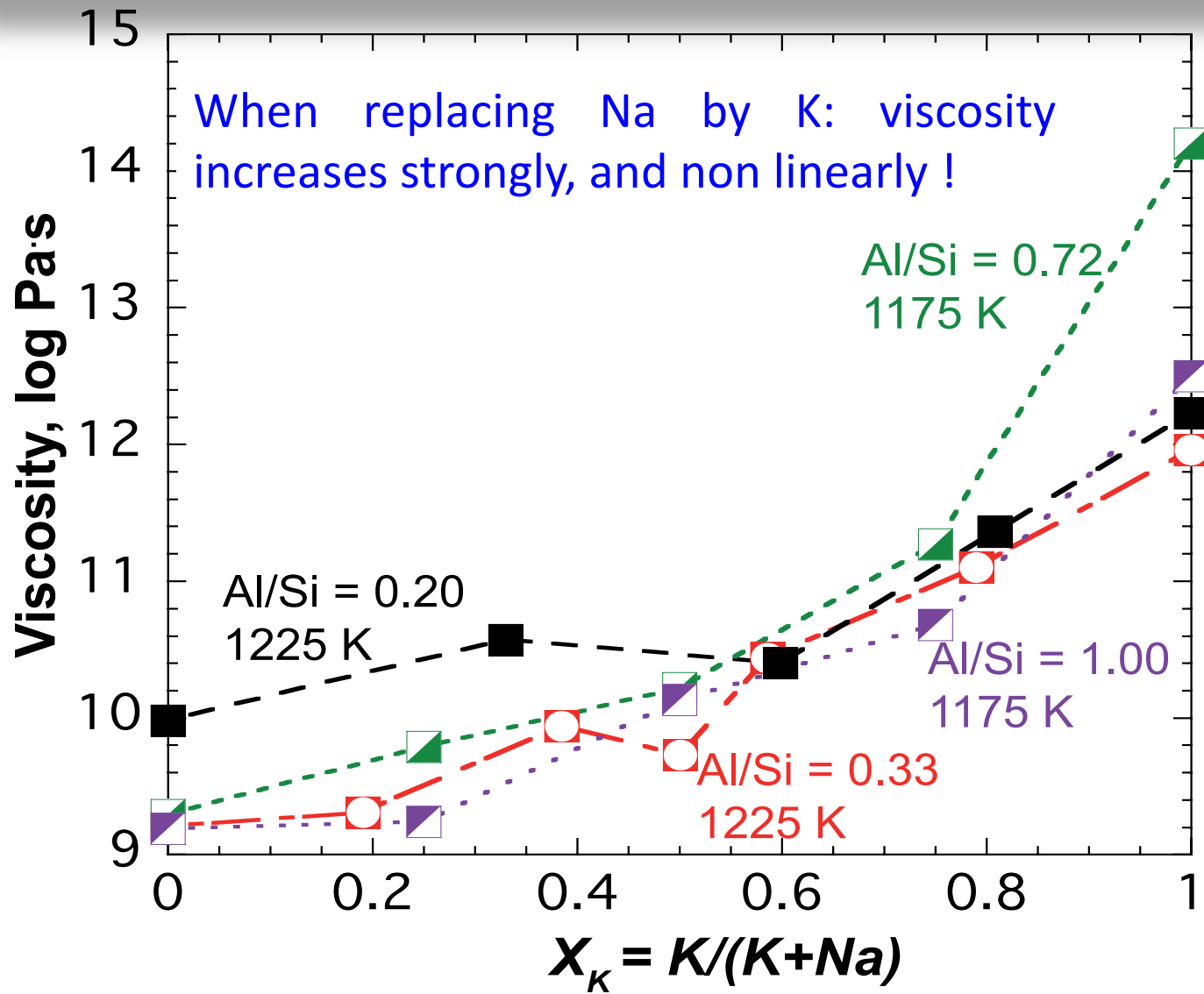


Cicconi M.R., de Ligny D., Gallo T. M., Neuville D.R. (2016) Ca Neighbors from XANES spectroscopy: a tool to investigate structure, redox and nucleation processes in silicate glasses, melts and crystals. *American Mineralogist*, 101, 1232-1236.



**Strong Relationships between structure,
and properties conductivity, viscosity,
alteration etc....**

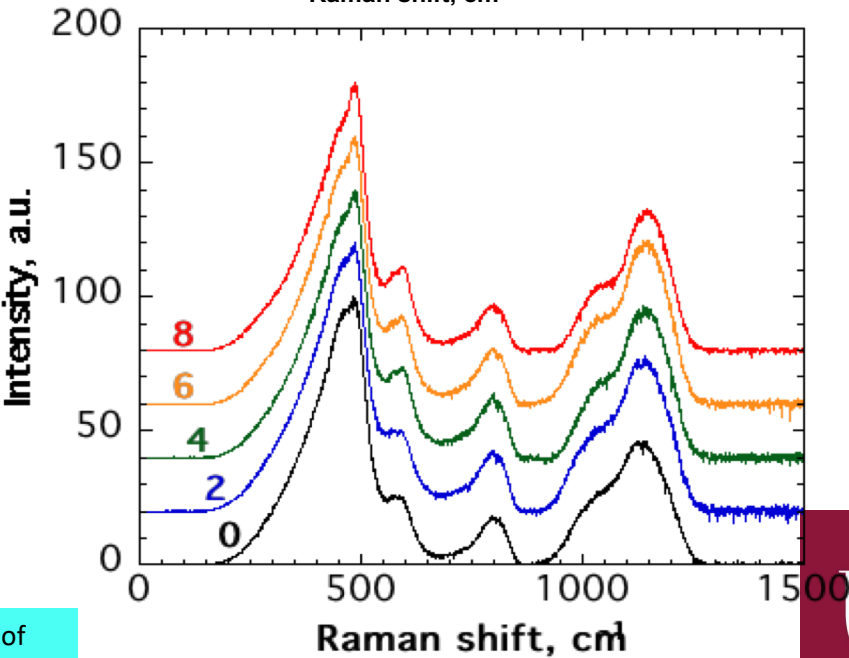
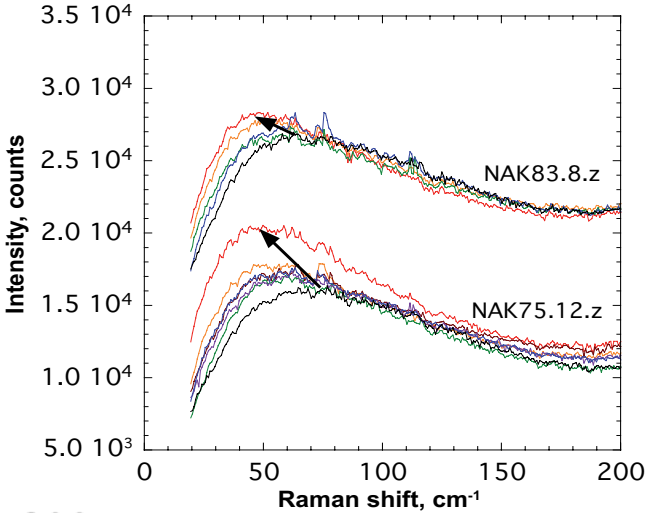
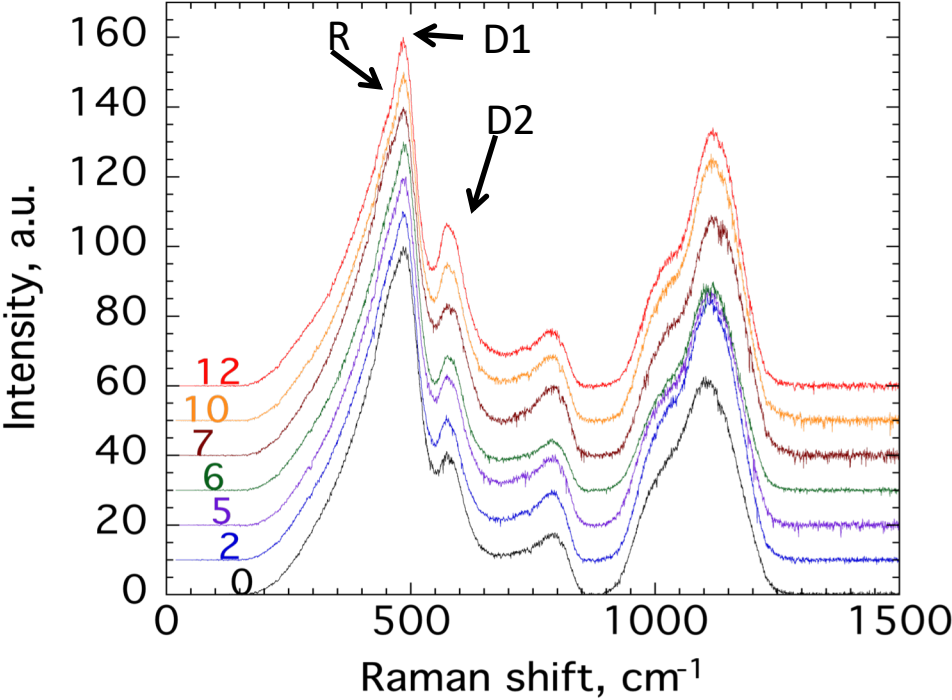




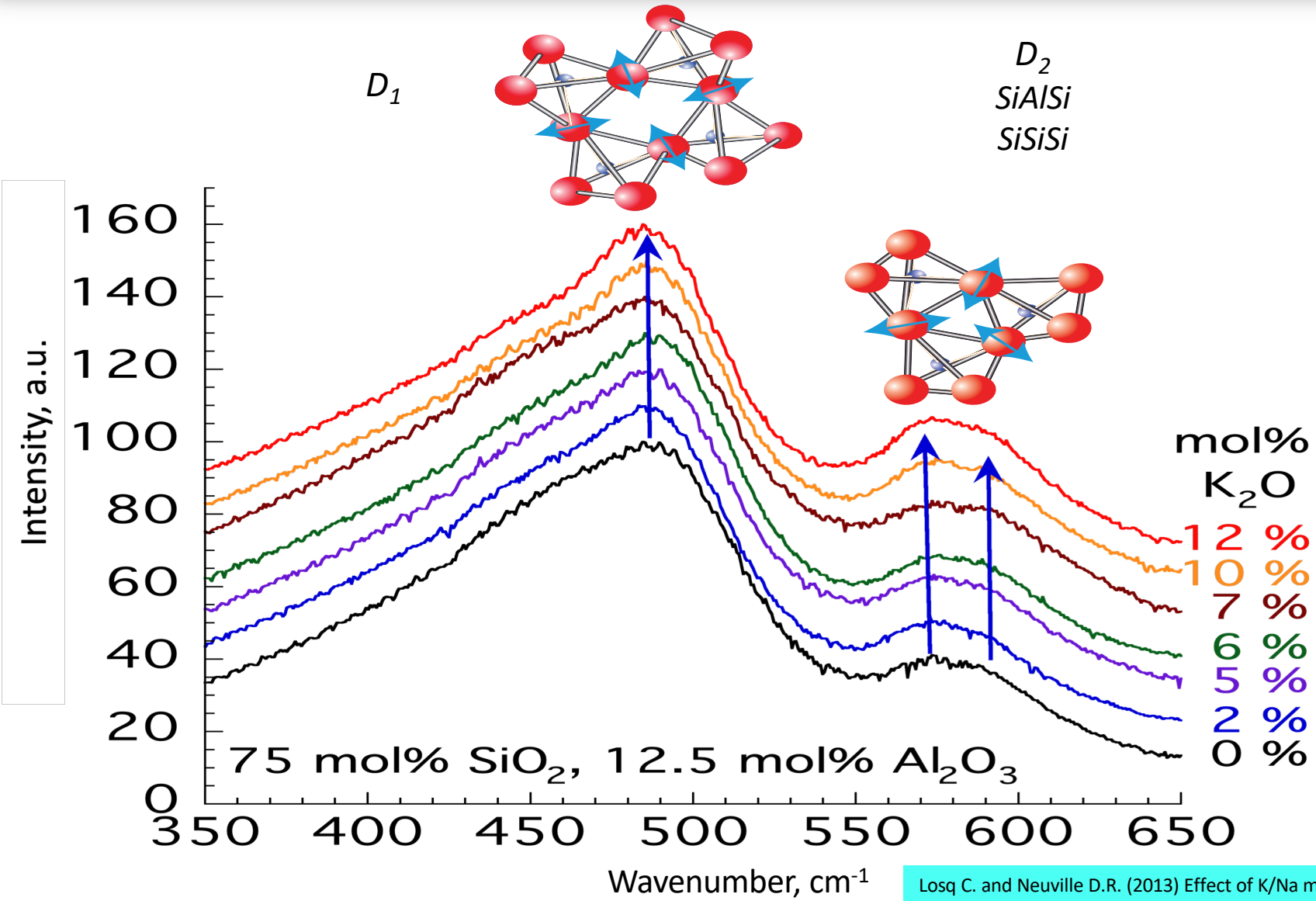
- Al/Si=0.11 =90%SiO₂
- Al/Si=0.20 =83%SiO₂
- Al/Si=0.33 =75%SiO₂
- Al/Si=0.50 =66%SiO₂
- Al/Si=0.72 =58%SiO₂
- Al/Si=1.00 =50%SiO₂

Le Losq C, Neuville D.R., Florian P., Massiot D., Zhou Z., Chen W., Greaves N. (2017) Percolation channels: a universal idea to describe the atomic structure of glasses and melts. Scientific Reports, 7, Article number: 16490, doi:10.1038/s41598-017-16741-3





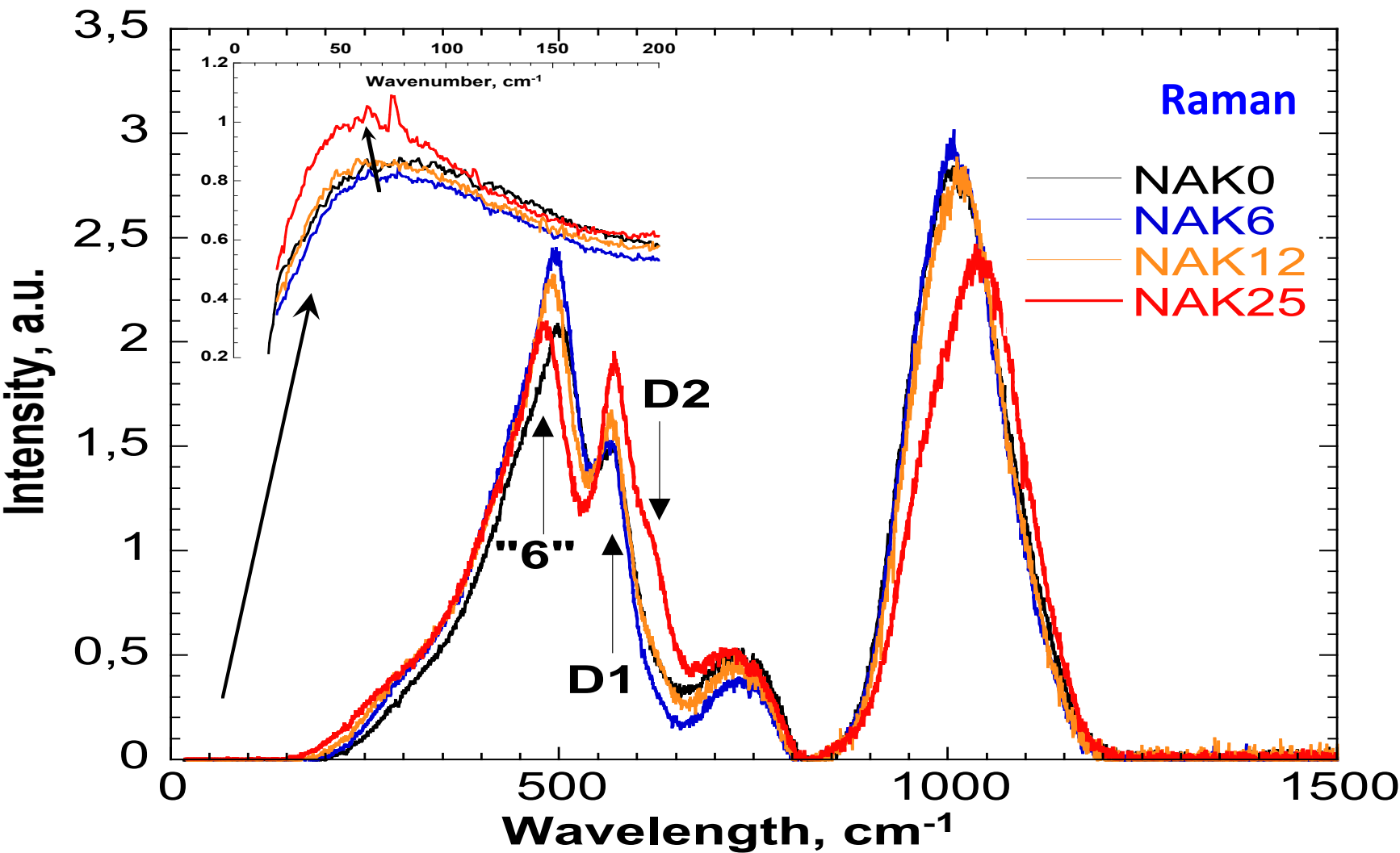
- Boson peak increase in I and decreases in frequency with K like close than SiO2
- D1 and D2 increase with K
- New D2 band

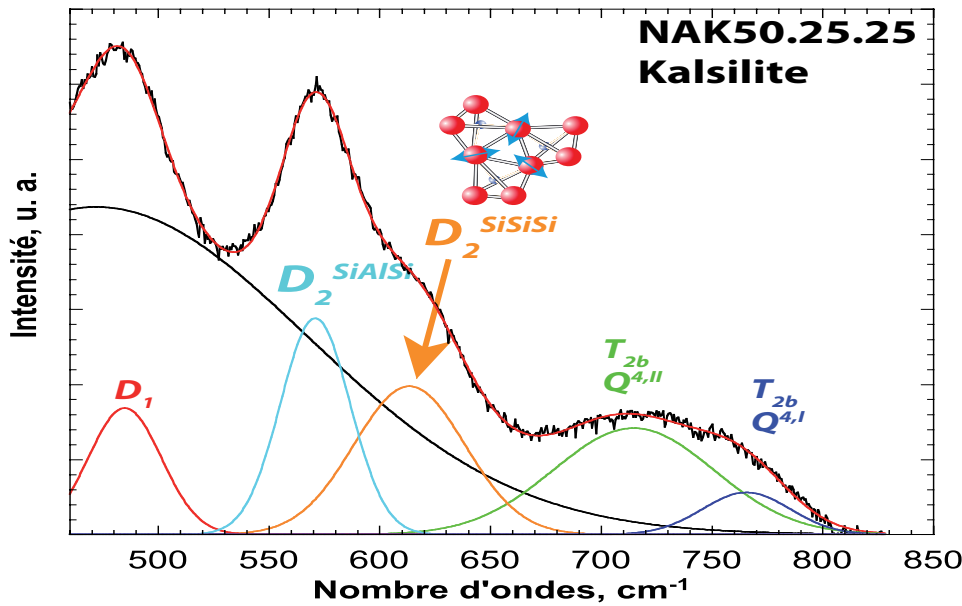
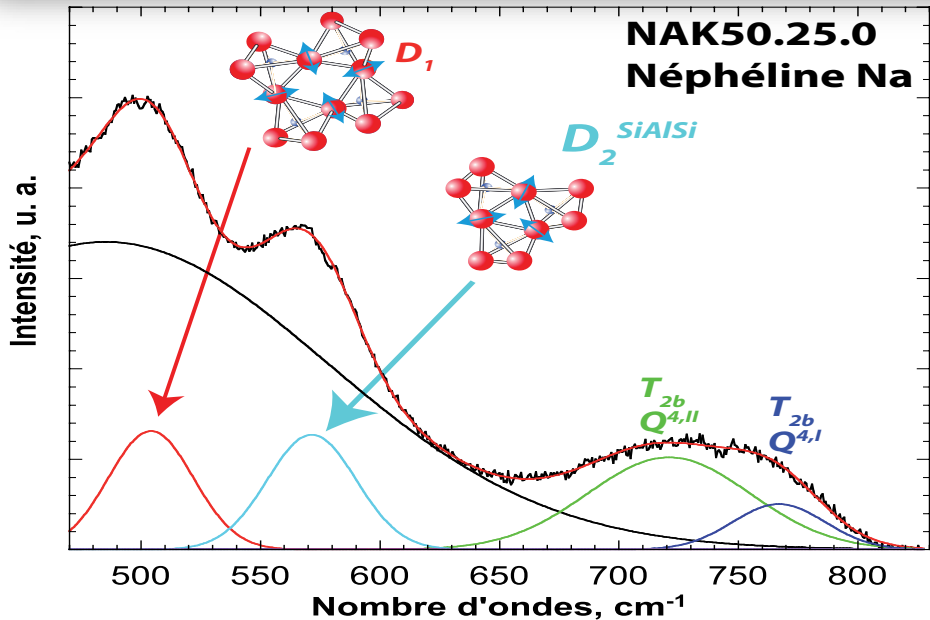


Losq C. and Neuville D.R. (2013) Effect of K/Na mixing on the structure and rheology of tectosilicate silica-rich melts. Chemical Geology, 346, 57-71.



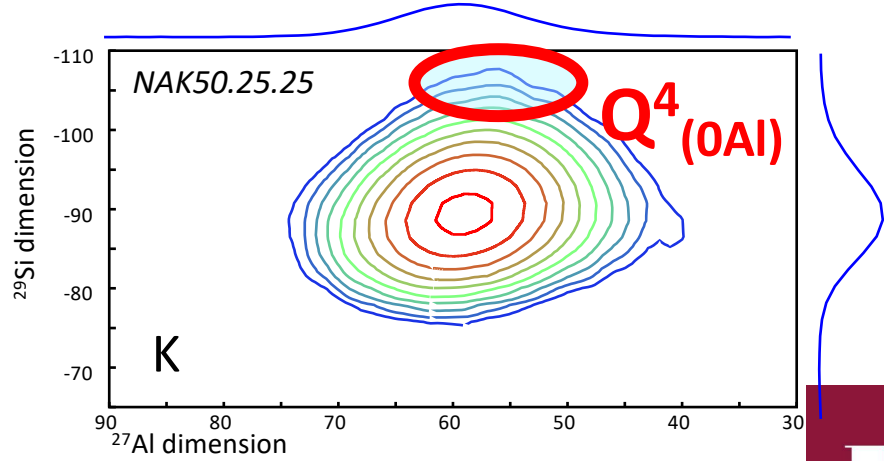
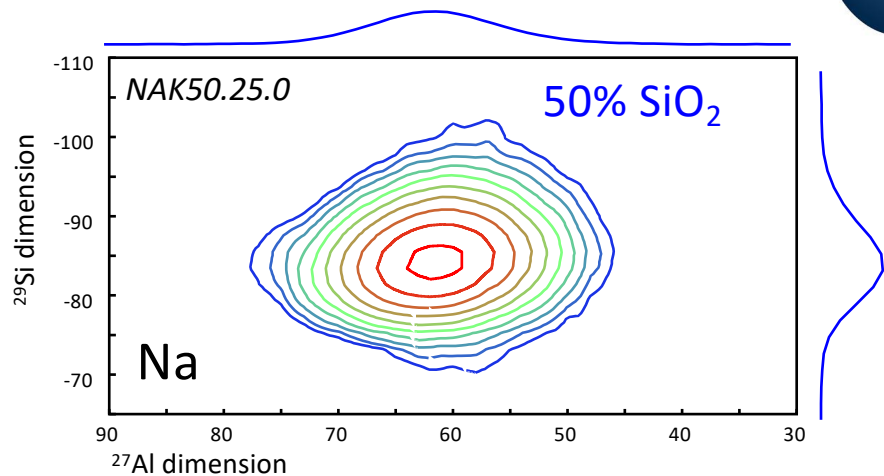
At lower SiO_2 concentration...

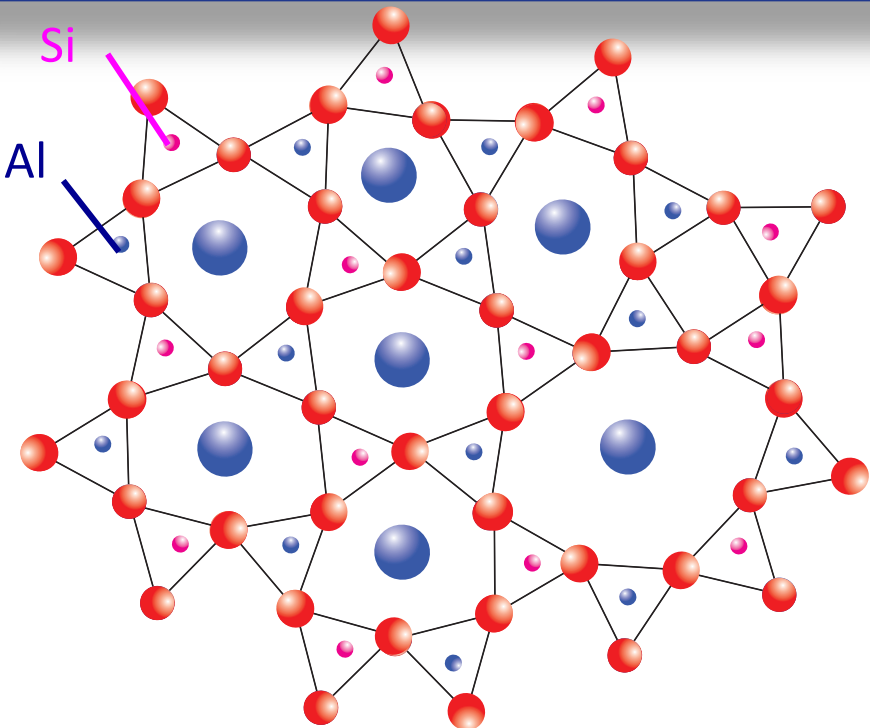




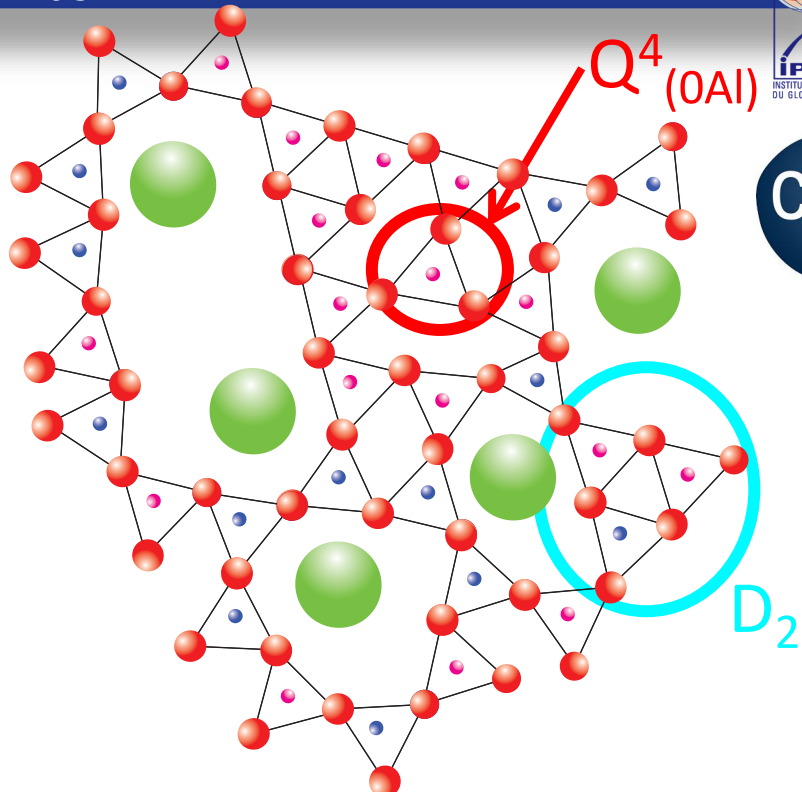
Glassy nepheline-kalsilite $\text{NaAlSiO}_4\text{-KAlSiO}_4$

$\{^{29}\text{Si}\}^{27}\text{Al}$ HMQC dipolar SR421 MAS 10kHz

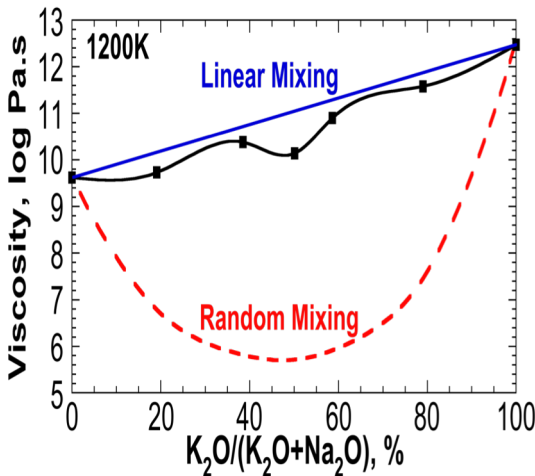




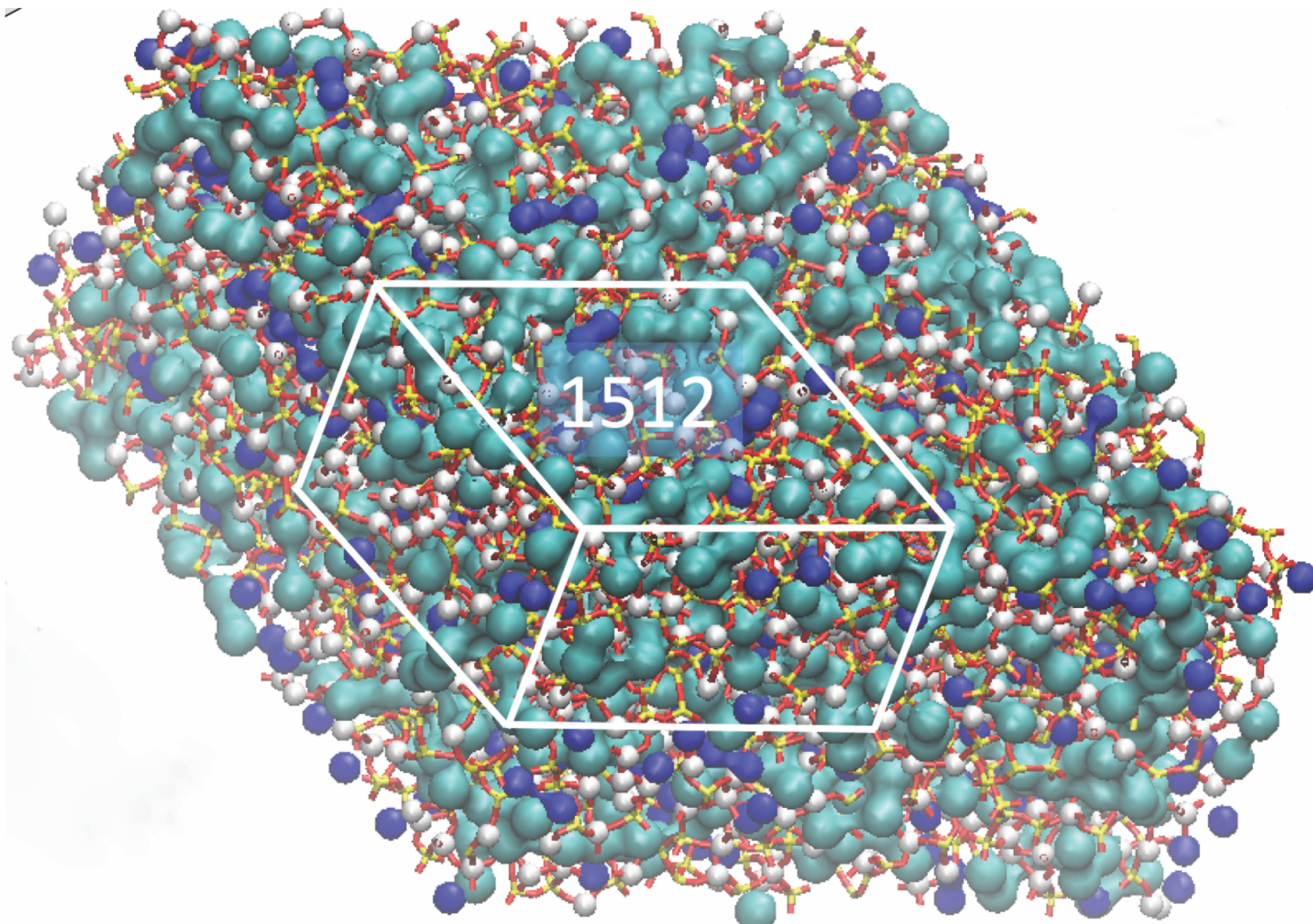
Na tectosilicates



K tectosilicates



Na and K are in different structural positions
⇒ Two different networks
⇒ Non random mixing



WELCOME ESRF & ILL Director of Research General Introduction - D. NEUVILLE	X-ray Emission Spectroscopy - P. GLATZEL	Raman Spectroscopy - B. HEHLEN	Static NMR - P. FLORIAN	Ab initio simulation - M. GONZALEZ
Coffee break	Coffee break	Coffee break	Coffee break	Coffee break
X-ray & neutron scattering - G. VAUGHAN & G. CUELLO	X-ray Imaging - M. COTTE	IR Spectroscopy - D. de SOUSSA M.	Dynamic NMR - D. MASSIOT	Access to instruments - F. d'ACAPITO & E. MITCHELL
Lunch	Lunch	Lunch	Lunch	Round Table and Conclusions
				Lunch
X-ray Spectroscopy - Y. JOLY	X-ray Photoemission Spectroscopy - D. FOIX	Practicals	XPCS - B. RUTA	
Coffee break	Coffee break		Coffee break	
Practicals	Practicals	Coffee break	Practicals	
		Visits to beamline		



