

LASER ET PROCÉDÉS DANS LES VERRERIES

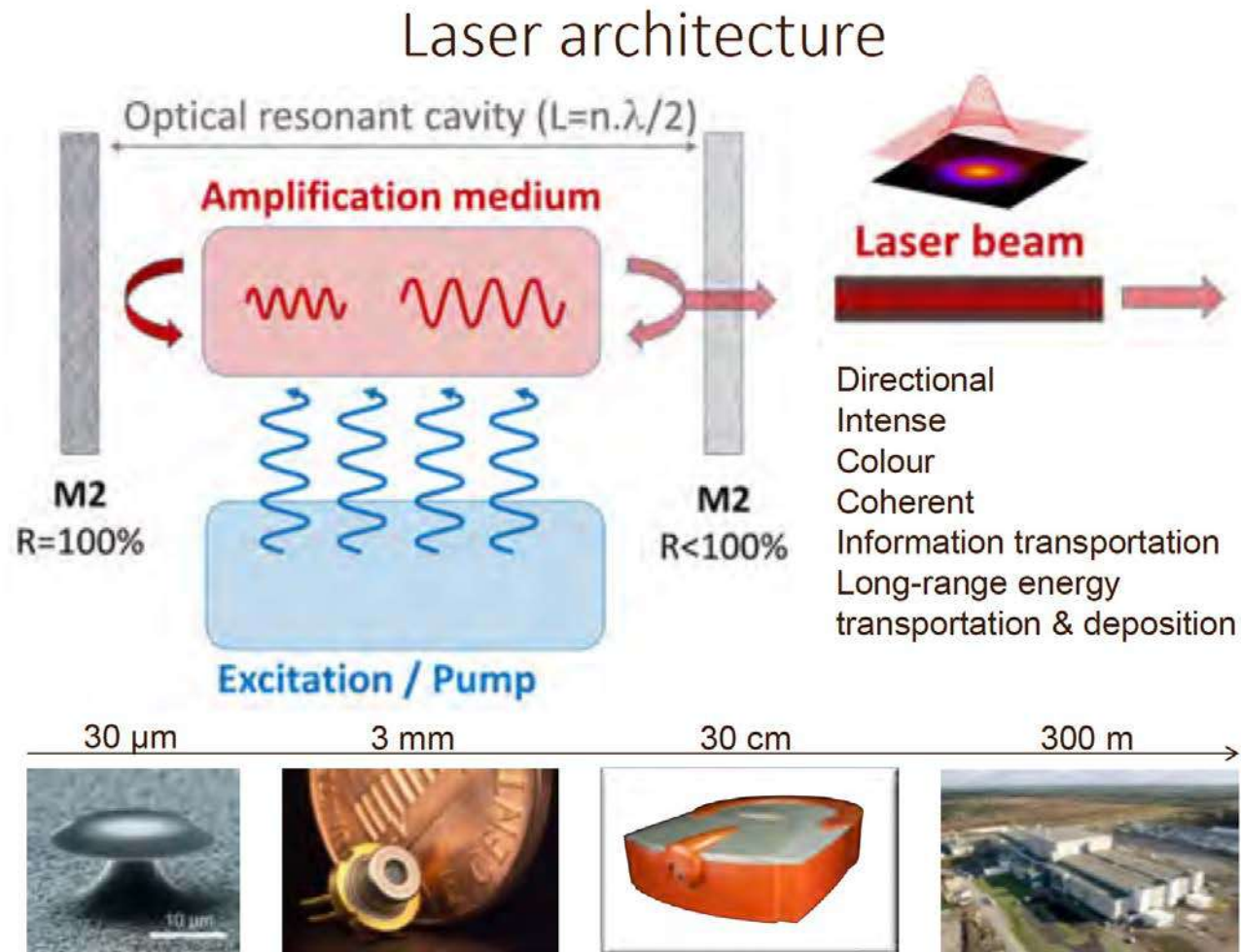
Matthieu Lancry



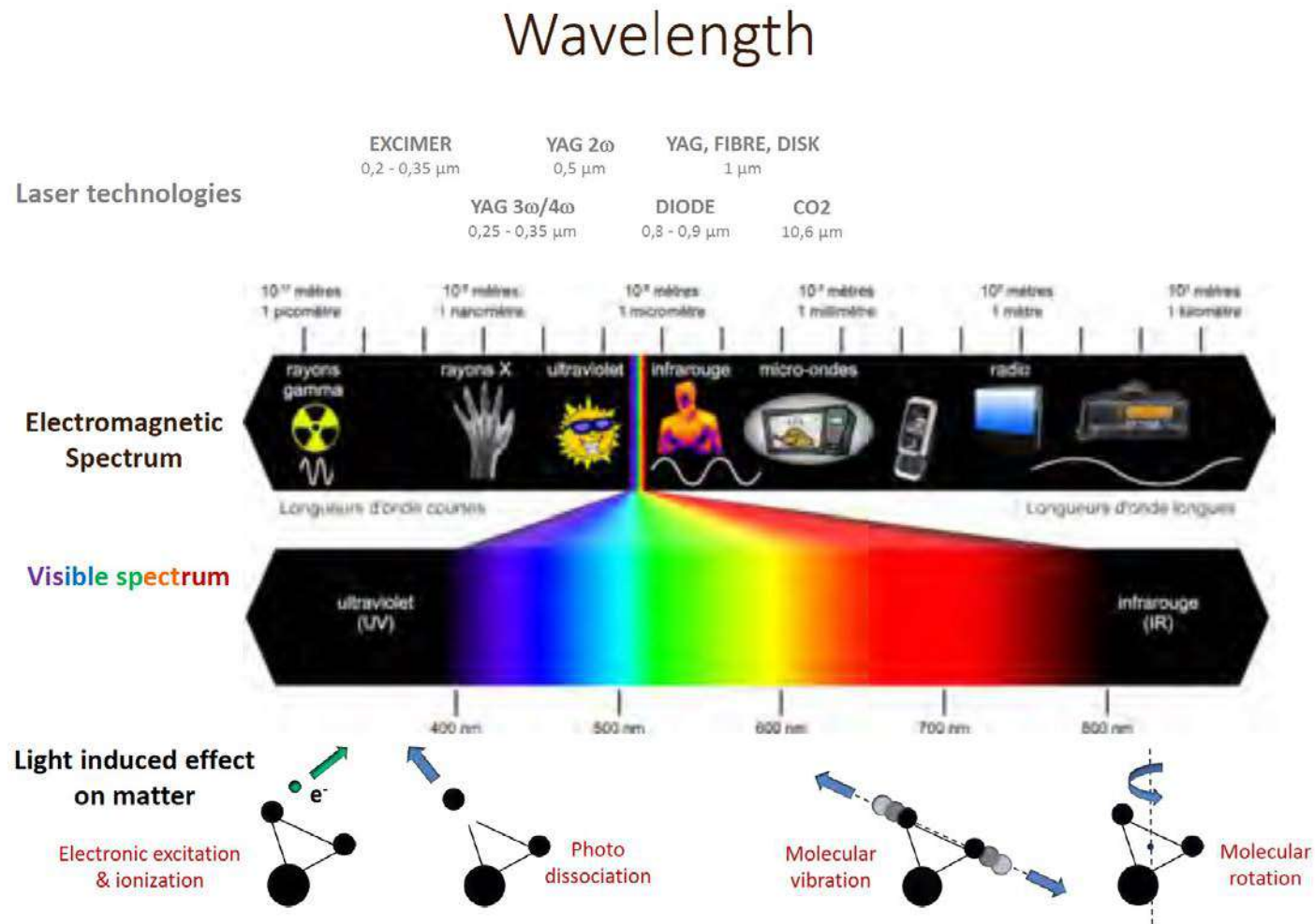
OUTLINE

- **Introduction**
- **Basic concepts**
- CO₂ laser
- Surface processing
- Volume processing
- Conclusions

CONTEXT AND INTRODUCTION

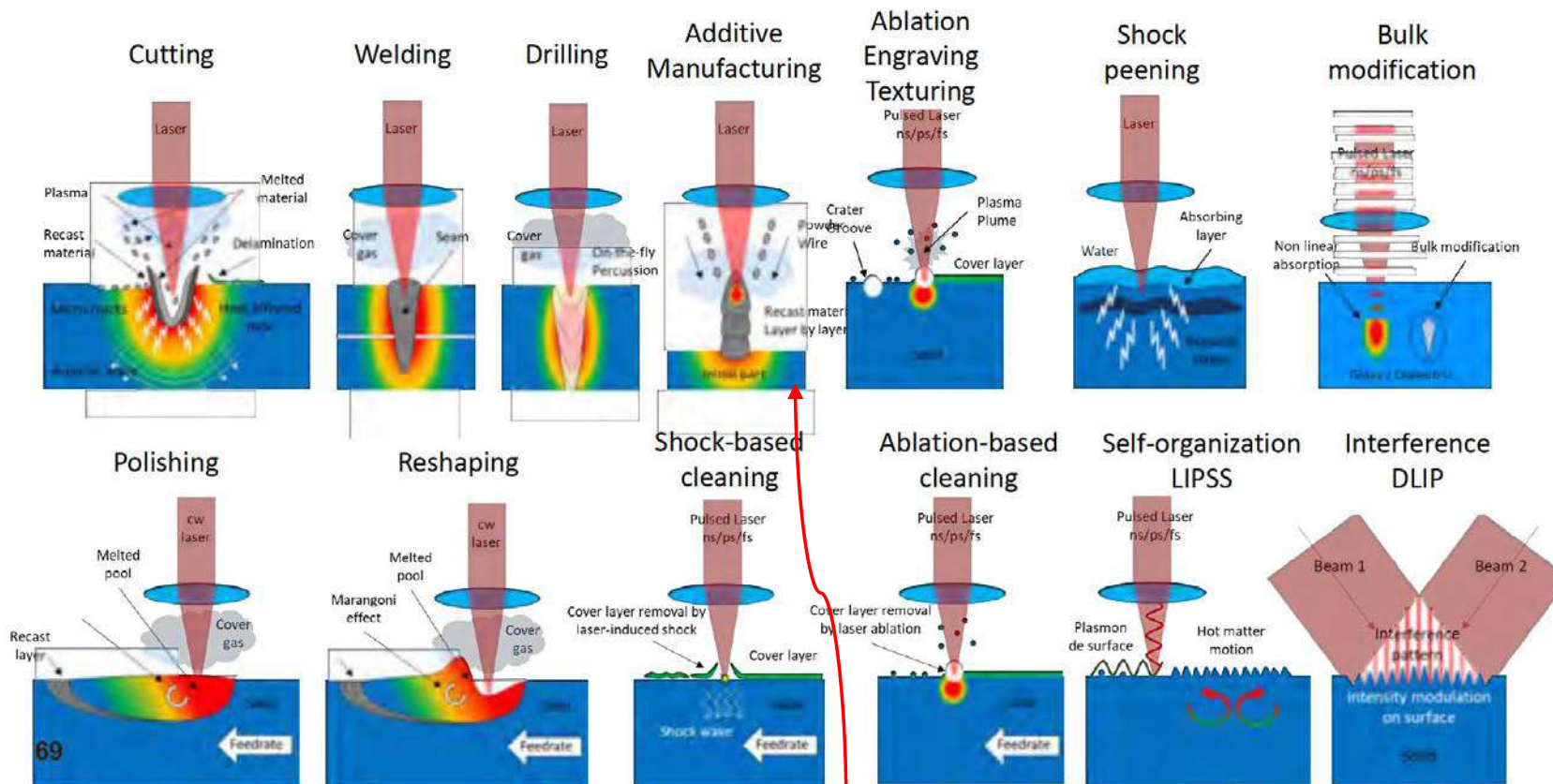


CONTEXT AND INTRODUCTION



CONTEXT AND INTRODUCTION

An universal tool for material processing...

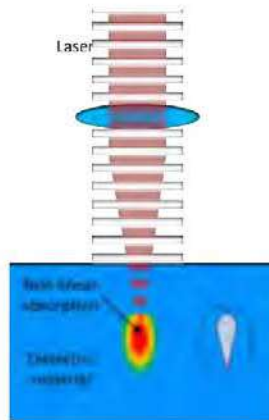


Cours de Yves Quiquempois

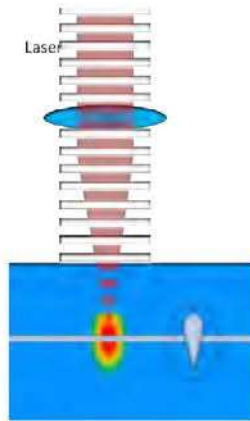
CONTEXT AND INTRODUCTION

... even for transparent materials

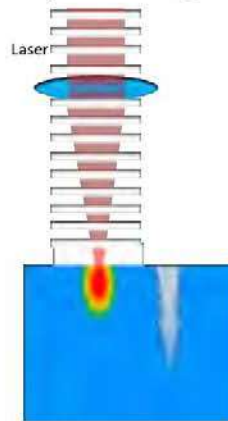
Bulk modification



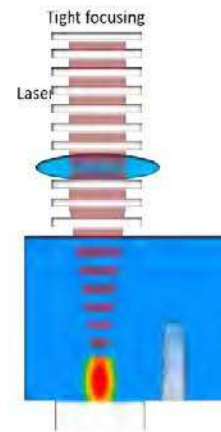
Transparent welding



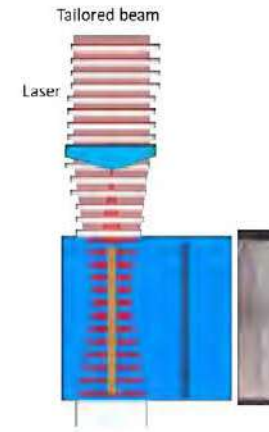
Top-down processing



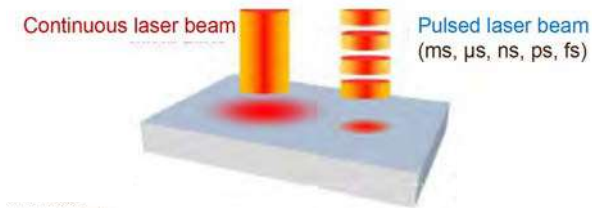
Bottom-up processing



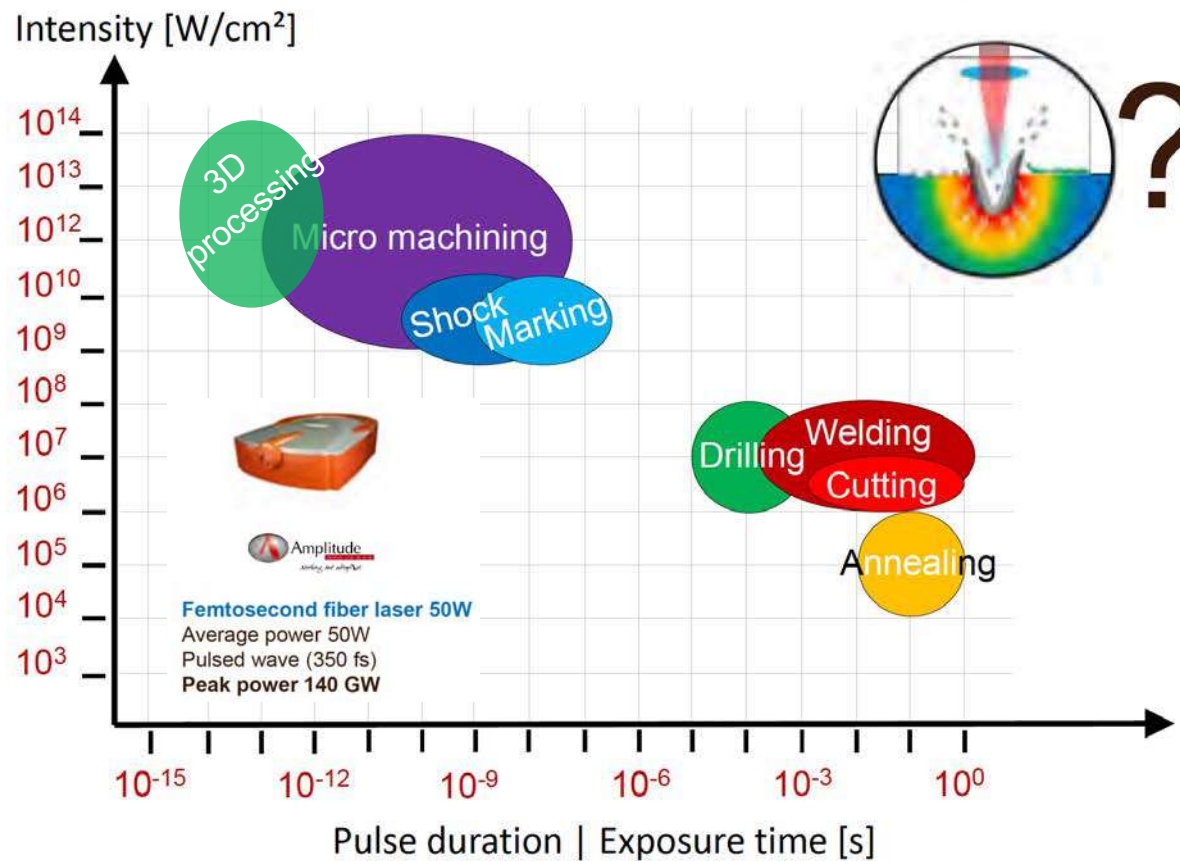
Non diffractive beam (Bessel, Airy)



CONTEXT AND INTRODUCTION



Pulse duration vs intensity



Fiber laser 100kW
Average power 100 kW
Continuous wave
Peak power 100 kW

Cours de
Maxime
Cavillon

OUTLINE

- Context and introduction
- **Basic concepts**
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BASIC CONCEPTS

- **Indice de réfraction (n) et biréfringence**

- Lien avec la constante diélectrique : $\varepsilon = \varepsilon_1 + i\varepsilon_2$, $N = n + i\kappa$,
- Milieu isotrope : n unique (scalaire).
- Milieu anisotrope : n devient un tenseur $\rightarrow$ indices propres selon les axes principaux.
- Biréfringence : $\Delta n = n_e - n_o$ (uniaxe) ; deux ondes orthogonales se propagent avec vitesses/indices différents $\rightarrow$ déphasage ou retard
- Retard : $\delta = (2\pi \Delta n \cdot e) / \lambda$ (e = épaisseur).

Matériau isotrope
(n ; κ)

Matériau anisotrope
(n_o, n_e ; $\kappa_{\parallel}, \kappa_{\perp}$)

- **Absorption (α) et dichroïsme linéaire**

- Coefficient d'extinction κ lié à l'absorption : $\alpha = 4\pi\kappa / \lambda$
- Isotrope : κ scalaire. Anisotrope : κ tenseur $\rightarrow$ absorption dépend de la polarisation.
- Dichroïsme linéaire (DL) : $\Delta\alpha = \alpha_{\parallel} - \alpha_{\perp}$; absorption différentielle entre polarisations orthogonales.

$$N = n + i\kappa,$$

$$\alpha = (4\pi \kappa / \lambda),$$

$$\text{Biréfringence : } \Delta n = n_e - n_o$$

$$\text{Retard } \delta = (2\pi \Delta n e) / \lambda$$

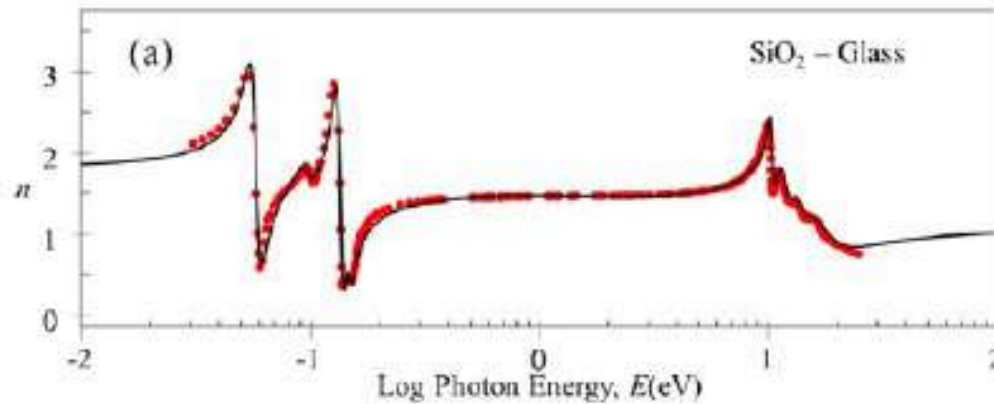
$$\text{Dichroïsme : } \Delta\alpha = \alpha_{\parallel} - \alpha_{\perp}$$

Lien cours de Wilfried Blanc



BASIC CONCEPTS

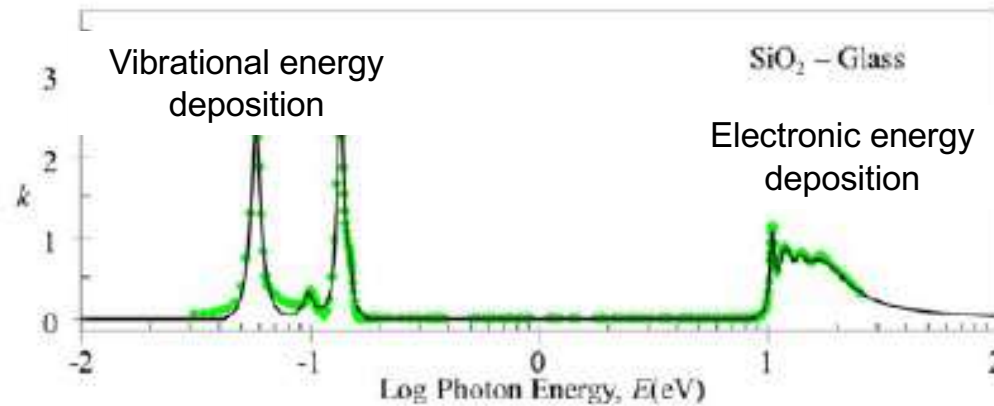
Everything start with light absorption



$$N = n + i\kappa,$$

$$N^2 = \epsilon_1 + i\epsilon_2$$

Partie réelle → Indice de réfraction n



Partie imaginaire → Indice imaginaire k

➡ SPECTRE D' ABSORPTION

$$\alpha = \frac{4\pi k}{\lambda}$$

Type de laser	CO_2	Ti:Sa, Nd:YAG, Yb :YVO ₄	Excimère
Longueur d'onde	10.6 μm	266nm-1064nm	157-351nm
Energie des photons	0.12eV	4.6-1.2eV	7.8-3.5eV
Régime	Continu	Continu, pulsé ns-fs	Pulsé 10ns

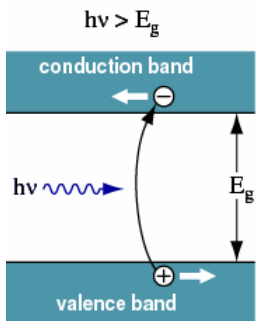
BASIC CONCEPTS

Everything start with light absorption

BANDGAP excitation

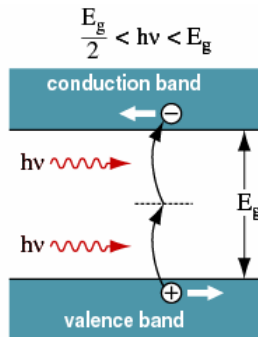
Two cases

Single photon
create (e-h) pair
Linear absorption



M. Lancry et al. Phys Reports 523 (2013)

Multiphotons
create (e-h) pair
Two-photons absorption



But multiphotons absorption (MPA)
needs high peak power $> \text{MW}/\text{cm}^2$
and thus ns-ps-fs lasers

REAH (Rare Earth Atom Heating)

R. Sato et al. J. Non-Cryst. Solids 289 (2001)

Sm^{3+} , Dy^{3+} , Nd^{3+} ...
 $> 5\text{mol}\%$ Sm_2O_3 , Dy_2O_3

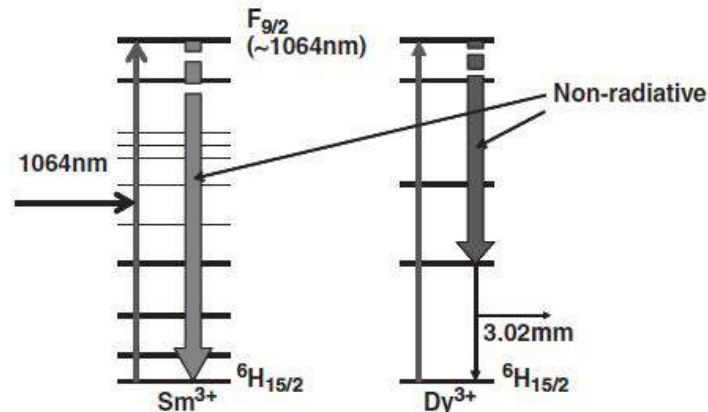


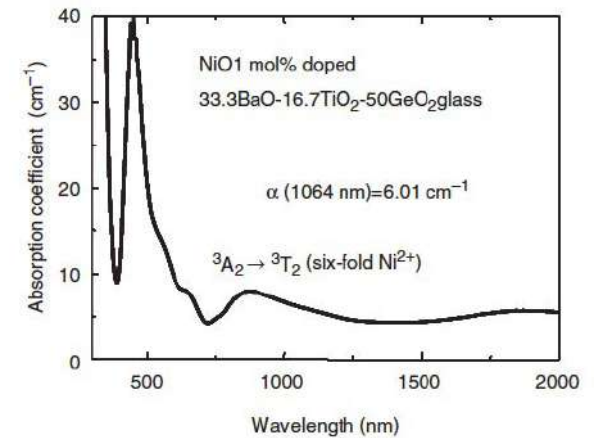
Fig. 1. Electronic energy levels of Sm^{3+} and Dy^{3+} ions.

T. Komatsu et al. J. Am. Ceram. Soc., 90 [3]
699–705 (2007)

TMAH (Transition Metal Atom Heating)

T. Honma et al. Appl. Phys. Lett. 88 (2006)

Ni^{2+} , Fe^{2+} , Cu^{2+} , ...
Typ. 0.01-1mol%

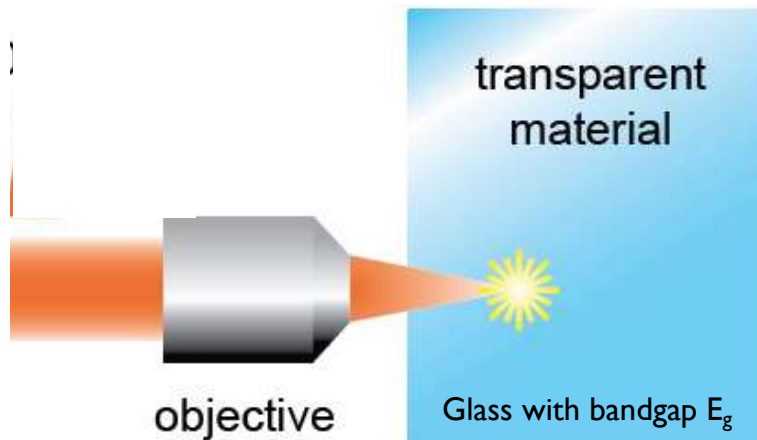


T. Komatsu et al. J. Am. Ceram. Soc., 90 [3]
699–705 (2007)

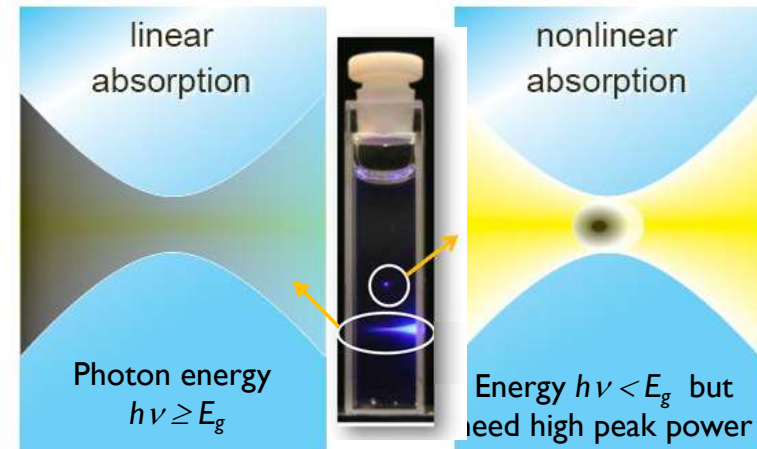
Cours « verres et couleurs » Laurent
Cormier, Gérald Lelong

BASIC CONCEPTS

Everything start with light absorption



Davis et. al, Opt. Lett., 21, 1729 (1996)



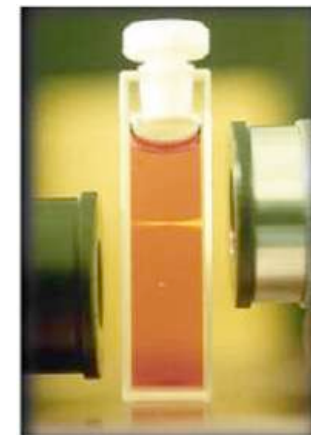
Amount of absorbed power in a thin layer of the medium is proportional to the I^n square of the light intensity (n is the number of photons).

Provide a 3D bulk confinement in a voxel of few μm^3

Example: 2 photons absorption (TPA)

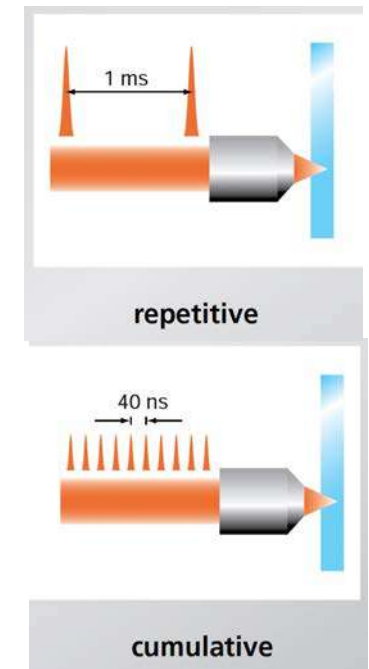
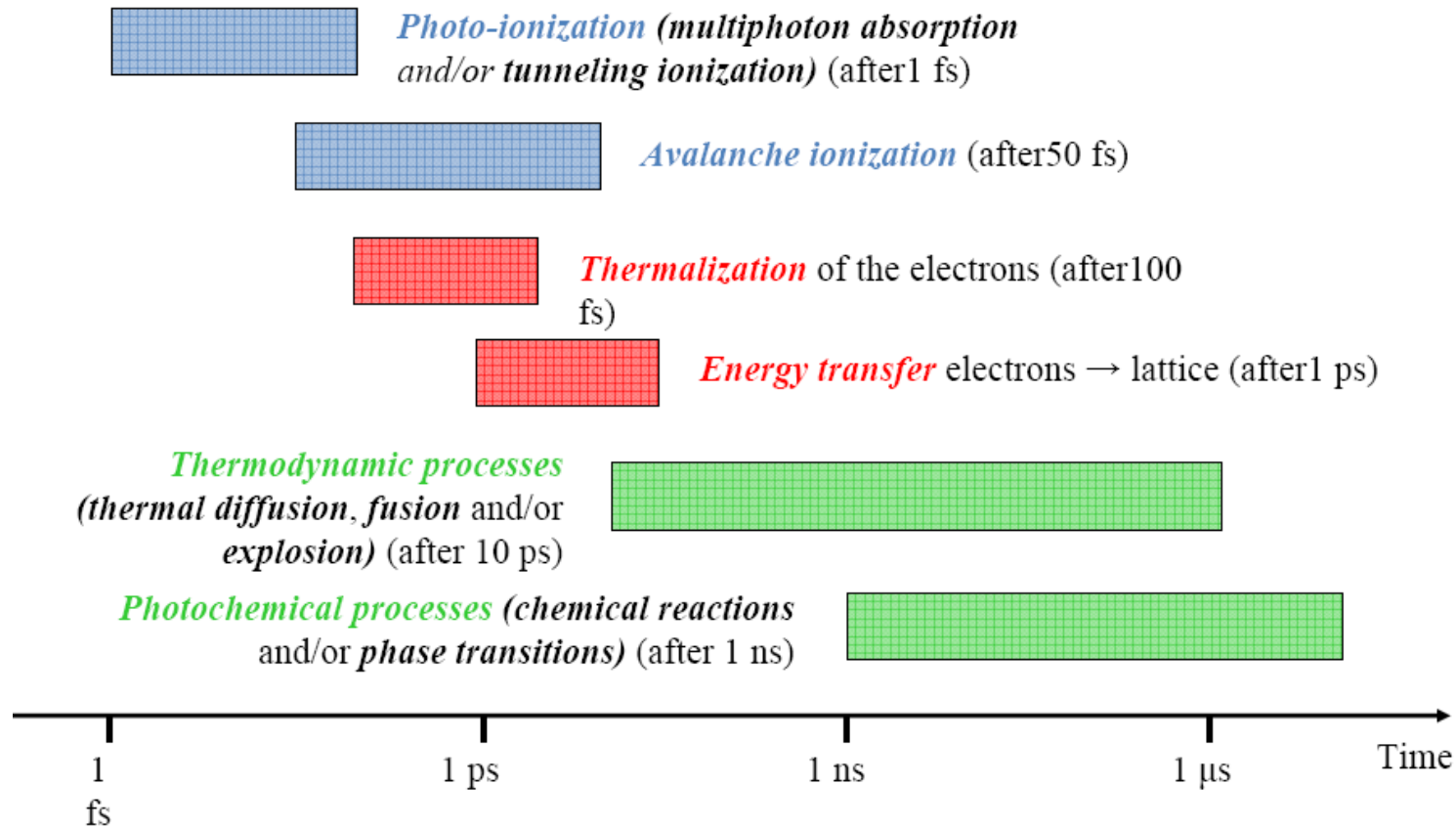
$$\beta = \frac{1}{I^2} \frac{dI}{dz} = \frac{3\omega}{2\epsilon_0 n^2 c^2} \text{Im}(\chi_{xxxx}^{(3)})$$

Absorption coefficient is proportional to the intensity of the light !



BASIC CONCEPTS

Light absorption followed by many relaxation pathways



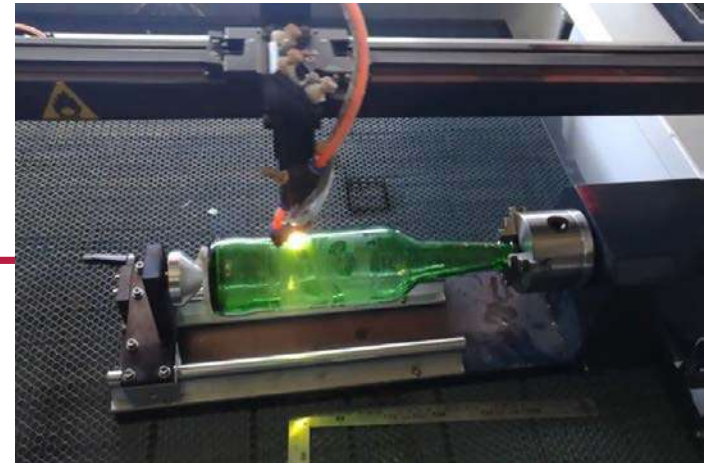
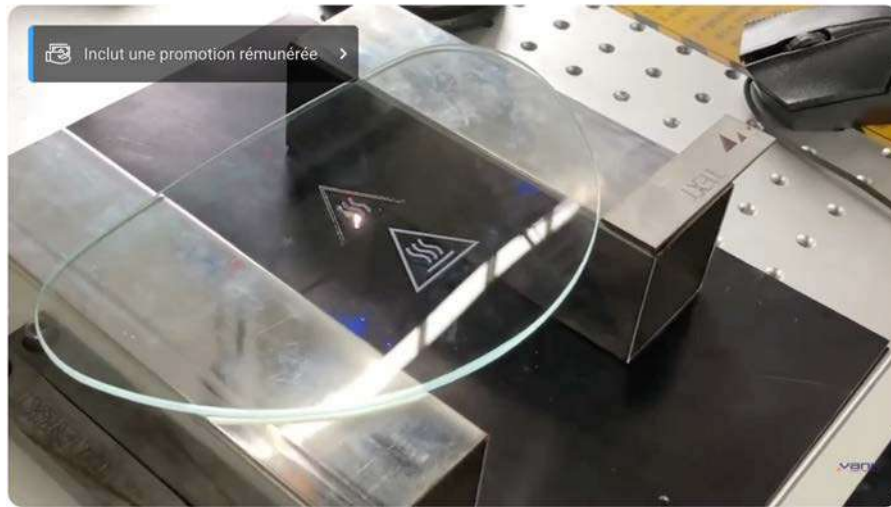
Schematics of main physical processes involved in the ultrashort (fs, ps) pulse laser processing of oxide glasses



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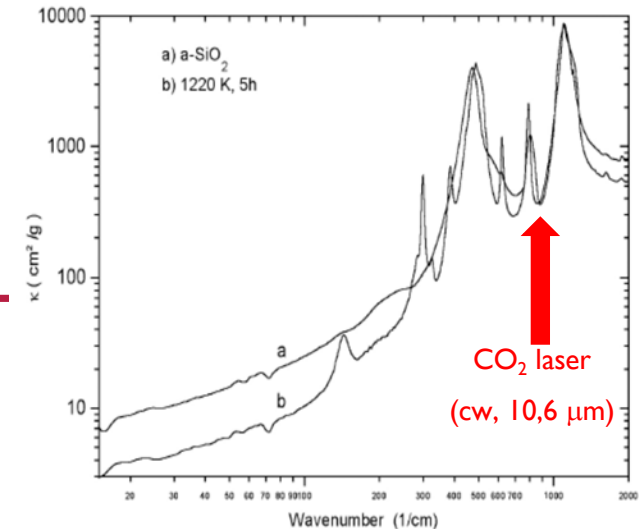
CO₂ LASER HEATING



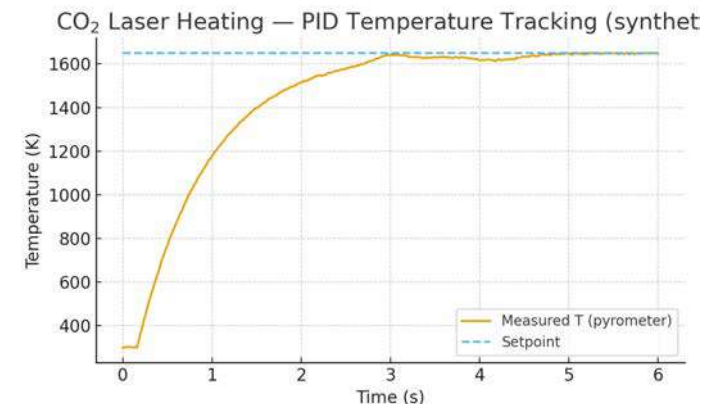
CO₂ LASER HEATING OF GLASS — FUNDAMENTALS

- Strong absorption at 10.6 μm $\rightarrow$ volumetric/surface heating
- Above T_g , viscosity drops $\rightarrow$ stress relaxation & surface flow.
- Then control cooling rate to set fictive temperature T_f and residual stress state.

- ΔT set by application (anneal vs morphing/polishing).
- Closed-loop control: pyrometer (e.g., 1300–2000 K) and bound thermal budget (avoid crystallization/devitrification).
- Process window : spot size (100–300 μm typical), irradiance W/cm^2 , dwell 0.1–5 s (scan speed, duty cycle); environment (air/He/N₂).



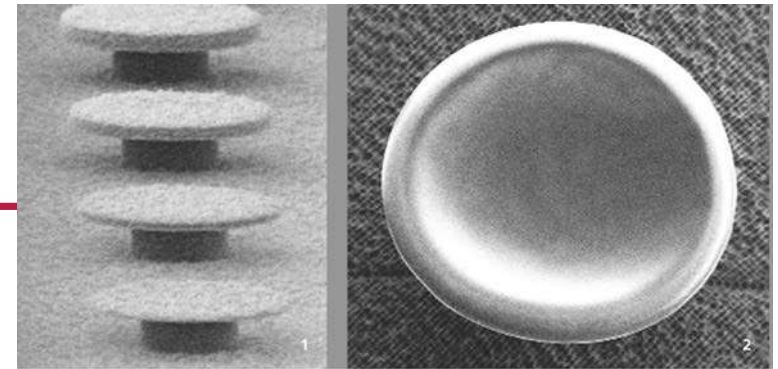
Strong linear absorption: $\alpha = 100\text{--}1000 \text{ cm}^{-1}$
i.e. 63% (1/e) of light absorbed within 10 μm ($\sim 1/\alpha$)



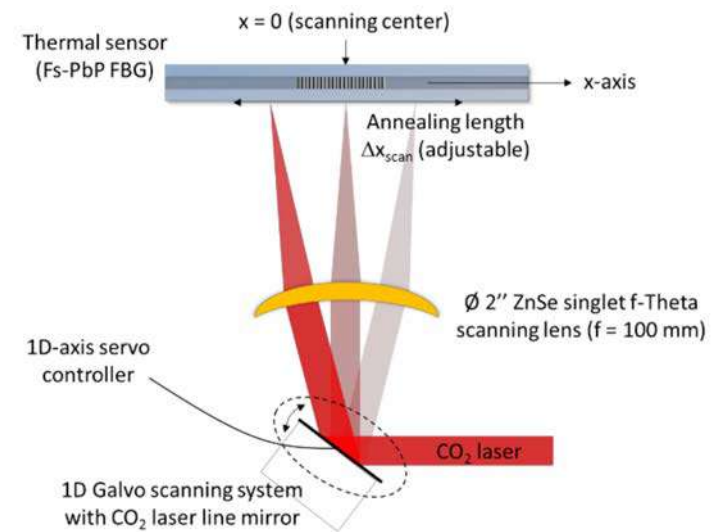
PROCESSES & APPLICATIONS

- Surface morphing: forming spheres and sealed cavities;
- Polishing of flats, micro-optics, and resonators: adaptable to molds and lens arrays.
- CO₂ pretreatment of fibers (few-mode, single-mode): relaxes frozen-in stresses → rapid cooling (higher T_f).
- CO₂ annealing of Fiber Bragg Gratings (=fiber sensors): → thermal stabilization; slower decay at 1050 °C

Cf Cours fibres de Maxime Cavillon



Microresonator (diameter of 50–200 μm , height of 2–10 μm)
Roughness is reduced to $R_a \sim 50 \text{ nm}$



Fiber Bragg Gratings (FBG) heating

CO₂ LASER GLASS POLISHING

- Mechanism: surface tension (σ) driven smoothing in the viscous regime; $\tau \sim \eta \cdot R / \sigma$
- Peaks flow faster than valleys (curvature-dependent), lowering Rugosity (R_q). $\sigma = \sigma_0 \left(1 - \frac{2\delta}{R}\right)$
- Regimes: (i) sub-melting viscous flow for ultra-low roughness; (ii) near-melting reshaping for recontouring; (iii) overexposure $\rightarrow$ crack risk.

- Optimization: choose dwell/overlap to hit R_q minimum; raster or spiral scan ($1 - 10 \text{ cm}^2/\text{s}$), multi-pass, cooling intervals.
- Metrology: white-light interferometry/AFM for R_q ; profilometry for form error; scattering/reflectance; micro-crack inspection; refractive-index homogeneity.

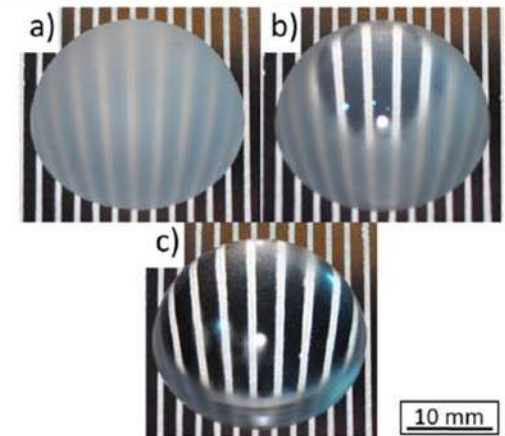
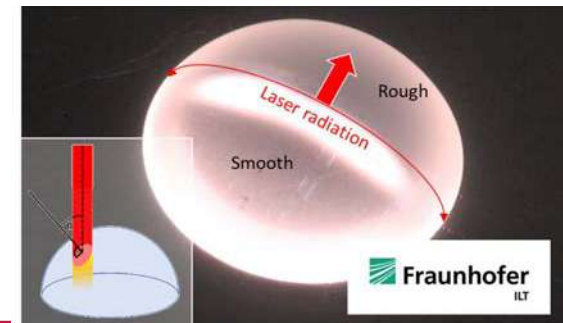
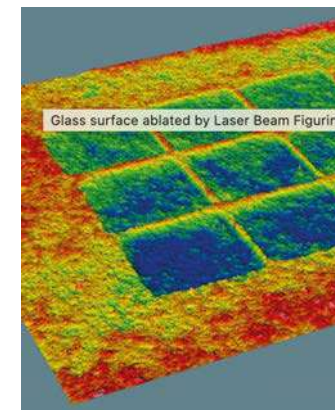


Fig. 8. Left: Images of samples with initial state (a), laser polished without correction (b) and with the presented model-based correction (c).



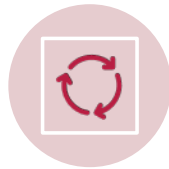
Roughness rms < 5 nm in SiO₂

Micro roughness rms < 0.4 nm

CO₂ LASER ENGRAVING ON GLASS



Mechanism — 10.6 μm absorption $\rightarrow$ localized heating; micro-fracturing / frosting/glazing of surface yields white contrast.



Typical settings — Spot 100–300 μm ; 5–25 W at part; 200–1200 mm/s; multi-pass to deepen without cracks.



Best materials — Soda-lime, borosilicate, lead crystal; avoid tempered glass (risk shatter). Masking improves edge quality.



Tips — Wet or dish soap to reduce chipping; large beam for smoother frost..



Example: personalized flute (text + symbol).



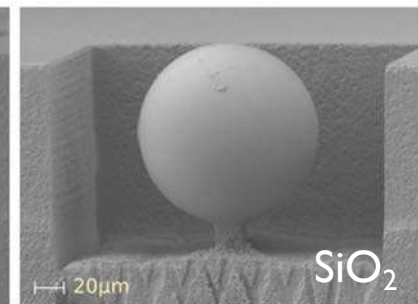
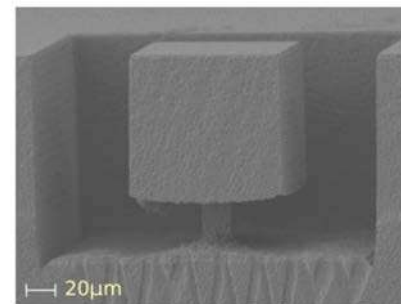
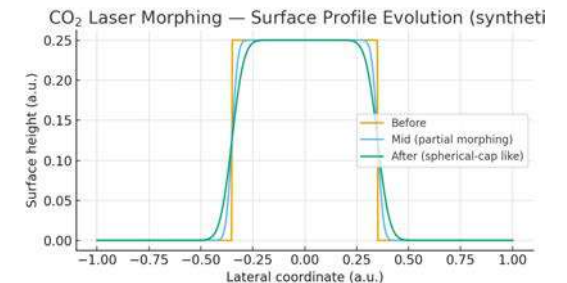
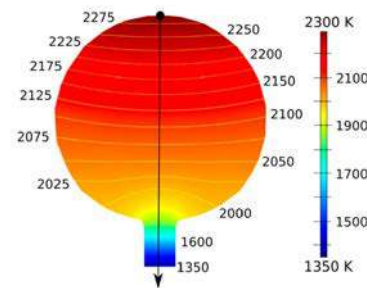
Example: patterned frosting on tumbler (rotary engraving).

- Notes: ranges are indicative and depend on optic size, coating/contamination, pyrometer emissivity calibration, and desired outcome (anneal vs morphing vs polishing)..



CO₂ LASER MORPHING — SURFACE TENSION DRIVEN SHAPING

- Principle — Above T_g , viscosity drops and surface tension σ drives curvature-dependent flow $\rightarrow$ features relax toward spherical caps. $\Delta P \approx 2\sigma/R$
- Process window — $\lambda = 10.6 \mu\text{m}$; spot 100–300 μm ; 2–20 W at part; dwell 1–60 s; scan (raster/spiral), multi-pass with cool-down.
- Applications— Recontouring micro-optics, sealing cavities, shaping microspheres, smooth polishing.
- Metrology — Interferometry/AFM for R_q & form error; check n homogeneity; watch for crystallization, ripple, microcracks (over-exposure).

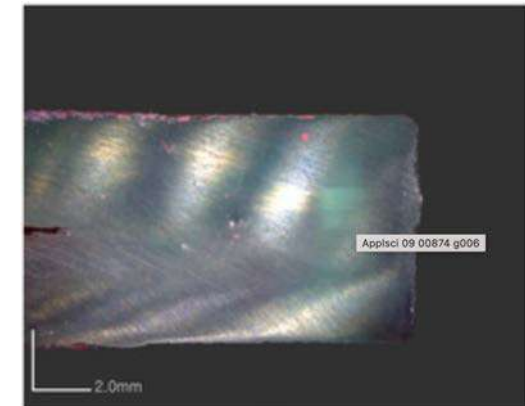
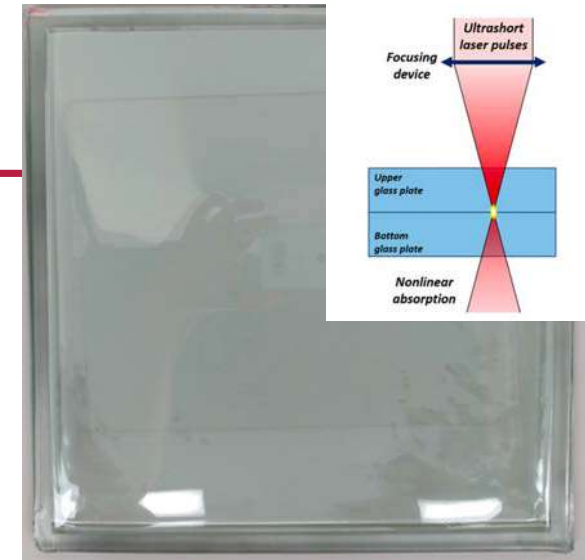


Jakub Drs, Opt. Express **23**, 17355-17366 (2015)

CO₂ LASER WELDING OF GLASS — EDGE/SURFACE SEALING

$$\tau \sim \eta \cdot R / \sigma$$

- Principle — Strong absorption at 10.6 μm heats the surface/edge above $T_{\text{softening}}$ + controlled cooling forms a welded seam.
- Process window — Spot 100–300 μm , 5–30 W, 0.1–5 mm/s; multi-pass
- Criticals — Contact (gap < few μm), clean interfaces, CTE mismatch $\rightarrow$ cracks (mitigate by preheat).
- Applications — Cover-glass sealing, edge joints on plates/tubes, simple glass-glass assemblies (not internal bulk welding).
- Metrology — He or liquid leak test; polarimetry (residual stress); check light scattering (bubbles/voids);



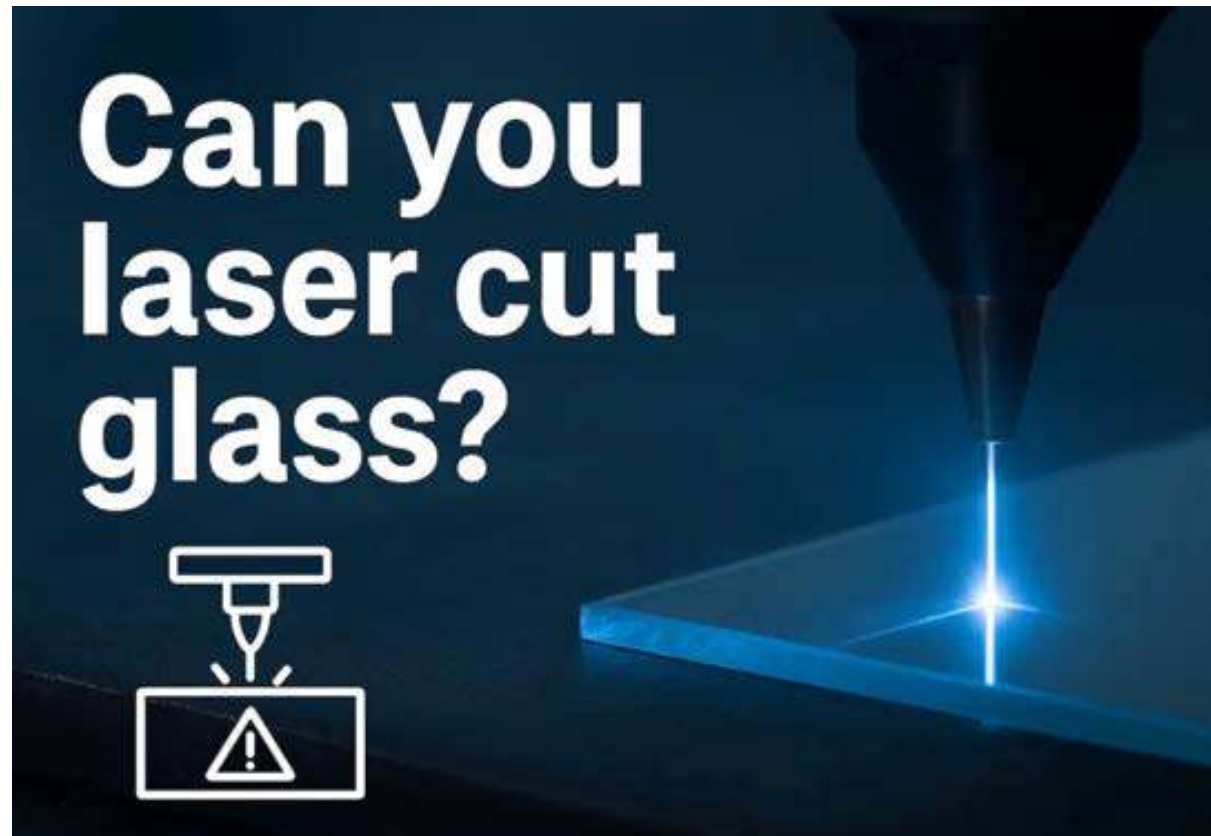
Schematic (not to scale): two glass plates clamped; CO₂ beam scans along the edge $\rightarrow$ molten seam; cool down under control.

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- **Surface processing (en Français)**
- Volume processing
- Conclusions

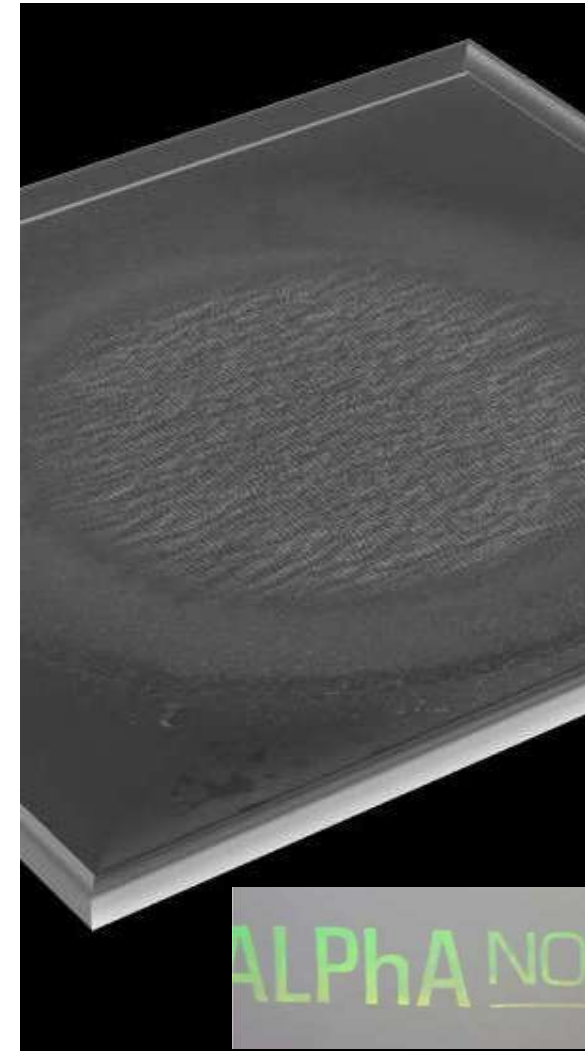
MODIFICATION DE SURFACE DES VERRES PAR LASER

MARQUAGE • ABLATION • DÉCOUPE • TEXTURATION • SOUDAGE



PANORAMA

- Procédés visés
 - Découpe
 - Usinage / ablation
 - Marquage (contraste/couleur)
 - Texturation fonctionnelle (mouillabilité, optique, antibactérien)
- Atouts du laser
 - Procédé sans contact
 - Précision μm – nm , faible HAZ (heat affected zone) with ps/fs
 - Moins d'étapes / pas de chimie (e.g. mask + chemical etching)
 - Intégrable en ligne de production (m^2/min)
- Matériaux cibles
 - Silice fondue, borosilicate (Borofloat, EagleXG..) verres alcalins
 - Saphir, ou dérivée, couches minces sur verre (ITO/Ag/oxydes)

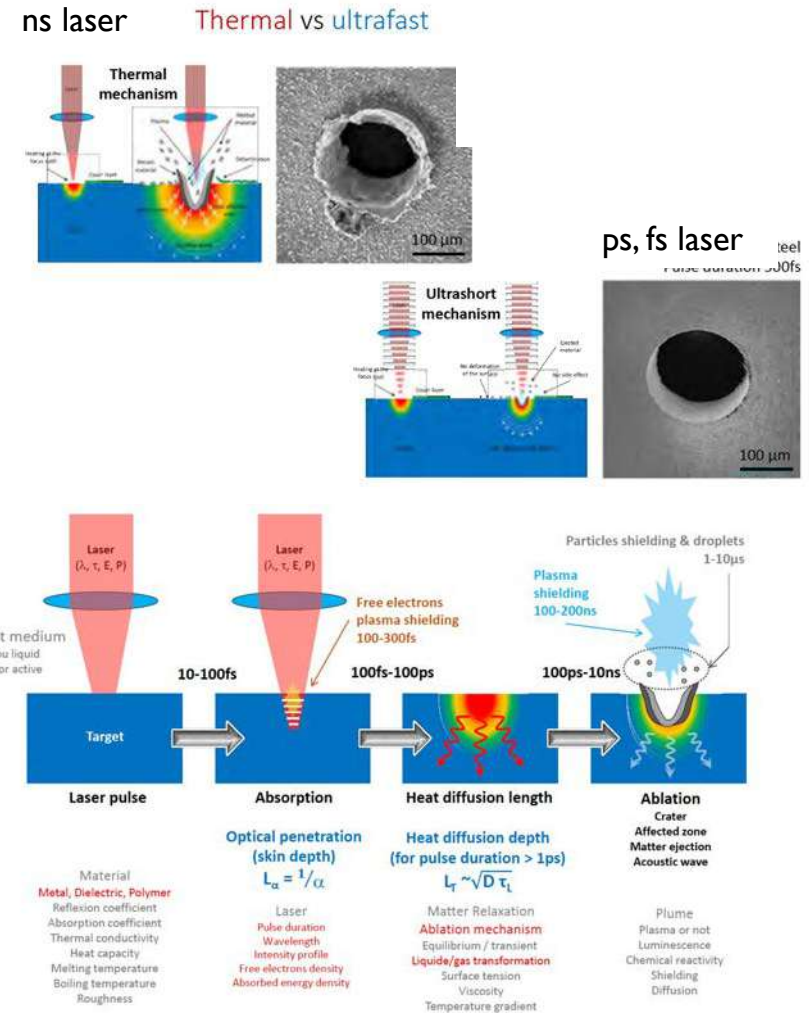


HAZ = Heat Affected Zone

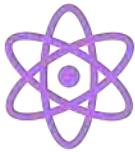


PHYSIQUE ESSENTIELLE (VERRES, IMPULSIONS NS/PS/FS)

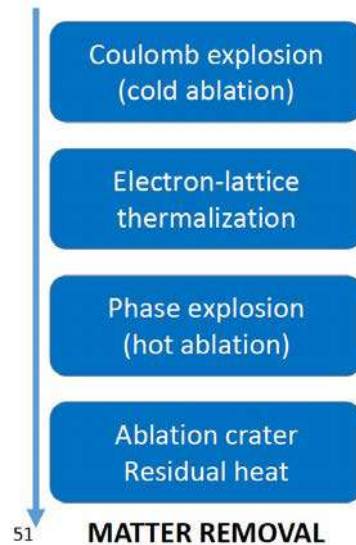
- Absorption non linéaire
 - Ionisation multiphotonique / tunnel (cf Keldysh)
 - Avalanche (impact) → plasma électronique transitoire
- Dépôt d'énergie – transfert au réseau vitreux
 - Électrons chauds → réseau (modèle à deux températures: $T_{\text{electronique}}$, T_{ionique})
 - Élévation locale de T , densification/expansion, ondes de choc, ondes de pression
- Phase explosion + effets de bords
 - Cratère d'ablation
 - Onde de pression, chaleur résiduelle, ...



PHYSIQUE ESSENTIELLE (VERRES, IMPULSIONS NS/PS/FS)



Mécanismes



Dielectric materials (non organic)

Semi conductor	Gap (eV)
Ge	0,7
Si	1,1
InP	1,35
GaAs	1,4
GaP	2,3
SiC	3,3
GaN	3,4
BN	5-6
Diamond	6,0
AlN	6,2
SiO ₂ quartz	8,8
Al ₂ O ₃ sapphire	9,0



Guides de choix λ

Higher photo ionization with shorter λ

Higher electrons heating with longer λ

UV : smaller penetration depth, precise ablation depth, marquage “sans chauffage”

Vert/IR : couches métalliques (décoating propre)



Paramètres

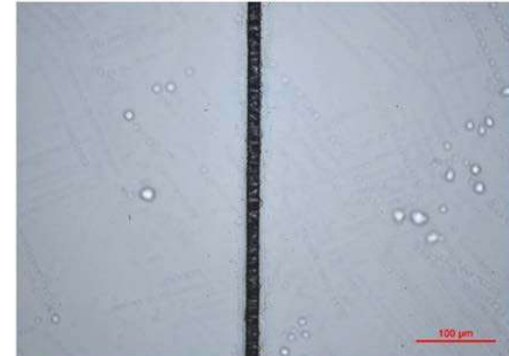
Fluence proche du seuil, durée/recouvrement

Défocus léger ou NA modérée pour marquage « doux »

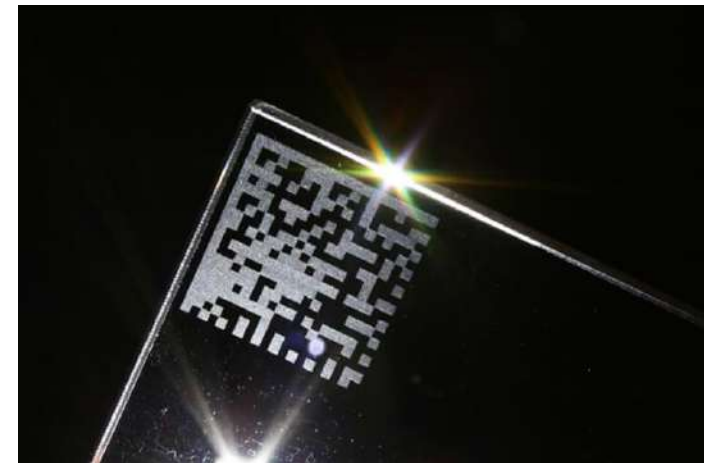
Matrices de trames (multicouches) pour niveaux de gris

MARQUAGE / ABLATION SUPERFICIELLE

- Ablation fs/ps
 - Enlèvement couche par couche (5nm - 10 μm ; mm^3/s), faible HAZ
- Rafales GHz (nouveaux laser fs)
 - Efficacité d'enlèvement $\uparrow$ (accumulation contrôlée)
- Paramètres clés
 - Fluence par pulse & par rafale
 - Nb de pulses/rafales, cadence (MHz–GHz)
- Applications
 - Scribing fin (0,1–10 μm de large /passe)
 - Cavités peu profondes (logos, QR, microstructures)



V-shape channel with a 15 μm width and a 20 μm depth.

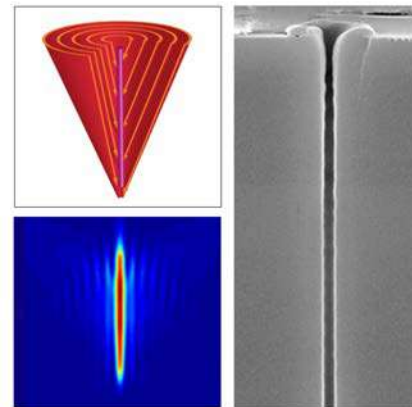
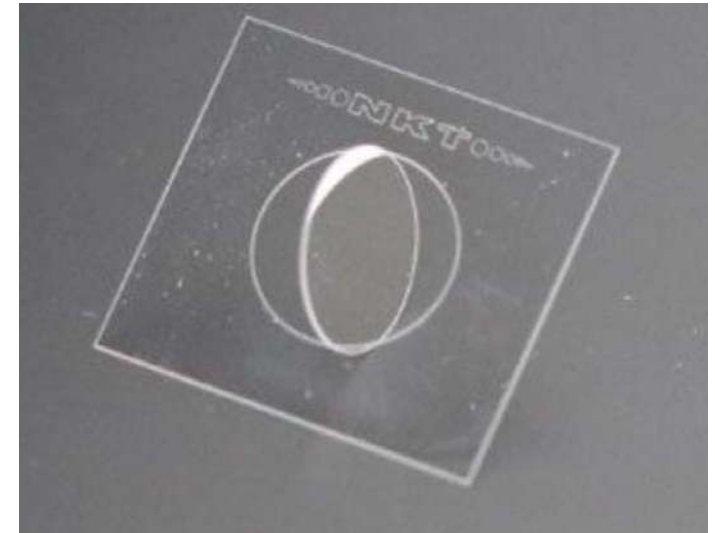


HAZ = Heat Affected Zone

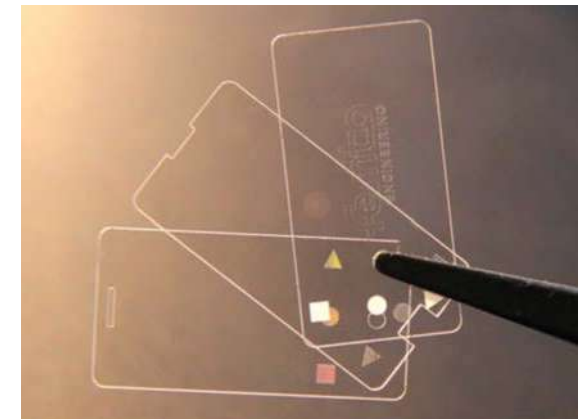


DÉCOUPE / SÉPARATION DU VERRE (FS)

- Approche surface
 - Scribe / ablation (chaud ou froid) + choc (jet, ultrason, flexion)
- Écriture en volume + clivage
 - Focalisation fs dans le bulk (NA adaptée, Bessel-Gauss beam)
 - Lignes de faiblesse (densification/vides) → clivage contrôlé
- Points de réglage
 - Profondeur focale vs épaisseur
 - Polarisation & vitesse de scan

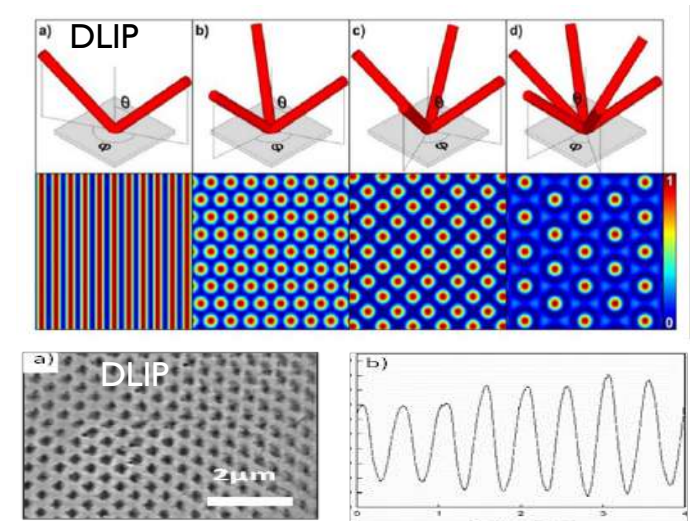
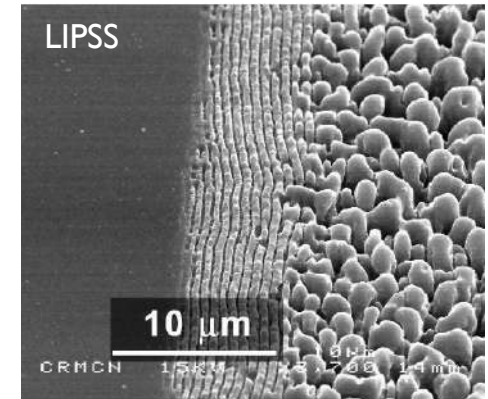
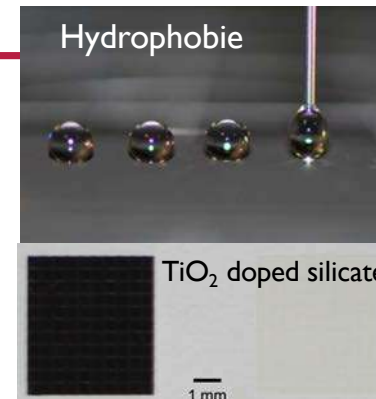


Bessel-Gauss beam



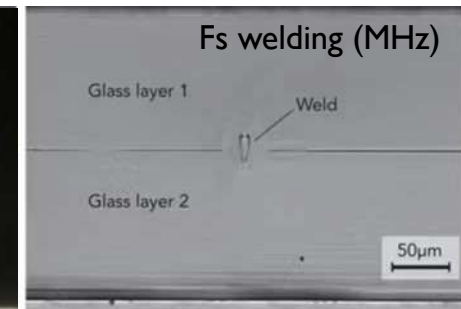
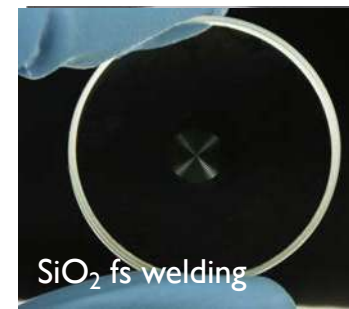
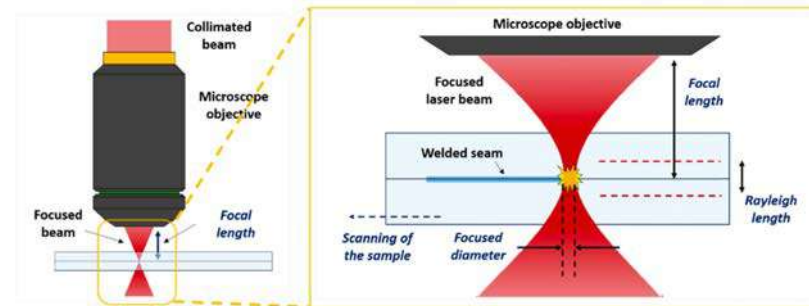
TEXTURATION FONCTIONNELLE (LIPSS & DLIP)

- LIPSS - Laser-Induced Periodic Surface Structures .
(auto-organisation)
 - Périodicité $\sim \lambda/2 \dots \lambda$ et sous- λ
 - Contrôle par polarisation, fluence, dose
 - Propriétés : super-hydrophobie, anti-givre, noir profond
(e.g. sub- λ ripples + nano-features trap light)
- DLIP - Direct Laser Interference Patterning
 - Réseaux 1D/2D en 1–2 tirs
 - Pas Λ typiques 0,5–50 μm
 - Très productif (grande surface)
- Intégration industrielle
 - Lignes prod kW, m^2/min
 - Réplication par moulage

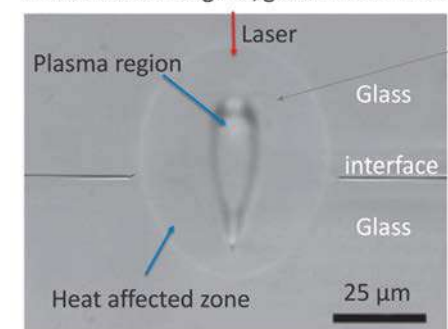


SOUDURE (CW, FS)

- Principe — L'absorption localisée augmente $T > T_{\text{soft}}$ ou T_{fusion} ; le refroidissement contrôlé crée une liaison permanente à l'interface.
- Plusieurs chemins d'absorption possibles —
 - (i) CO_2 à $10.6 \mu\text{m}$ (surface);
 - (ii) Proche-IR Cw avec couches absorbantes (métal, C);
 - (iii) Ultracourt (fs/ps): absorption non-linéaire à l'interface
- Contrôle de l'interface (écart, planéité), CTE induit des contraintes : le préchauffage et/ou la stratégie de scan atténuent cracks ou la biréfringence résiduelle.
- Applications — Fermeture hermétique (capteurs, MEMS, PICs), puce microfluidique, assemblages verre-verre/Si, emballage

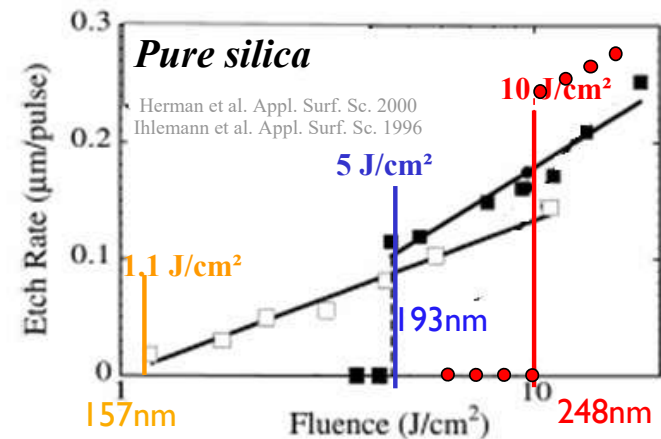
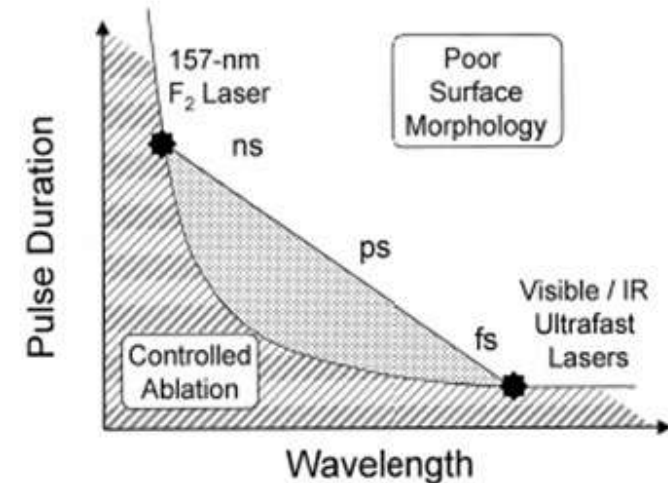


Cross-section of glass/glass fs laser weld



FENÊTRES PROCÉDÉ — ORDRES DE GRANDEUR

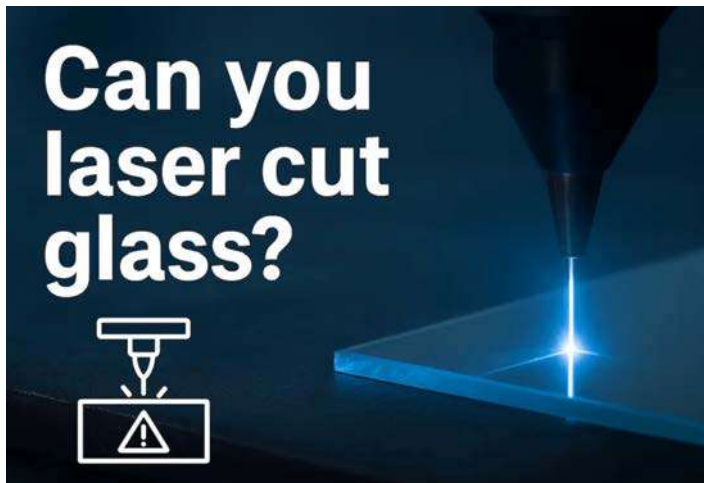
- Durées d'impulsion
 - 100 fs – 10 ps (qualité, HAZ ↓)
 - 5-10 ns : économique mais HAZ ↑
- Fluences
 - Proche du seuil d'ablation (0.5–10 J/cm², selon matériaux et longueur d'onde)
 - Plus bas pour contraste/marquage sans enlèvement
- Optique / Focalisation
 - NA 0.1–0.4 (marquage classique, soudure, ...)
 - NA ≥ 0.4 ou Bessel (écriture en volume)
- Cadence
 - 100 kHz – multi-MHz
 - Rafales GHz pour ablation efficace (nouveaux lasers fs!)



HAZ = Heat Affected Zone



CHECK-LIST



1) Définir l'objectif : contraste, profondeur, débit, rugosité



2) Choisir λ (UV/Vis/IR) et durée (fs/ps) selon matériau/couches/épaisseurs, HAZ



3) Balayer : fluence, recouvrement, cadence, pulses/rafale



4) Optimiser polarisation (LIPSS/DLIP) et la focalisation (NA, defocus, Bessel)



5) Métrologie : interférométrie, profilométrie, MEB/AFM, angle de contact, endurance, indice, absorption

HAZ = Heat Affected Zone



APPLICATIONS

« PRÊT-À-PRODUIRE »



Marquage noir profond sur verre (LIPSS, oxide + slight melt, nanoparticules photoreduction (Ag ou TiO_2))



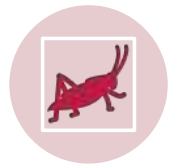
Décoating sélectif (ITO/Ag/oxydes) pour vitrages fonctionnels



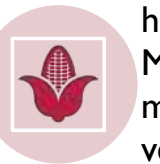
Marquage blanc, coloré: QR code & microgravure haute lisibilité,



Scribing & découpe via lignes Bessel + clivage

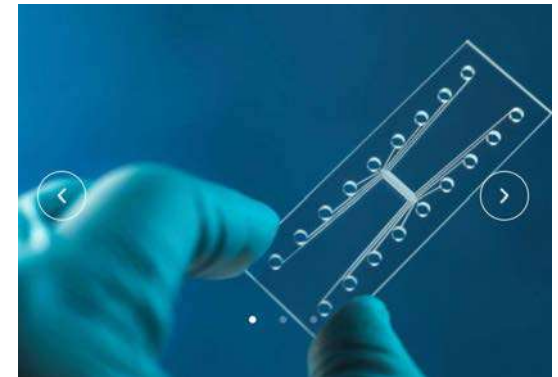
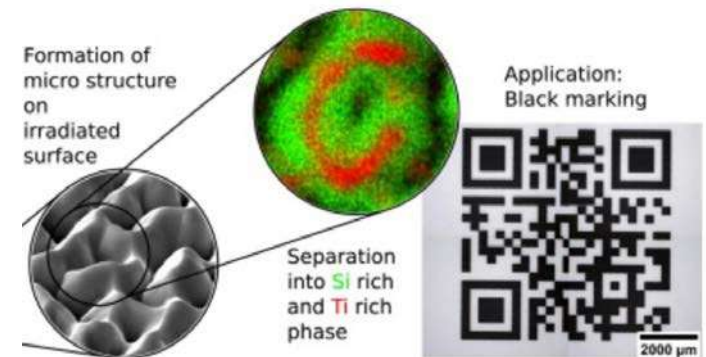


Fonctionnalisation de larges surfaces (hydrophobe, anti-givre, anti-bactérien)



Soudure : Scellage hermétique (capteurs, MEMS, PIC), puces microfluidiques, assemblages verre-verre/Si, soudures d'emballage

Schott SF11, N-SF11 (TiO_2 doped silicate)



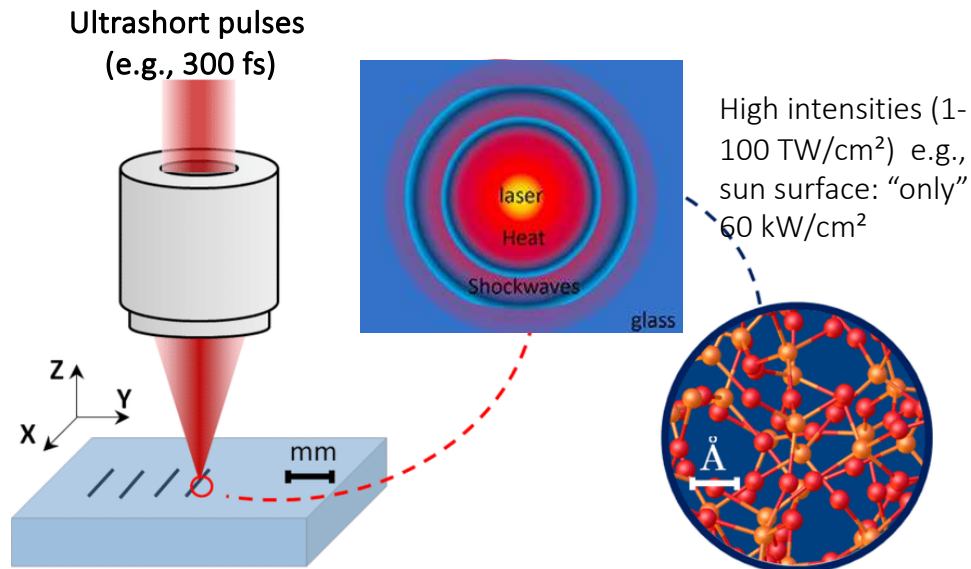
OUTLINE

- Introduction
- Basic concepts
- CO₂ laser
- Surface processing
- **Volume processing**
- Conclusions

Femtosecond laser direct writing (FLDW)

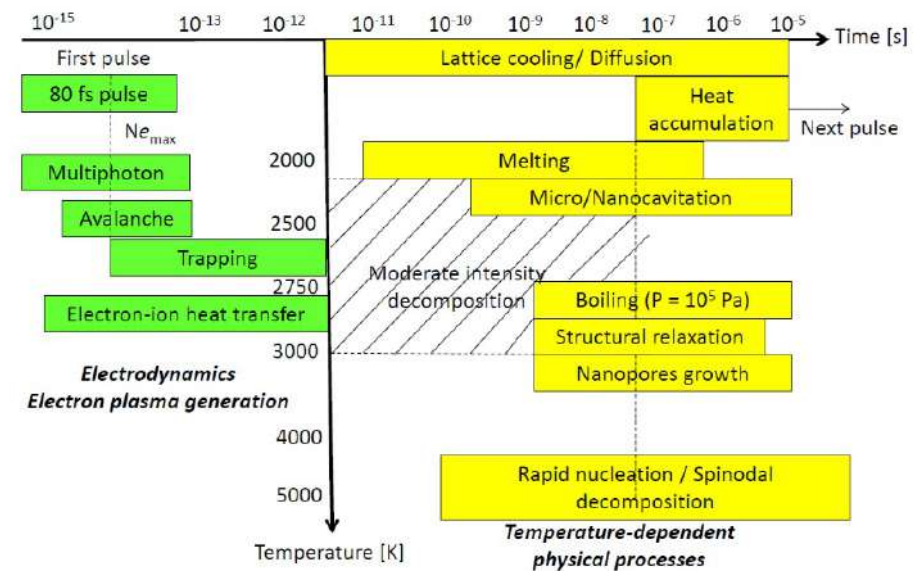
3D laser processing in glasses

Ultrashort pulse : 3D confined HP-HT micro-reactor



- High Temperatures (1000's K)
- High Pressures (>100's GPa)
- Each pulse contain $E >$ glass formation enthalpy
- Strong gradients (T, P, I, E_{dc})

Few words on fs – matter interaction

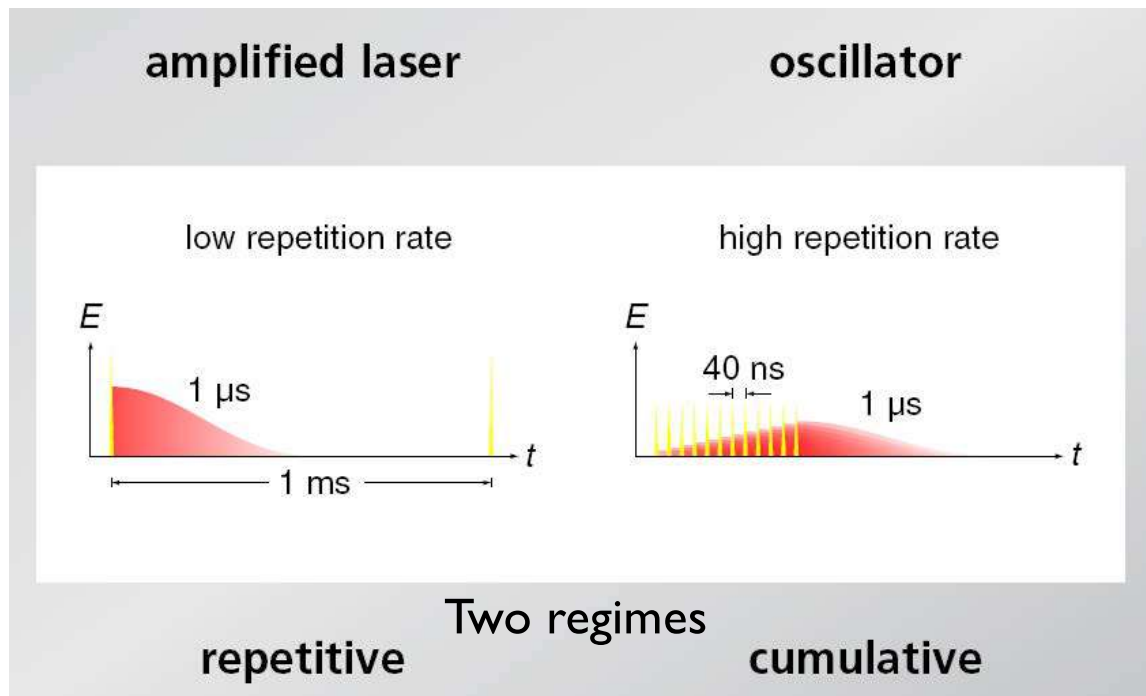


A plasma mediated process

Femtosecond laser direct writing (FLDW)

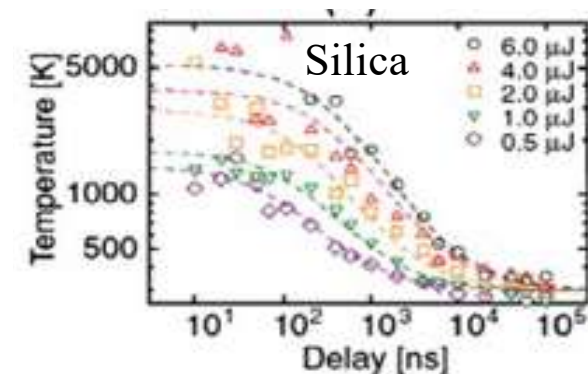
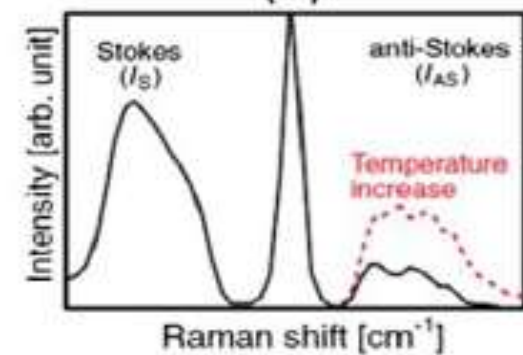
Femtosecond-materials interactions: Temperature effects

Thermal effects are somehow decoupled from optical one



Heat diffusion time a few μs in silica
So, relaxation time has to be taken into account

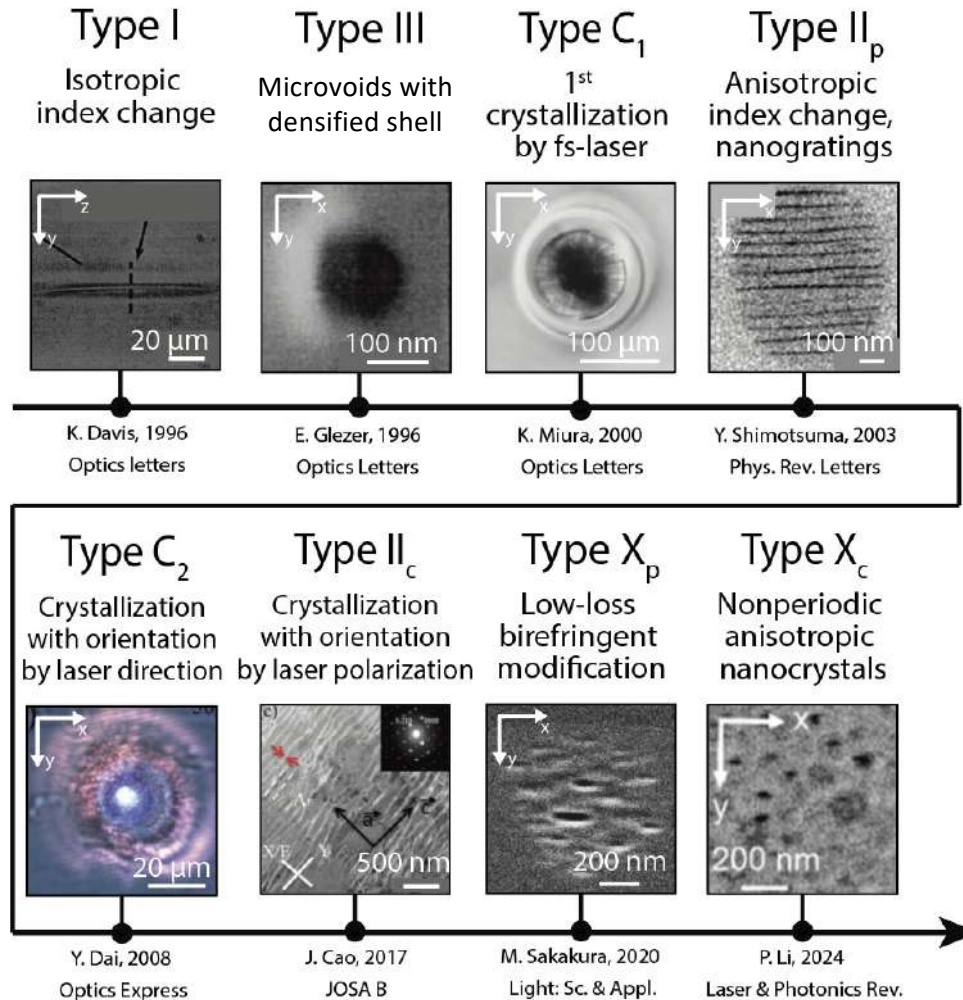
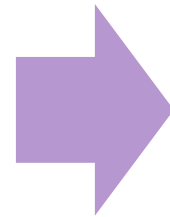
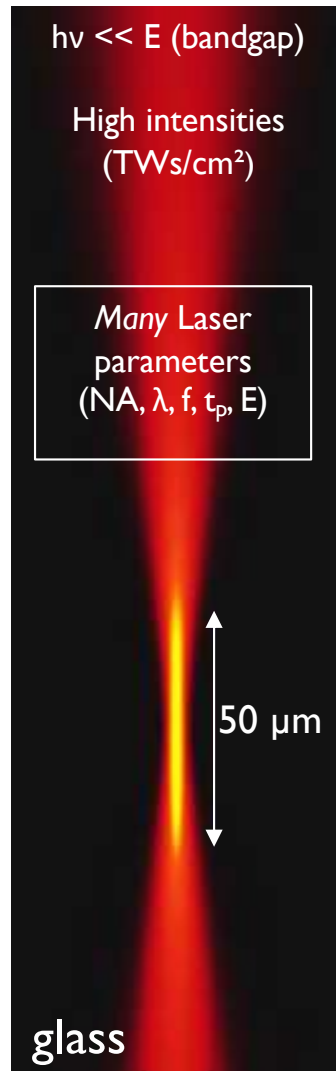
Single pulse



Fernandez, T. T. et al. (2018). *Progress in Materials Science* 94
Tomoki, Yoshino et al. (2012). *Japanese JAP* 51(10R)

Femtosecond laser direct writing (FLDW)

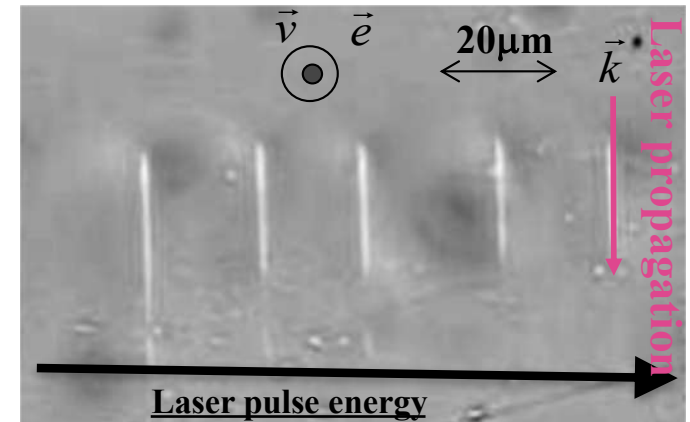
So many possibilities for glass microstructuring



TYPE I: 3D ENGINEERING OF « ISOTROPIC » INDEX CHANGES

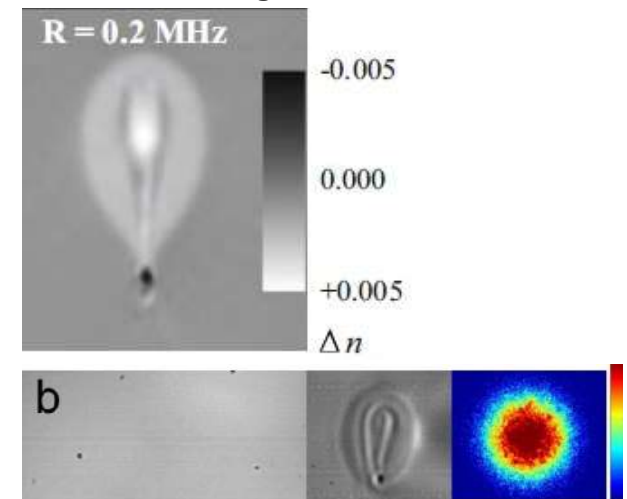
- Definition — Smooth, permanent refractive-index changes (Δn) with no cracks or voids, minimal stress birefringence
- Nonlinear absorption (multiphoton, avalanche) $\rightarrow$ free carriers; sub-ns relaxation.
- 3D localization (up to cm below the surface)
- “Voxel” longitudinal size (typ. 5-100 μm) but depends on pulse duration, NA, aberrations, pulse shaping (e.g. Bessel beam)...
- Heat accumulation at high rep-rates can increase Δn but must avoid damages, bubbles, ...

Cross section (xz) - Repetitive regime

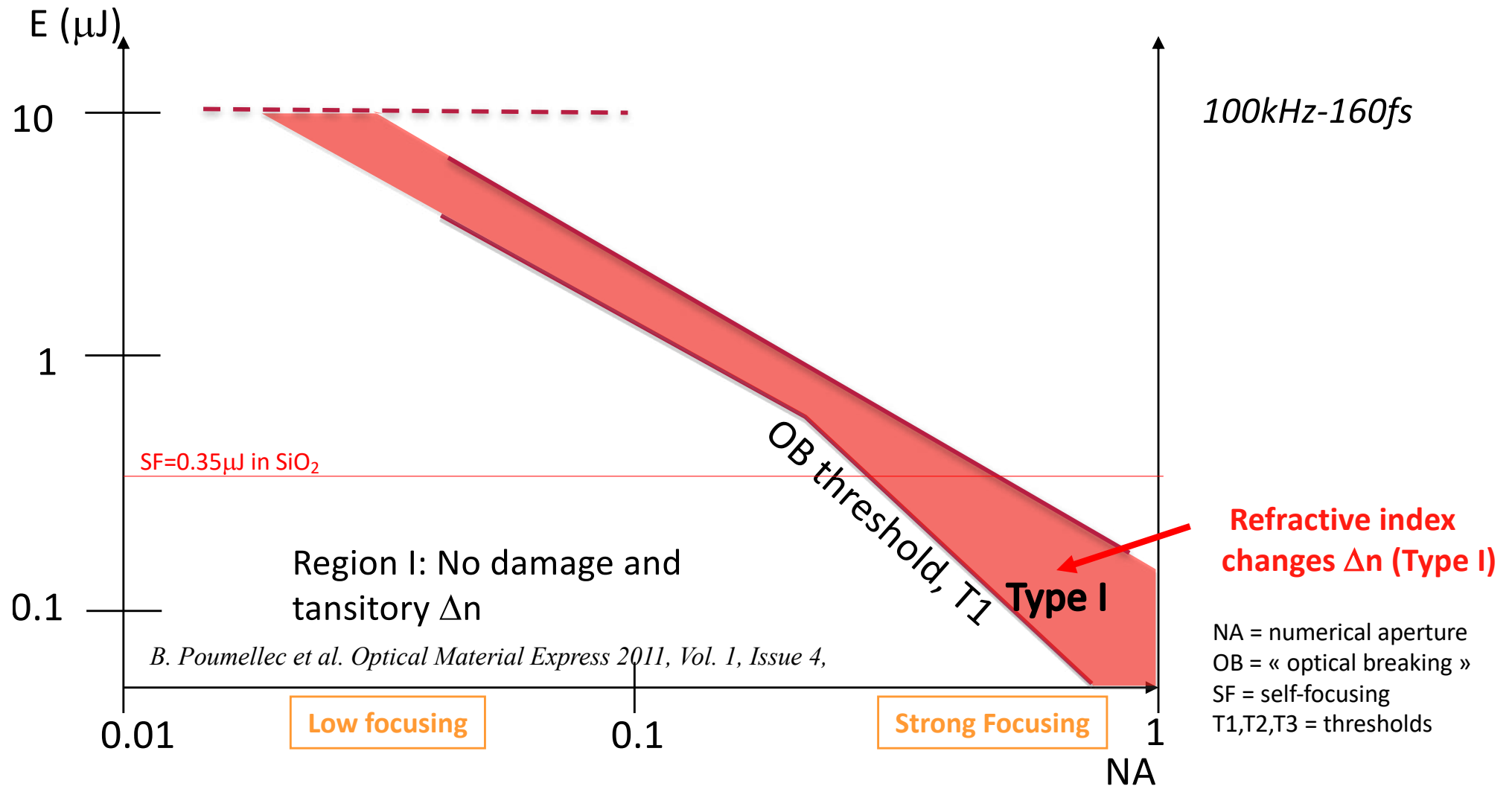


Lancry et al. OME (2011)

Cumulative regime



TYPICAL WRITING WINDOW (MATERIAL-DEPENDENT)

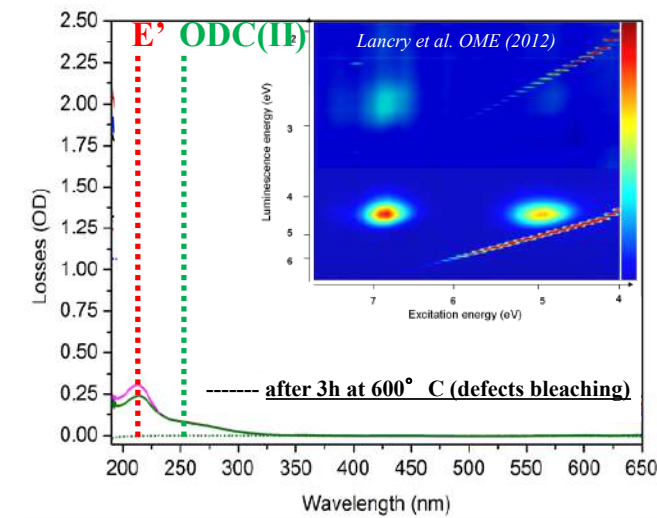


Pulse: 50–300 fs | λ : 800–1100 nm | NA: 0.3–0.65 (buried focus).
Rep-rate: 100 kHz–5 MHz | Scan: 0.1–20 mm/s |

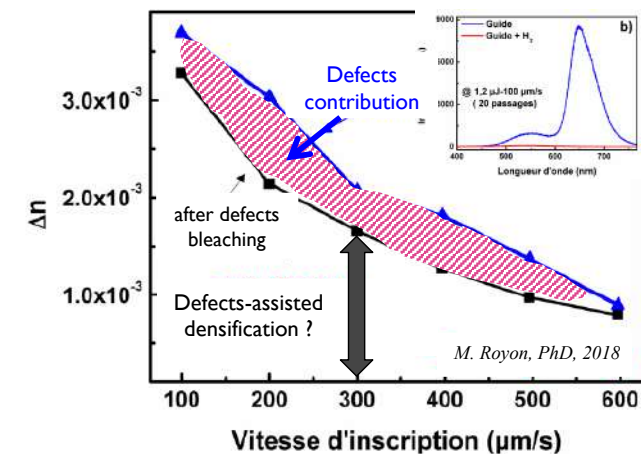
TYPE I: MECHANISMS

Point/color centers $\rightarrow \Delta n$

- Defects centers so called colors centers
- NBOHC, (Non Bridging Oxygen Hole center) ODC (Oxygen Deficient center), E'
- seen by PL (UV excited photo-luminescence) and OA (optical absorption)
- Points defects (color centers) contribution : $< 5.10^{-4}$. (nearly like in UV ns laser glass photosensitivity)
- Main contribution: « Defects-assisted » densification



Cf Cours N. Ollier



SiO_2 , 800 nm, 160 fs, 100 kHz, 100 $\mu\text{m/s}$, 0.05-0.4 μm ,



TYPE I: MECHANISMS

Densification $\rightarrow \Delta n$

- Vibrational (Raman or FTIR) signature of a permanent densification in SiO₂ cf cours de N Ollier

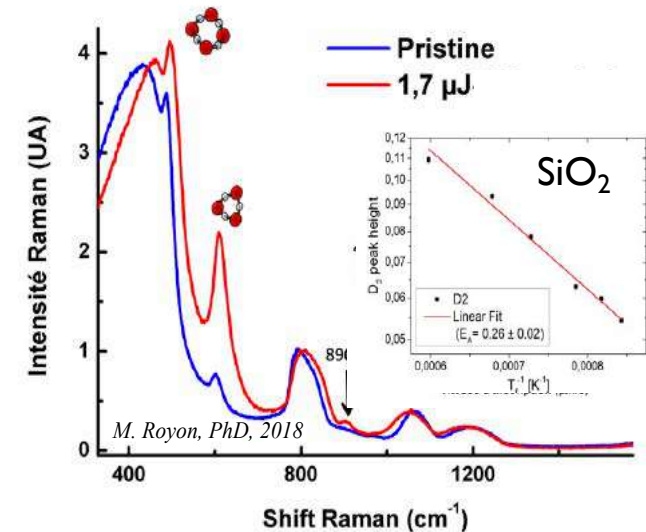
- Mechanical signature of glass densification (AFM, PSI, profilometer)

$$\Delta n_{aa} = -\frac{n^3}{2} \left[p_{11} \varepsilon_{aa}^e + p_{12} (\varepsilon_{\beta\beta}^e + \varepsilon_{\gamma\gamma}^e) \right] - \frac{(n^2 - 1)(n^2 + 2)}{6n} \cdot (1 - \Omega) \cdot \text{Tr}(\varepsilon^p)$$

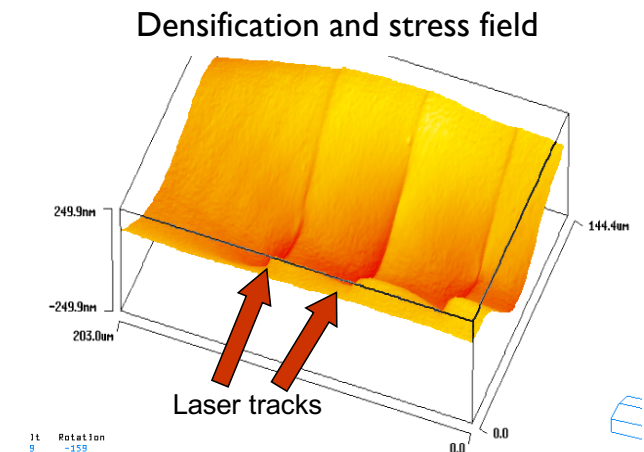
Photo-elastic relation Lorentz-Lorenz

- Type I origin: defects assisted densification likely assisted by heat effects (fictive temperature increase)

SiO₂, 800 nm, 160 fs, 100 kHz, 100 $\mu\text{m/s}$, 0.05-0.4 μJ ,



M. Lancry et al. Opt. Material Express, Vol. 1, (2011)



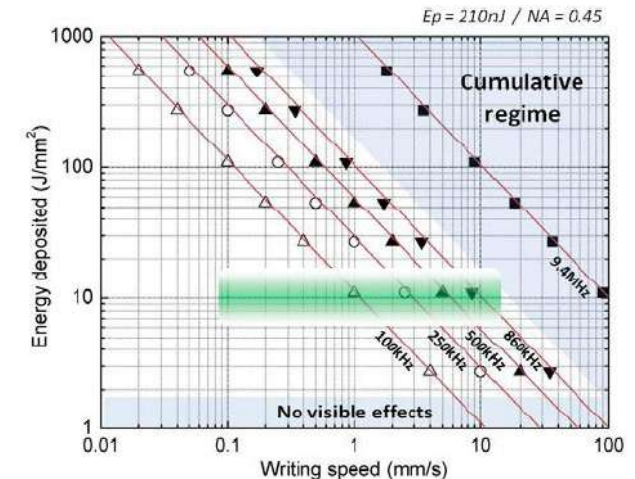
Poumellec et al. Opt. Express (2008)

TYPE I: TECHNOLOGICAL ASSETS

- $\Delta n \approx 10^{-4} - 10^{-2}$; (m-lines, near field refractometry, QPM, DHM, QPI, Brewster ...)
- Waveguides with low loss (< 0.5 dB/cm in fused silica).
- Ultralow losses down to 0,04dB/cm in SiO_2 ; 0,052dB/cm in Gorilla glass (Corning)

*J. Lapointe,, "Making smart phones smarter with photonics,"
Opt. Express 22, 15473-15483 (2014)*

- Industrial potential - single mode 3D waveguides :
 - 1 cm/s (repetitive regime)
 - 10 cm/s (cumulative regime)
 - 1m long – 30 cm/s reported in Gorilla glass



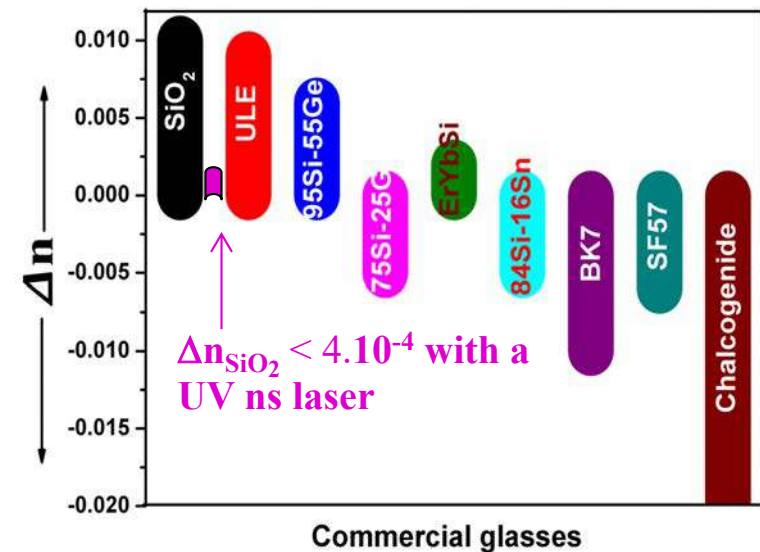
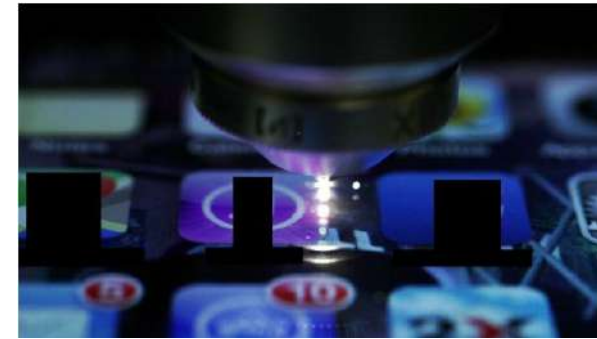
QPM (Quantitative Phase Measurements), DHM (Digital Holography Microscope), QLSI (Quadriwave Lateral Shearing Interferometry)



TYPE I: GLASS MATERIALS

VIS, NEAR-IR SPECTRAL RANGE

- Works in all glasses (contrarily to UV ns or cw light)
 - $\Delta n \approx$ up to 1×10^{-2} ; (even more in mid-IR glasses)
 - Nearly polarization-insensitive (low stress)
-
- Δn positive / negative vs composition (T_f related)
 - Best $\Delta n > 0$: high-purity fused silica but also mid-IR (see next slide).
 - Network modifiers (Na, K) or B, P lower thresholds but may raise losses; « optimize trade-off.»

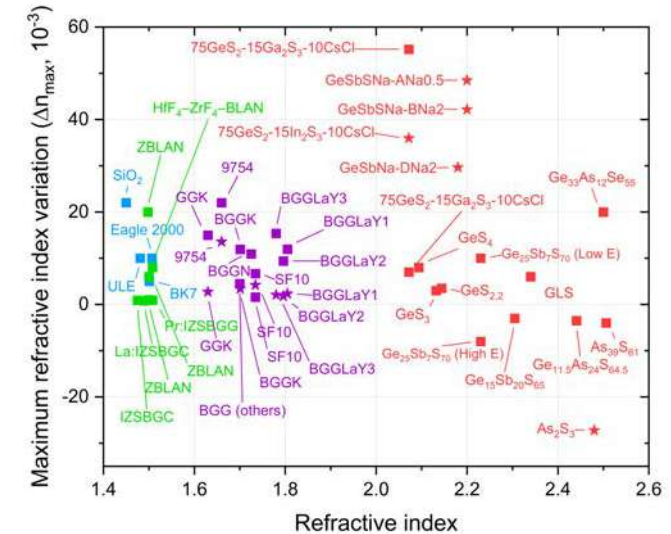


TYPE I: GLASS MATERIALS TOWARDS IR SPECTRAL RANGE

- Best Δn / high reflection loss in chalcogenide
- $\Delta n > 5 \times 10^{-2}$; in some Chalcogenide (GeS₂ based)
- Δn up to 2×10^{-2} ; in Fluoride
- Δn up to 2×10^{-2} ; in Heavy Oxide (BGG = Barium Gallo-Germanate) or Alumino-germanate

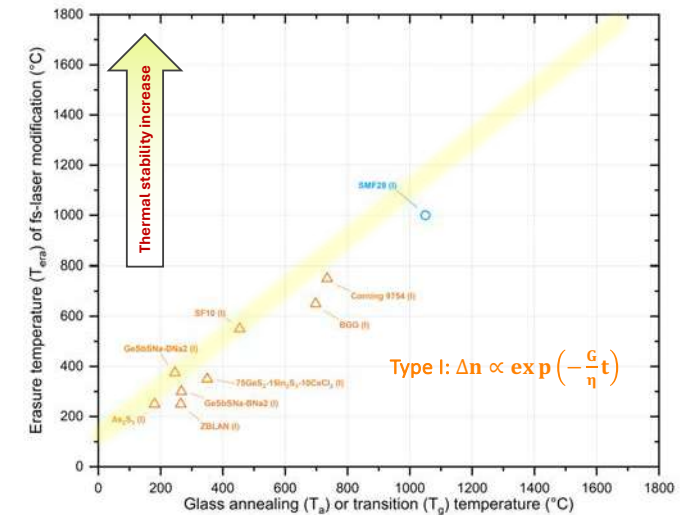
cf cours verres IR de Laurent Calvez

- Thermal stability mostly driven by glass structural relaxation $\eta(T)/G(T)$
- High thermal stability in heavy oxide glasses (Corning 9754 or BGG)
- Fluoride and Chalcogenide owns quite limited thermal stability (typ. below 300°C).

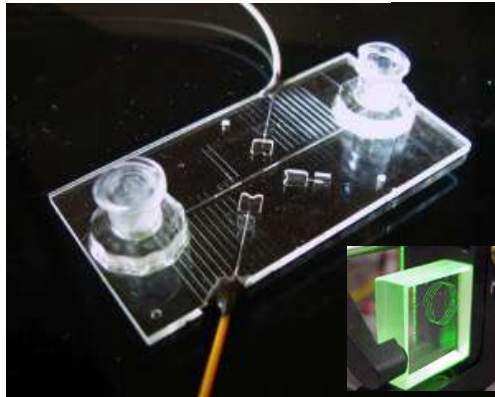


J Ari et al., Optical Materials (2025)

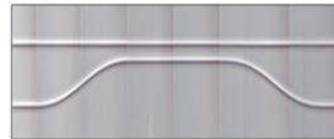
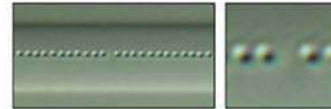
Compilation of results – Isochronal annealing experiments ($\Delta t = 30$ min)



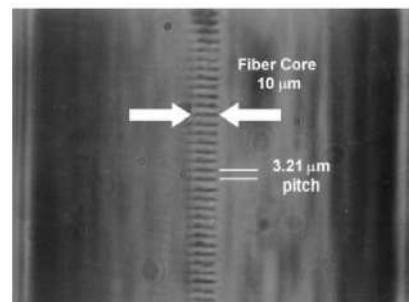
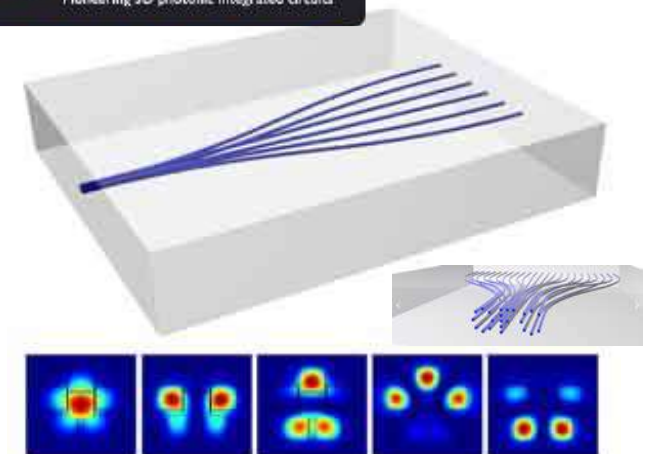
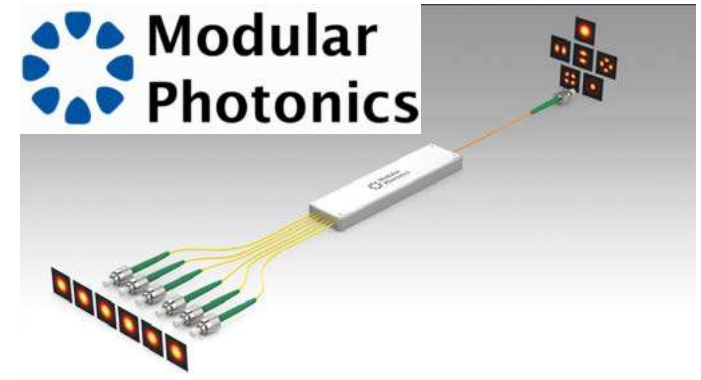
TYPE I: commercial applications



Fibre Bragg Gratings (FBGs)



Laser written photonic circuits



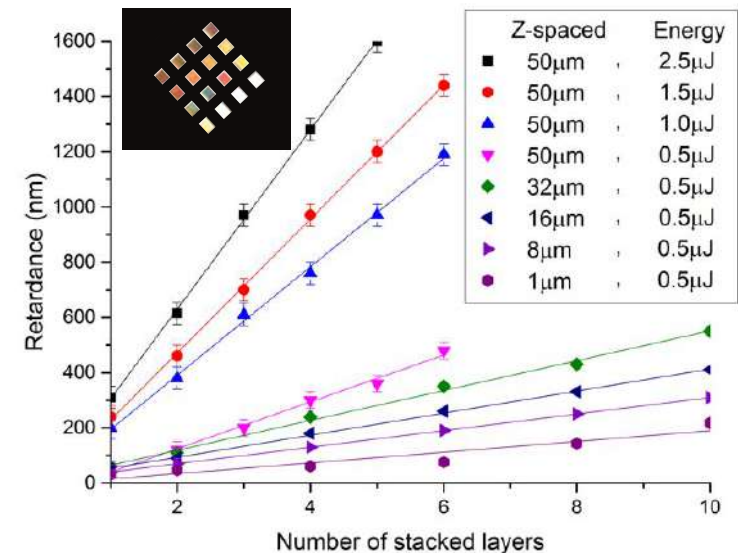
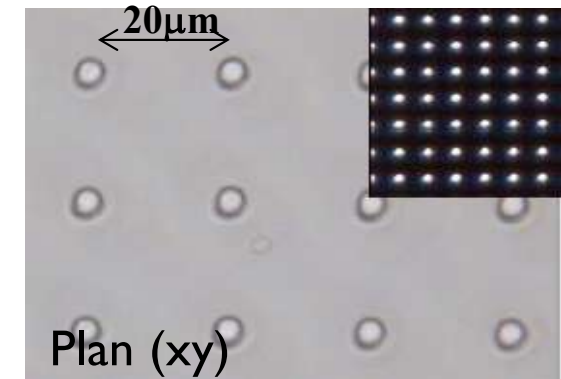
- 3D buried waveguides, splitters/ couplers, MZIs, laser waveguides.
- Optofluidic chips, lab-on-chip;
- Sensors (RI, temperature) via Δn control or Bragg grating period.
- Lenses (GRIN, Fresnel)
- Phase mask

TYPE II: 3D ENGINEERING OF « ANISOTROPIC » INDEX CHANGES

- Definition — Self-organized, sub-wavelength periodic index modulation (nanogratings), producing strong birefringence

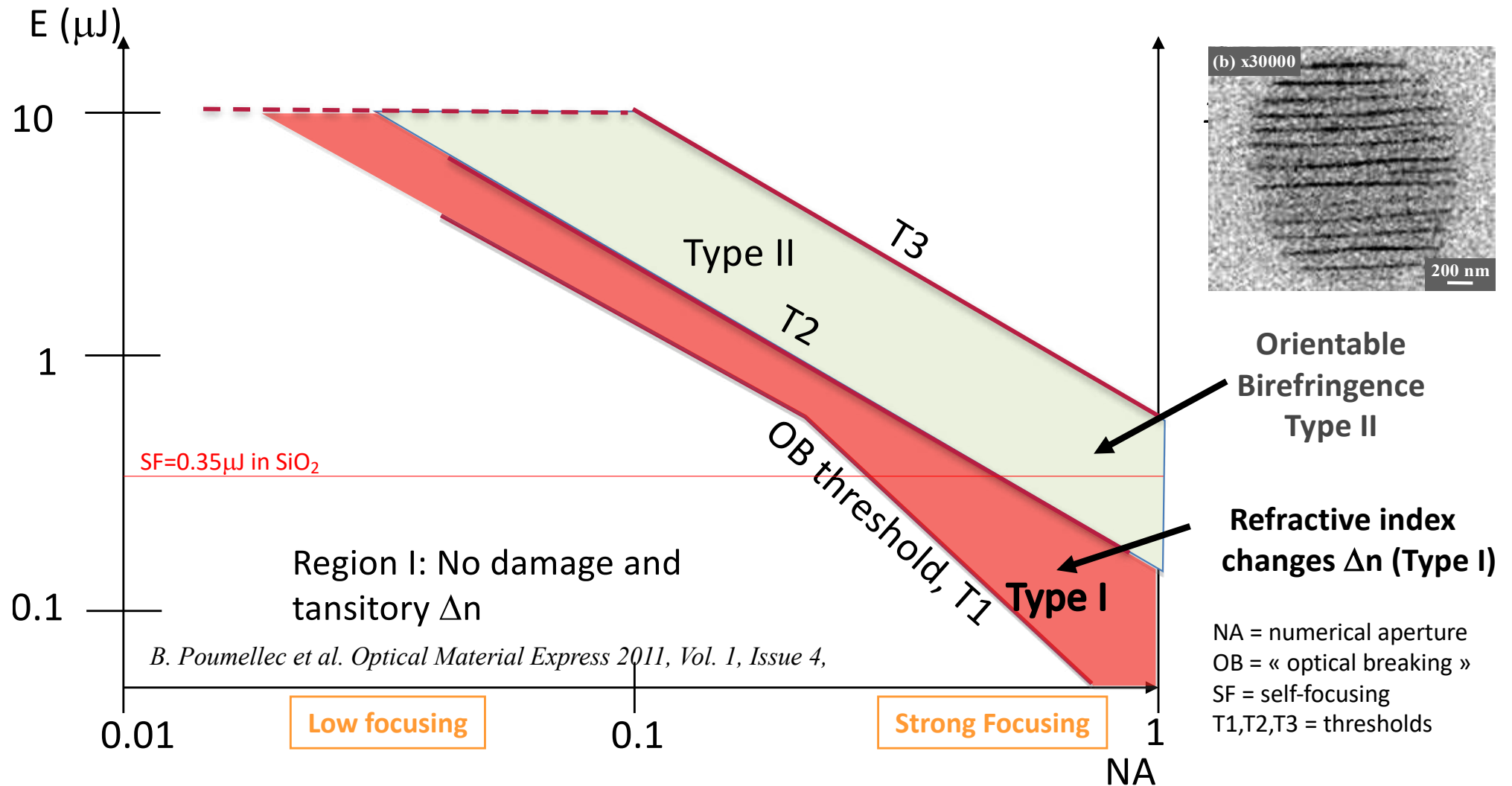
Main Features:

- Negative uniaxial birefringence ($n_e - n_o$) up to -10^{-2} in SiO_2
- Retardance $(n_e - n_o) \cdot L$ up to 600nm for a single layer
- On demand slow/fast axis orientation
- High laser damage threshold ($>23 \text{ J/cm}^2 @ 1064\text{nm}, 5\text{ns}$)
- High thermal stability (years at 800°C)
- Medium writing speed (up to 1 cm/s)



M. Lancry et al. Micromachines (2014)

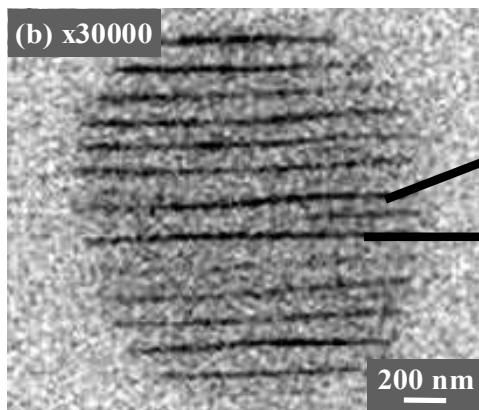
TYPICAL WRITING WINDOW (MATERIAL-DEPENDENT)



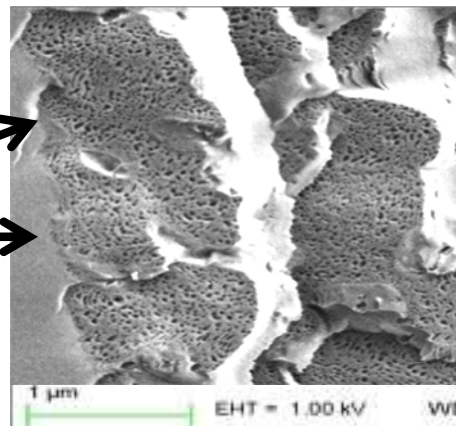
Pulse: 50–2000 fs | λ : 800–1100 nm | NA: 0.1–0.65 (buried focus).
 Rep-rate: 100 kHz–1 MHz | Scan: 0.1–10 mm/s |

TYPE II: MECHANISMS

- The smallest « self-organized » nanostructures created by light in the glass bulk (first observed in SiO_2)
- Ultrafast decomposition into $x\text{O}_2 + \text{SiO}_{2(1-x)}$ in less than $1\ \mu\text{s}$
- Need multiple pulses (from few pulses to thousands and more)
- Polarization oriented

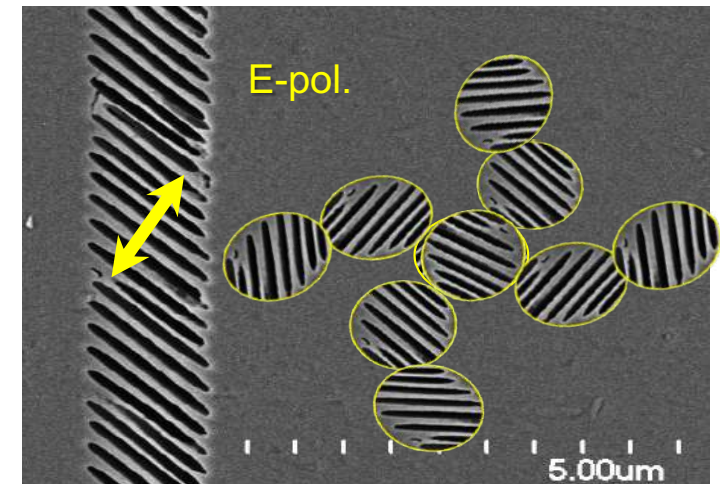


Shimotsuma et al. Phys. Rev. B 91 (2003)



M. Lancry et al. Laser Photonics Rev. 7 (2013)

Controllable orientation (in 3D)

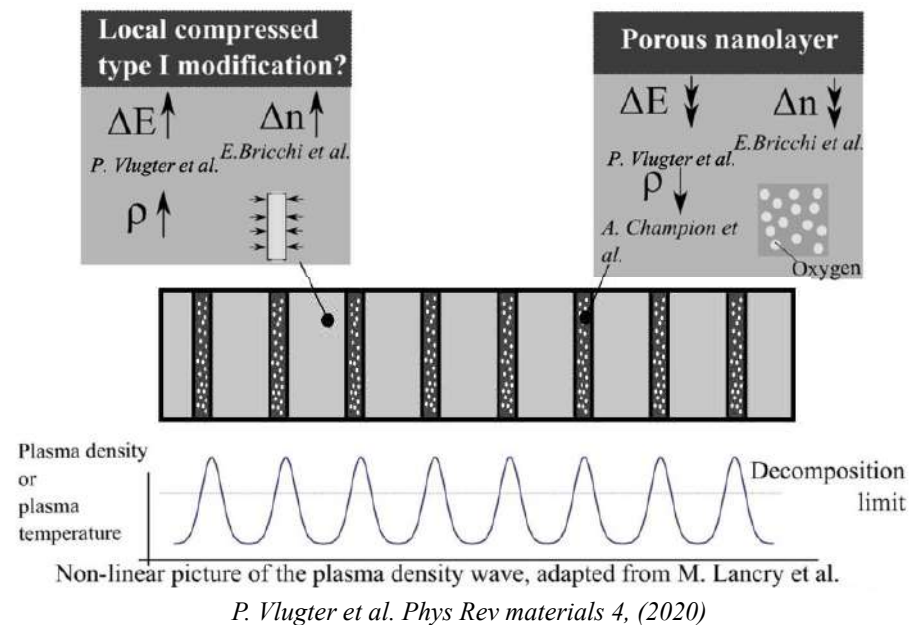
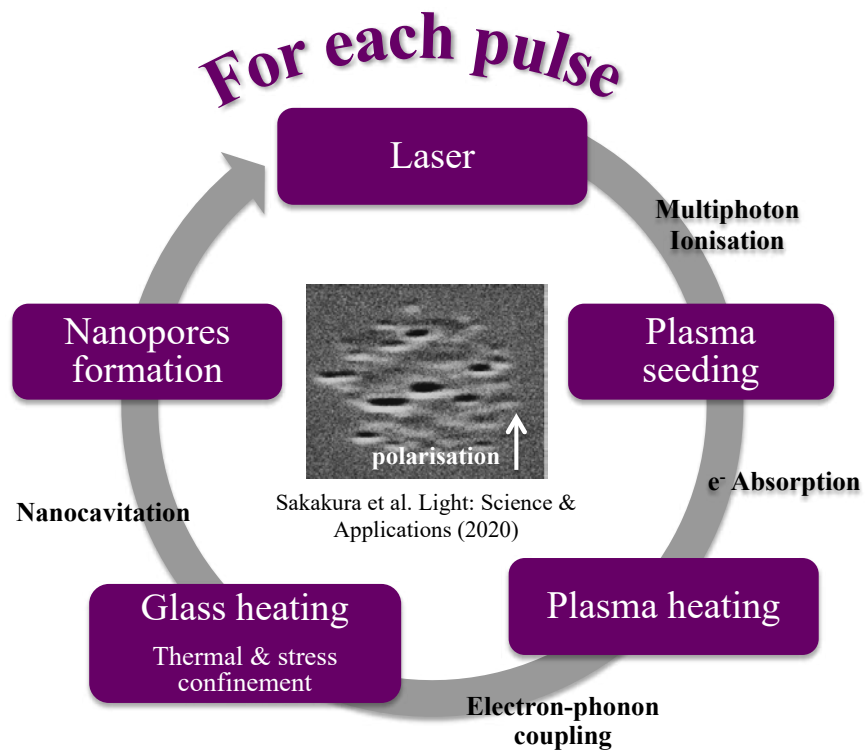


S. Juodkazis, Sci Report 7 (2017) J. Lu et al. APL (2023)

TYPE II: MECHANISMS

Nonlinear absorption → dense free-electron plasma spatial structuration

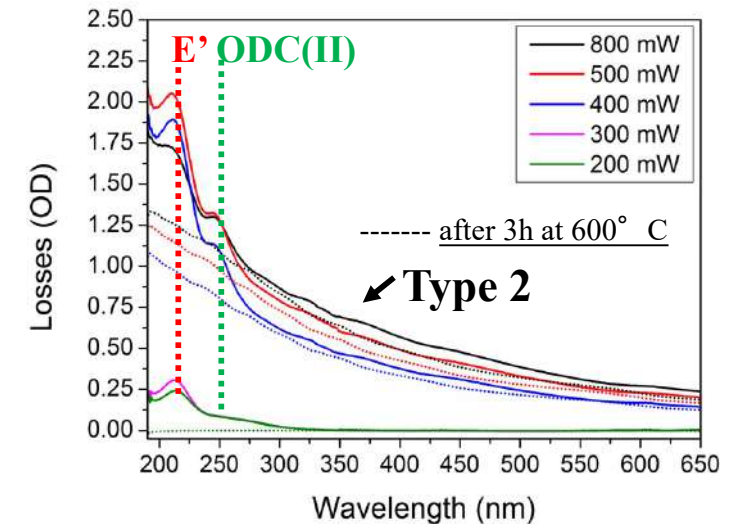
Interference/scattering of the writing field and plasma mediated nanocavitation process



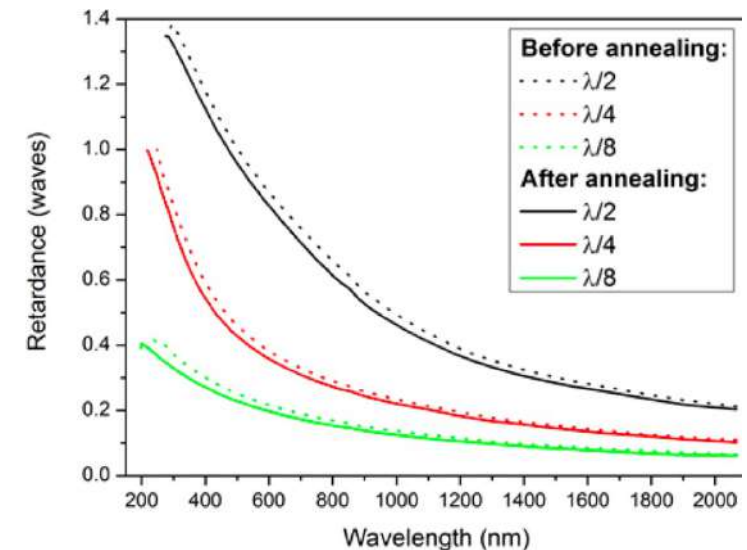
TYPE II: TECHNOLOGICAL ASSETS

- Strong light scattering remains e.g. 75% transmission for $\lambda/2$ @ 532nm but 90-95% in the near-IR
- Anisotropic and oriented light scattering
- Better Transmission after heat treatment at 600°C: erasure of points defects/color centers

- Broadband birefringence,
- Strongly chromatic
- Only slight birefringence decrease after heat treatment (typ 3%)
- Defects play a role in the “seeding mechanisms” but low contribution to the birefringence



M. Beresna et al Appl Phys Lett 103 (2013)



TYPE II: GLASS MATERIALS

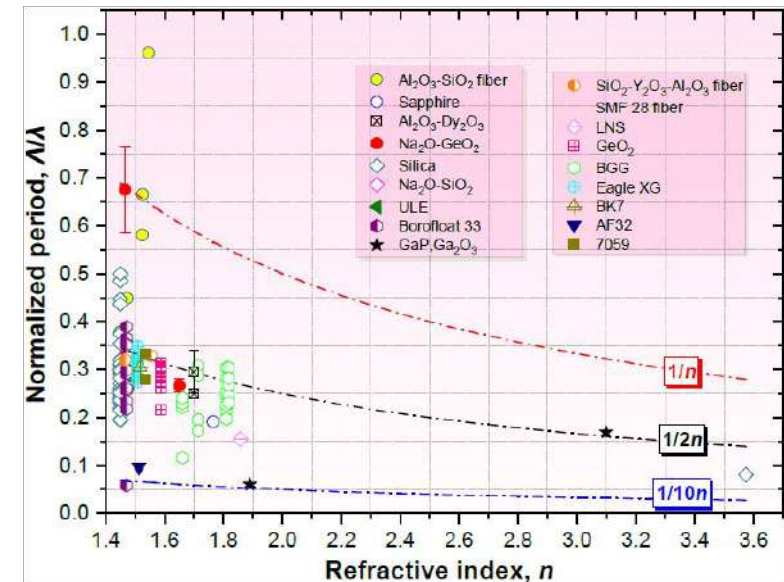
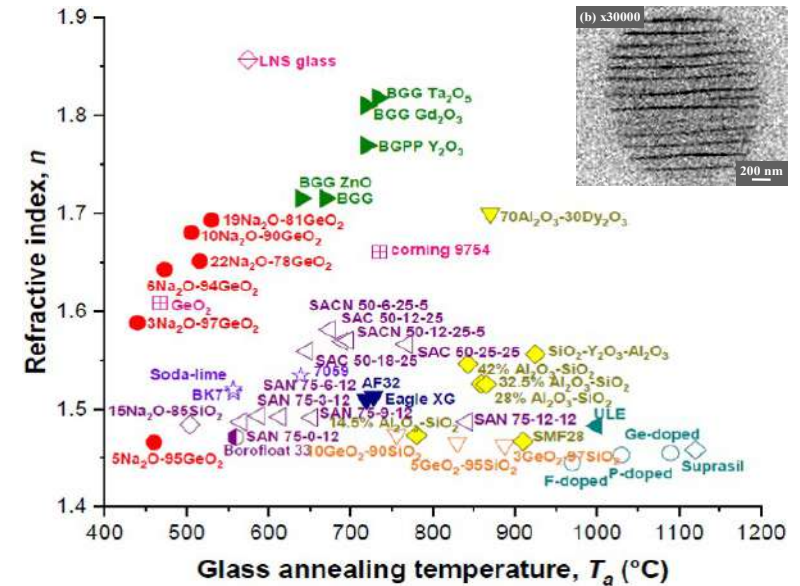
- Initially only in silica / doped silica
- Nanogratings can be achieved in any oxyde glasses !
- Looking forward to see them in Fluoride, chalcogenide etc... ???

- Nanogratings period $\Lambda/\lambda_{\text{laser}}$

$$\lambda/n < \Lambda < \lambda/10n$$

- Period Λ depends on the concentration C_i of “inhomogeneities” (thus glass composition i.e. AF32, B33)
- Higher pulse, number higher C_i shorter Λ !

A Rudenko et al. Phys Rev B (2016)

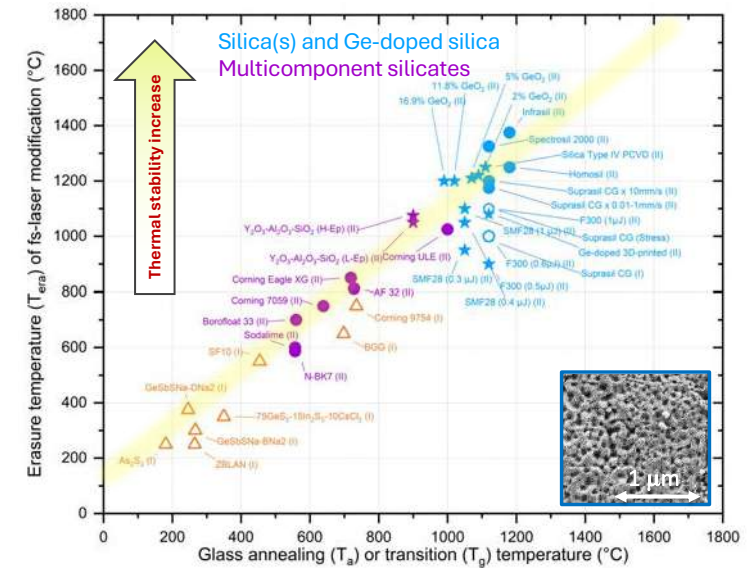
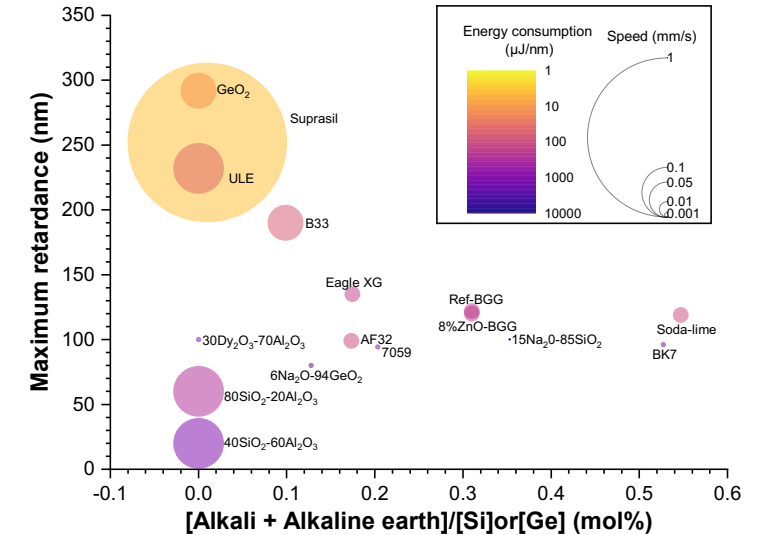


TYPE II: GLASS MATERIALS

- Up to now, SiO_2 best candidate to unlock industrial potential (low energy consumption, high writing speed, high retardance)
- Alkali and alkaline earth are detrimental for nanogratings formation.
- Alumino-borosilicate: I:I cationic ratio between Ca and Al is better (charge compensation) : viscosity(T) driven !
- Thermal erasure can be modeled using a Rayleigh-Plesset model
- Viscosity driven (mostly) but also related to σ and " Φ " init
- So SiO_2 would be the best performer ? Not fully true ;)

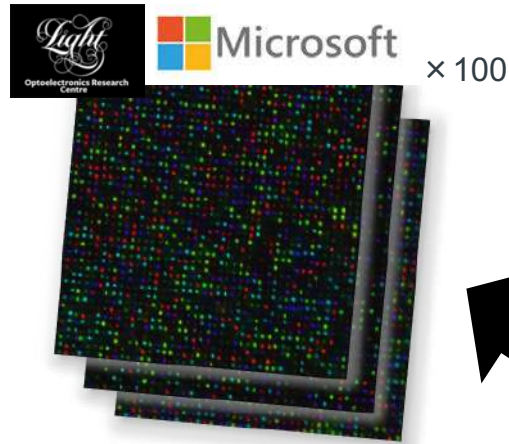
cf cours fibres de Maxime Cavillon

Xie et al., *Applied Optics* (vol. 62, 2023), Shuen et al., *Sensors* (vol. 20, 2020)

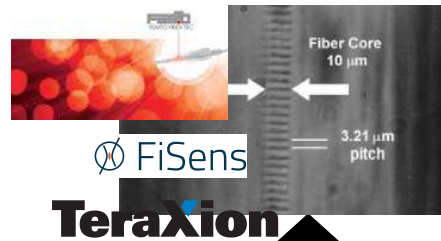


TYPE II: commercial applications

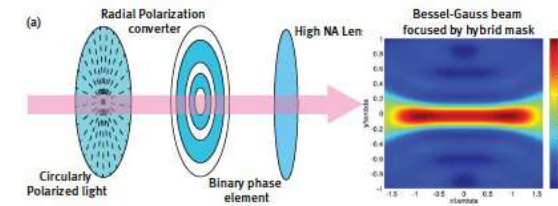
5D optical data storage



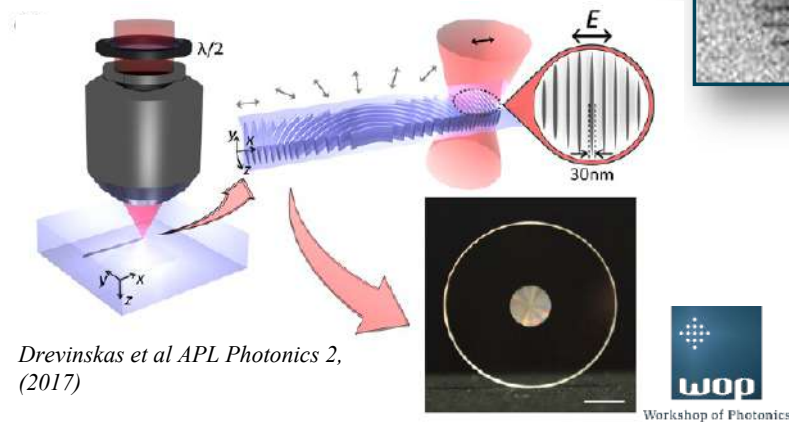
High T fiber sensors



2D- 3D geometric phase elements



3D space variant birefringent devices



E



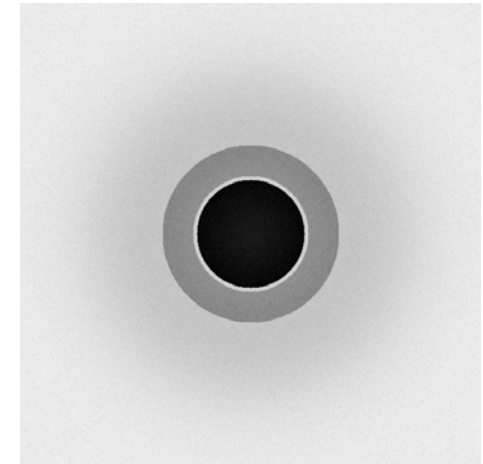
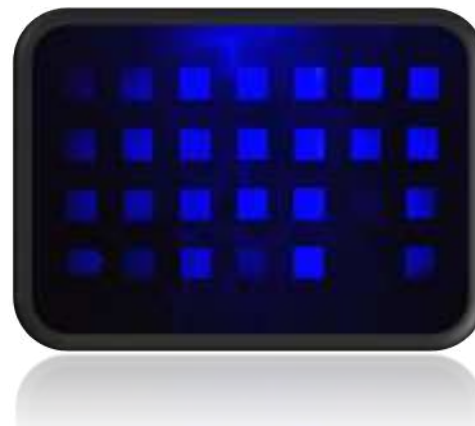
Optomechanics and optofluidics
using selective laser etching

TYPE III — VOIDS & 3D ENGINEERING OF LIGHT SCATTERING

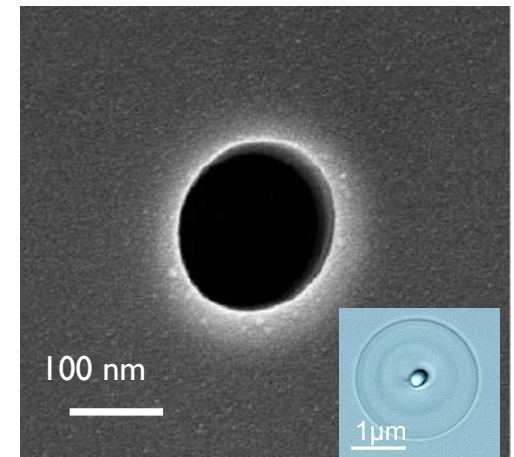
- Definition — Nanoscale to microscale voids/cavities formed by micro-explosions; surrounded by densified shell and stress.

Main Features:

- Single or chains (“pearl strings”)
- Very high index changes (Δn decreases -0,5 in SiO_2 !)
- Strong light scattering
- High writing speed (typ. up to few m/s)
- Residual stress-birefringence and possible microcracks



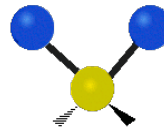
Schematic cross-section: central cavity with densified shell and stress halo



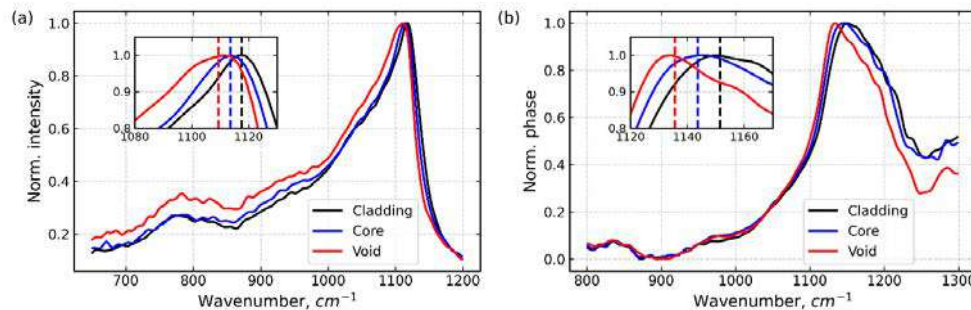
Gamaly et al. *PHYSICAL REVIEW B* 73, 214101 2006

TYPE II: MECHANISMS

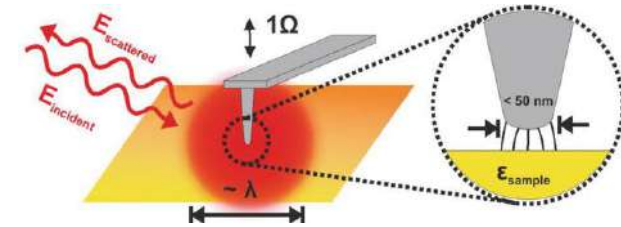
- Surface topography (e.g. AFM) depression : mechanical signature of densification
- Vibrational signature of densification using IR s-SNOM at $\sigma = 1130 \text{ cm}^{-1}$ (Si-O-Si asym stretching),
- Phase contrast at 1130 cm^{-1} : Δn increase



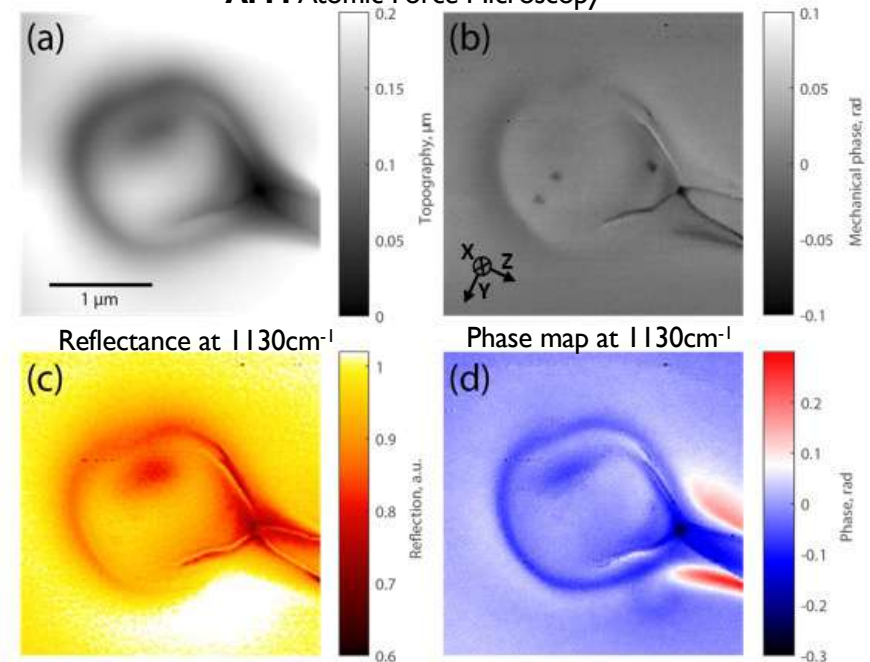
N. Shchedrina 2025, <https://doi.org/10.1111/ijag.16706>



s-SNOM scattering-type scanning near field optical microscopy made at SOLEIL Synchrotron (SMIS)



AFM Atomic Force Microscopy



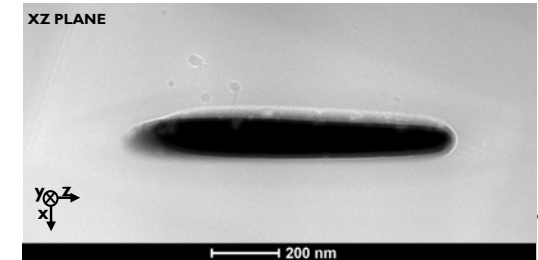
M. Sosa et al. *Nanomaterials* 2024, 14, 1228.

- Densification is estimated 6% which involved High Pressure (>10 GPa) mechanisms

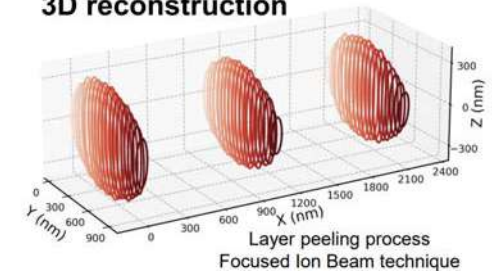


TYPE II: MECHANISMS

- Generation of a dense plasma (multiphoton ionization), with an electron density near or above the critical density ($\sim 10^{21}$ electrons/cm³).
- Localized energy deposition (ultrafast heating, 1–10 ps).
- Supersonic material expansion (CTE-driven) creating a shock wave.
- Void formation (a cavity with a densified surrounding shell).

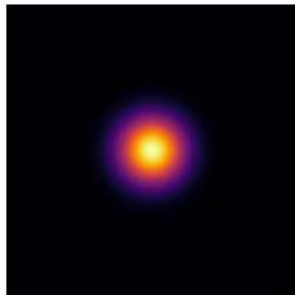


3D reconstruction

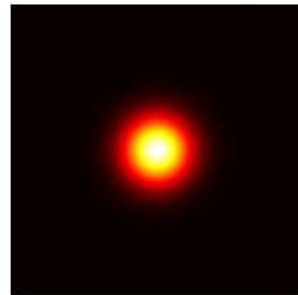


M. Sosa et al. Nanomaterials 2024, 14, 1228.

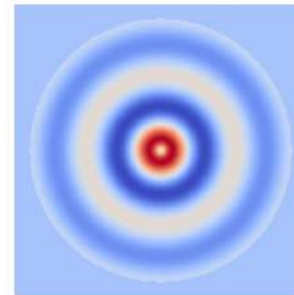
1. Plasma generation
(multiphoton ionization)



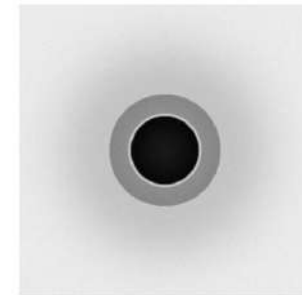
2. Energy deposition
(local heating)



3. Microexplosion
(shockwave & expansion)



4. Void + densified shell
(Schematic cross-section)



TYPE III: TECHNOLOGICAL ASSETS

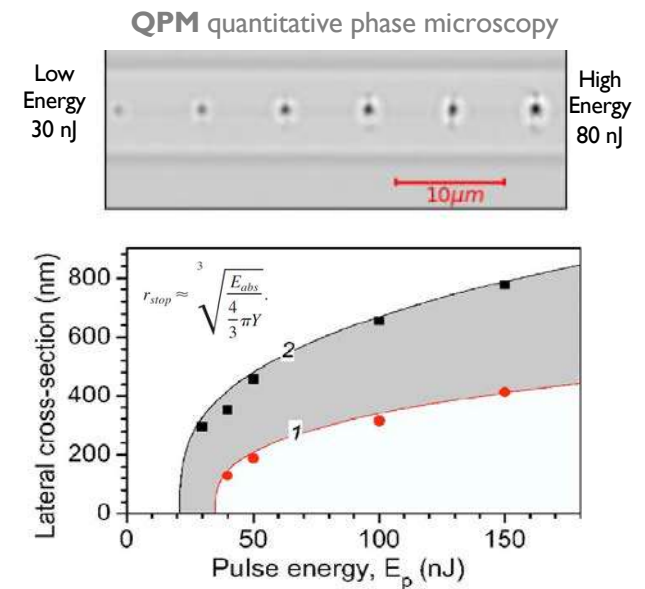
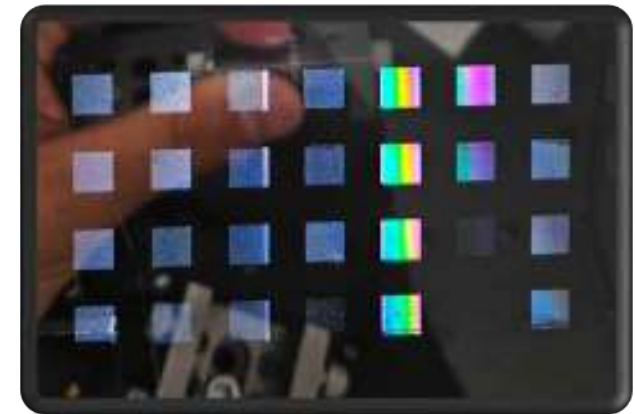
- High NA > 1 (oil immersion objective): single void



- Low NA : voids chain



- Single pulse writing → high writing speed e.g. 100 mm/s using 100 kHz laser repetition rate
- Strong light scattering (Mie regime) → forward scattering by voids: $\lambda/10 < \text{size} < \lambda$
- Void radius and shock-affected zone scale up with absorbed energy (and elastic modulus)
- High Δn shell (few 10^{-2}) : densified shell (6% $\Delta \rho/\rho$)



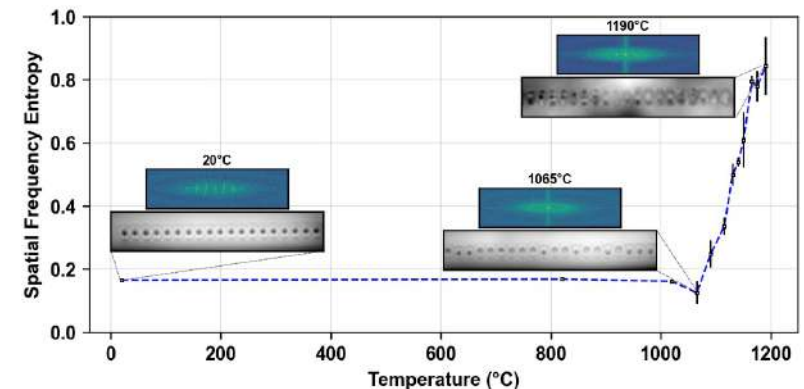
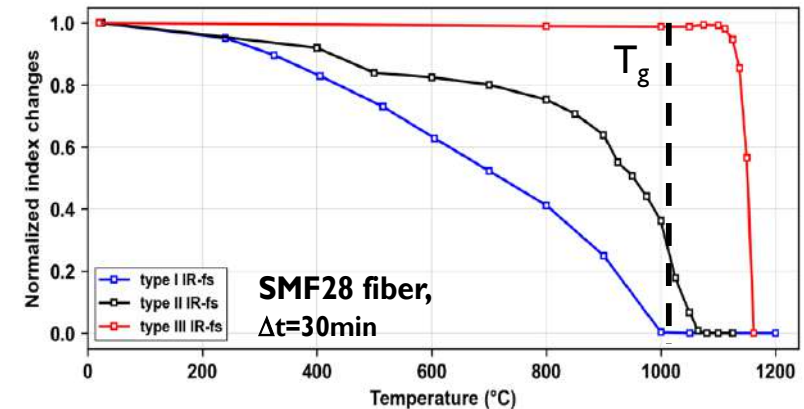
Gamaly et al. PHYSICAL REVIEW B 73, 214101 2006

TYPE III: TECHNOLOGICAL ASSETS

Thermal stability

Around T_g

- Type I: Δn erased (defects, densification relaxation).
 - Type II: Nanogratings collapse $\rightarrow$ major birefringence loss., small nanopores radius closes faster (e.g. Rayleigh-Plesset).
 - Type III: Voids still stable i.e. larger voids persist longer at the same T .
-
- Below T_g (~ 1050 °C). $\Rightarrow$ voids remain stable
 - Near T_g : viscous flow becomes significant $\rightarrow$ voids round and shrink; smaller R closes faster ($\Delta P \approx 2\gamma/R$),
 - Above T_g : partial/total closure vs regrowth can occur depending on voids shape and internal pressure (PO_2)

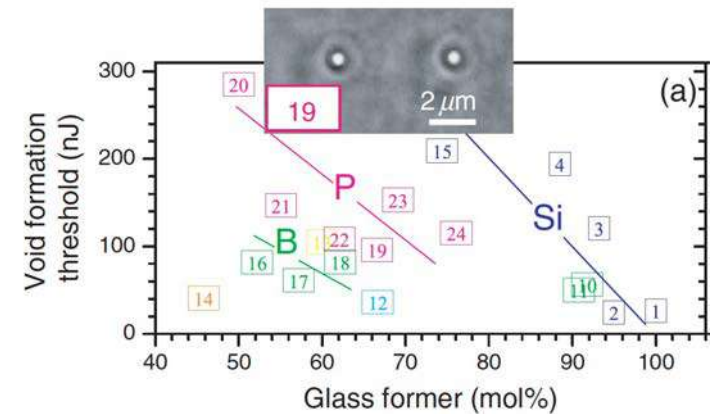


M. Sosa et al., Micromachines (2025)

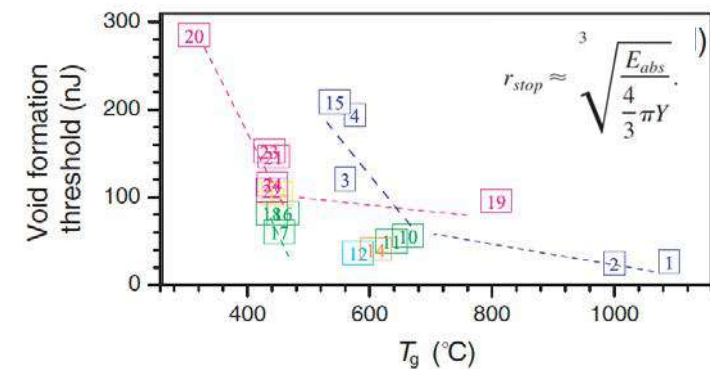
Scaling: $\tau \approx \eta(T) \cdot R / \gamma$ (η : viscosity; R : void radius; $\gamma \sim 0.3 \text{ N m}^{-1}$)

TYPE III: GLASS MATERIALS

- **Lower threshold** with higher network-former fraction ($\text{SiO}_2/\text{P}_2\text{O}_5/\text{B}_2\text{O}_3$), T_g , and Young's modulus (trend across 20+ glasses)
- **n & n₂ increase** (e.g. heavy oxides) → stronger self-focusing → less confined energy due to filamentation and clamping
- **Shell nature depends on chemistry:**
 - SiO_2 or GeO_2 shell mimics high-P high-T amorphous phase (fictive T and fictive P changes),
 - Al_2O_3 - SiO_2 shell is partly crystallized
- **Clean, “spherical”, voids** — choose network-former rich glasses (high-purity silica, some phosphates/borates), use high NA to counter self-focusing; low-energy (just above threshold)

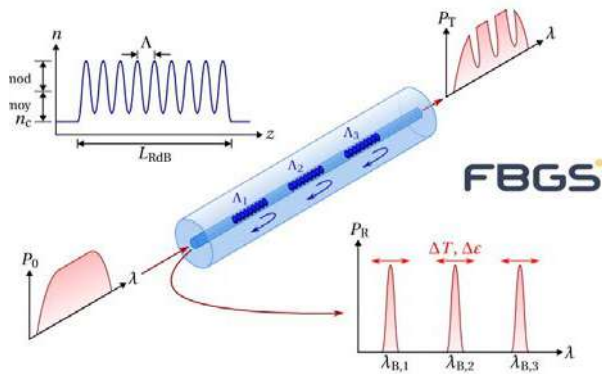
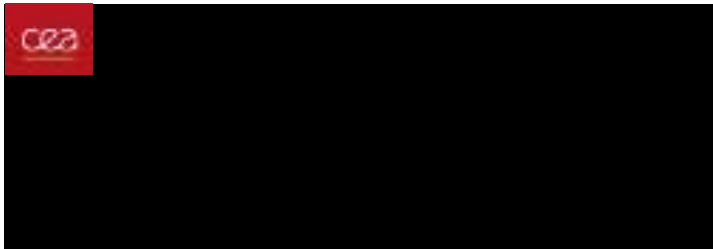


Hashimoto et al. New Journal of Physics 9 (2007) 253



TYPE III: towards applications

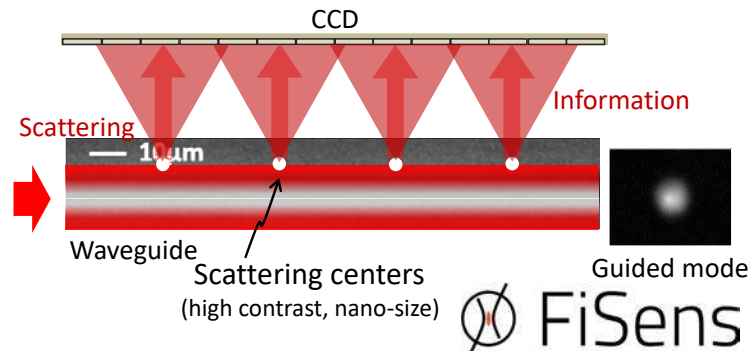
Optical Fiber Sensors



λ_B sensitive to temperature (or σ , ϵ)
Sensitivity: $\frac{d\lambda_B}{dT} \approx 11,2 \text{ pm}/^\circ\text{C}$ at $\lambda \approx 1550 \text{ nm}$

cf cours fibres de
Maxime Cavillon

Spectrometer function embedded on a glass chip!



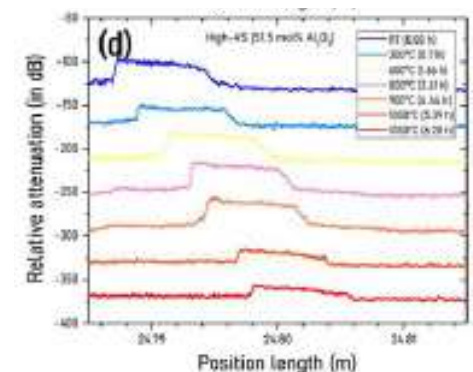
3D marking/QR-like, optical Data Storage



Rayleigh/Mie enhanced scattering, directional scattering



Vision glazing applications



Scattering based sensors



OTHERS OPTICAL MODIFICATIONS IN VOLUME

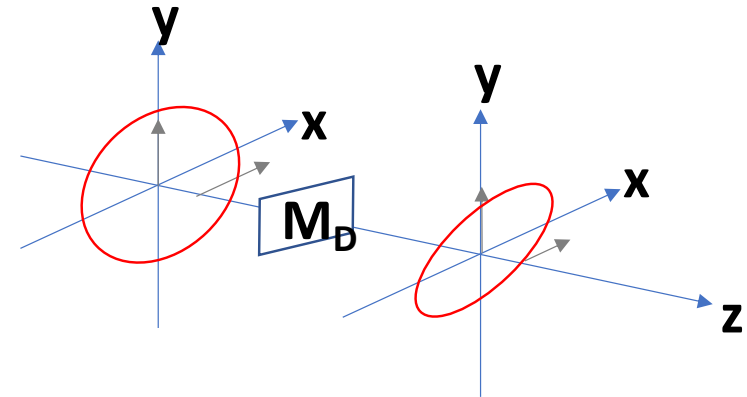
DICHROISM • NANOPARTICULES • CRYSTALLIZATION • OTHERS



DIATTENUATION – DICHROISM (LD, CD)

Definitions

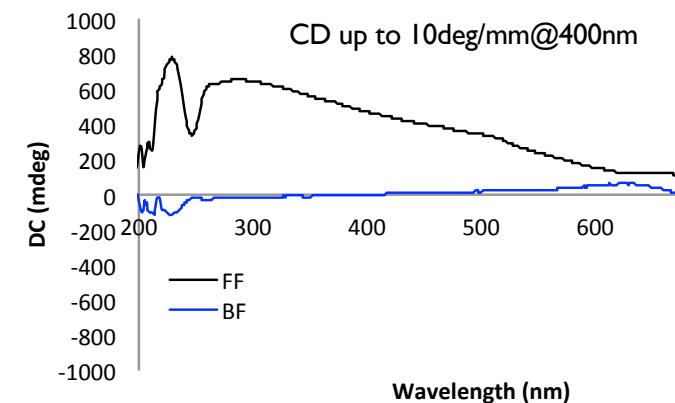
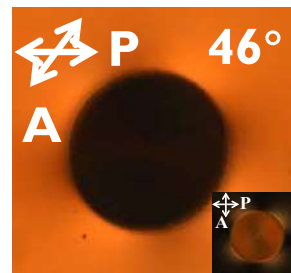
- Linear dichroism: different absorption for two orthogonal linear polarizations.
- Circular dichroism: different absorption for left vs right circular polarization.



Potential applications:

- All integrated optical isolator
- Achromatic polarization rotator
- Chiral sensitive sensors
- Magneto-optics crystal photoprecipitation

Circular dichroism and optical rotation in silica



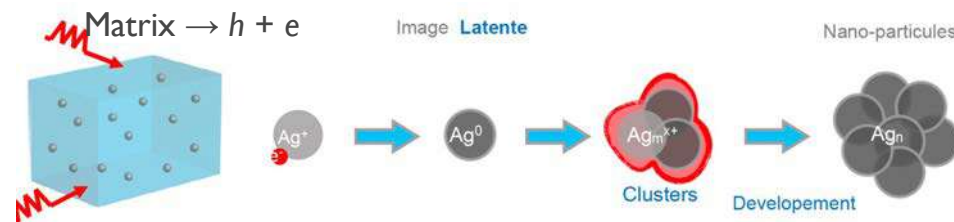
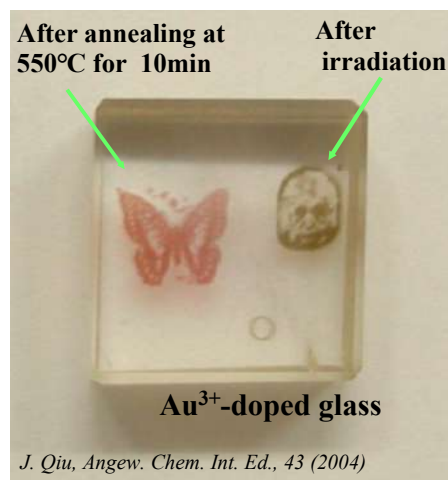
PRECIPITATION, DISSOLVING AND SHAPING OF METAL NANOPARTICLES IN GLASSES

Photochromic glasses, Photo Thermo Refractive Glass

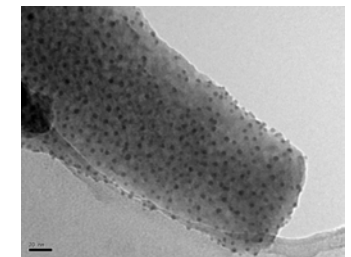
- **Approach:** Laser radiation + Heat treatment
- Size and space, density control
- Color control with doping ions: Au^{3+} , Ag^+ , Pb^{2+} , Pd^{2+} , Cu^{2+} ions etc..

cf cours de L. Cormier

Surface plasmon resonance

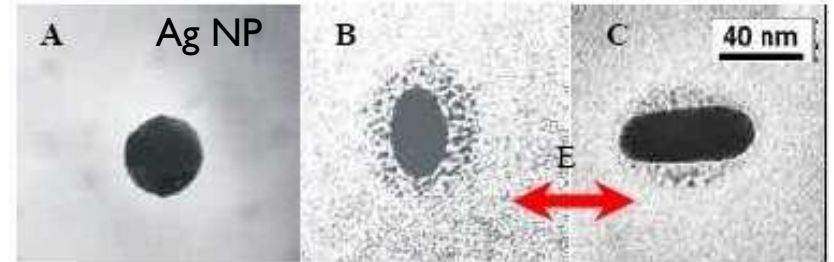


- Nonlinear interaction of laser with glass:
Matrix $\rightarrow h + e$ (option: add $\text{Ce}^{3+} \rightarrow \text{Ce}^{4+} + e^-$)
- Formation of metal atoms + colors centers:
 $\text{Au}^{3+} + 3e^- \rightarrow \text{Au}^0$, $\text{Ag}^+ + e^- \rightarrow \text{Ag}^0$, $\text{Pd}^{2+} + 2e^- \rightarrow \text{Pd}^0$
- Aggregation of metal atoms under heat treatment:
 $\text{Au}^0 \rightarrow \text{Au}_n$, $\text{Ag}^0 \rightarrow \text{Ag}_n$, $n\text{Pd}^0 \rightarrow \text{Pd}_n$



Note PTR glass : add Ce^{3+} as electron donor + formation of NaF or Li_2SiO_3 nanocrystals (e.g. Schott Foturan, alkaline alumino silicate)

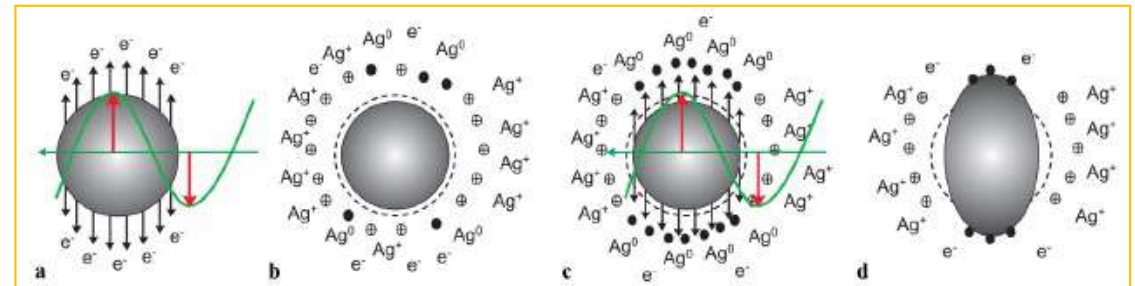
SHAPING OF METAL NANOPARTICLES IN GLASSES



Fs: 400nm, 150fs, 1kHz, 0,5-2 TW/cm²

Modification in low intensity mode

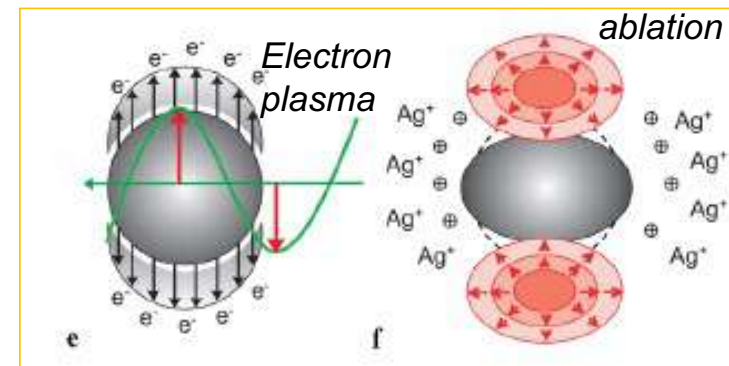
- 1: e^- emission parallelly to laser polarization
- 2: Ag^+ trap electrons forming Ag^0
- 3: Ag^0 is amplified by the next pulse
- 3: **Spherical NP grows in the polarization direction $\rightarrow$ elongated (prolate)**



A. Stalmashonak, Opt. Lett. 32(2007) 3215

Modification in high intensity mode

- 1: High intensity $\rightarrow$ high plasma density at the nanoparticle-matrix interface
- 2: **Local ablation of NP $\rightarrow$ oblate**

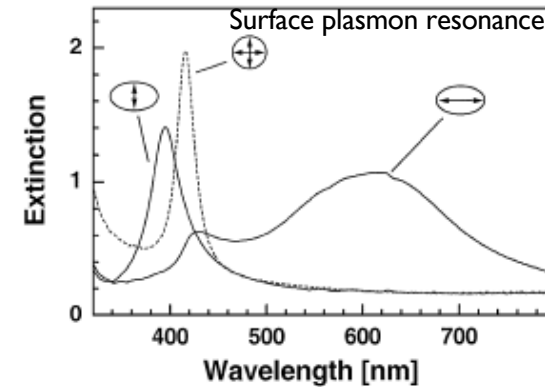


Interest ? Strong linear dichroism in the visible !

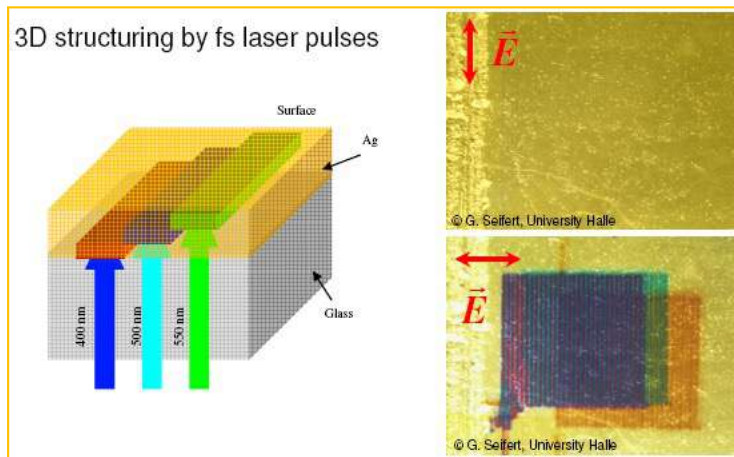


PRECIPITATION, DISSOLVING AND SHAPING OF METAL NANOPARTICLES IN GLASSES

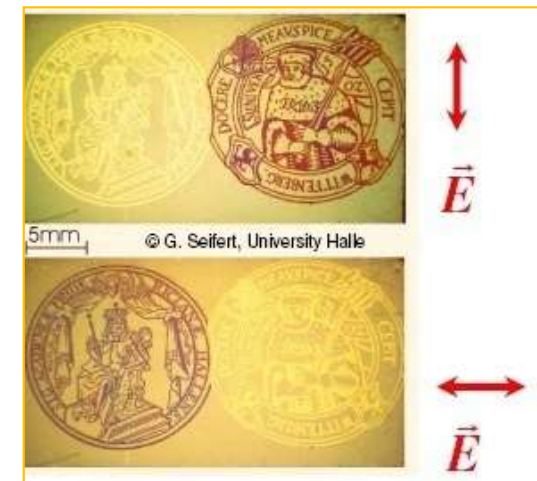
- Polarization dependent absorption: linear dichroism in the visible
- Tunable in wavelength / wavelength dependent reading
- Polarization dependent light scattering
- Polarization dependent emission (Fluorescent Ag clusters Ag_m^{x+} .)
- Optical contrast controlled (multilayers, etc...)



A. Stalmashoneak, Opt. Lett. 32 (2007)



Seifert et al Univ. Halle, Germany



NANO-MICRO CRYSTALLIZATION

Cw laser crystallization

Main Features:

- 2D (cw) / 3D (fs/ps) crystallization
- Size control (10's nm up to 10's mm)
- Oriented crystallization
- Polar axis parallel to laser scanning direction but not only (polarization controlled as well)
- β -BaB₂O₄, Ba₂TiGe/Si₂O₈, LiNbO₃, LaBGeO₅, etc...

LaBGeO₅ (Ferroelectric, congruent crystallization)

A. Stone et al. / JNCS 356 (2010)

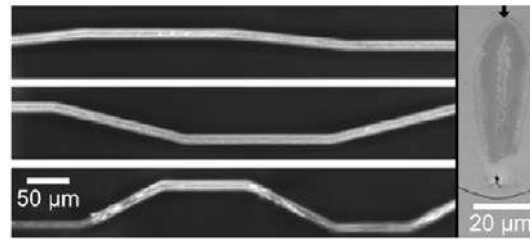
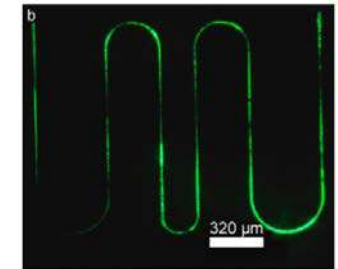


Fig. 2. Left: Polarized light micrographs of fs laser-crystallized LaBGeO₅ lines written in XY plane with bends of 6°, 14°, and 27°. Right: SEM backscattered electron image of a line cross-section. Arrow indicates incident beam direction.

Gupta et al. J. Am. Ceram. Soc., 91 (2008)



Fs laser crystallization

SmBiBO₃

T. Komatsu et al. J. Am. Ceram. Soc., 90 (2007)



LiNbO₃ High $\chi^{(2)}$

C. Fan et al. Opt. Lett. 37 (2012)



Polarized optical image of the written curved waveguides

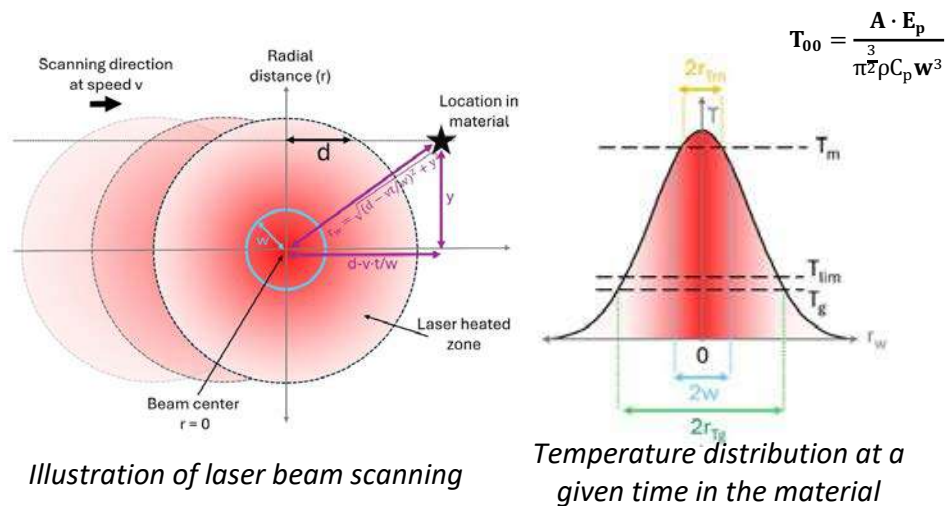
Some applications:

- SHG waveguides, Optical switch
- Acousto-optic modulators
- Phase modulation waveguides
- Saturable absorption

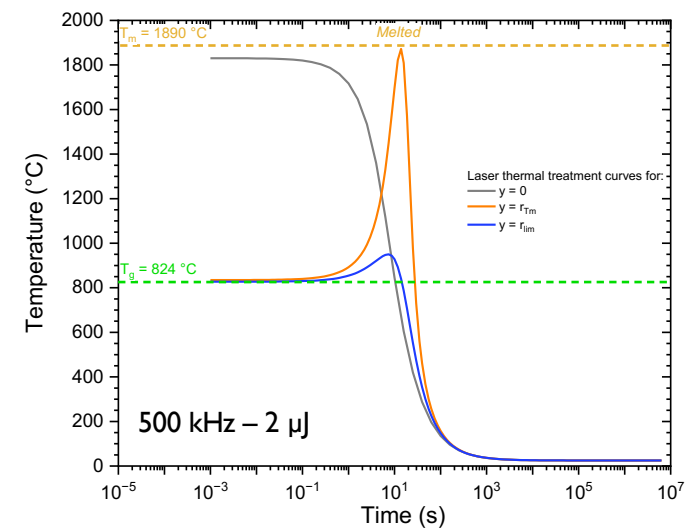
Cf Cours ONL de M Dussauze

NANO-MICRO CRYSTALLIZATION

Thermal treatment during the irradiation process



Cw laser or high repetition rate approach



- Computation of heating-cooling curves vs laser conditions

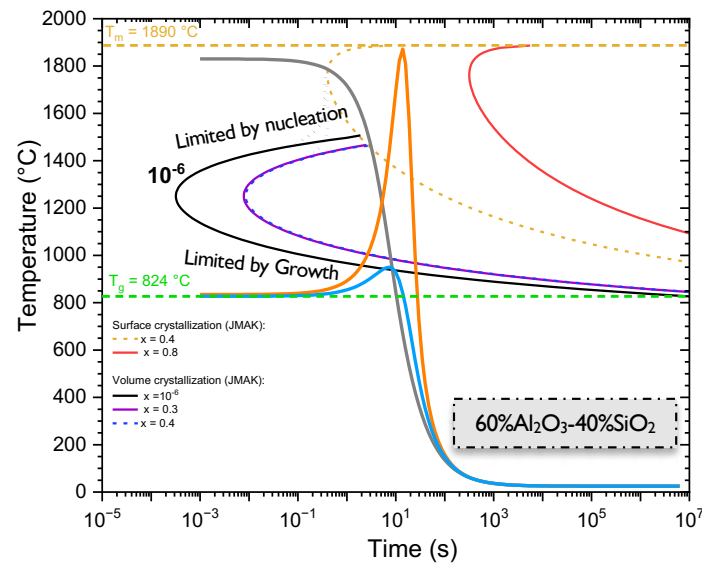
- Initial energy deposition $T_{00} = \frac{A \cdot E_p}{\pi^2 \rho C_p w^3}$ Note: $C_p(T)$, $\kappa(T)$, $\rho(T)$

- Solving Fourier heat equation $\rho T \left(\frac{\partial C_p}{\partial T} \frac{\partial T}{\partial t} \right) + \rho C_p \frac{\partial T}{\partial t} - \frac{\partial}{\partial r} \left(k \frac{\partial T}{\partial r} \right) = 0$

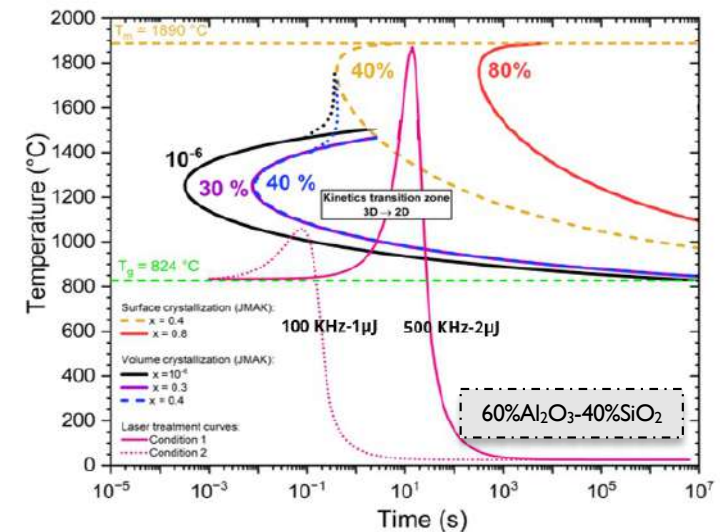
$$D_{th} = \frac{\kappa}{\rho \cdot C_p}$$

NANO-MICRO CRYSTALLIZATION

Thermal treatment during the irradiation process



Time-Temperature-Transformation (TTT) domain



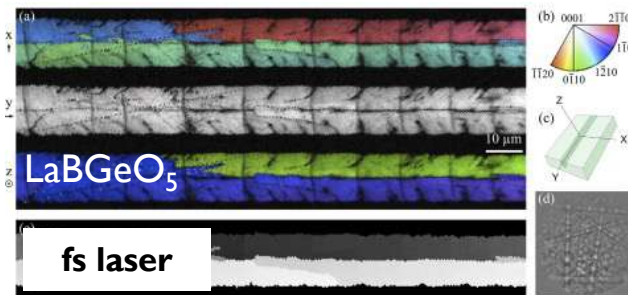
- Comparison of T variation (induced by laser) as a function of time (with the TTT diagram)
- Managing glass to crystal transformation (liquid-solid or solid-solid crystallization depending on $T_{\max}$)
- Example: 30% crystallization for 1 μJ – 100 kHz vs 60% for 2 μJ - 500 kHz



ORIENTED CRYSTALLIZATION

- Probing crystallization orientation : Second-harmonic generation, Raman, or EBSD.
- Polar axis parallel to laser scanning direction with cw laser (gradient of T)
- Polar axis $\perp$ to fs laser polarization (controlled by $E_{\omega} \wedge P_{\omega}$)

EBSD (electron backscatter diffraction)



A. Stone et al. *Sci. Rep* 5 10391 (2015)

Polarized microscopy

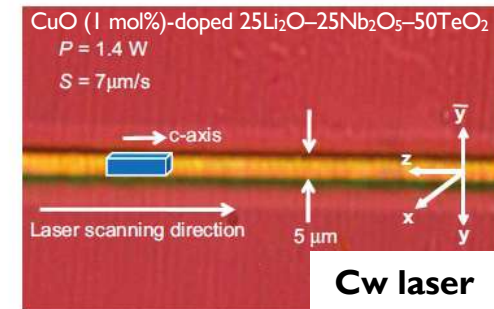


Fig. 9. Polarized optical microscope photograph for the sample obtained by Yb:YVO₄ laser ($\lambda=1080$ nm) irradiations with $P=1.4$ W and $S=7$ $\mu\text{m/s}$ in the Cu-LNS glass.

H. Sugita et al. / *Solid State Communications* 143 (2007) 280–284

Polarized SHG

(second harmonic generation)

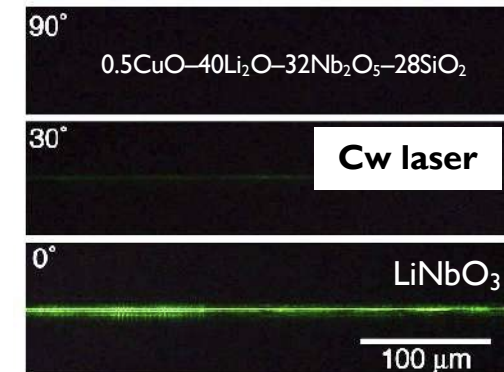


Fig. 6. Second harmonic (SH) wave images (532 nm) at different configuration (H-H mode) for the crystal line which written in the sample No. 2.

T. Honma et al. / *Optical Materials* 31 (2008) 315–319

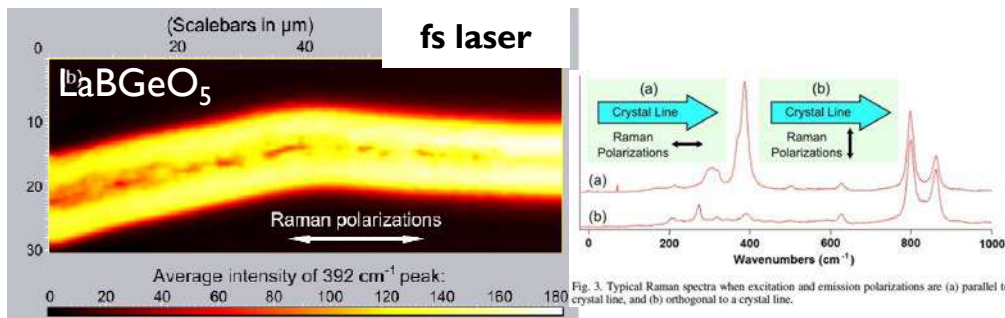
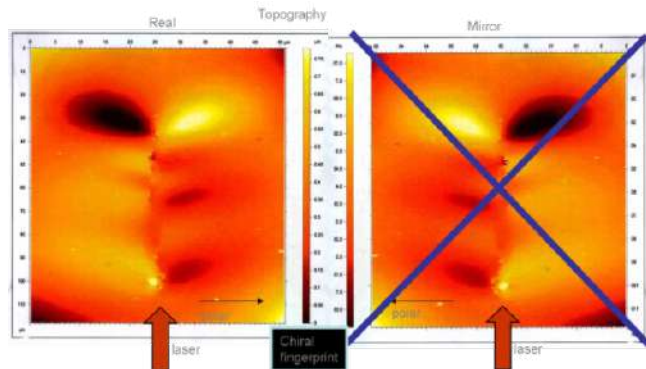


Fig. 3. Typical Raman spectra when excitation and emission polarizations are (a) parallel to a crystal line, and (b) orthogonal to a crystal line.

A. Stone et al. *Optics Express* 17(25):23284–23289 (2009)

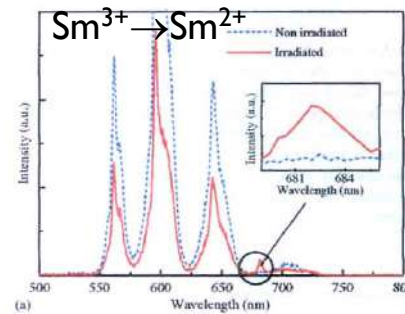
MANY OTHERS

Chiral mechanical structures



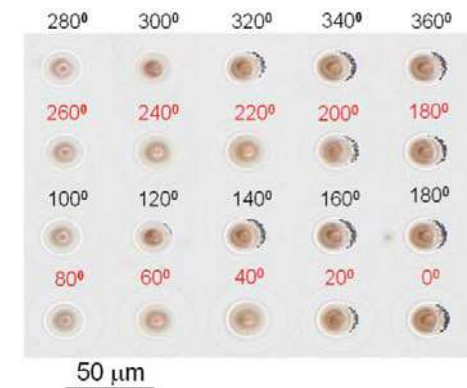
Poumellec et. al, *Opt. Express* 2003 & 2008

Oxydo-reduction



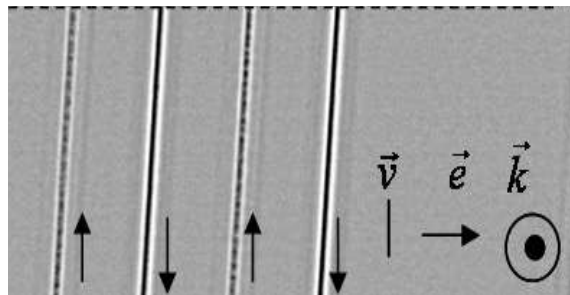
You et al. *JNCS*, Vol 352 (26–27), 2006

Photothermal effect



Zhang, F., et al. (2016). *Optics Express*, 24(19)

Orientational dependent writing



Poumellec et al. (2013). *Optical Materials Express*, 3

Kazansky et al. *APL* 2006

Oriented chemical migration

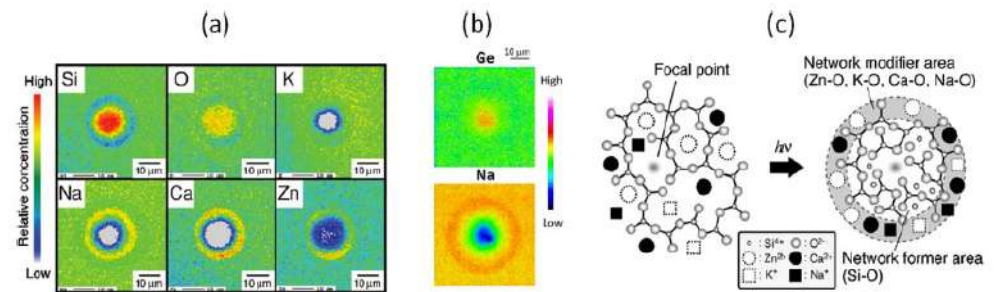


Fig. 3-15. EPMA mapping perpendicular to femtosecond laser irradiation at 200 kHz repetition in (a) 67SiO₂-7CaO-9K₂O-9Na₂O-4ZnO glass and (b) GeO-Na₂O glass. (c) Schematic of the tendency of femtosecond laser induced ion migration (perpendicular to laser irradiation direction) inside a silicate glass, as proposed by Kanehira et al.

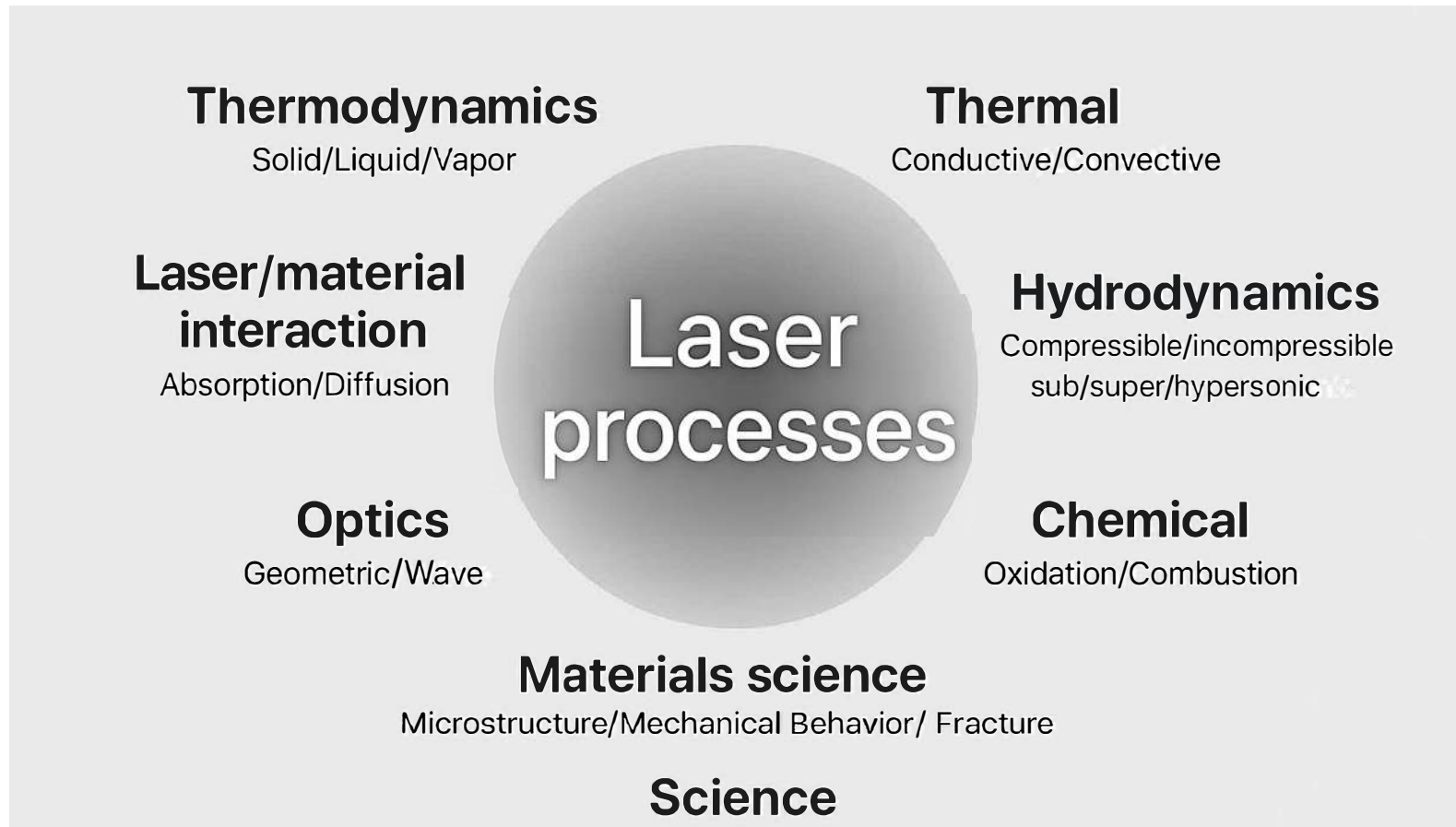
Fernandez, T. T., et al. (2018). *Progress in Materials Science*, 94, 68

Outline

- Context and introduction
- Basic concepts
- CO₂ laser
- Surface processing
- Volume processing
- **Conclusions**



CONCLUSIONS – MECHANISMS



CONCLUSIONS - APPLICATIONS

Modulators, optical switch, SHG waveguides

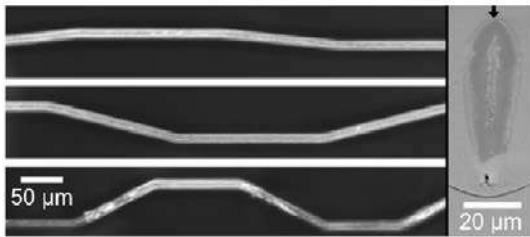
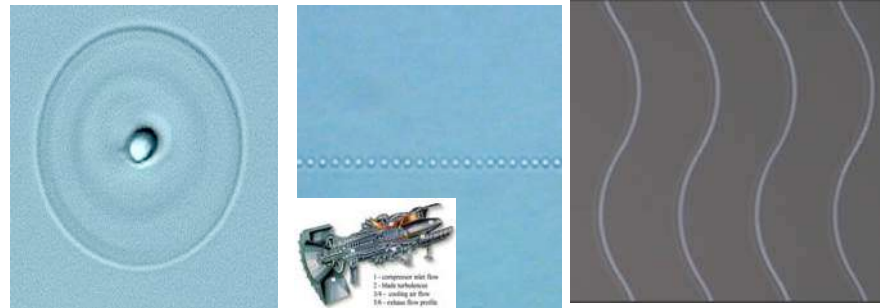
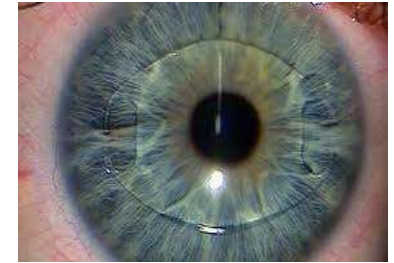


Fig. 2. Left: Polarized light micrographs of fs laser-crystallized LaBGeO₃ lines written in XY plane with bends of 6°, 14°, and 27°. Right: SEM backscattered electron image of a line cross-section. Arrow indicates incident beam direction.

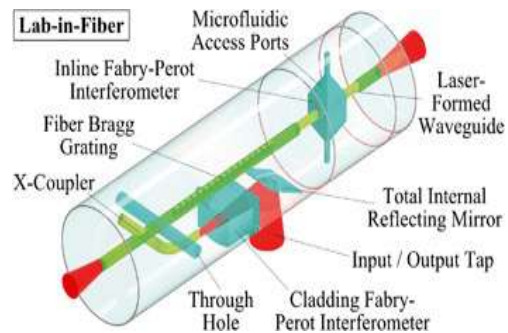
Fiber Bragg gratings, waveguides, Sensors, photonics integrated circuits



Fresnel or graded index Lens

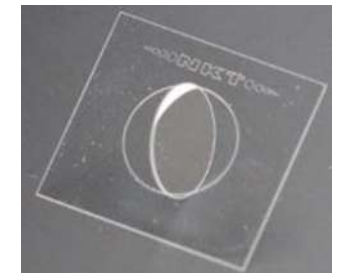


3D marking and optical data storage,

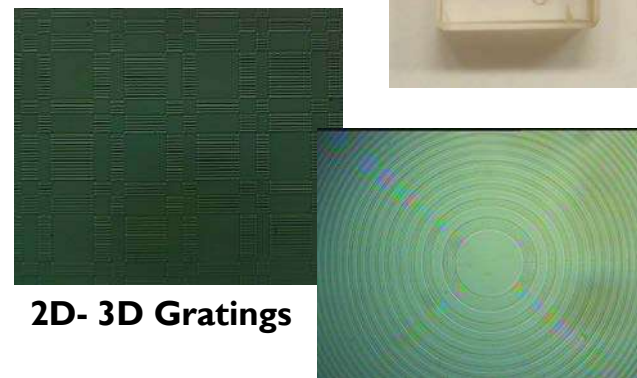


Lab-on-chip and Lab-in-fiber

Polarizers, SPR



Cutting, ablating, polishing



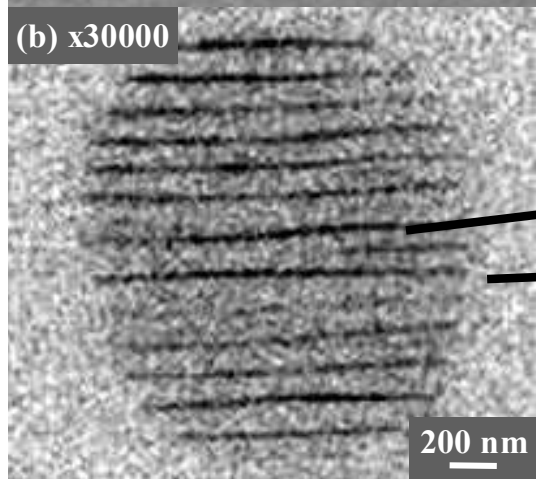
2D- 3D Gratings

Micro-optics

Welding, sealing



Thanks for your
attention !



1 μ m

EHT = 1.00 kV

WD = 1.9 mm

SE2