



# Modélisation des propriétés thermodynamiques - Prise en compte des effets de pression et de volume

Dominique de Ligny

# *2 countries - 2 towns - 2 years*



## DGG ↔ USTV JOINT MEETING 2019 - 2020



13-15 of May 2019

MARITIM Hotel Nürnberg  
Frauentorgraben 11  
D-90443 Nürnberg  
<http://www.hvg-dgg.de/en/>



15-19 of June 2020

Orléans University  
<http://www.ustverre.fr>

**International School: Thermodynamic of Glass  
(TC3 ICG, USTV, DGG)**

Erlangen Sunday 12 May 2019

Comity of organization:

Reinhard Conradt, Natalia Vedishscheva, Daniel Neuville, Dominique de Ligny

**[www.hvg-dgg.de/en/](http://www.hvg-dgg.de/en/)**

# Content

- **Généralité sur le volume**
- **Evolution de la densité avec la composition**
- **Volume et température fictive**
- **Densification et pression fictive**
- **Anomalie élastique et composition**

# Pressure and Volume with the Gibbs free energy

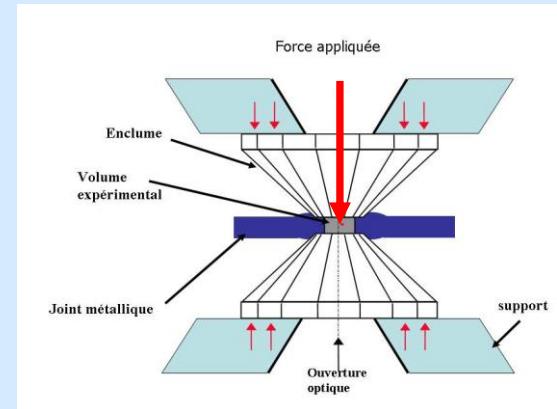
$$dG = VdP - SdT + \sum \mu_i dn_i$$

Volume is the derivative of G by the Pressure

We need: Equation of State  
 $V=f(T,P)$

Direct determination or by integration of the bulk modulus K

**in situ Brillouin**



$$\Delta\nu_{180} = \frac{2n \cdot c}{\lambda}$$

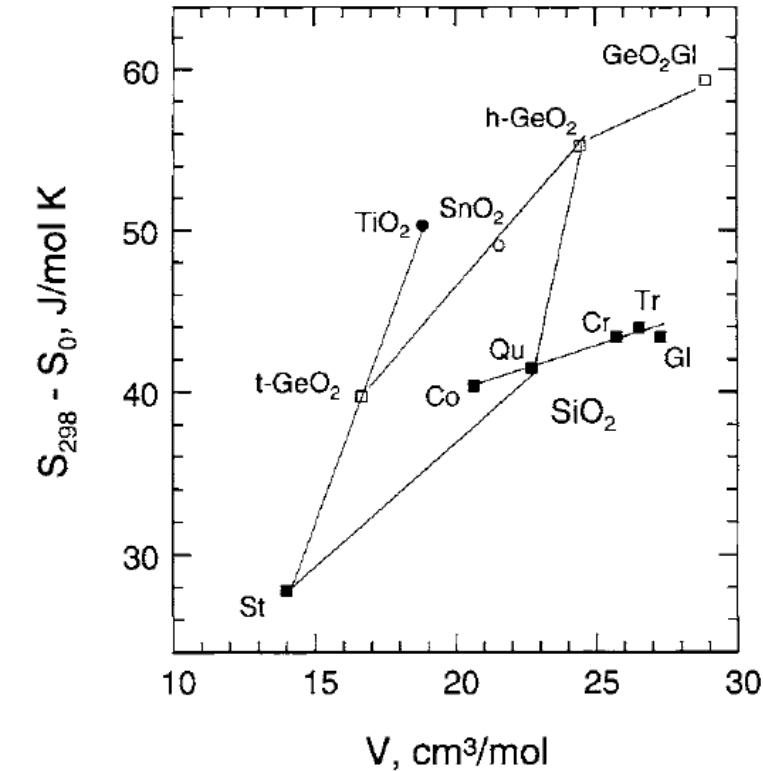
$$c_l = \sqrt{\frac{K + \frac{4}{3}G}{\rho}}$$

# Entropy versus Volume

Entropy is sensitive:

- Vibrational properties
- Short range order
- Coordination number

Sens of evolution known

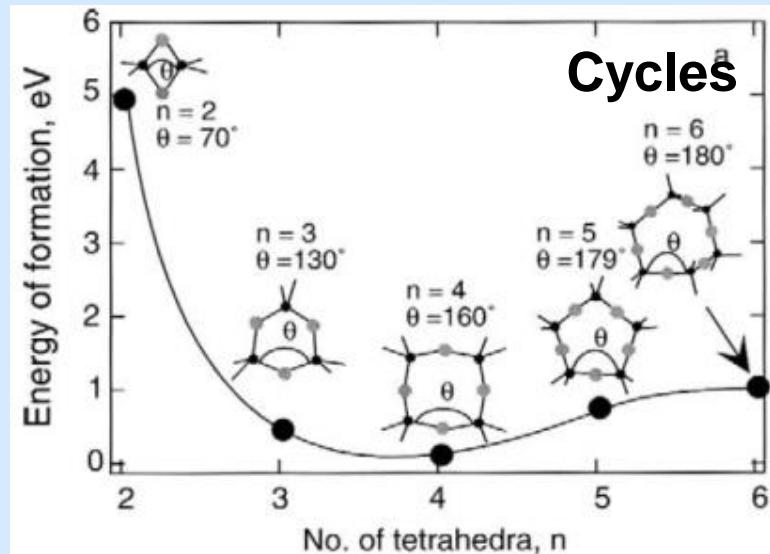


Volume is sensitive:

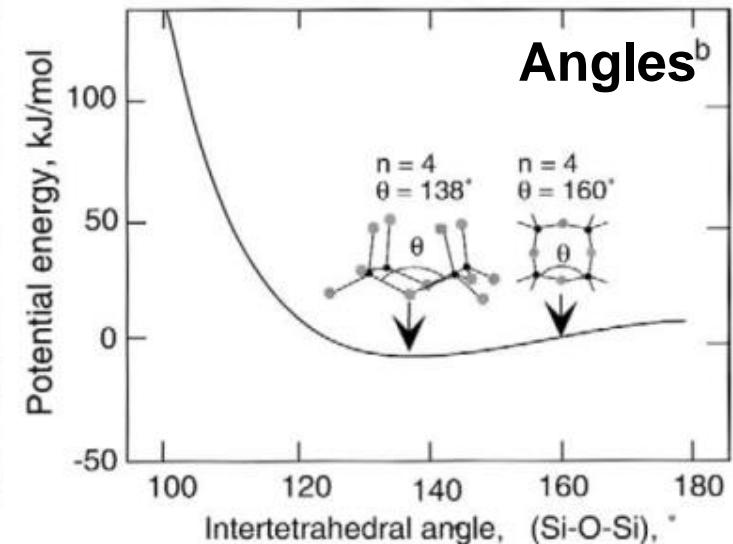
- Short range order
- Medium range order
- Angles between units

Sens of evolution unknown

# Why so much possibilities for $\text{SiO}_2$ ?



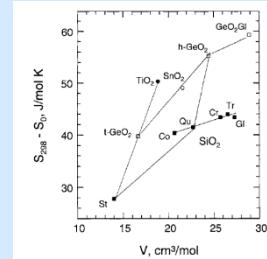
Galeener 82



Gibbs et al. 81

Network very flexible as shown with  $\text{SiO}_2$  polymorphs:

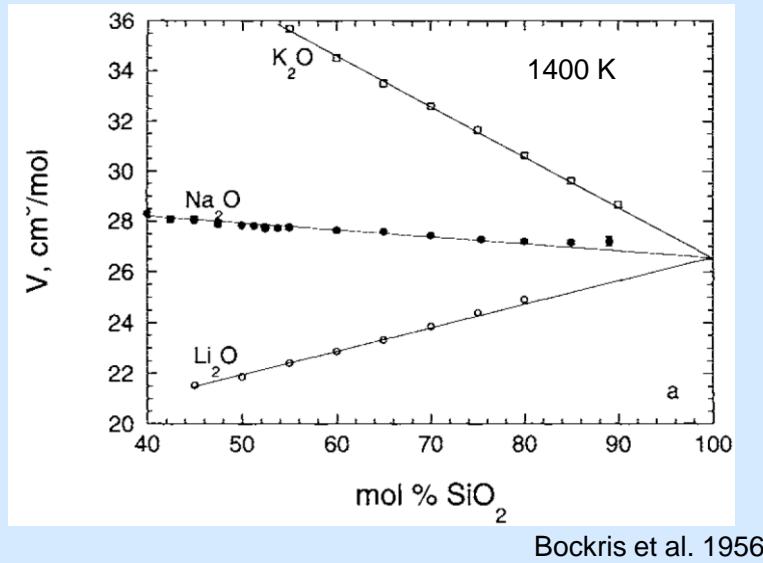
- coesite ( $n=4$ )
- quartz (spiral of 6)
- tridymite ( $n=6$ )
- cristobalite ( $n=6$ )
- zeolites ( $n=6$ )



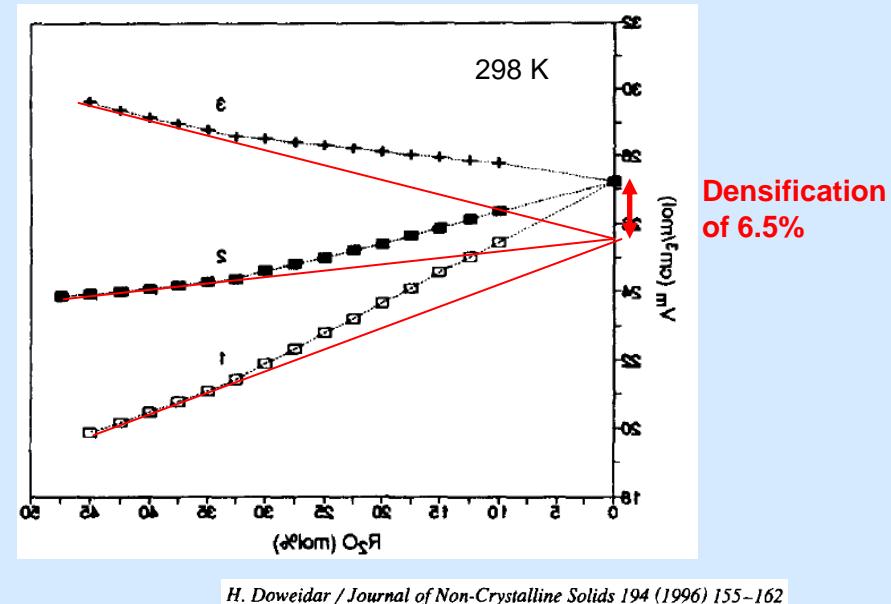
No significant energetic effect

# Evolution of volume with the composition

Liquid



Glass



Liquid additivity of oxides

$$v(T) = \sum_i x_i \bar{v}_i(T)$$

no additivity from oxides

# Modelisation of the volume with composition

Using the database SciGlass  
Priven and Fluegel

$$\text{Density} = b_o + \sum_{i=1}^n \left[ b_i C_i + \sum_{k=i}^n \left( b_{ik} C_i C_k + \sum_{m=k}^n b_{ikm} C_i C_k C_m \right) \right]$$

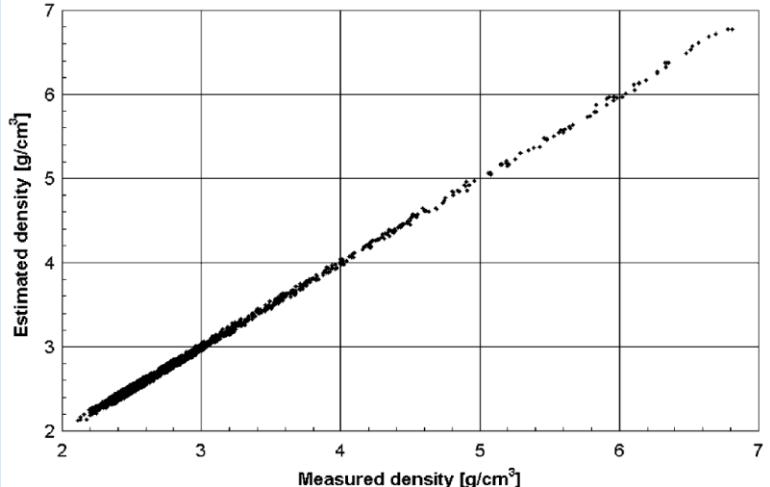


Fig. 2. Plot of 6719 measured versus estimated density values according to the model in Table II.

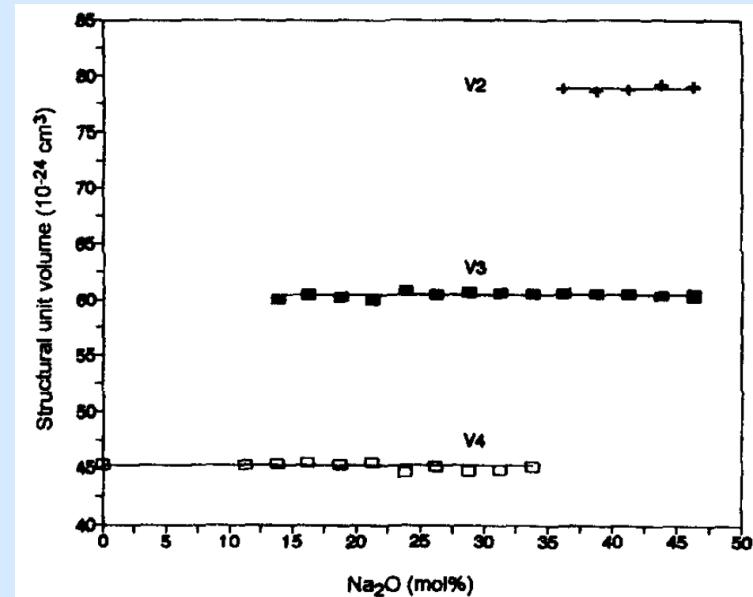
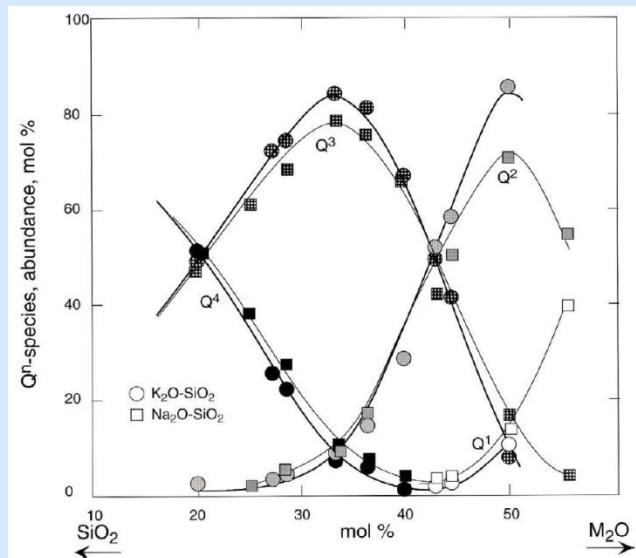
*J. Am. Ceram. Soc.*, **90** [8] 2622–2625 (2007)

| Variable                                       | Coefficient   | Variable   | Coefficient  | Variable   | Coefficient  |
|--|---------------|--|--------------|--|--------------|
| Intercept                                      | 2.121560704   | CdO  | 0.052945783  | K <sub>2</sub> O × MgO   | -0.000337747 |
| Al <sub>2</sub> O <sub>3</sub>                 | 0.010525974   | La <sub>2</sub> O <sub>3</sub>                                 | 0.10643194   | K <sub>2</sub> O × CaO   | -0.000349578 |
| (Al <sub>2</sub> O <sub>3</sub> ) <sup>2</sup> | -0.000076924  | Nd <sub>2</sub> O <sub>3</sub>                                 | 0.090134135  | K <sub>2</sub> O × SrO   | -0.000425589 |
| B <sub>2</sub> O <sub>3</sub>                  | 0.00579283    | NiO  | 0.024289113  | K <sub>2</sub> O × BaO   | -0.000392897 |
| (B <sub>2</sub> O <sub>3</sub> ) <sup>2</sup>  | 0.000129174   | ThO <sub>2</sub>   | 0.090253734  | Al <sub>2</sub> O <sub>3</sub> × CaO                                 | -0.000102444 |
| (B <sub>2</sub> O <sub>3</sub> ) <sup>3</sup>  | -0.000019887  | U <sub>x</sub> O <sub>y</sub>                                  | 0.063297404  | Al <sub>2</sub> O <sub>3</sub> × PbO                                 | -0.000651745 |
| Li <sub>2</sub> O                              | 0.012848733   | Sb <sub>x</sub> O <sub>y</sub>                                 | 0.044258719  | Al <sub>2</sub> O <sub>3</sub> × TiO <sub>2</sub>                    | -0.000563594 |
| (Li <sub>2</sub> O) <sup>2</sup>               | -0.0000276404 | SO <sub>3</sub>  | -0.044488661 | Al <sub>2</sub> O <sub>3</sub> × BaO                                 | -0.000273835 |
| (Li <sub>2</sub> O) <sup>3</sup>               | 0.000000259   | F  | 0.00109839   | Al <sub>2</sub> O <sub>3</sub> × SrO                                 | -0.000177761 |
| Na <sub>2</sub> O                              | 0.018129123   | Cl   | -0.006092537 | Al <sub>2</sub> O <sub>3</sub> × ZnO                                 | -0.000109968 |
| (Na <sub>2</sub> O) <sup>2</sup>               | -0.000264838  | Remainder <sup>†</sup>   | 0.02514614   | Al <sub>2</sub> O <sub>3</sub> × ZrO <sub>2</sub>                    | -0.002381651 |
| (Na <sub>2</sub> O) <sup>3</sup>               | 0.000001614   | Na <sub>2</sub> O × K <sub>2</sub> O                           | -0.000395491 | Na <sub>2</sub> O × PbO  | -0.000036455 |
| K <sub>2</sub> O                               | 0.019177312   | Na <sub>2</sub> O × Li <sub>2</sub> O                          | -0.00031449  | Na <sub>2</sub> O × TiO <sub>2</sub>                                 | -0.00014331  |
| (K <sub>2</sub> O) <sup>2</sup>                | -0.000319863  | K <sub>2</sub> O × Li <sub>2</sub> O                           | -0.000329725 | Na <sub>2</sub> O × ZnO  | -0.000155275 |
| (K <sub>2</sub> O) <sup>3</sup>                | 0.000000191   | Na <sub>2</sub> O × B <sub>2</sub> O <sub>3</sub>              | 0.000242157  | Na <sub>2</sub> O × ZrO <sub>2</sub>                                 | -0.000126728 |
| MgO  | 0.01210604    | K <sub>2</sub> O × B <sub>2</sub> O <sub>3</sub>               | 0.000259927  | Na <sub>2</sub> O × Fe <sub>2</sub> O <sub>3</sub>                   | -0.000371343 |
| (MgO) <sup>2</sup>                             | -0.000061159  | Li <sub>2</sub> O × B <sub>2</sub> O <sub>3</sub>              | 0.000106359  | K <sub>2</sub> O × PbO   | -0.000525213 |
| CaO  | 0.017992367   | MgO × B <sub>2</sub> O <sub>3</sub>                            | -0.000206488 | K <sub>2</sub> O × TiO <sub>2</sub>                                  | -0.000386587 |
| (CaO) <sup>2</sup>                             | -0.00005478   | CaO × B <sub>2</sub> O <sub>3</sub>                            | -0.000032258 | K <sub>2</sub> O × ZnO   | -0.000329812 |
| SrO  | 0.034630735   | PbO × B <sub>2</sub> O <sub>3</sub>                            | -0.000186195 | CaO × PbO  | -0.00084145  |
| (SrO) <sup>2</sup>                             | -0.000086939  | Fe <sub>2</sub> O <sub>3</sub> × B <sub>2</sub> O <sub>3</sub> | -0.000720268 | ZnO × Fe <sub>2</sub> O <sub>3</sub>                                 | -0.001536804 |
| BaO  | 0.049879597   | ZrO <sub>2</sub> × B <sub>2</sub> O <sub>3</sub>               | -0.000697195 | Na <sub>2</sub> O × K <sub>2</sub> O × B <sub>2</sub> O <sub>3</sub> | -0.000032967 |
| (BaO) <sup>2</sup>                             | -0.000168063  | Al <sub>2</sub> O <sub>3</sub> × B <sub>2</sub> O <sub>3</sub> | -0.000735749 | Na <sub>2</sub> O × MgO × CaO  | -0.000009143 |
| ZnO  | 0.025221567   | Li <sub>2</sub> O × Al <sub>2</sub> O <sub>3</sub>             | -0.000116227 | Na <sub>2</sub> O × MgO × Al <sub>2</sub> O <sub>3</sub>             | -0.000012286 |
| (ZnO) <sup>2</sup>                             | 0.000099961   | Na <sub>2</sub> O × Al <sub>2</sub> O <sub>3</sub>             | -0.000253454 | Na <sub>2</sub> O × CaO × Al <sub>2</sub> O <sub>3</sub>             | -0.000005106 |
| PbO  | 0.070020298   | K <sub>2</sub> O × Al <sub>2</sub> O <sub>3</sub>              | -0.000371858 | Na <sub>2</sub> O × CaO × PbO  | 0.000100796  |
| (PbO) <sup>2</sup>                             | 0.000214424   | MgO × CaO  | 0.000057248  | K <sub>2</sub> O × MgO × CaO   | -0.00001217  |
| (PbO) <sup>3</sup>                             | -0.000001502  | MgO × Al <sub>2</sub> O <sub>3</sub>                           | 0.000167218  | K <sub>2</sub> O × MgO × Al <sub>2</sub> O <sub>3</sub>              | -0.000041908 |
| Fe <sub>x</sub> O <sub>y</sub>                 | 0.036995747   | MgO × ZnO  | 0.000220766  | K <sub>2</sub> O × CaO × Al <sub>2</sub> O <sub>3</sub>              | -0.000012421 |
| Mn <sub>x</sub> O <sub>y</sub>                 | 0.016648722   | Li <sub>2</sub> O × CaO  | -0.00008792  | K <sub>2</sub> O × CaO × PbO   | 0.000125759  |
| TiO <sub>2</sub>                               | 0.018820343   | Na <sub>2</sub> O × MgO  | -0.000300745 | MgO × CaO × Al <sub>2</sub> O <sub>3</sub>                           | -0.000011236 |
| ZrO <sub>2</sub>                               | 0.043059714   | Na <sub>2</sub> O × CaO  | -0.000228249 | CaO × Al <sub>2</sub> O <sub>3</sub> × Li <sub>2</sub> O             | -0.000016177 |
| (ZrO <sub>2</sub> ) <sup>2</sup>               | -0.000779078  | Na <sub>2</sub> O × SrO  | -0.00023137  | Al <sub>2</sub> O <sub>3</sub> × B <sub>2</sub> O <sub>3</sub> × PbO | 0.000030116  |
| Ce <sub>x</sub> O <sub>y</sub>                 | 0.061277268   | Na <sub>2</sub> O × BaO  | -0.000171693 |  |              |

# Modelisation of the volume with composition

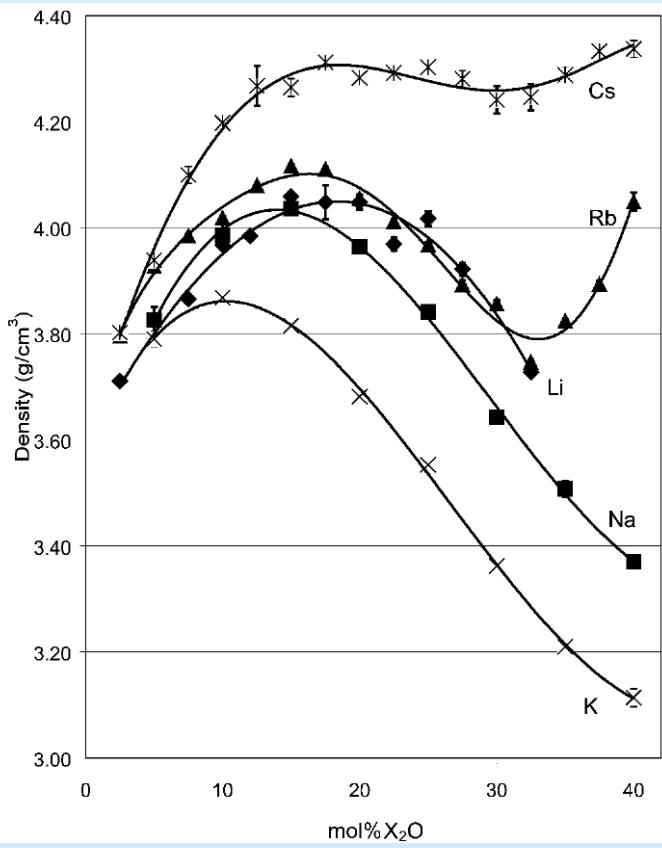
Molecular Approach  
Doweidar

Additive model on the  
Qn species



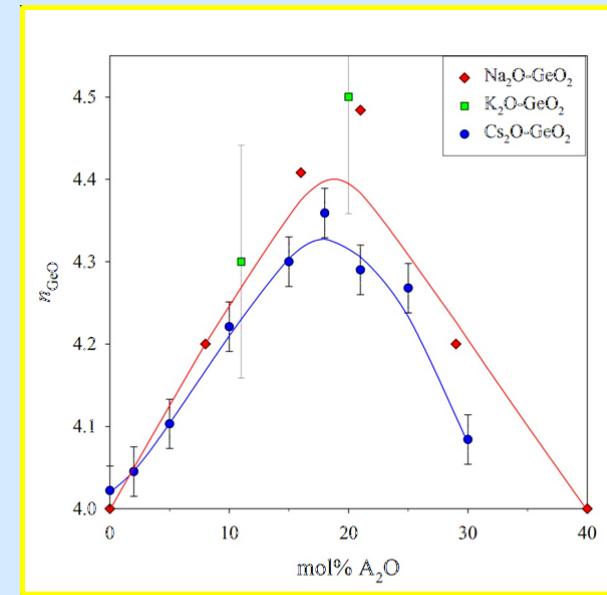
Here a structural model is needed

# Modelisation of the volume with composition



Henderson and Wang  
Eur. J. Mineral.  
2002, 14, 733-744

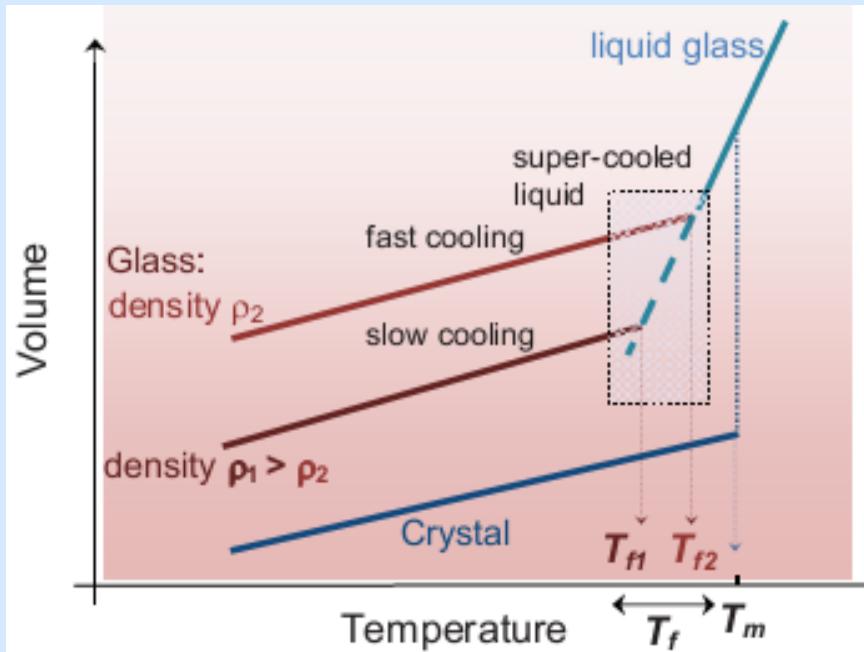
Effect more complex with change of coordination



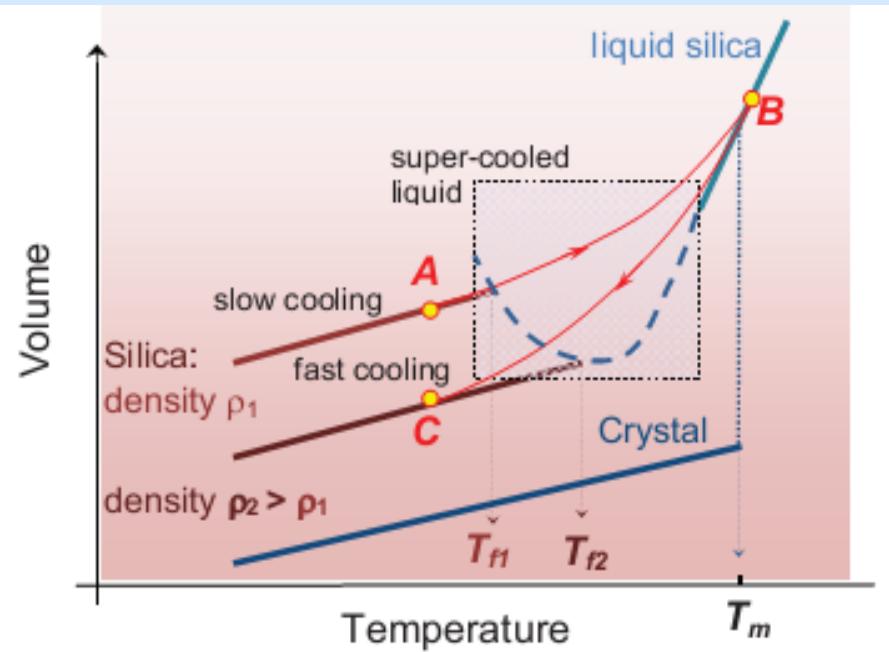
$\text{Na}_2\text{O}$ : (Ueno et al, *Physica B*, 1983)  
 $\text{K}_2\text{O}$ : (Hoppe et al, *J Non-Cryst Sol*, 1999)  
 $\text{Cs}_2\text{O}$ : (Hannon et al, *J Phys Chem B*, 2007)

Mixing terms can take in account these variations

# Variation of volume with cooling rate



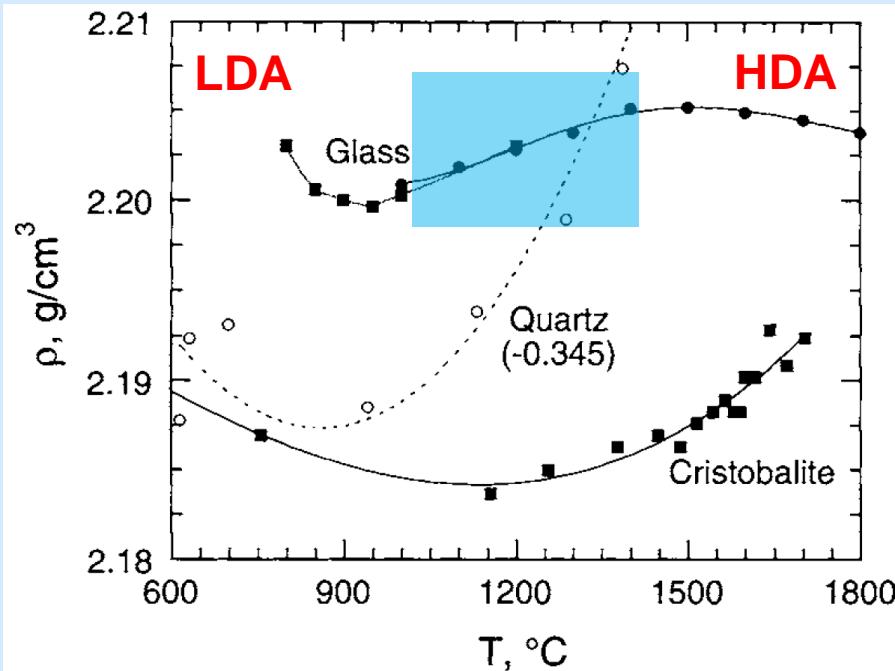
Normal ( $\text{GeO}_2$ )



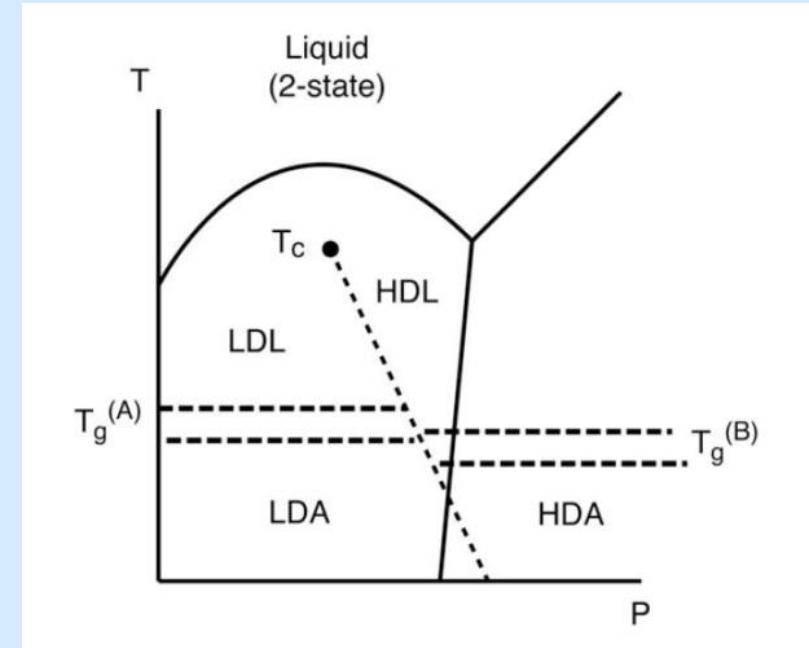
Anormal ( $\text{SiO}_2$ )

Anomalous relationship of silica glass with cooling rate explain by Sen with polyamorphism

# Silica glass and polyamorphism



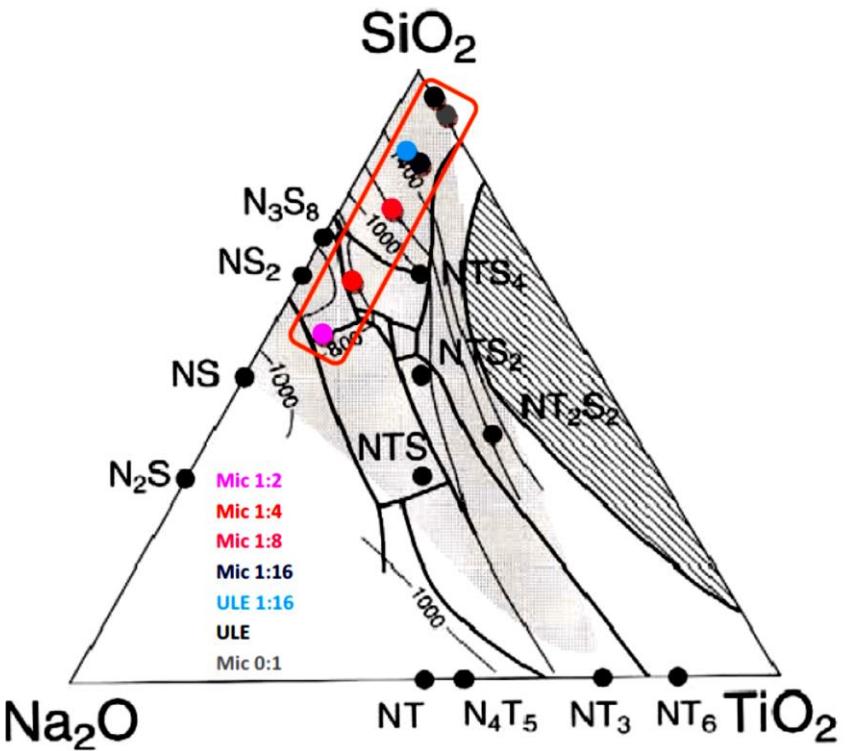
Mysen Richet 08  
Sen et al. PRL 04



McMillan et Wilding JNCS09

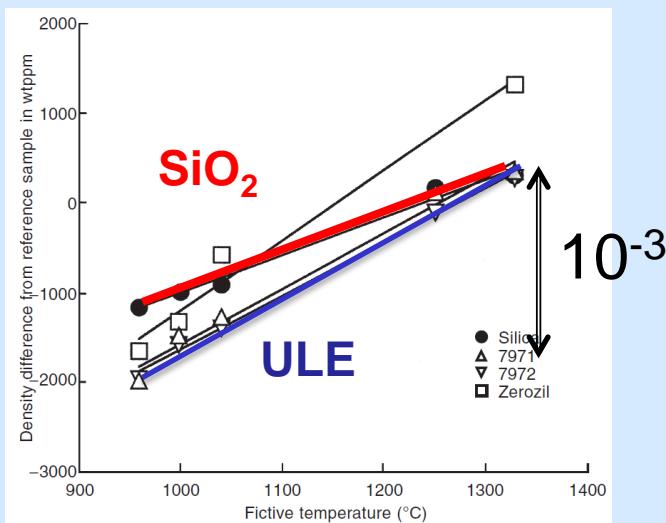
Silica polyamorphic transition smeared on a large  $T$  range?  
Continuous evolution?

# Titanosilicates



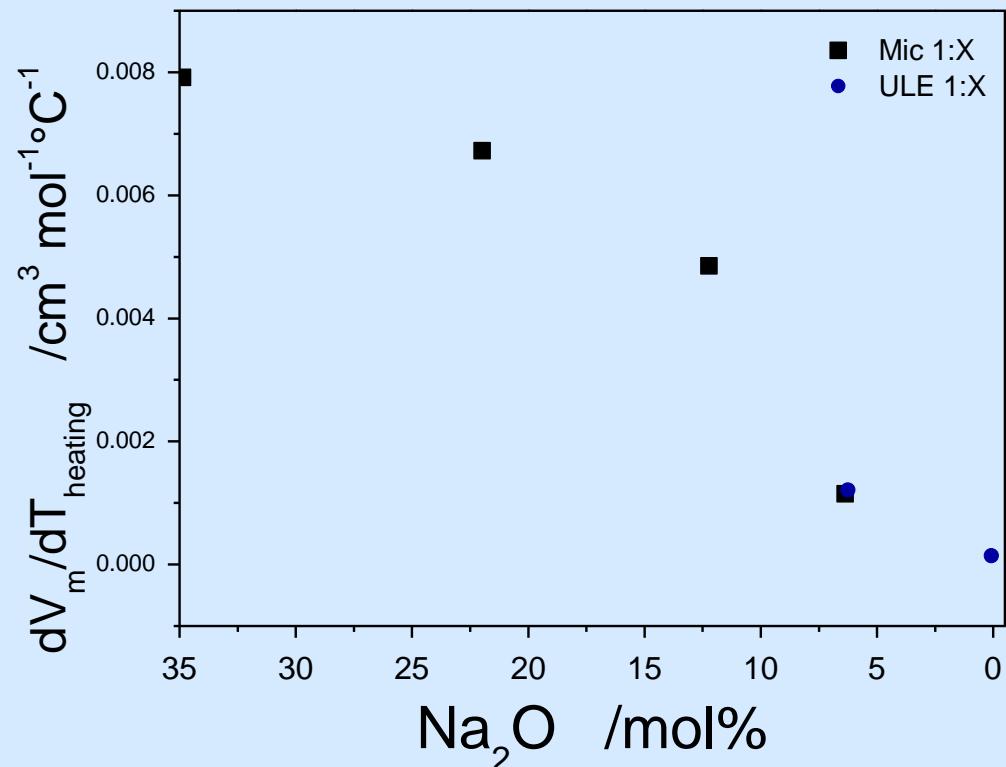
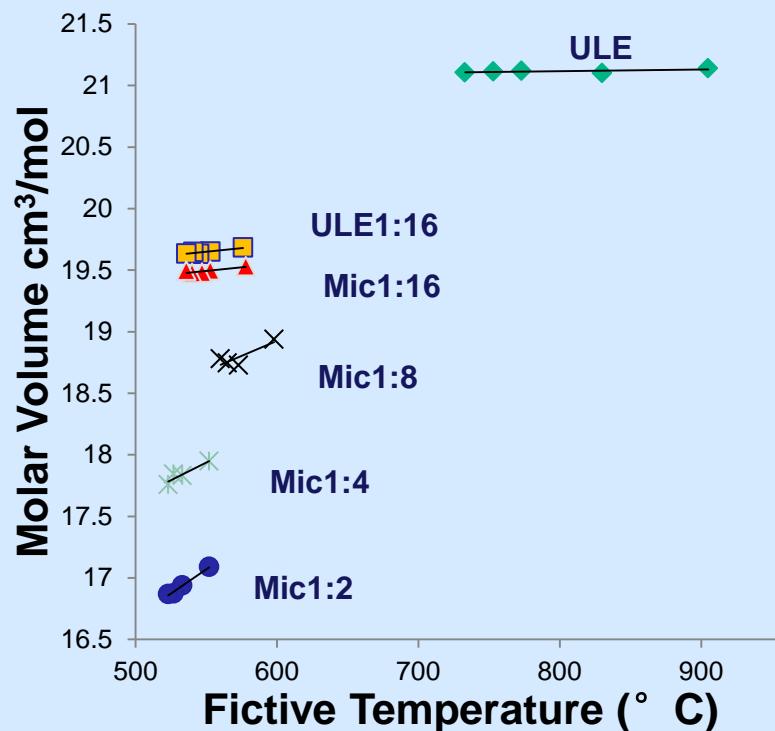
Mysen and Richet 2005

| Sample   | $\text{Na}_2\text{O}$<br>/mol% | $\text{SiO}_2$<br>/mol% | $\text{TiO}_2$<br>/mol% | M<br>$/g \text{ mol}^{-1}$ |
|----------|--------------------------------|-------------------------|-------------------------|----------------------------|
| Mic 1:2  | 35                             | 56                      | 9                       | 43,91                      |
| Mic 1:4  | 22                             | 68                      | 10                      | 45,00                      |
| Mic 1:8  | 12                             | 78                      | 10                      | 45,79                      |
| Mic 1:16 | 6                              | 85                      | 9                       | 46,20                      |
| ULE      |                                |                         |                         |                            |
| 1:16     | 6                              | 87                      | 7                       | 46,05                      |
| ULE      | 0                              | 94                      | 6                       | 46,45                      |



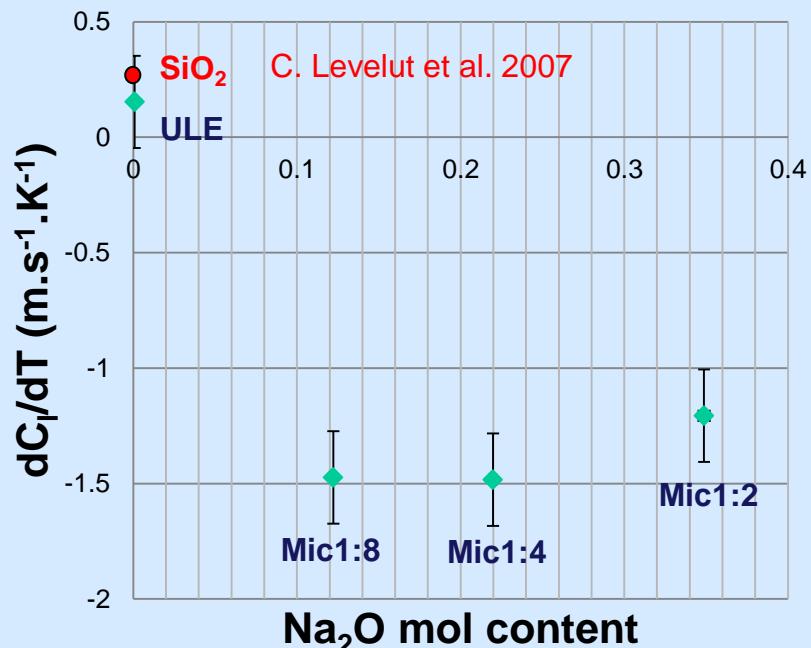
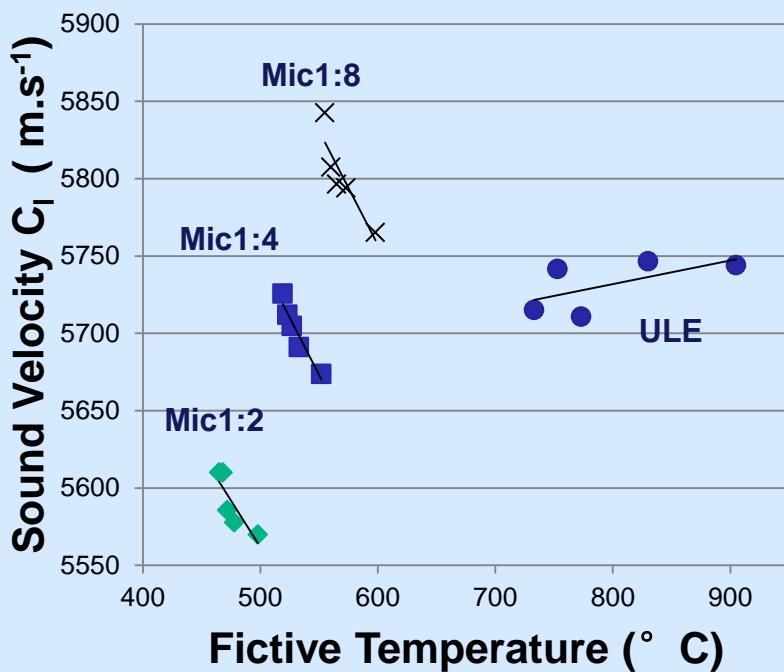
Shelby Phys. Chem. of Glasses 205

# Molar volume versus fictive temperature



- The variation of the Molar volume with  $T_f$  is increasing with the concentration of  $\text{Na}_2\text{O}$
- Due to the precision of the buoyancy measurement the ULE glass here appears constant

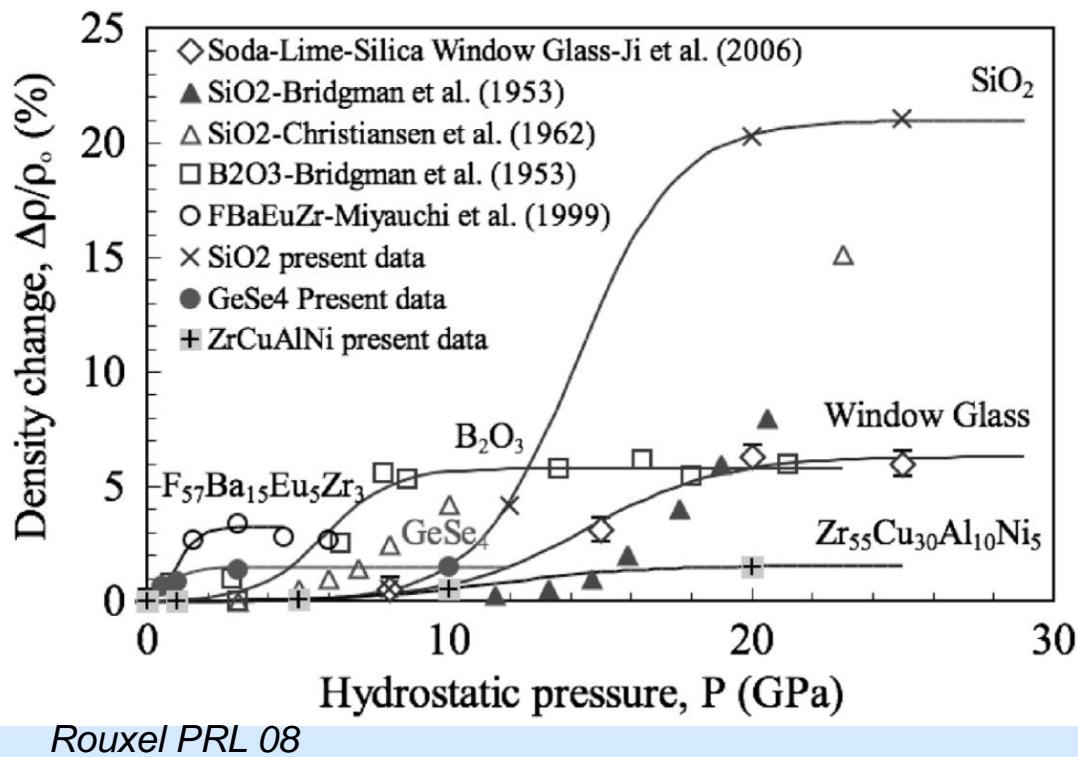
# Evolution of longitudinal sound velocity



ULE glass behaves very differently from the Na containing glasses

ULE has more or less the same behavior than pure silica glass  
Could it be associated with HDA to LDA transition?

# Densification and Energy



Variations with density ( $\text{g}/\text{cm}^3$ ) of the enthalpy (kJ/mol) of oxide glasses

| Glass                              | $\partial H / \partial \rho$ |
|------------------------------------|------------------------------|
| $\text{GeO}_2$                     | -32.3 (14.1)                 |
| $\text{SiO}_2$                     | -44 (20)                     |
| $\text{K}_2\text{Si}_4\text{O}_9$  | -105.2 (10.7)                |
| $\text{Na}_2\text{Si}_4\text{O}_9$ | -52.1 (13.6)                 |
| $\text{NaAlSi}_3\text{O}_8$        | -9.1 (10.4)                  |

P. Richet et al. / Chemical Geology 213 (2004) 41–47

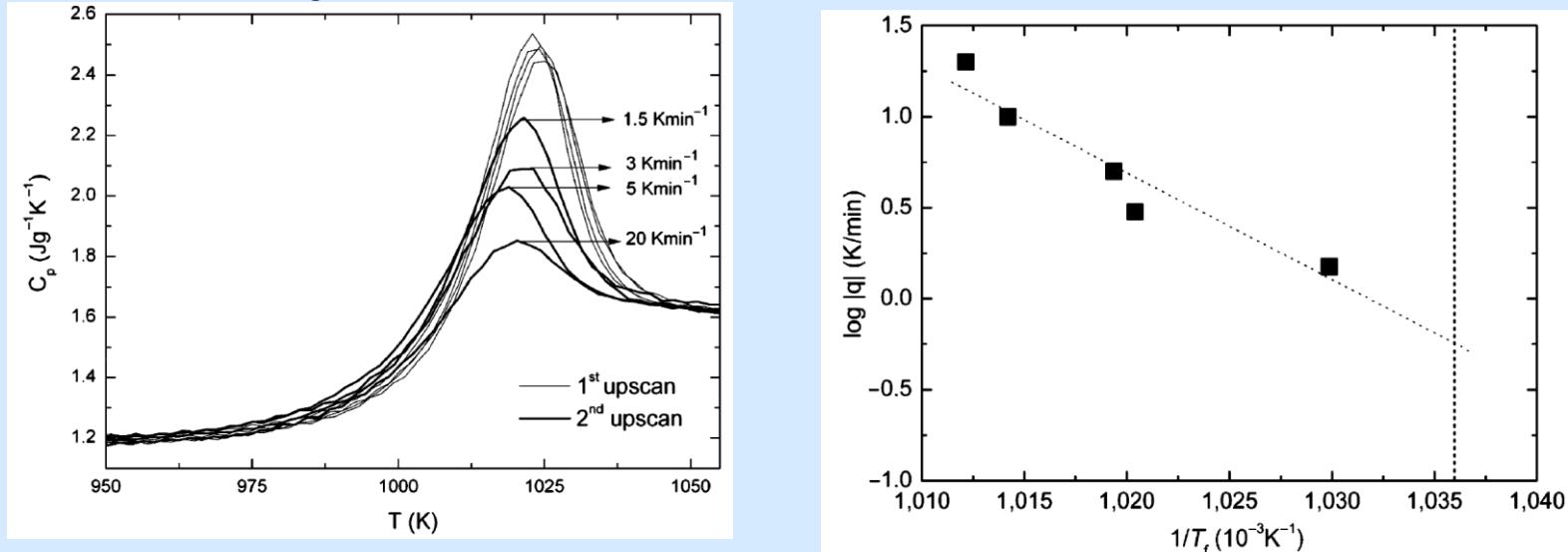
$$T_{\text{fictive}} = \left( \frac{\partial U}{\partial S} \right)_V = T - A \left( \frac{\partial \xi}{\partial S} \right)_V ,$$

$$p_{\text{fictive}} = - \left( \frac{\partial U}{\partial V} \right)_S = p + A \left( \frac{\partial \xi}{\partial V} \right)_S .$$

Schmelzer and Gutzow 2011

# Evolution of Tg with densification

CaMgSi<sub>2</sub>O<sub>6</sub> densified at 500MPa at 1040K for 10 min

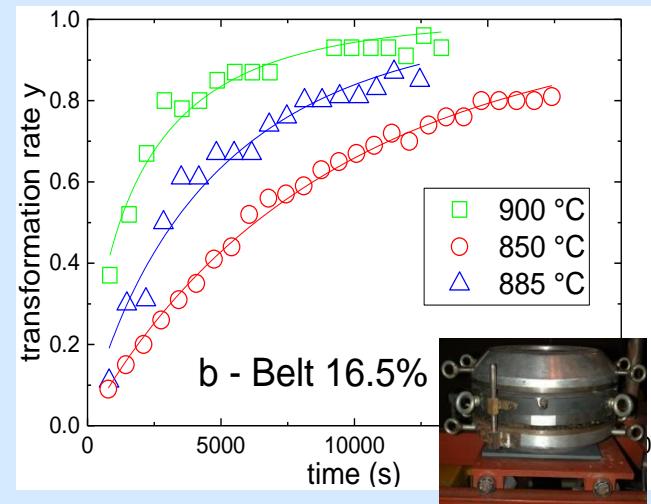
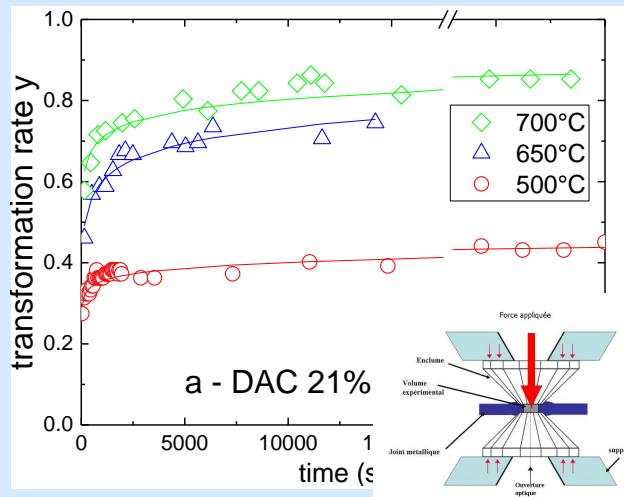
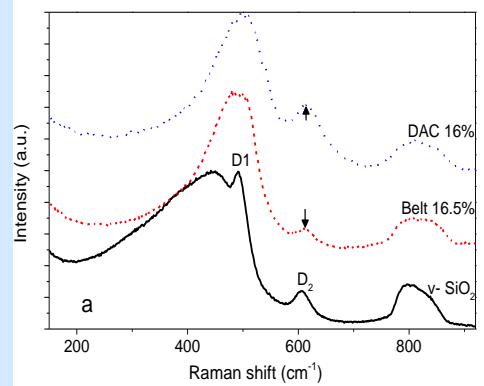


Wondraczek *et al.* *J. Am. Ceram. Soc.*, **90** [5] 1556–1561 (2007)

$$T_{f0} = T_{fA} \left[ 1 + \frac{V(p_0 - p)(\alpha_{Vl} - \alpha_{Vg})}{C_{pl} - C_{pg}} \right]$$

Continuity between Tf and Pf

# Relaxation of densified silica glass



Cold and hot compression lead to two different glass with the same density

At least two order parameters are needed to define the densified state

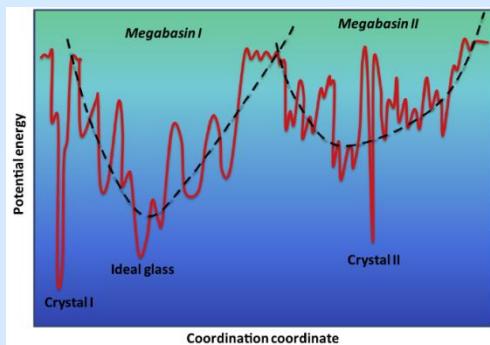
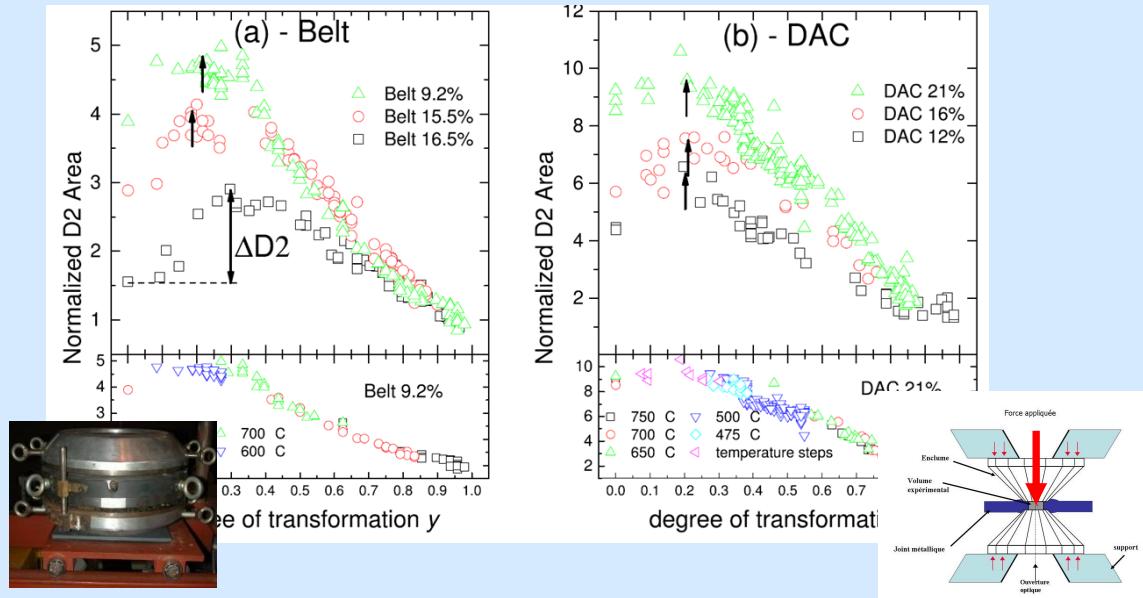
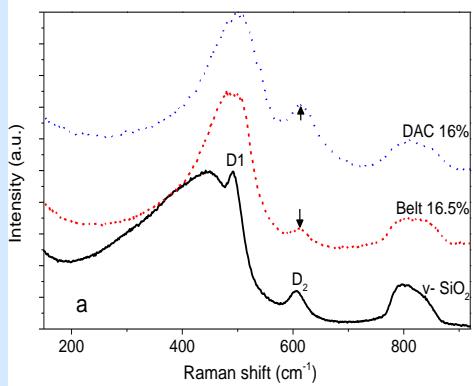
No link with  $T_g$ ....

Very fast evolution at temperature a lot below  $T_g$   
Transformation does not reach the final relaxation state

Slow evolution at higher temperature but still below  $T_g$

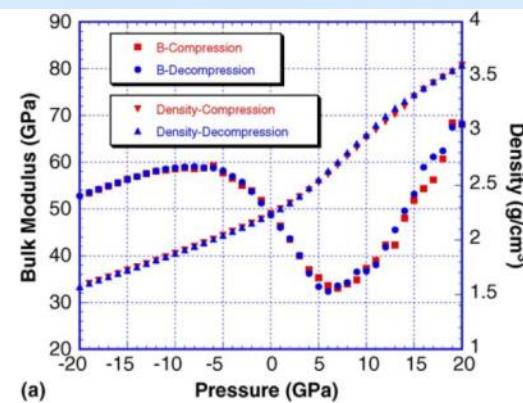
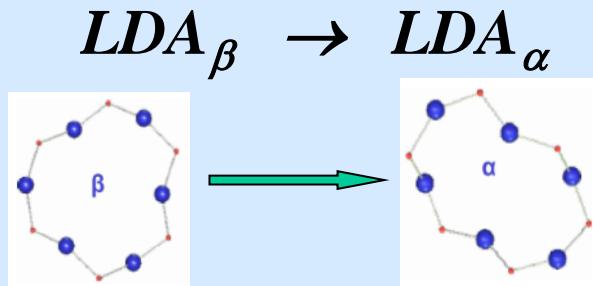
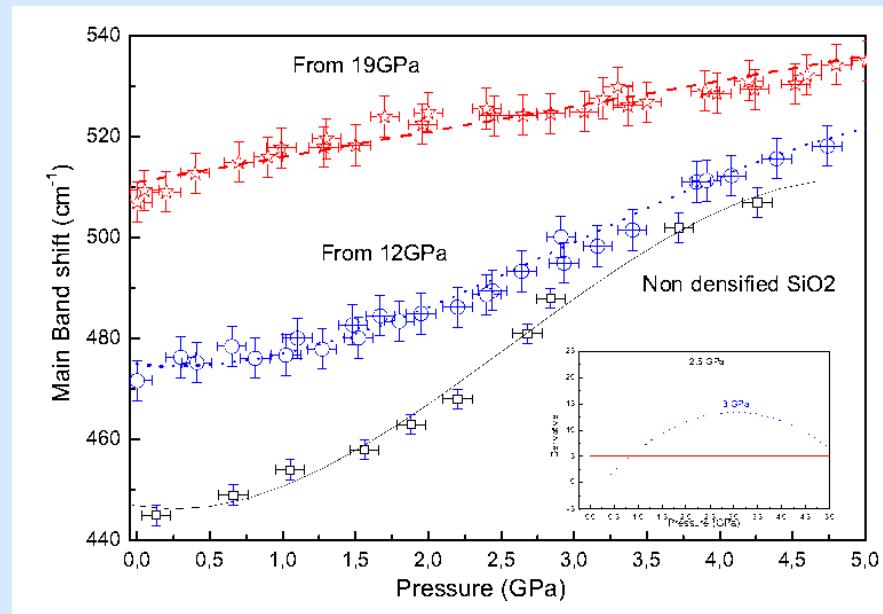
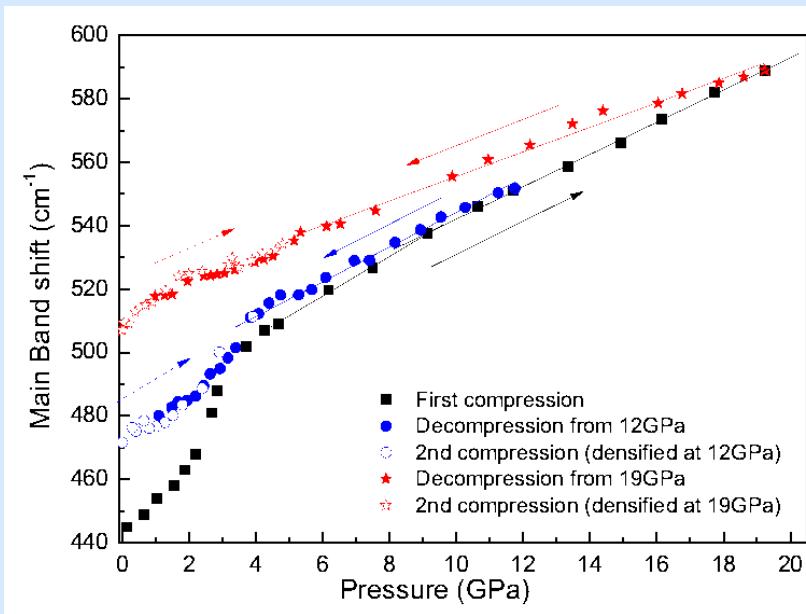
Transformation reaches the final relaxation state

# Existence of an intermediate exited state



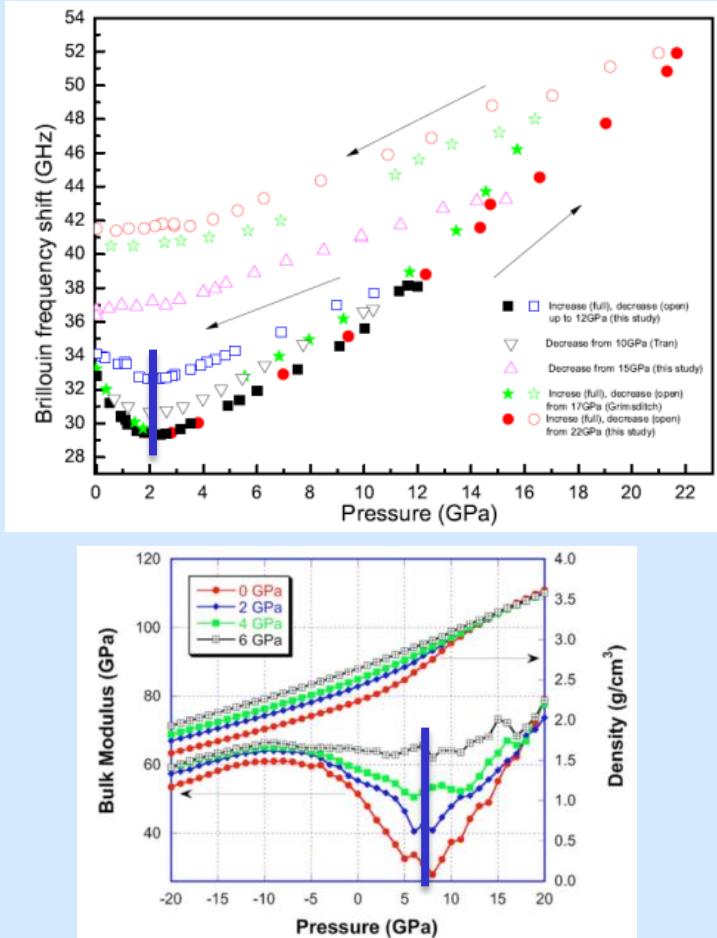
Existence of at least two megabassins  
Perhaps then change of the order parameter

# Elastic anomaly of $\text{SiO}_2$ glass



L. Huang et al. / Journal of Non-Crystalline Solids 349 (2004) 1–9

# Densification and elastic anomaly



progressive densification

*LDA* → *HDA*  
 $2,20 \text{ g cm}^{-3}$   
 $2,66 \text{ g cm}^{-3}$

Disappearance of elastic anomaly  
at 40% of the maximum densification  
The minimum does not shift

L. Huang and J. Kieffer Appl. Phys. Lett. **89**, 141915 (2006)



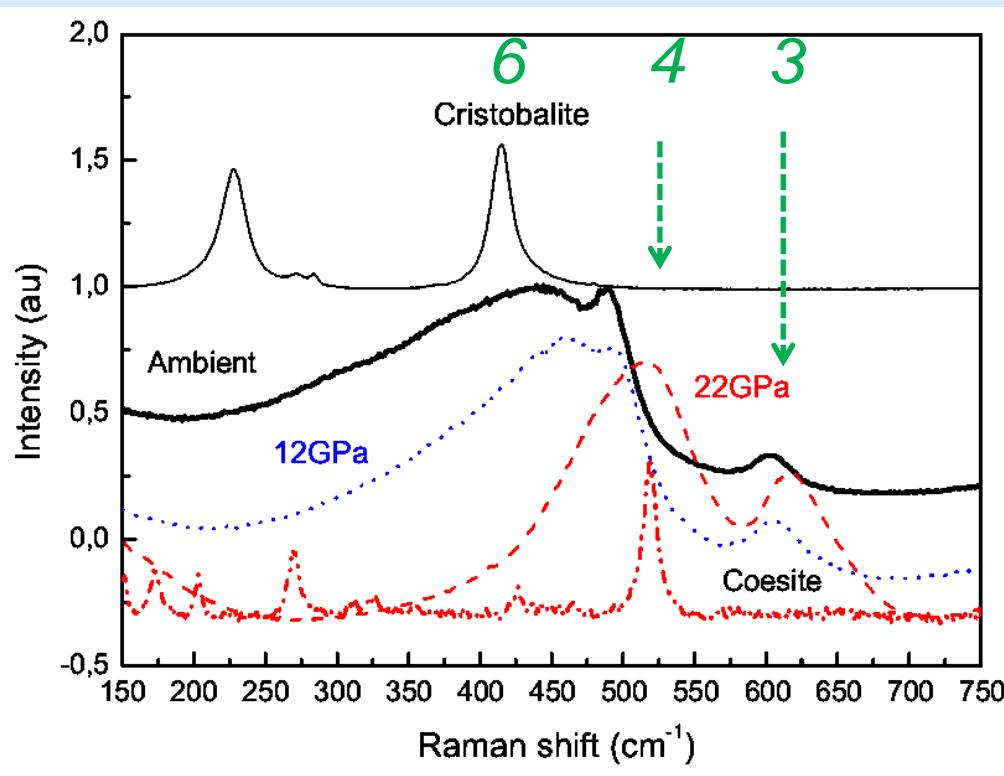
FRIEDRICH-ALEXANDER  
UNIVERSITÄT  
ERLANGEN-NÜRNBERG  
TECHNISCHE FAKULTÄT

i L M  
INSTITUT LUMIÈRE MATIÈRE

UFB Lyon 1

# $\text{SiO}_2$ glass ex situ

## Raman measurements ex situ



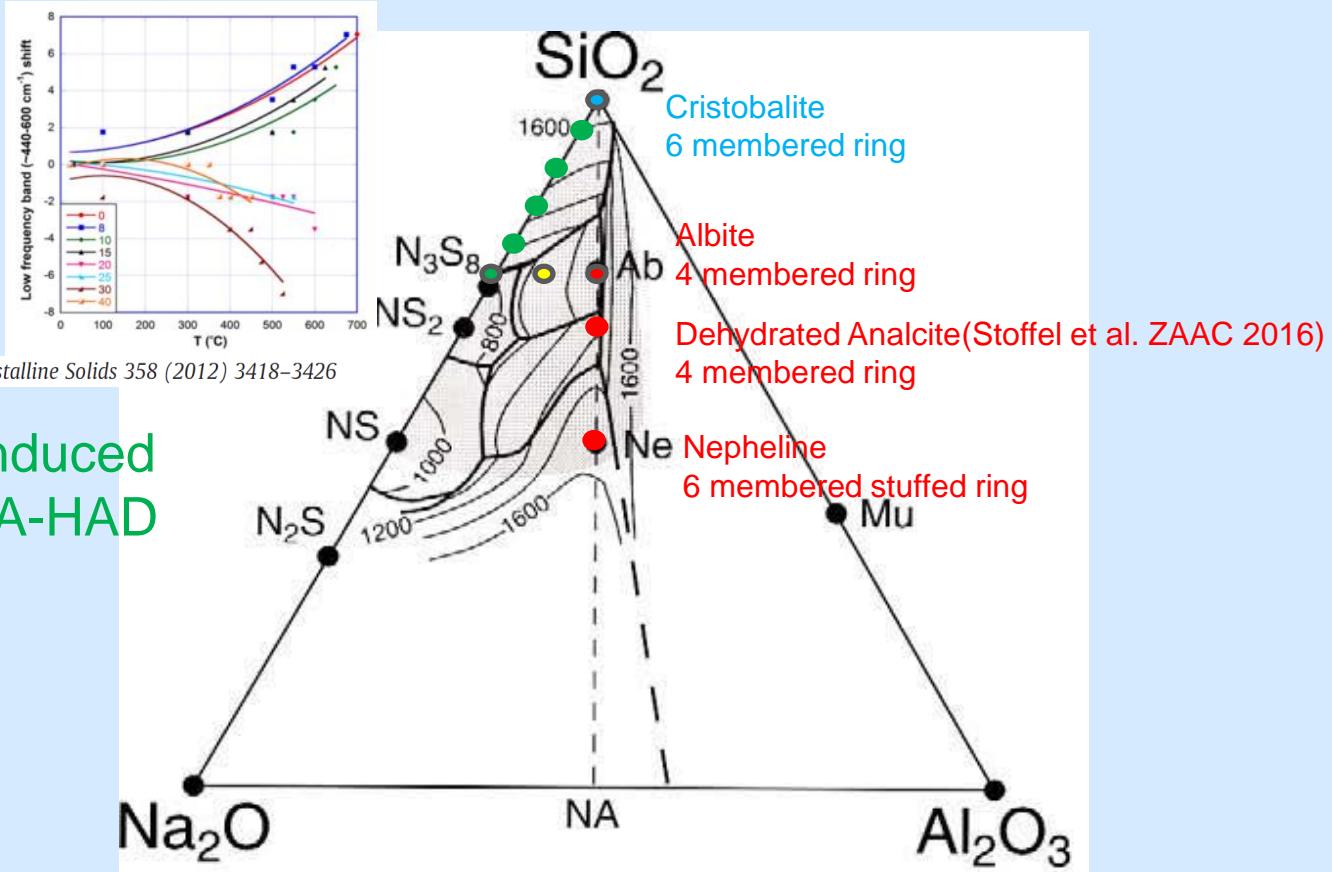
$LDA \rightarrow HDA$

$2,20 \text{ g cm}^{-3}$   
 $2,66 \text{ g cm}^{-3}$

*6 membered rings  
to 4 membered rings*

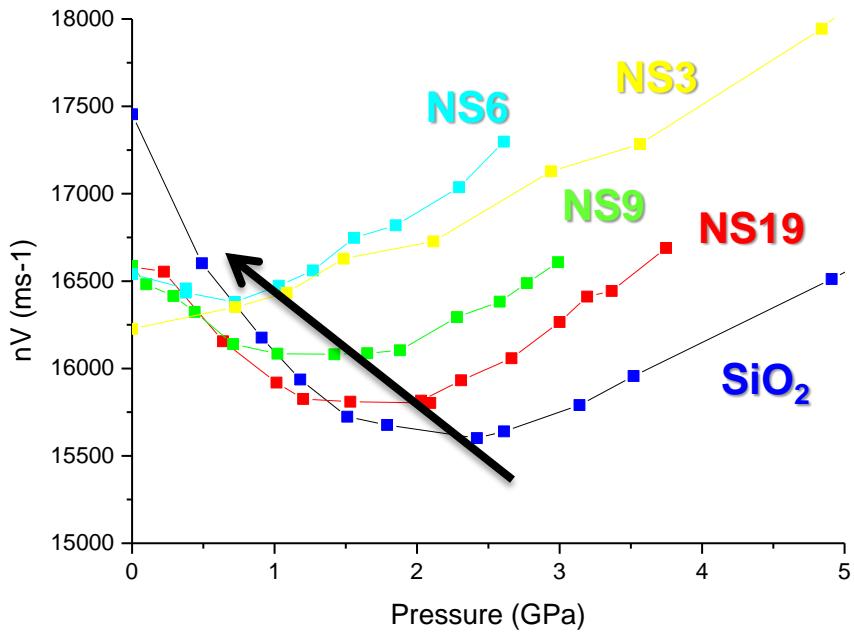
Could be a good candidate  
for a change of mega basin

# $\text{Na}_2\text{O}-\text{Al}_2\text{O}_3-\text{SiO}_2$ system and samples



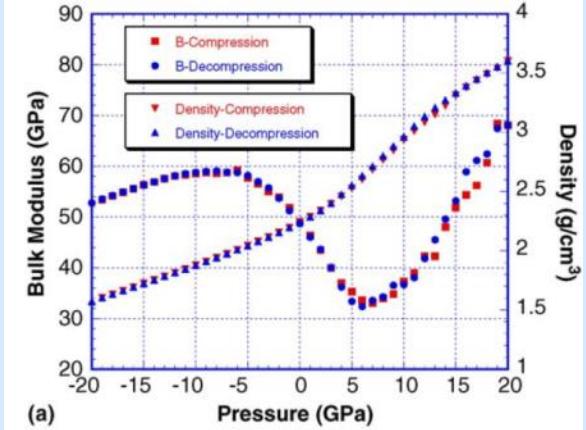
Chemically induced  
transition LDA-HAD  
Limit for NS4

# Binary $\text{SiO}_2\text{-Na}_2\text{O}$



Two possible interpretations:

- Same slope at high pressure shift of the position of the minimum with increasing Na content: shift of the O
- Progressive increase of HDA with Na content

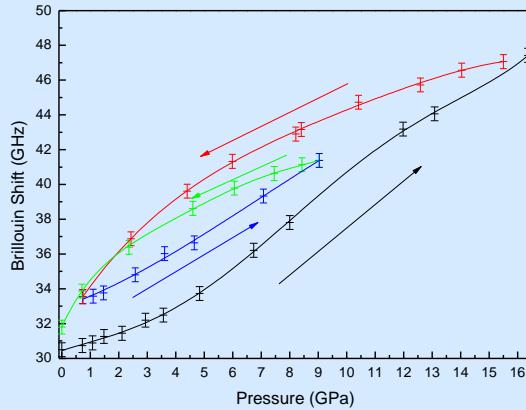


L. Huang et al. / Journal of Non-Crystalline Solids 349 (2004) 1–9

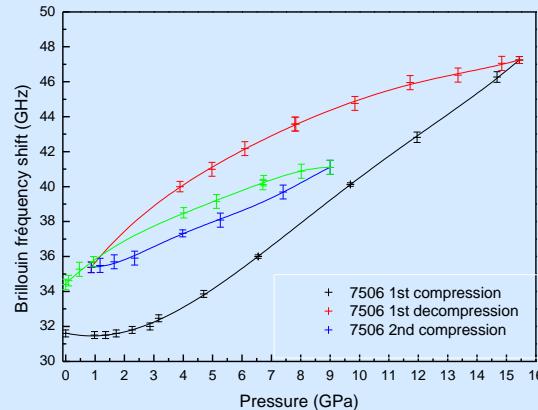
The prediction of Huang's group is correct

# Effect of the Na/Al ratio on permanent densification

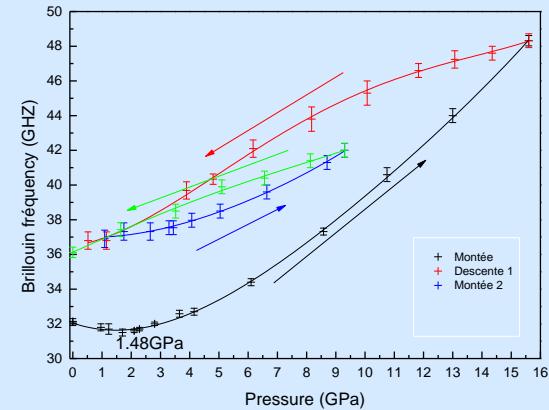
NA 75.00 / NS3



NA 75.06

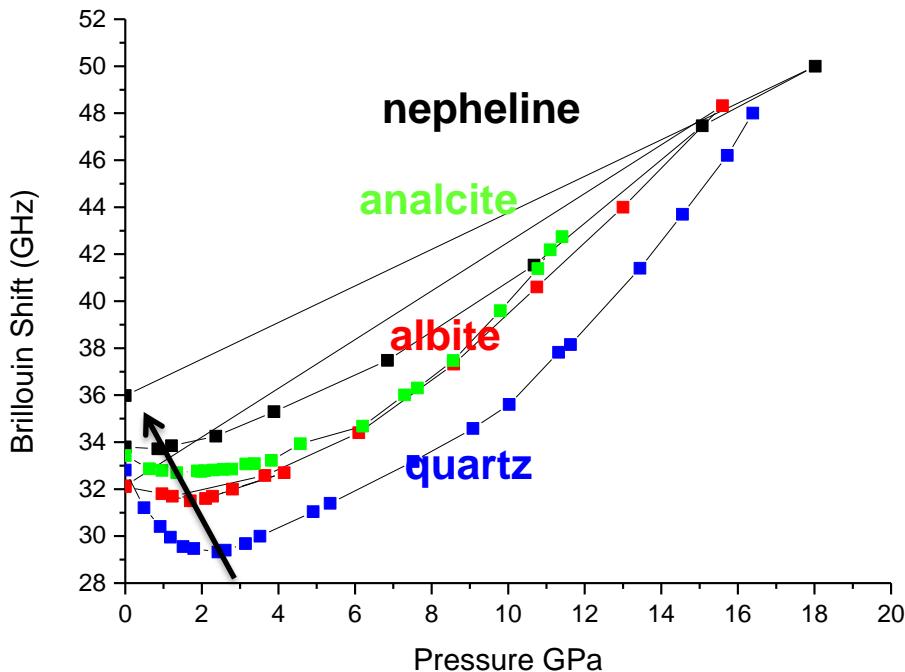


NA 75.12 / Albite



- Elastic hardening with densification higher for polymerized compositions
- NS3 no significant densification so already 100% HDA
- Hysteresis cycles: probably coordination change of Si and Na, bigger when depolymerized

# Decreasing SiO<sub>2</sub> at Al/Na=1

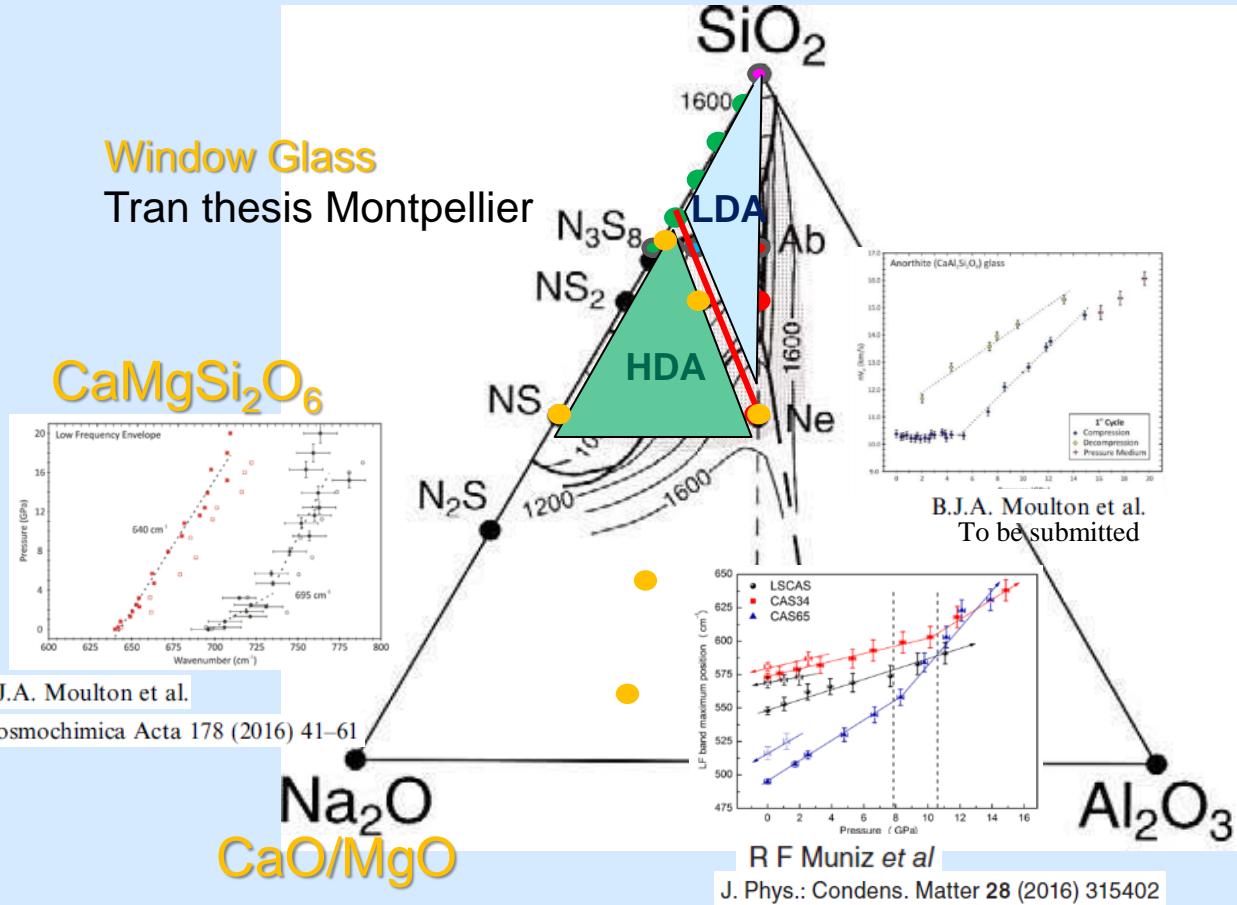


Three possible effects:

- Anomaly can take place with smaller rings
- Al is as good as Si in rings
- Filling the holes by Na

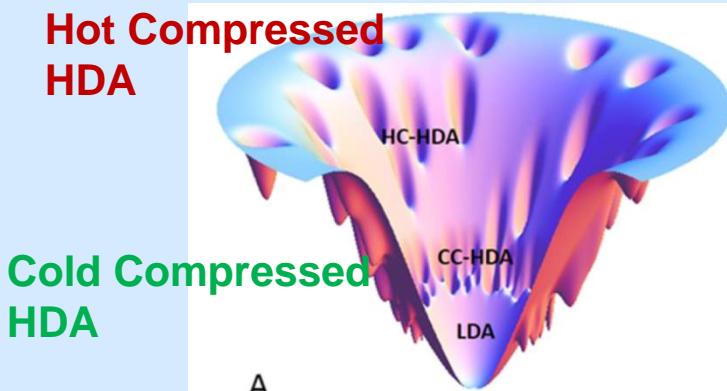
Here the minimum strongly flattened suggesting the disappearance of the structure responsible of the elastic anomaly... HDA formation?

# Limit of the elastic anomaly



# Conclusion

- Modélisation compliquée
- Evolution en Pression très riche en information
- Volume et température fictive liée à la polymerisation
- Polyamorphisme multiple
- HAD/LDA approche qui pourrait peut être aussi aider à la modélisation?



Guerette et al. J. Chem. Phys. 148 (2018)