



Structural role of elements in glasses and melts : Link between structure and properties



Daniel R. Neuville

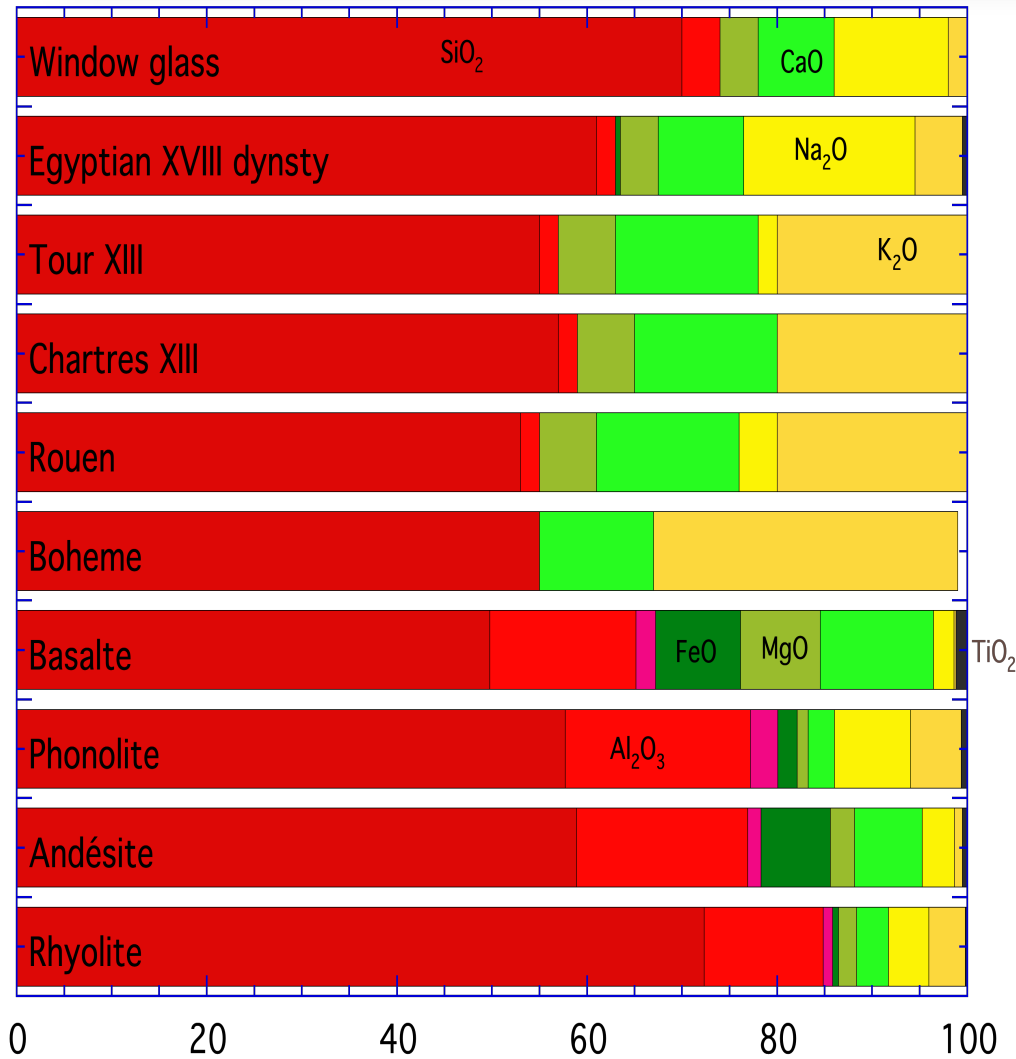
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Rita Cicconi (U. Erlangen),
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Sohei Sukenaga (Tohoku U., Sendai)
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$\text{TO}_2\text{-MO-M}'_2\text{O}$, with $\text{T} = \text{Si, Al, Fe}^{3+}$, $\text{M} = \text{Mg, Ca, Fe}^{2+}$, $\text{M}' = \text{Li, Na, K}$



$\Leftrightarrow$



Properties versus Structure ?



$\text{TO}_2\text{-MO-M}'_2\text{O}$, avec $\text{T} = \text{Si, Al, Fe}^{3+}$, $\text{M} = \text{Mg, Ca, Fe}^{2+}$, $\text{M}' = \text{Li, Na, K}$



Glasses, melts = *network former* + *alkali or earth-alkaline elements* + *transition elements*

What is a network modifier or charge compensator ?



Why alkaline or earth-alkaline element changes role?

What happens in the case of transition elements?



What happens during nucleation processes?



$\text{TO}_2\text{-MO-M}'_2\text{O}$, avec $\text{T} = \text{Si, Al, Fe}^{3+}$, $\text{M} = \text{Mg, Ca, Fe}^{2+}$, $\text{M}' = \text{Li, Na, K}$



How network former can be mixed?

What is the redox effect on glass forming?

How made an invert glasses?

How elaboration processes can influence glass forming ability?

Fragility and ability of glass forming?

Role of element can change as a function of their content?



$\text{TO}_2\text{-MO-M}'_2\text{O}$, avec $\text{T} = \text{Si, Al, Fe}^{3+}$, $\text{M} = \text{Mg, Ca, Fe}^{2+}$, $\text{M}' = \text{Li, Na, K}$



Is there and universal definition of glass former?

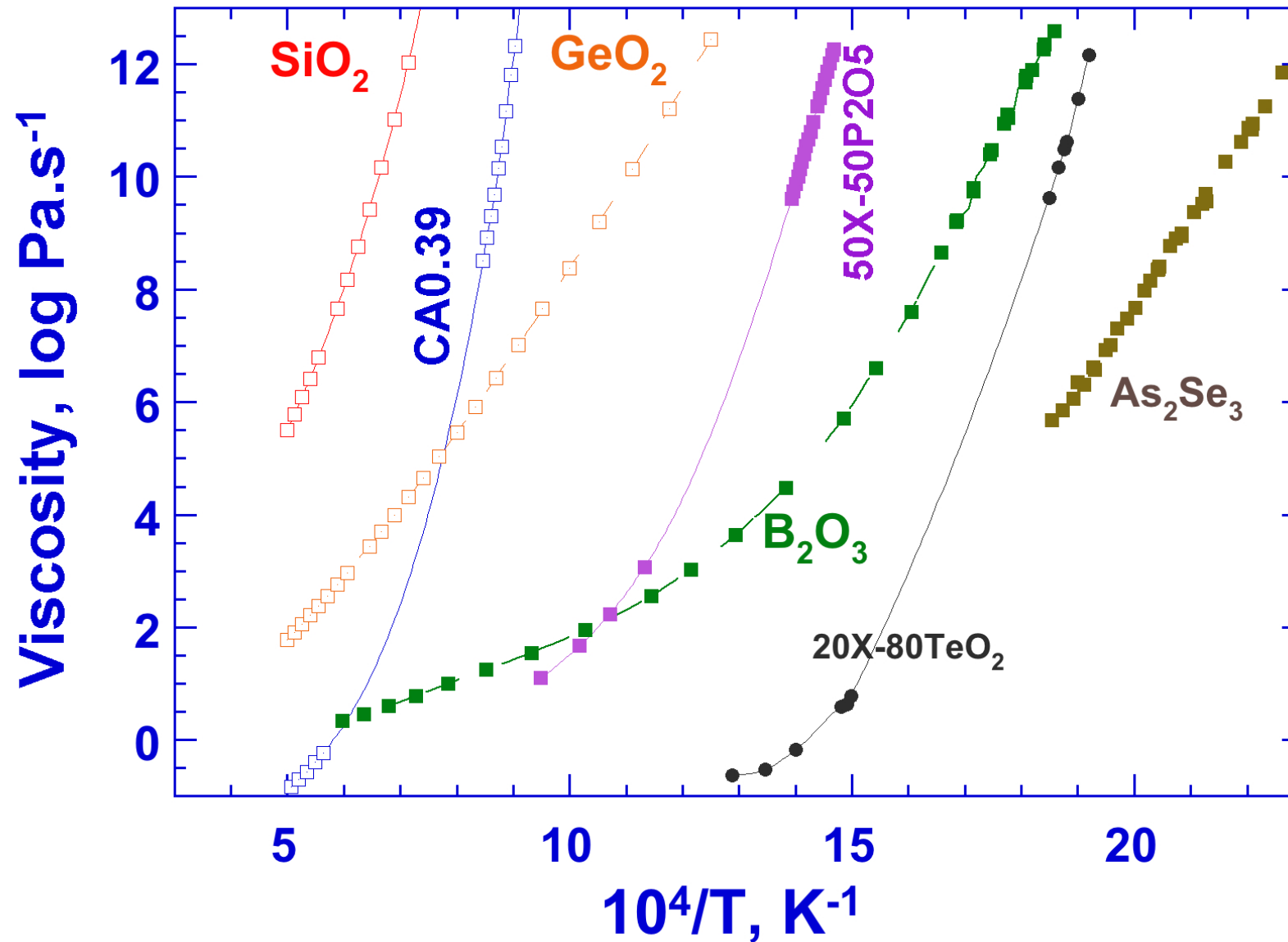
Does the definition of glass formers depends on the type of glass systems?

Does the definition of glass formers evaluate with new analytical tools?

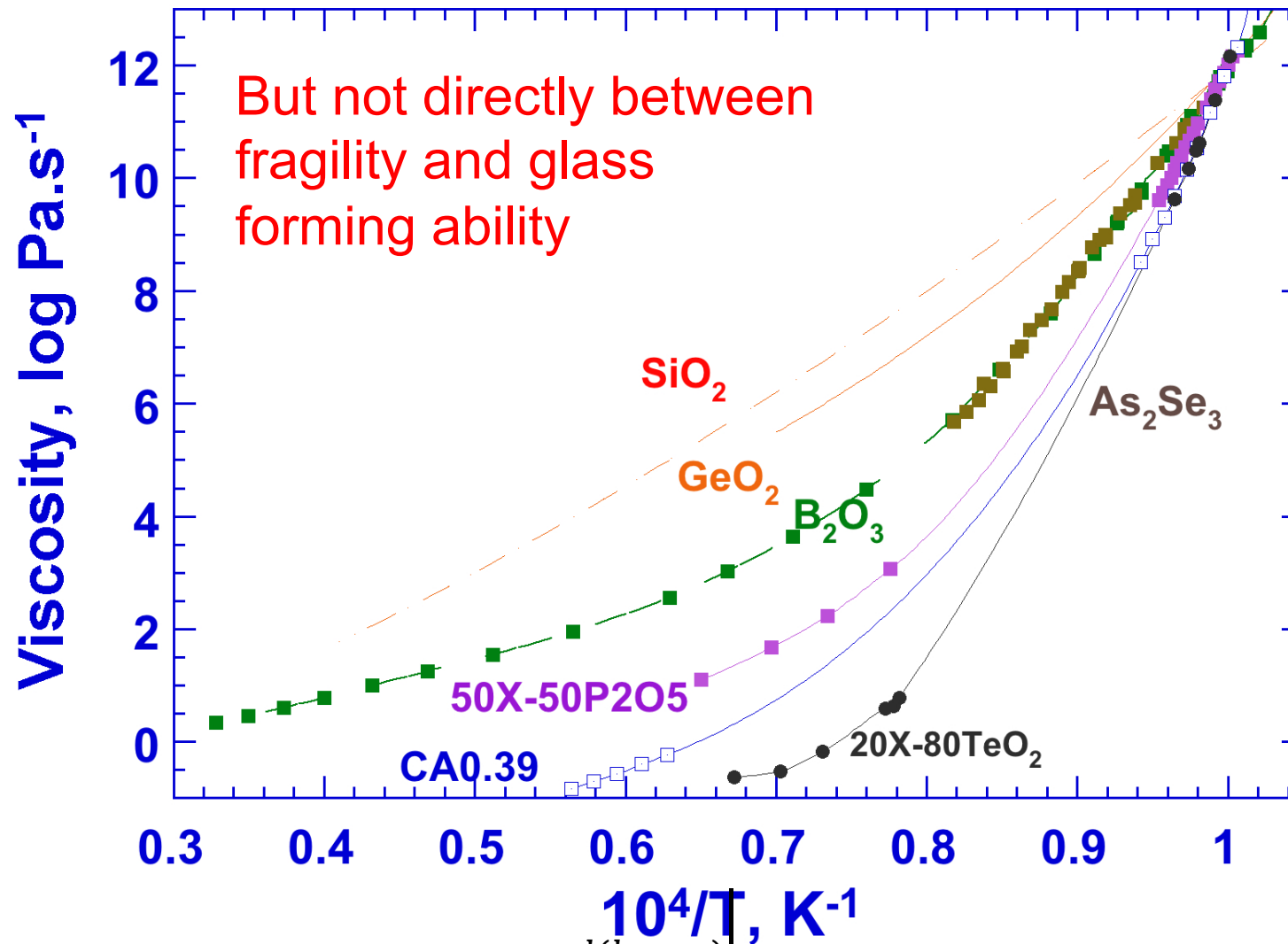
Does modelisation enable to get a different view of glass forming effect?



Viscosity versus chemical composition



Fragility versus chemical composition

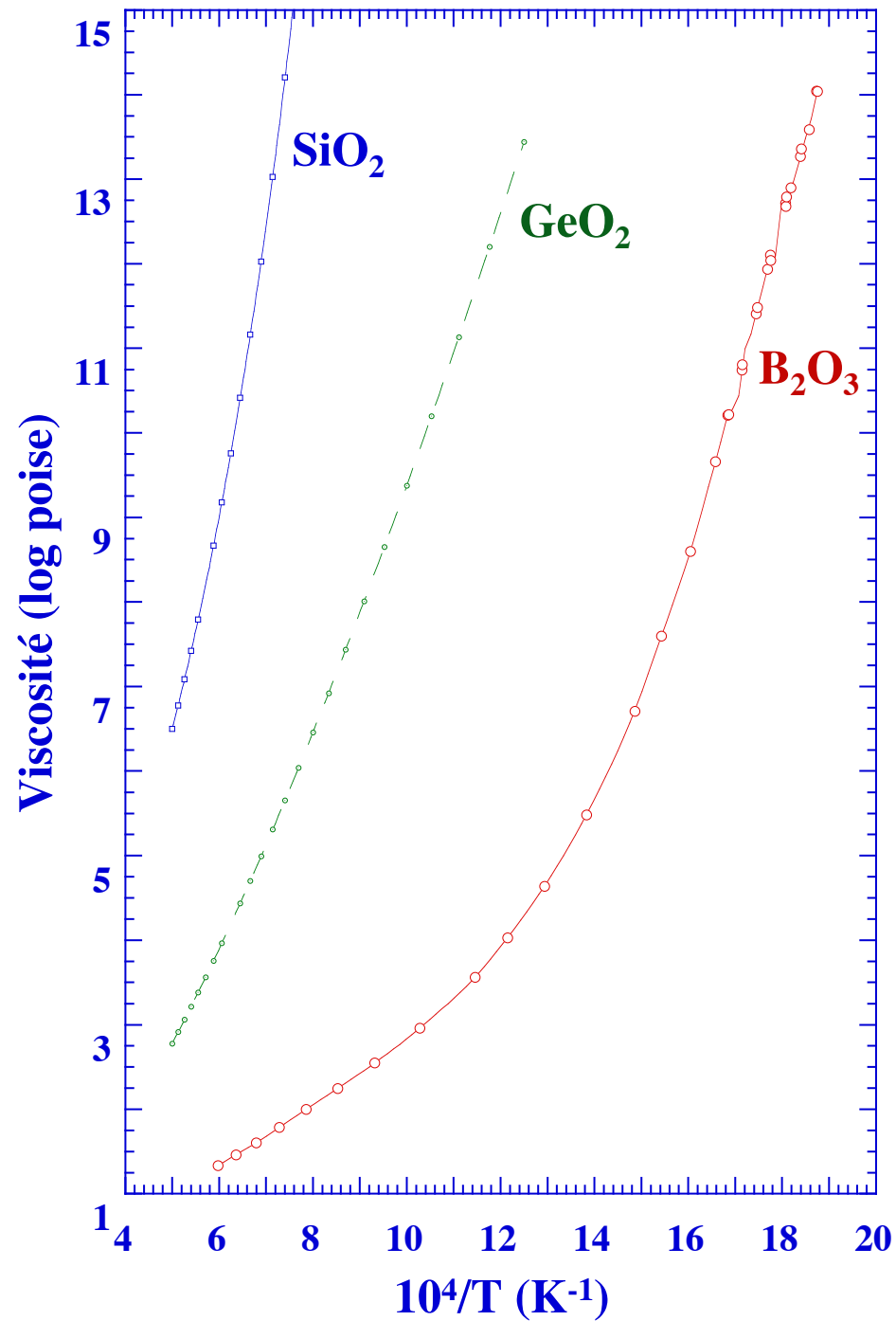


Fragility can be express : $m = \frac{d(\log_{10}\eta)}{d(\frac{T_g}{T})} \Big|_{T=T_g}$ And also by $C_p/C_{pg}(T_g)$



Network former...





Formateurs de réseau

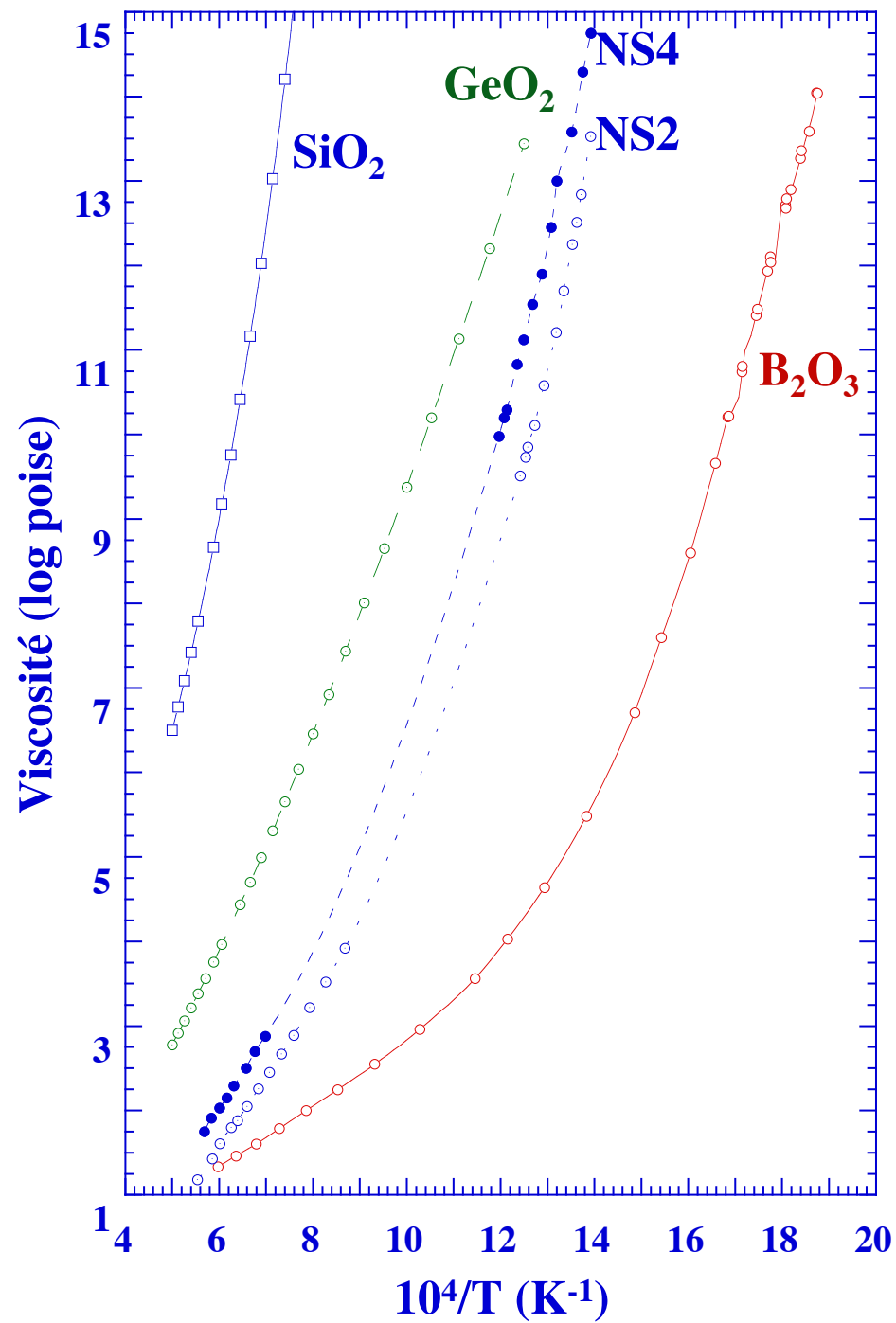


SiO₂ liaison covalente
tétraèdre SiO₄

GeO₂ liaison covalente
tétraèdre GeO₄

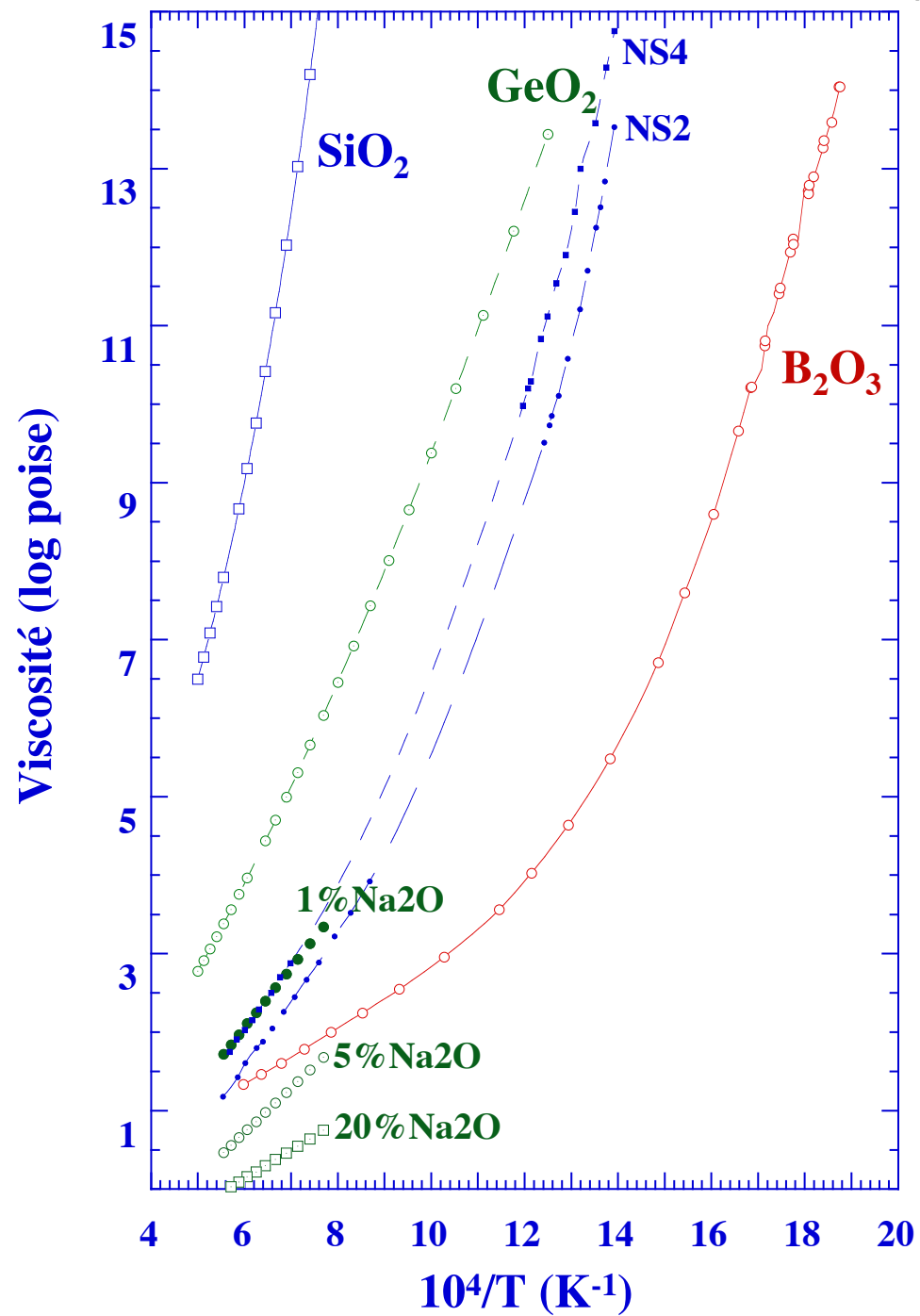
B₂O₃ liaison covalente
Triangle BO₃ et
tétraèdre BO₄





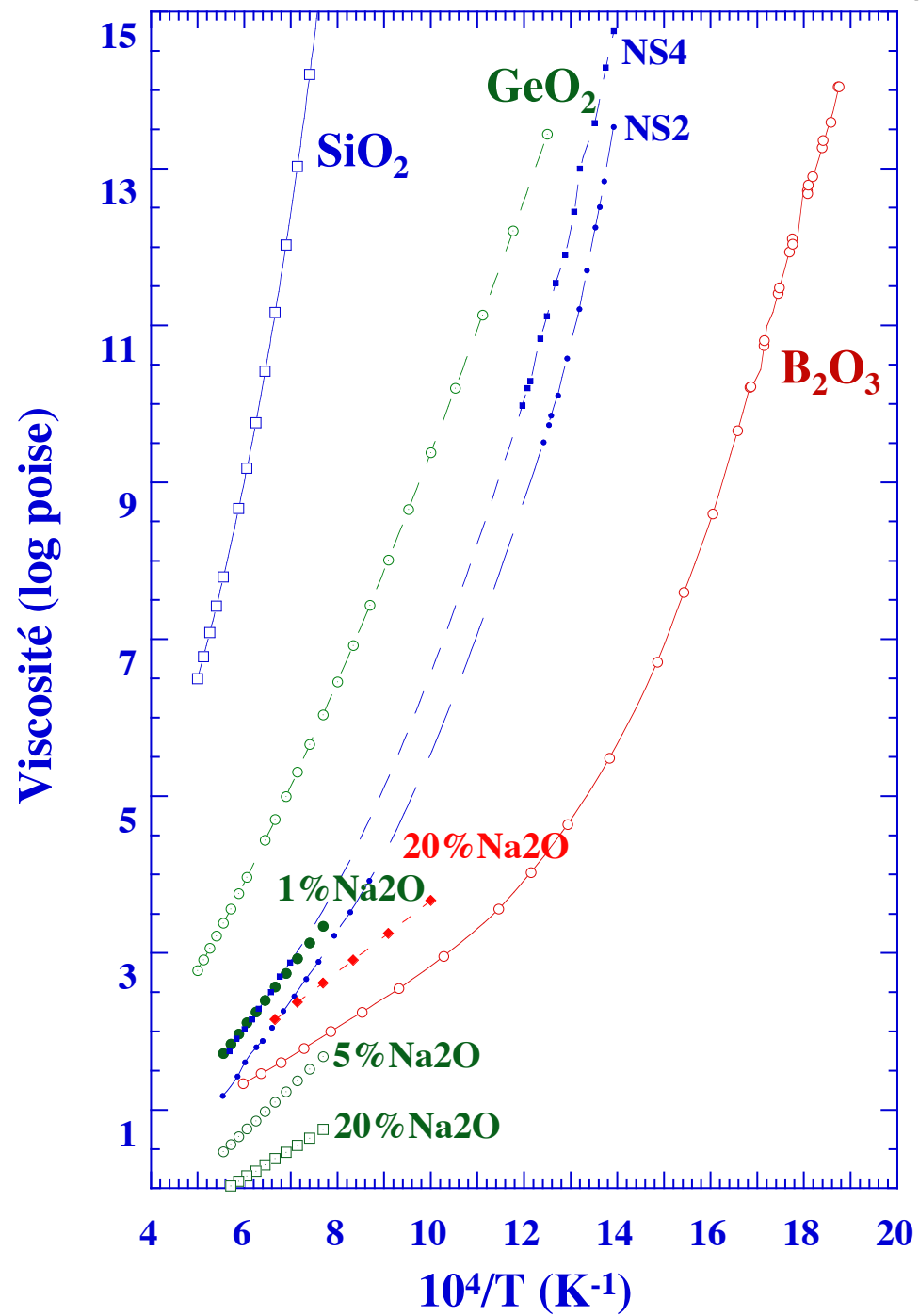
GDR-Verre-USTV-TC3-Neuville- Link between structure and properties





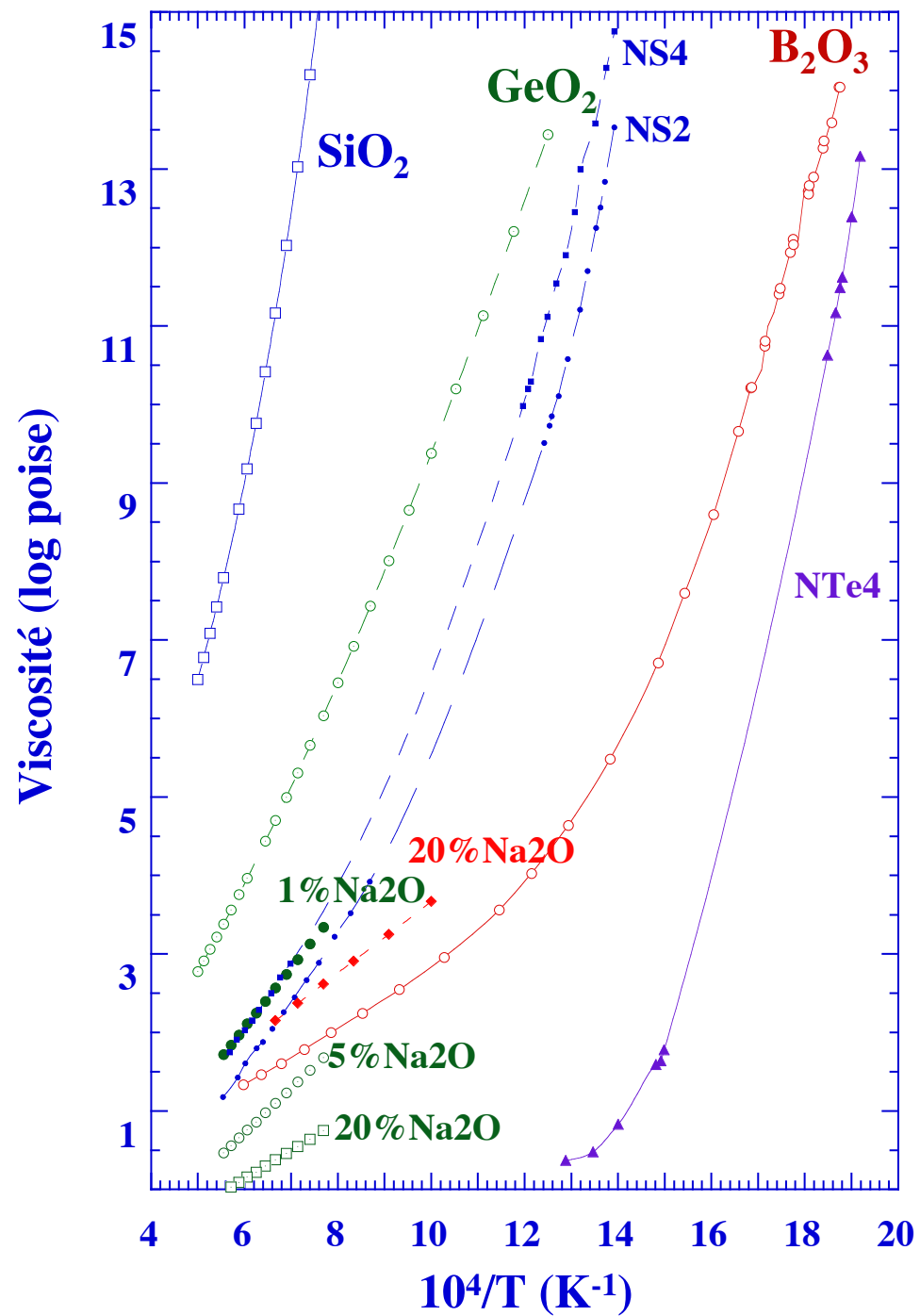
GDR-Verre-USTV-TC3-Neuville- Link between structure and properties





GDR-Verre-USTV-TC3-Neuville- Link between structure and properties





DR-Verre-USTV-TC3-Neuville- Link between structure and properties



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Network former

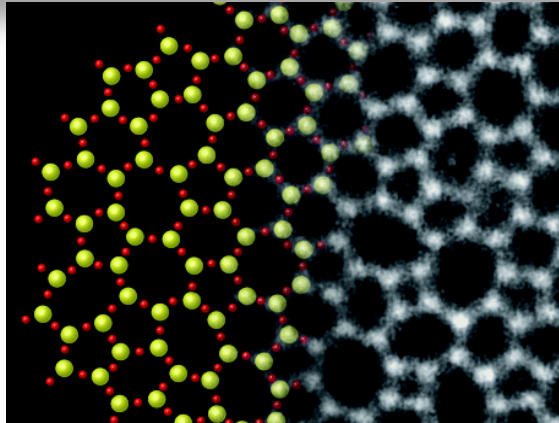
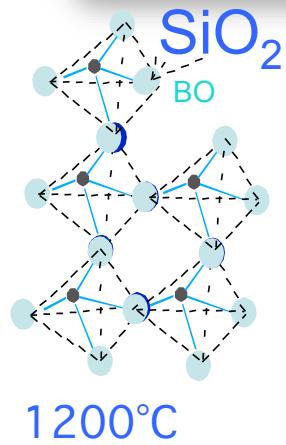


SiO_2 : SiO_4 tetrahedra, increase connectivity, viscosity and T_g , = strong liquid
Alkaline broke network, viscosity decreases, T_g

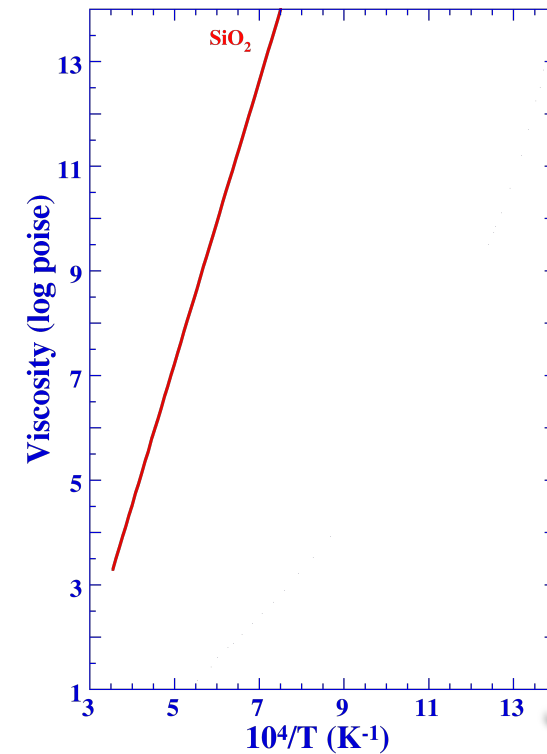
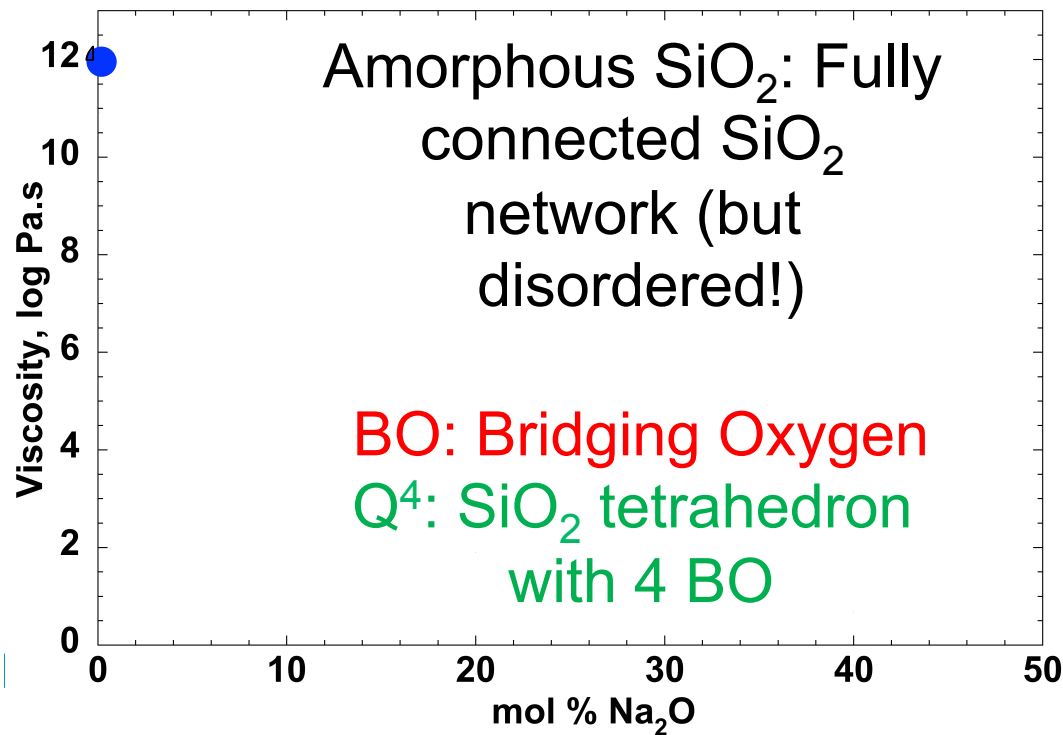
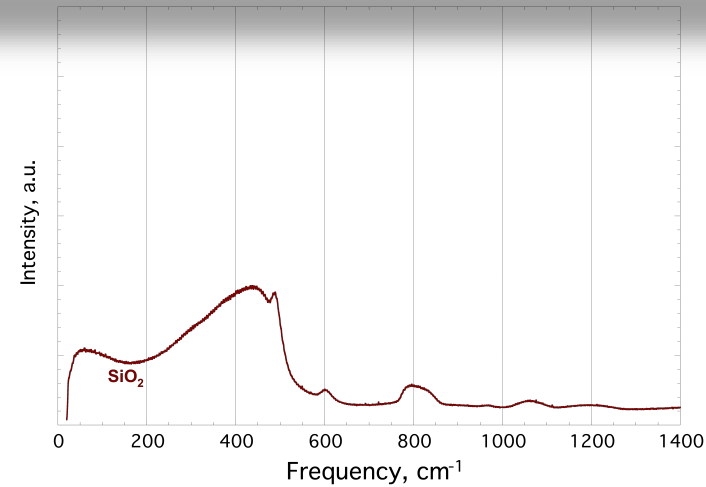
GeO_2 : GeO_4 tetrahedra, increase connectivity, viscosity and T_g , = strong liquid
Alkaline broke network, viscosity decreases, T_g

B_2O_3 : BO_3 triangle, 2D connectivity, low viscosity, T_g ...
fragile liquid
With M_2O $\text{BO}_3 \Rightarrow \text{BO}_4$

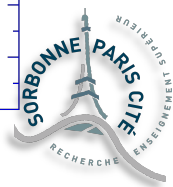
Structure versus properties of silicate melts



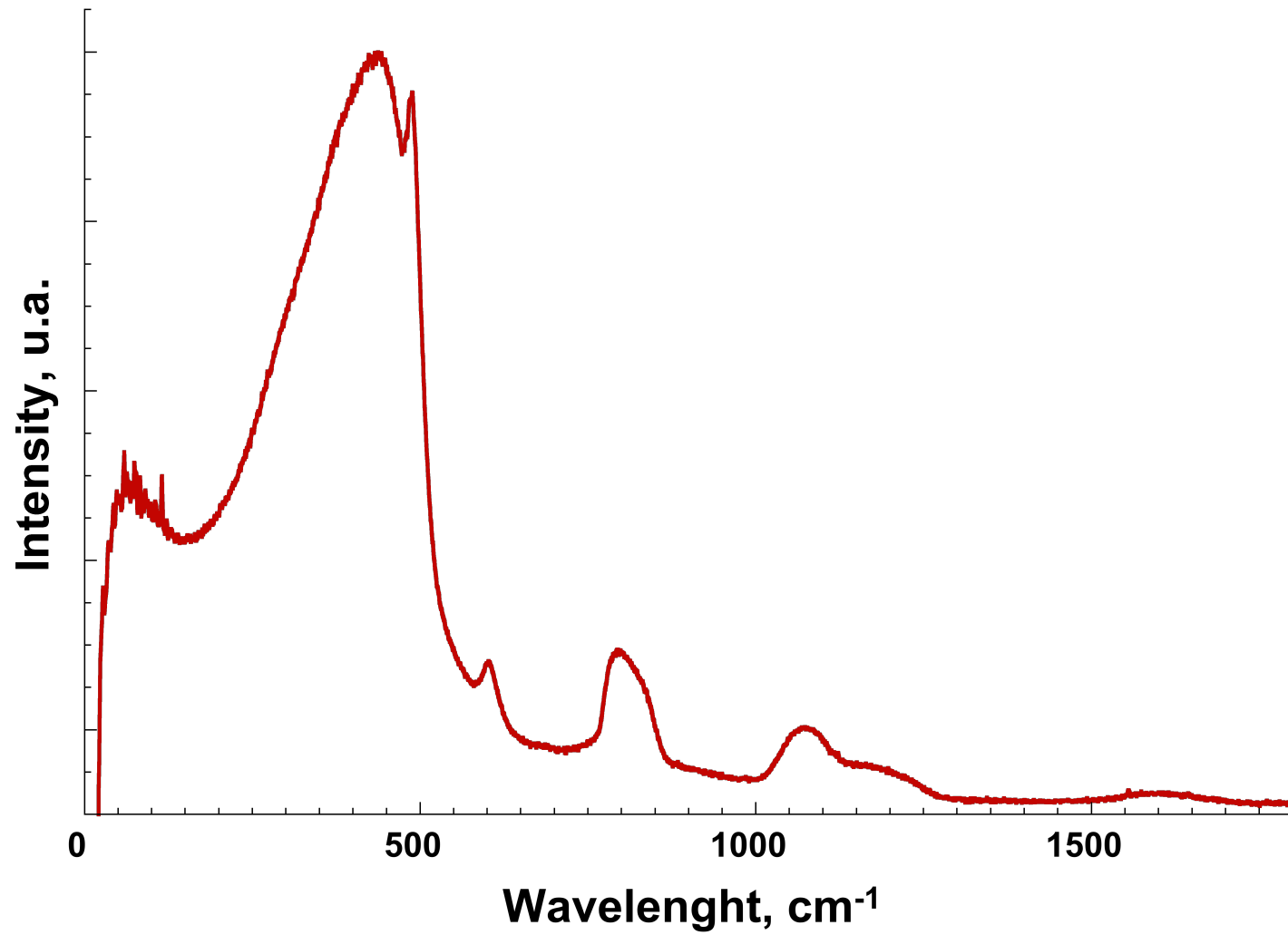
Huang et al. (2012): HRTEM picture of thin film of amorphous SiO_2



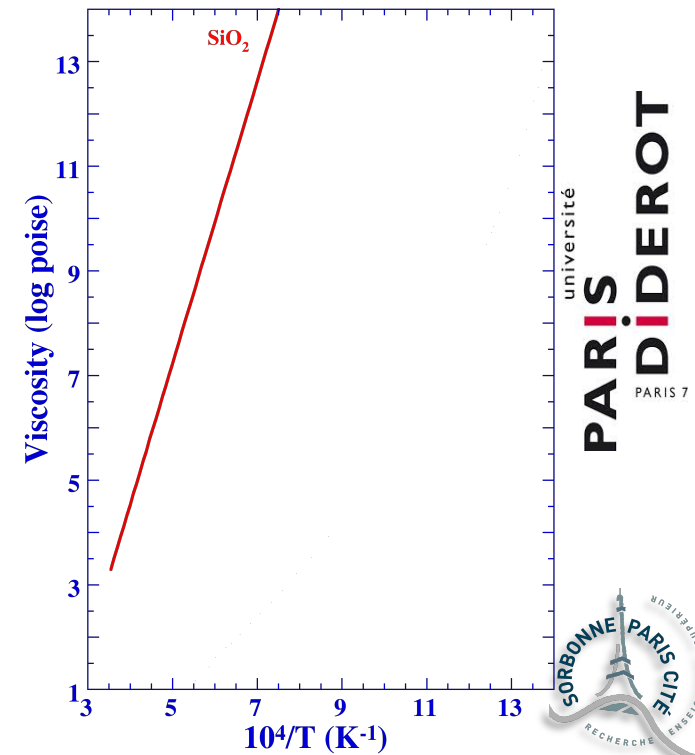
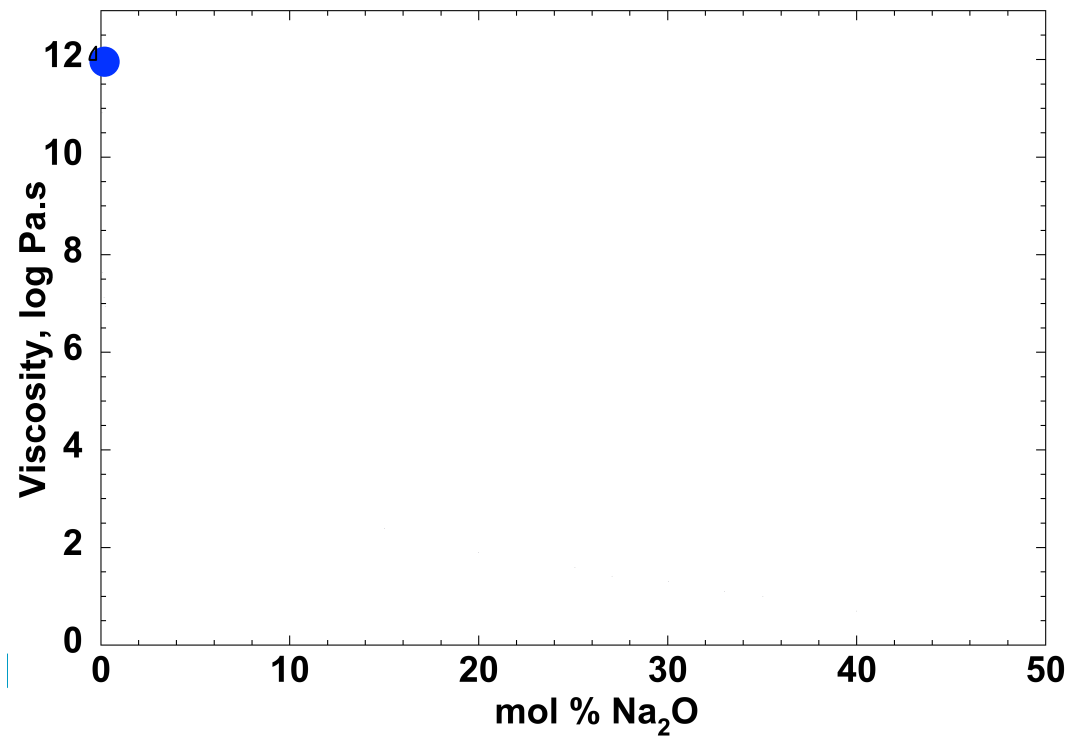
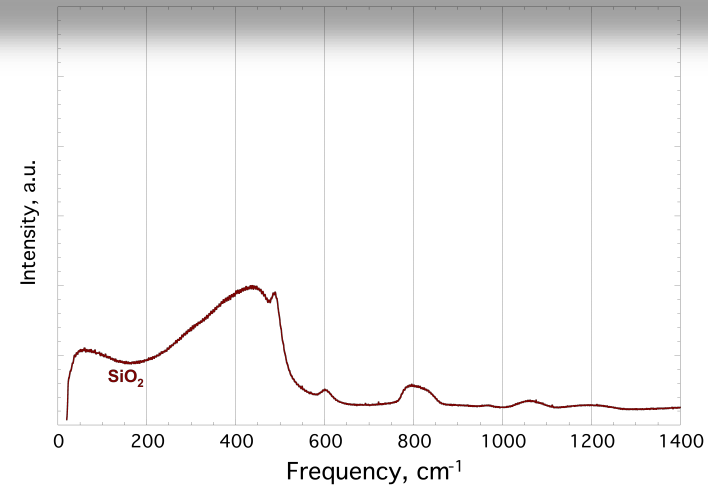
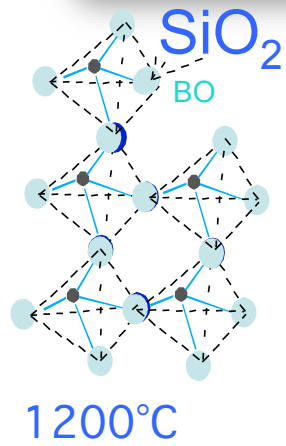
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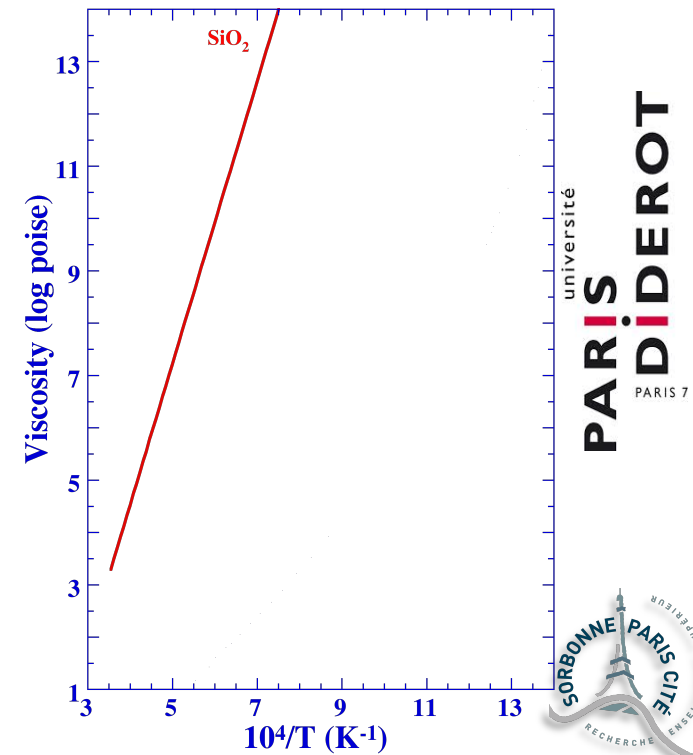
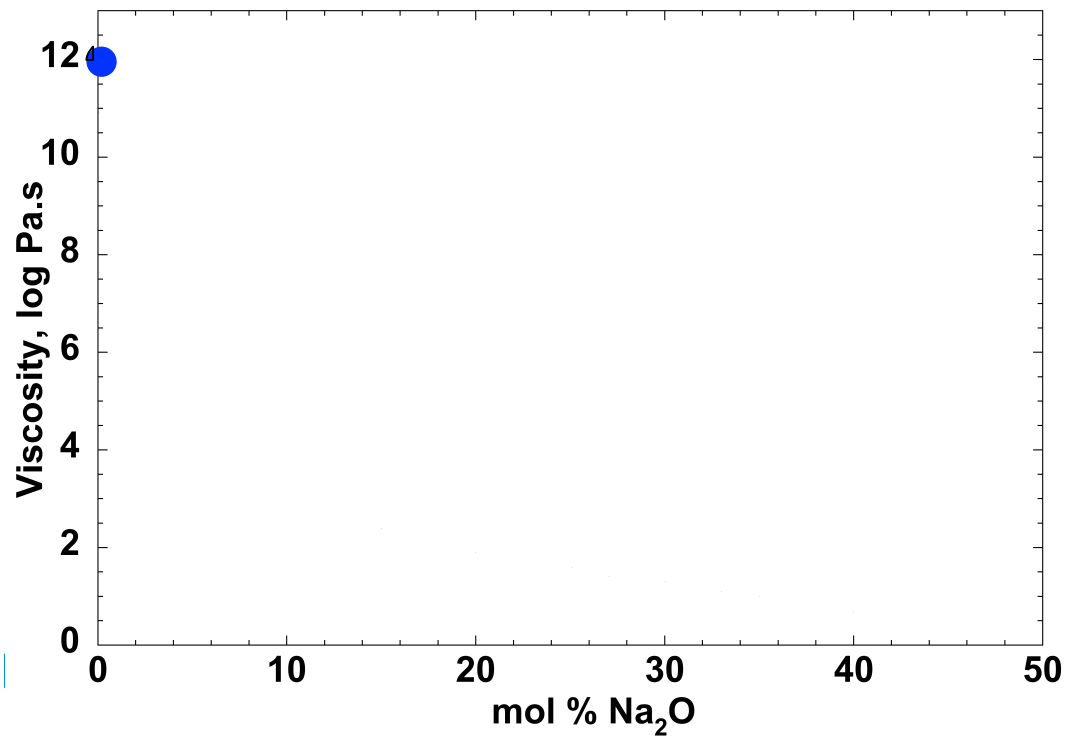
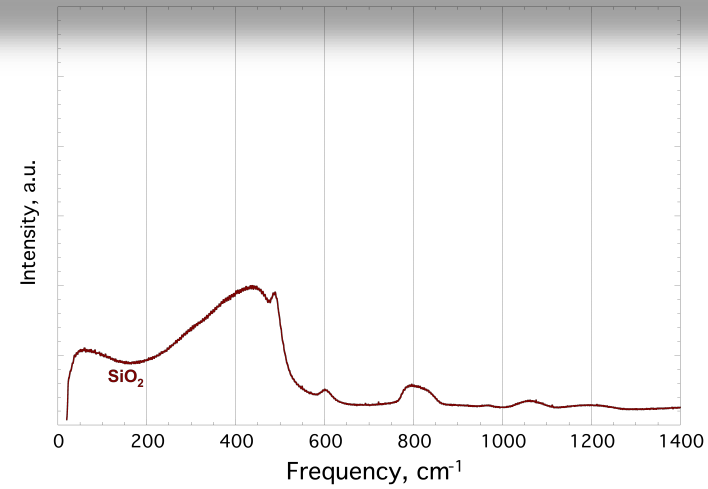
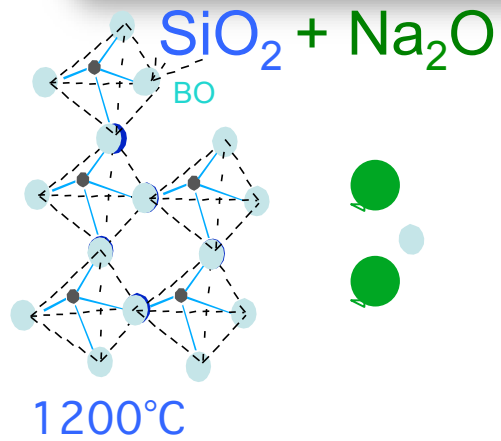
Raman : vibrations of ν -SiO₂



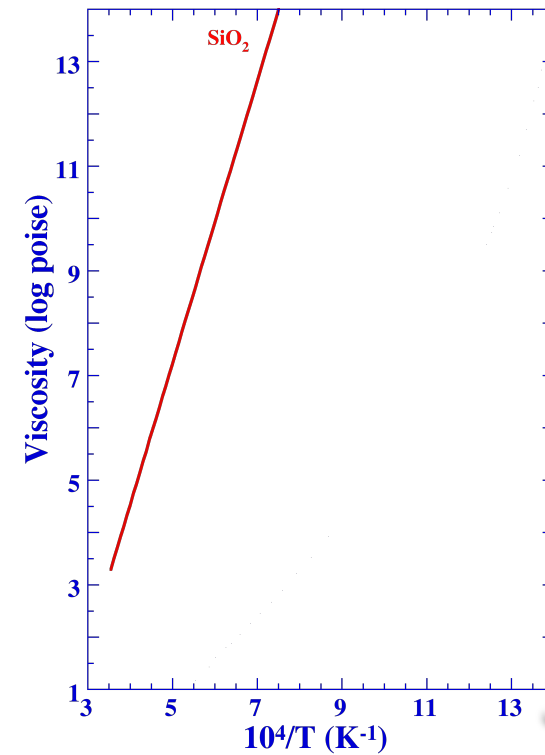
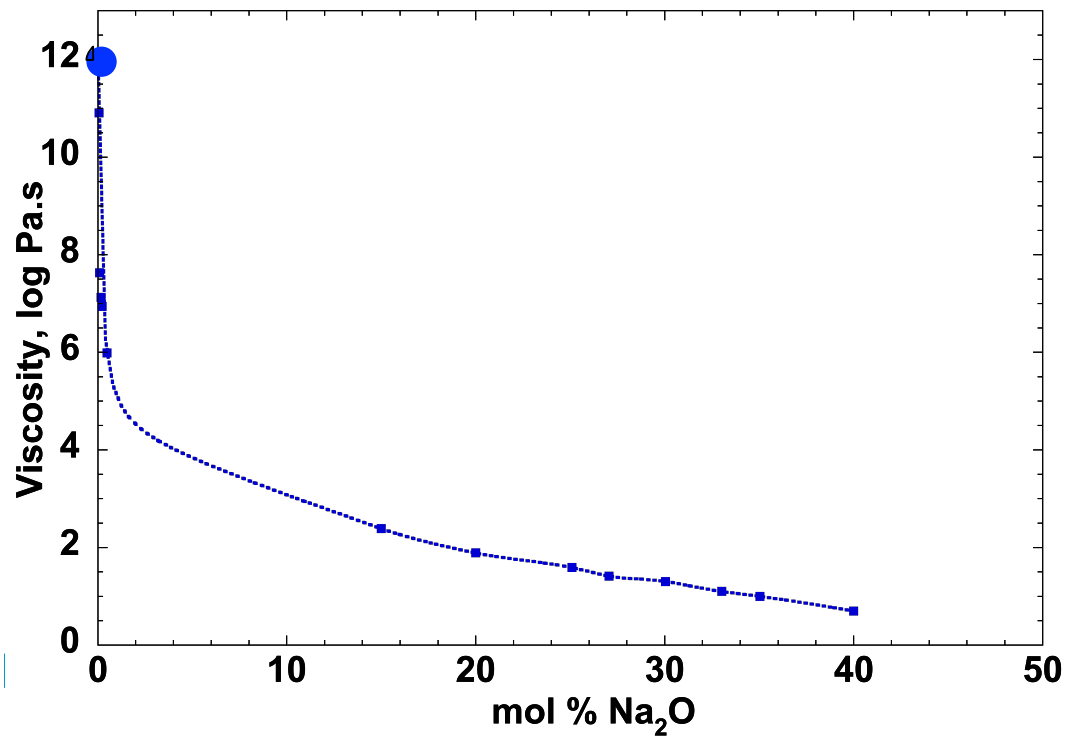
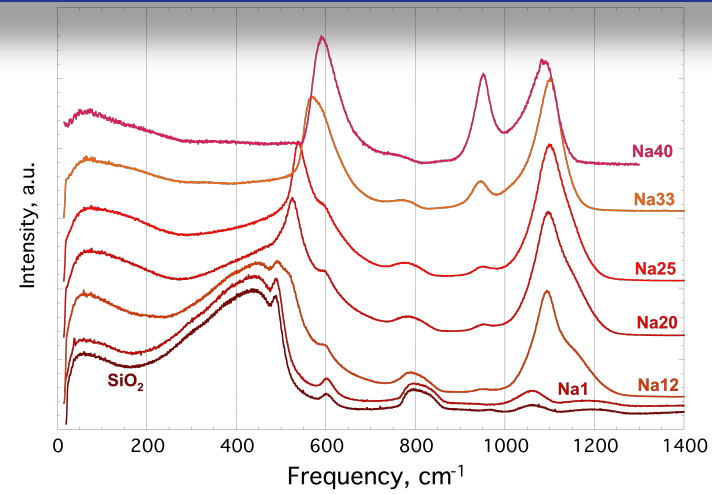
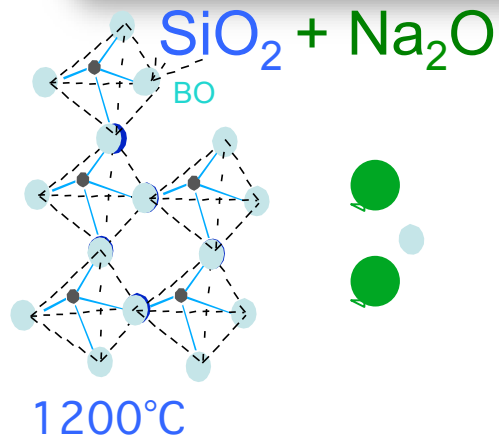
Structure versus properties of silicate melts



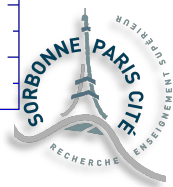
Structure versus properties of silicate melts



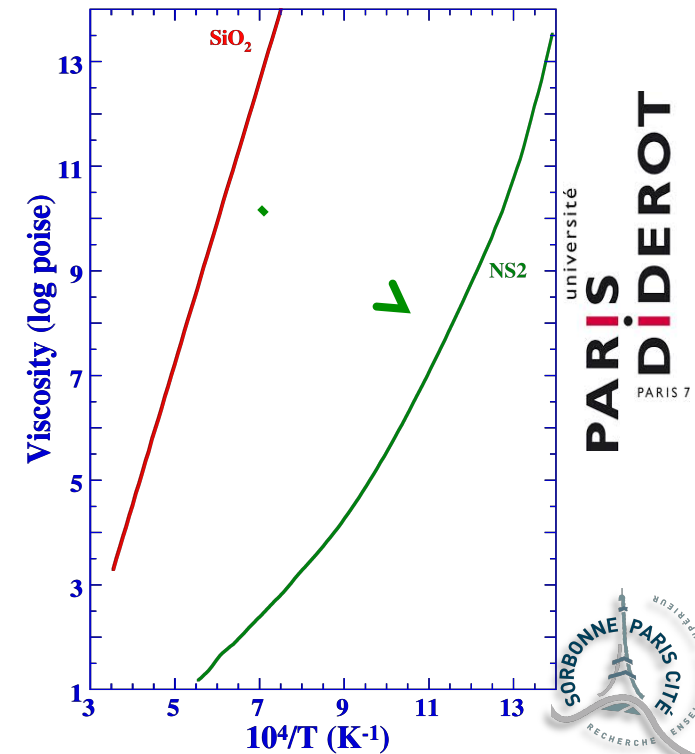
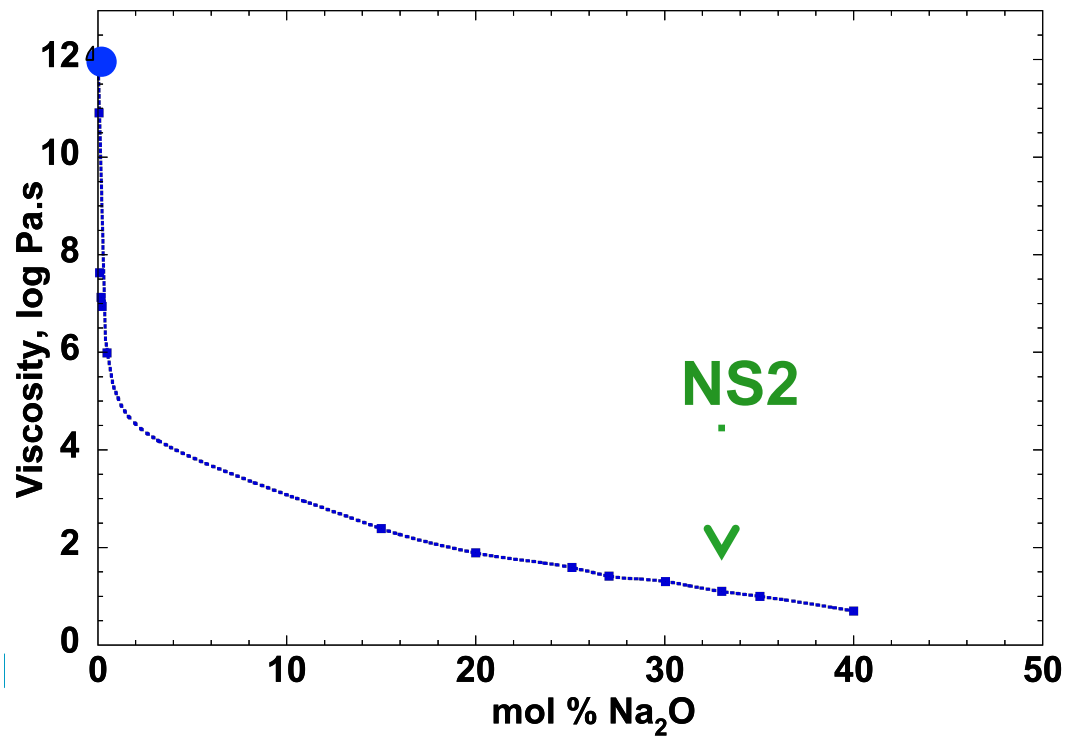
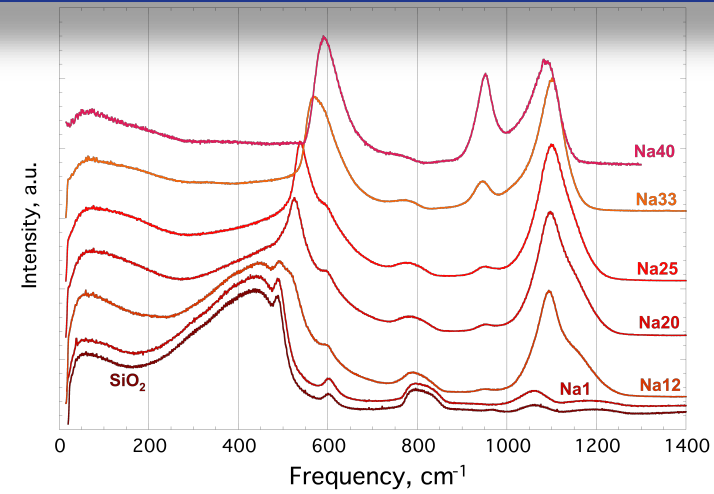
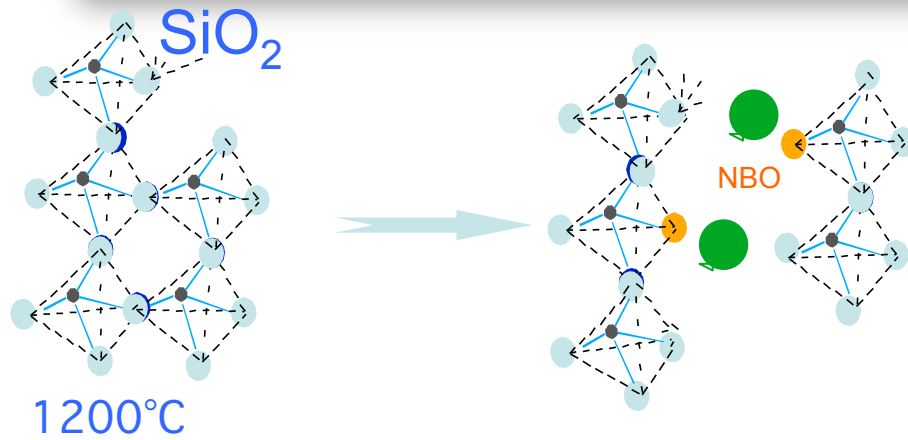
Structure versus properties of silicate melts



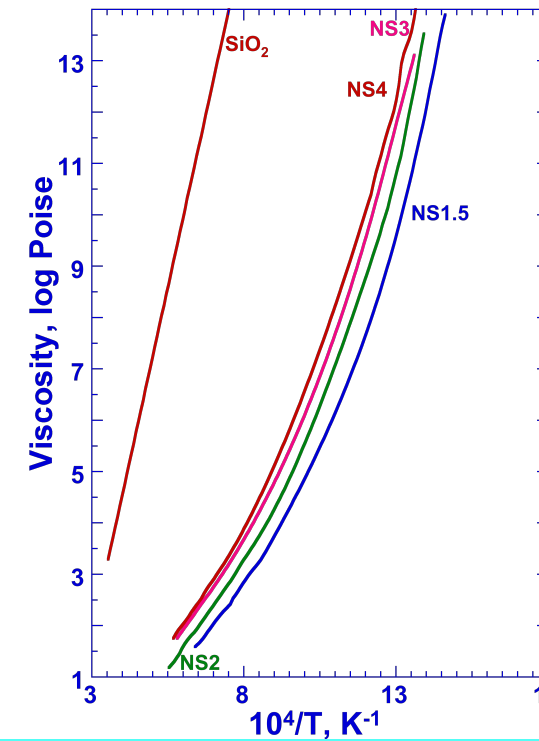
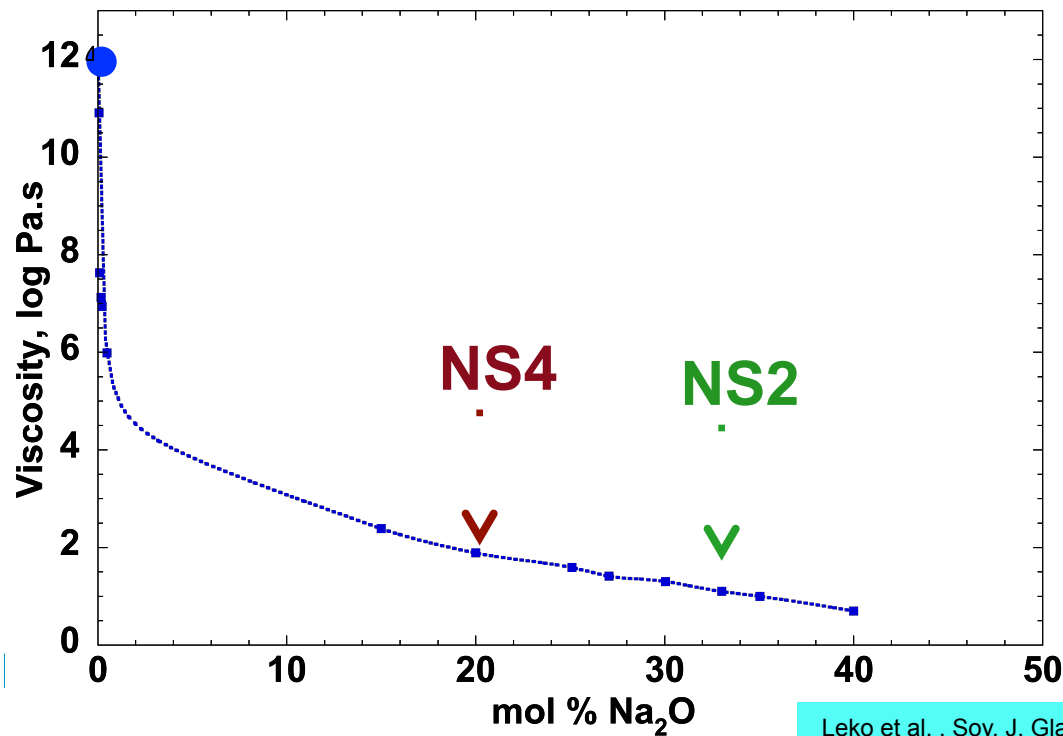
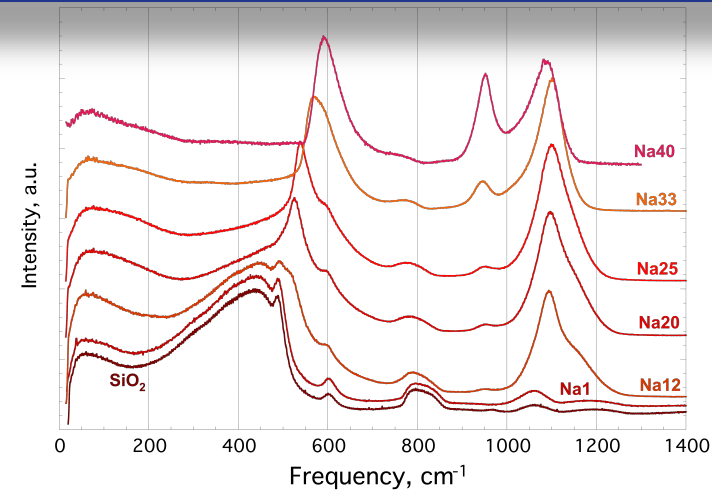
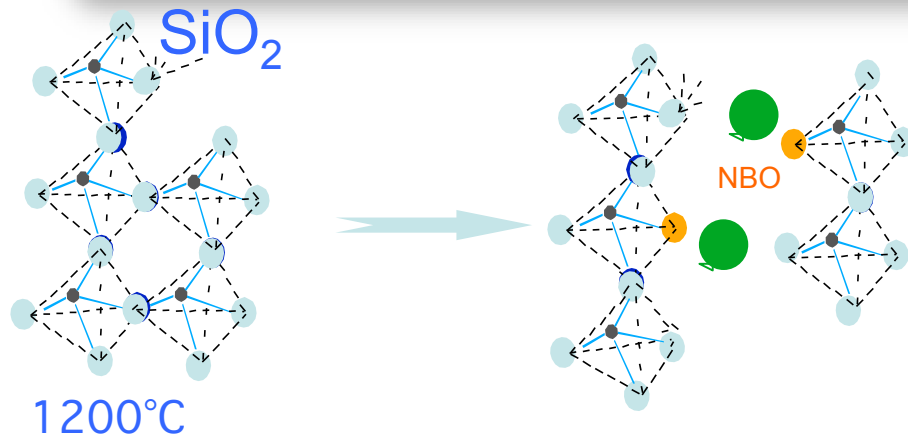
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Structure versus properties of silicate melts



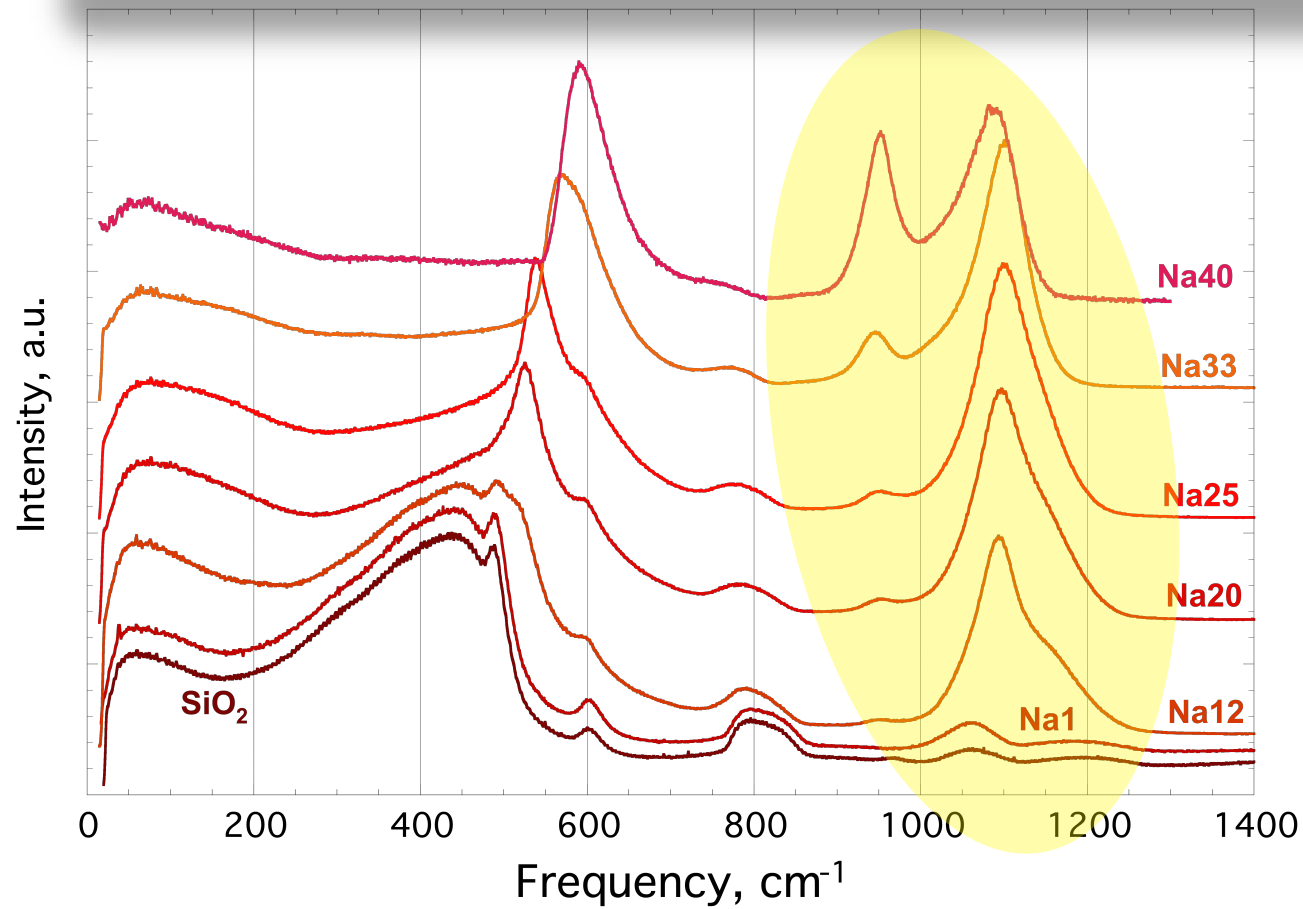
Structure versus properties of silicate melts



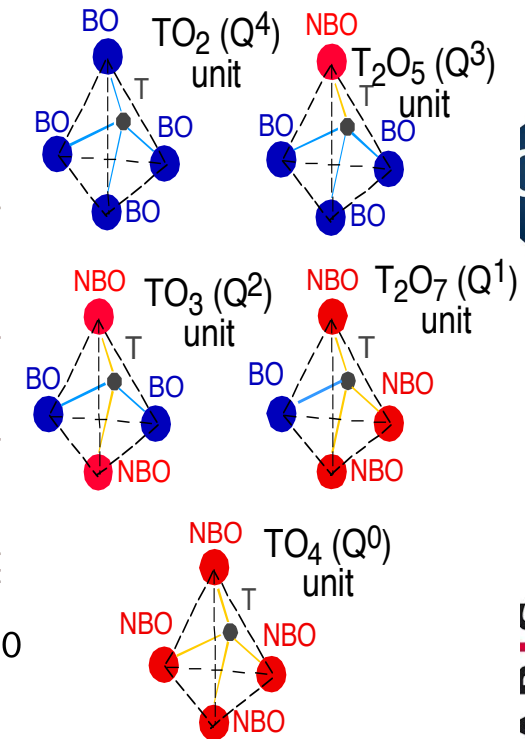
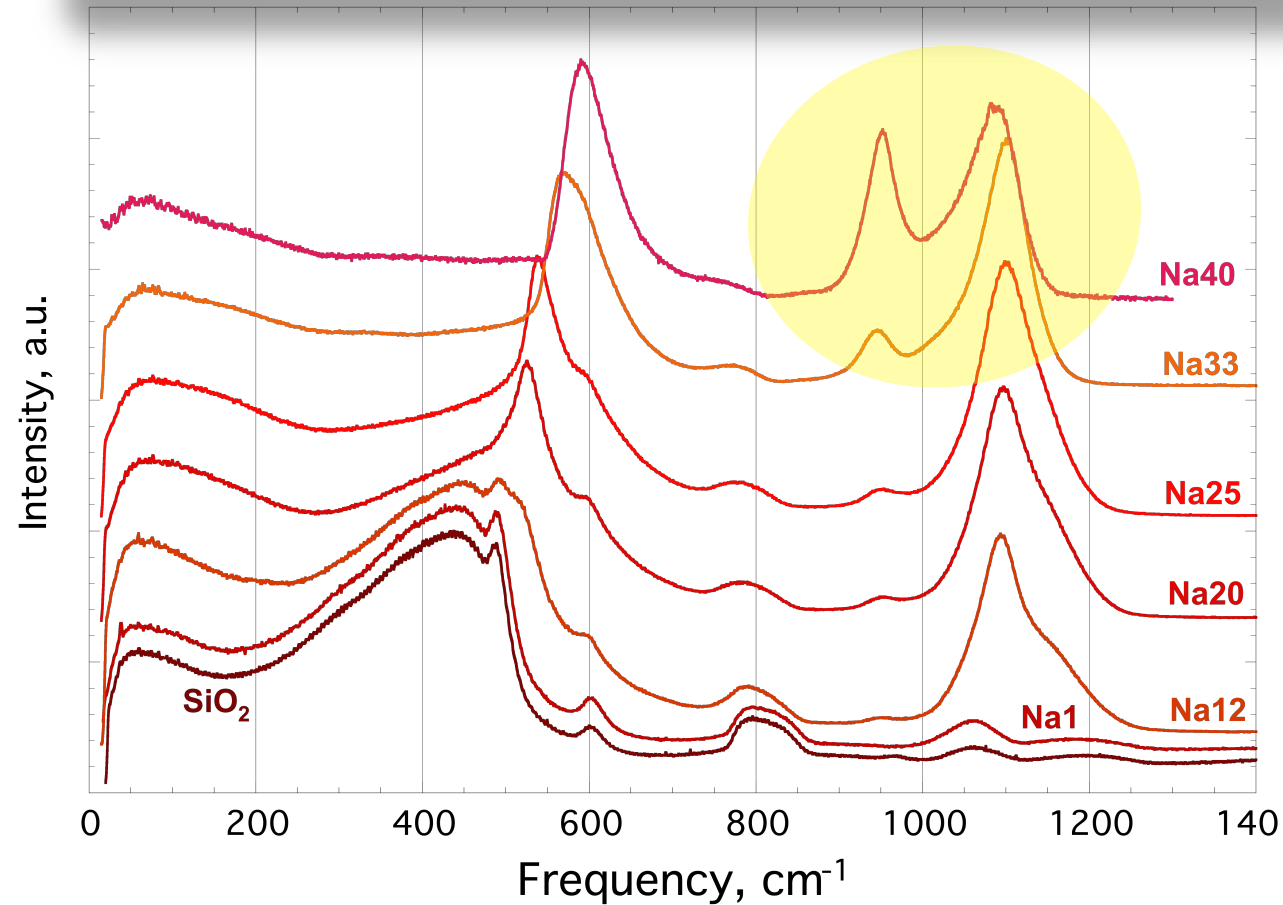
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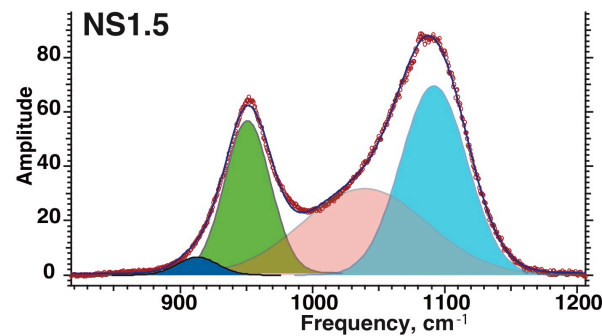
Raman spectra of SiO_2 - Na_2O glass



Raman spectra of $\text{SiO}_2\text{-Na}_2\text{O}$ glass

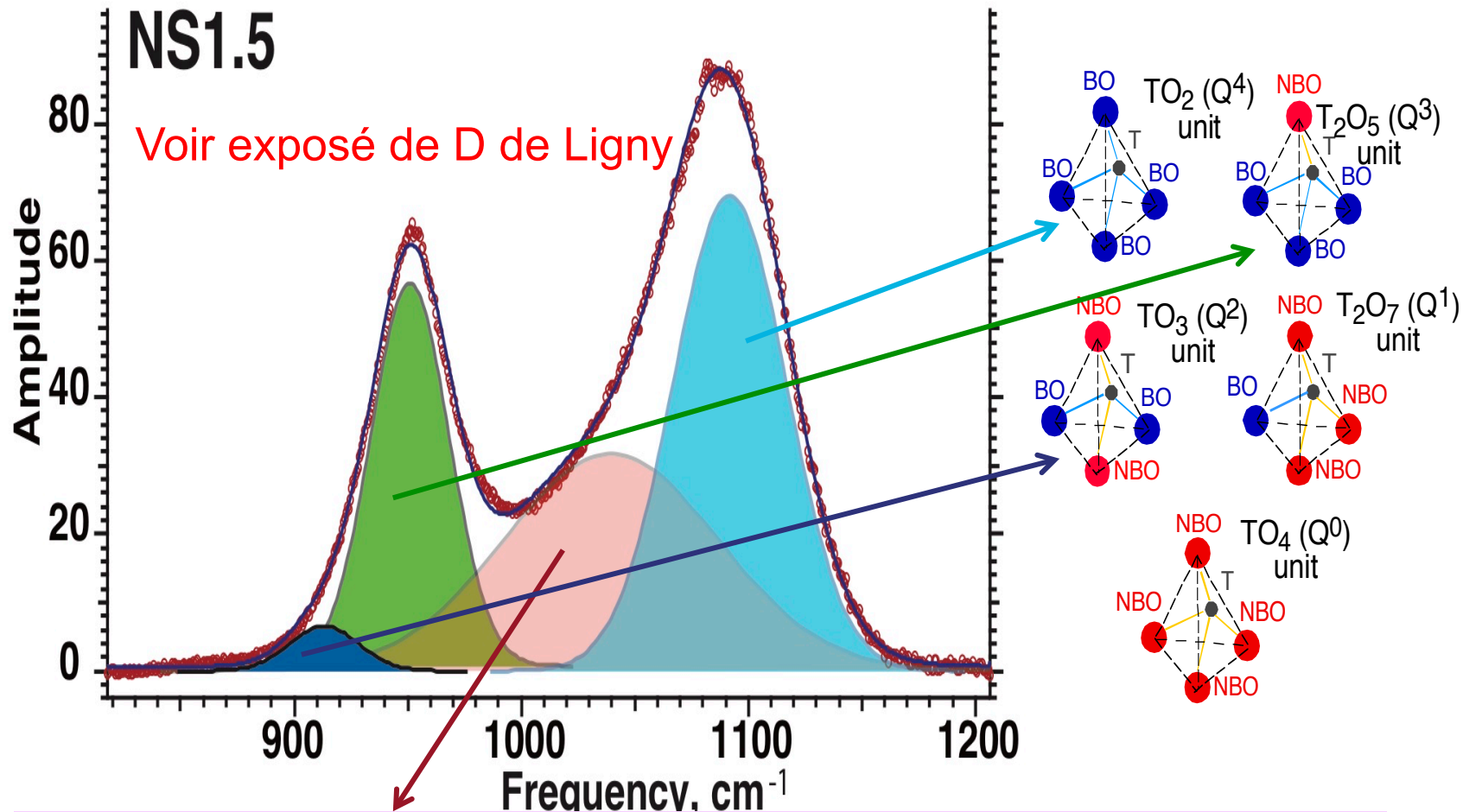


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Raman spectra of $\text{SiO}_2\text{-Na}_2\text{O}$ glass Q^n species

T-O-T stretching vibration



T_2 mode of TO_4 tetrahedra, Q^4 , two oxygen atoms moving closer to the central Si atom while the two others oxygen atoms are moving away (Sarnthein et al. 1997; Taraskin and Elliott 1997; Pasquarello et al. 1998)



NMR

 Q^n species

X-ray absorption

Voir exposé de G. Calas



Voir exposé de P. Florian

^{29}Si MAF Spectrum of $\text{K}_2\text{O}-2\text{SiO}_2$ glass
Davis et al., J. Phys. Chem. A, 2010, 114 5503–5508

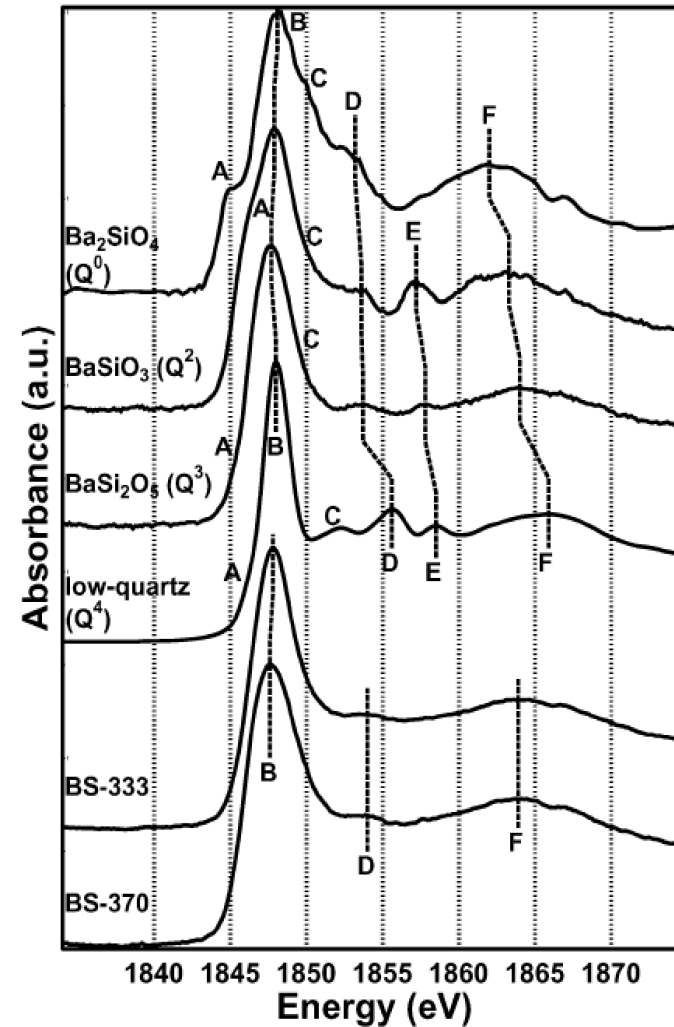
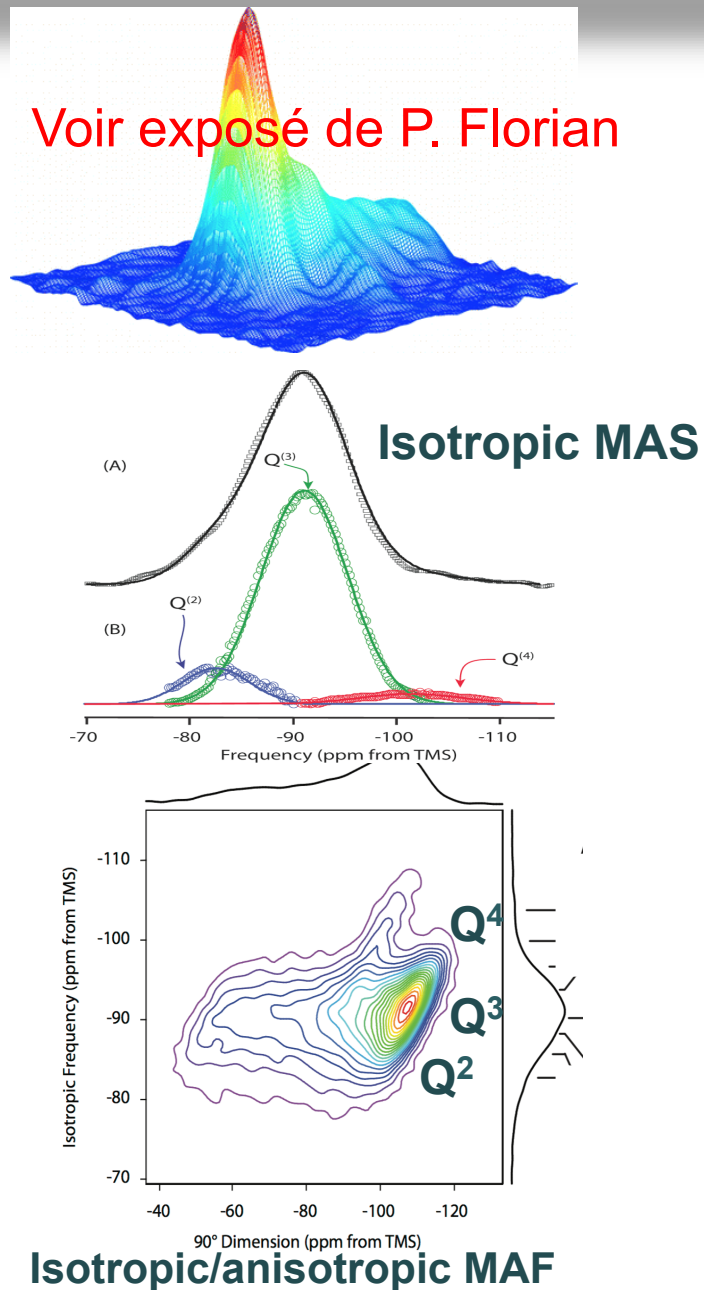


Fig. 1. Si K-XANES spectra of barium silicates, low quartz and the amorphous samples BS-333 and BS-370.

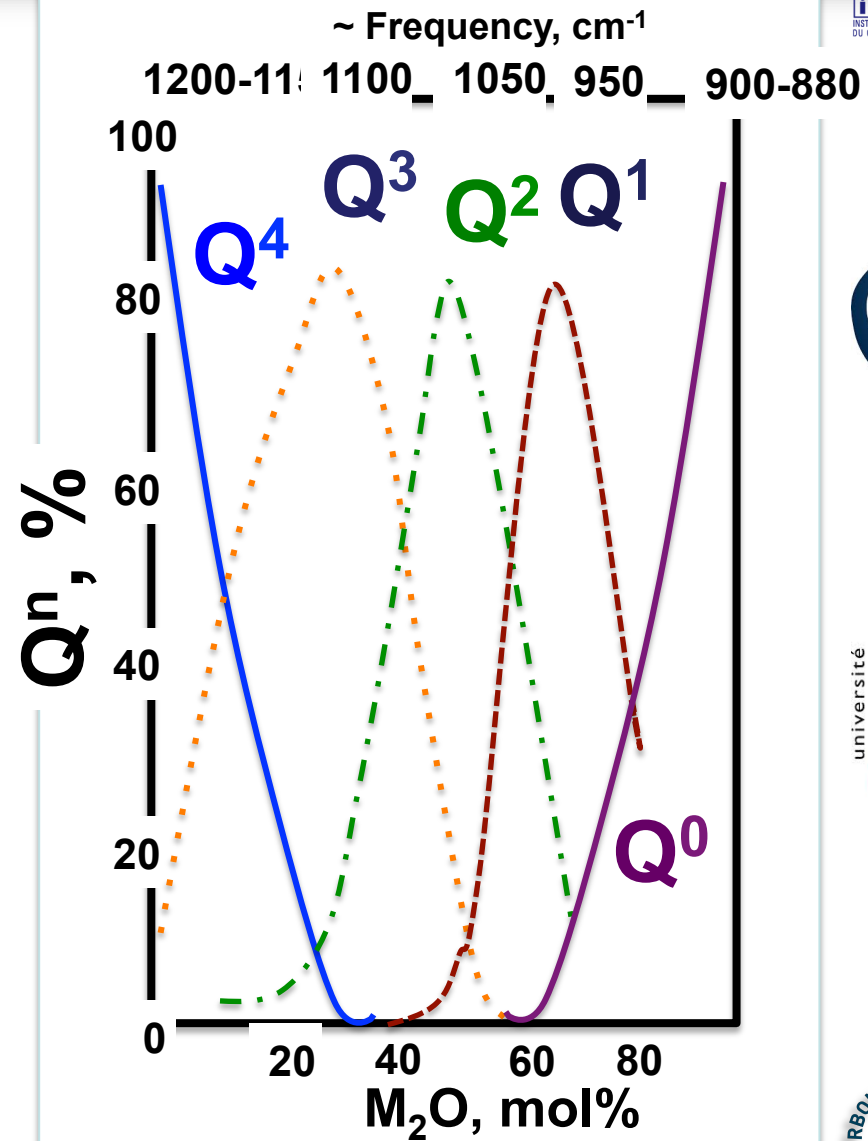
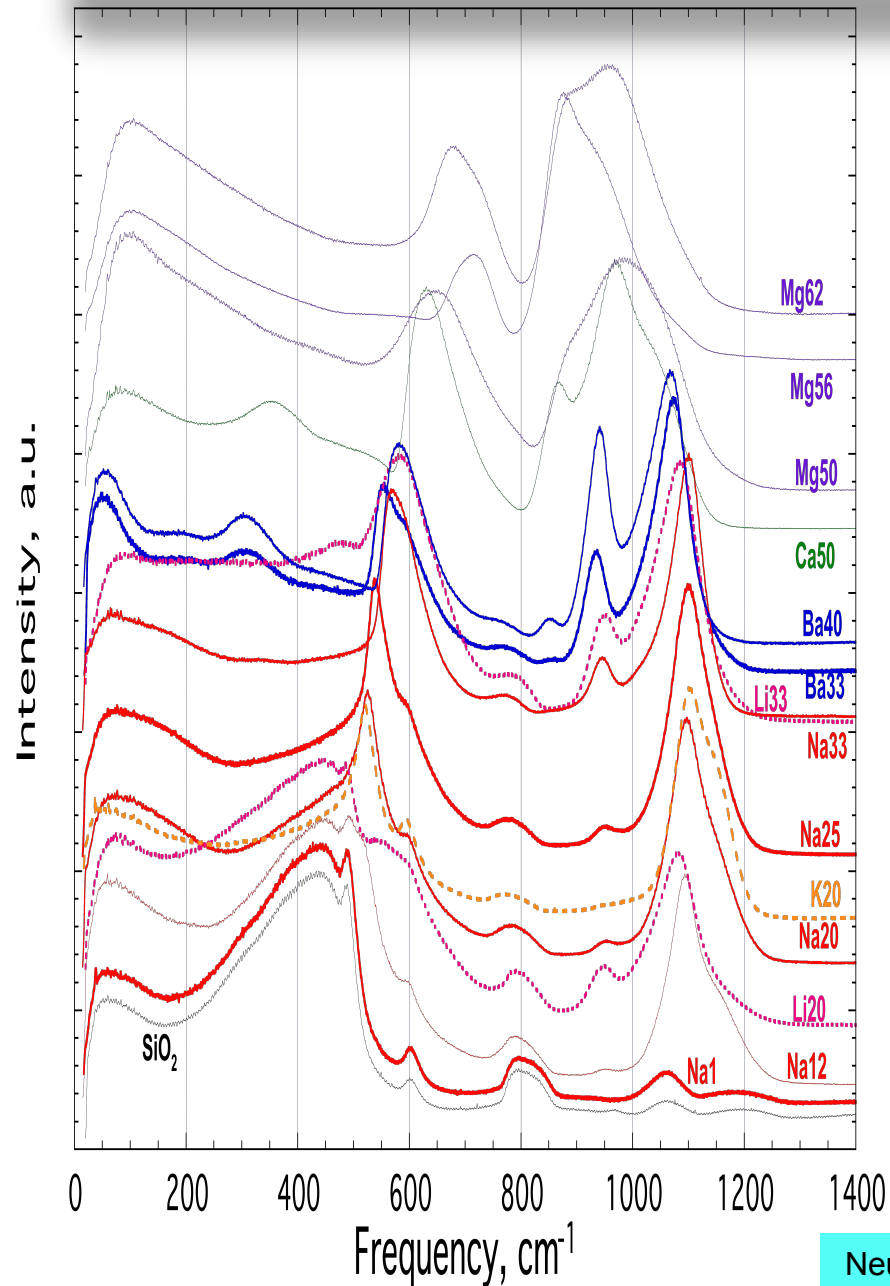
Si K-edge of $\text{BaO}-\text{SiO}_2$ glass
Bender et al., JNCS, 2002, 298, 99-108



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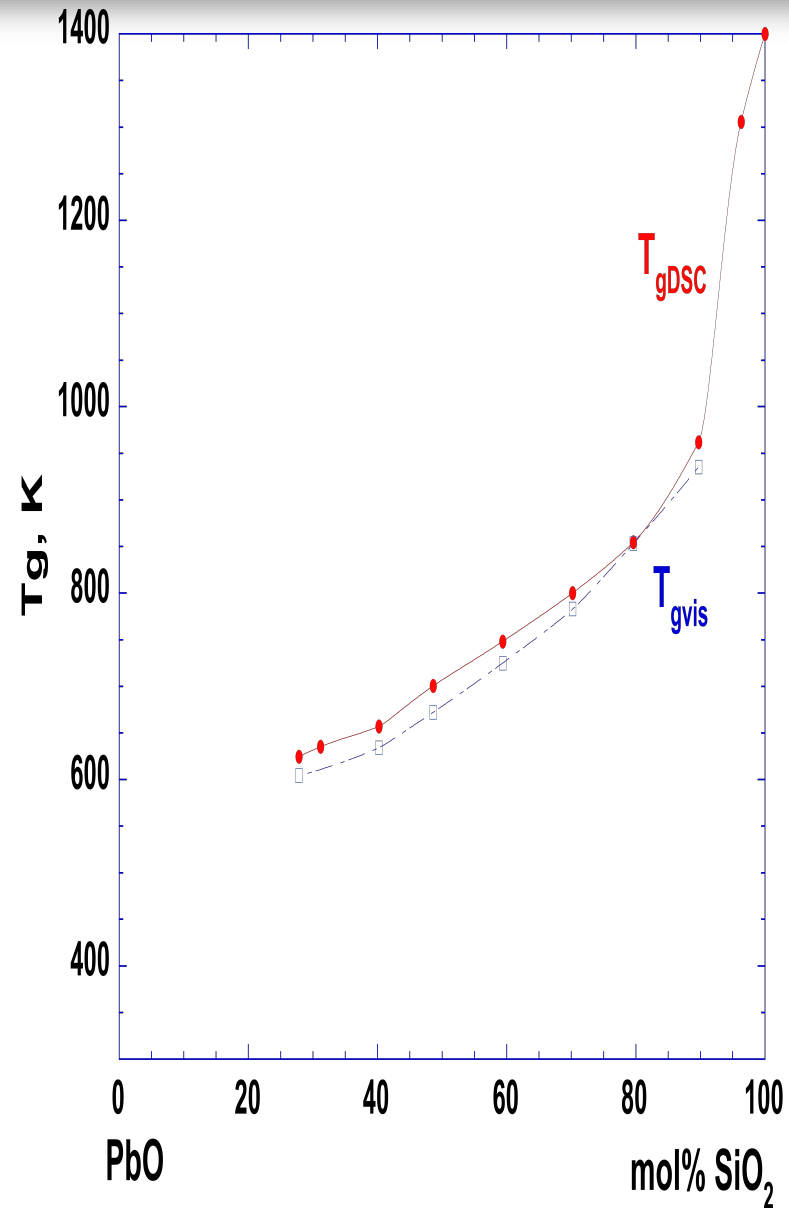
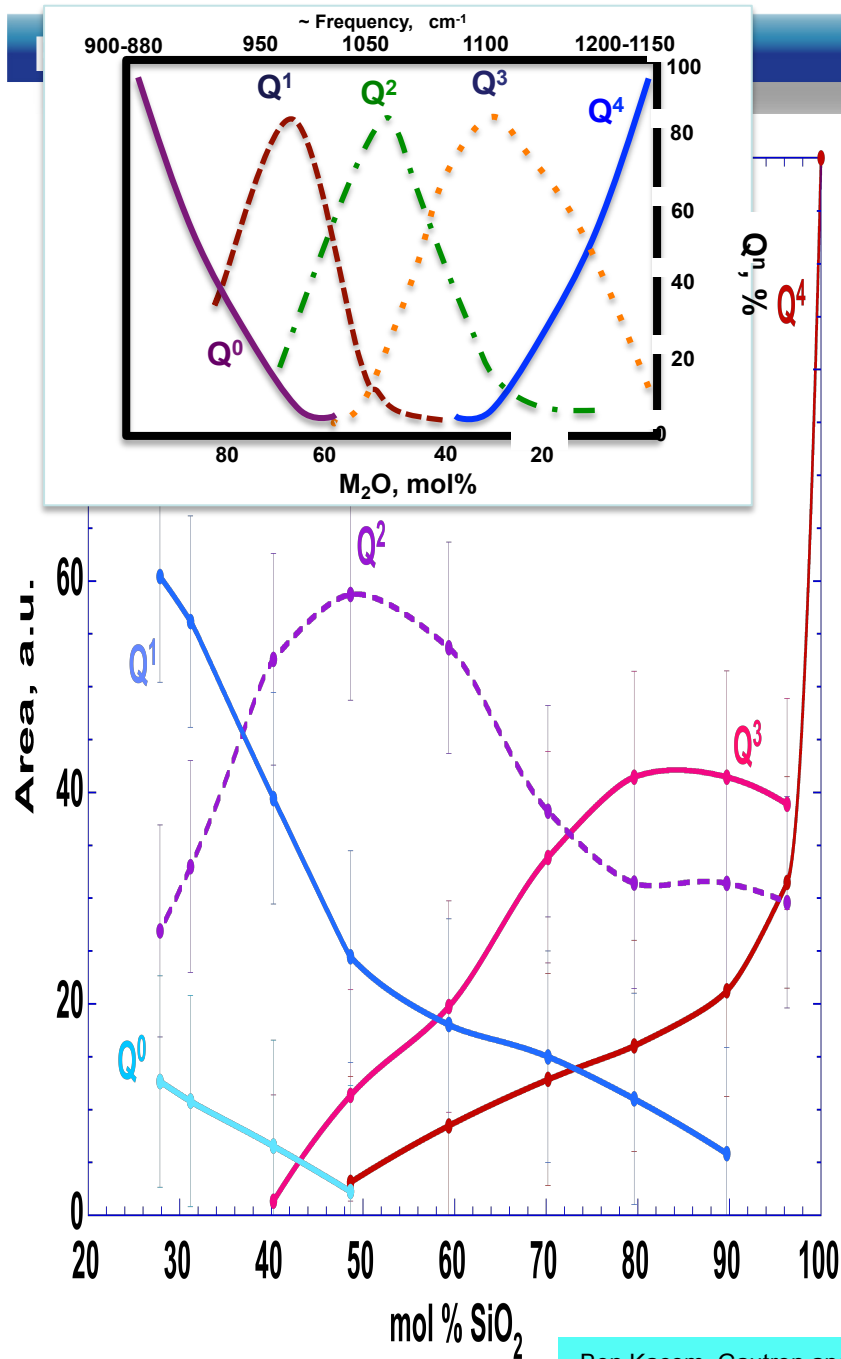


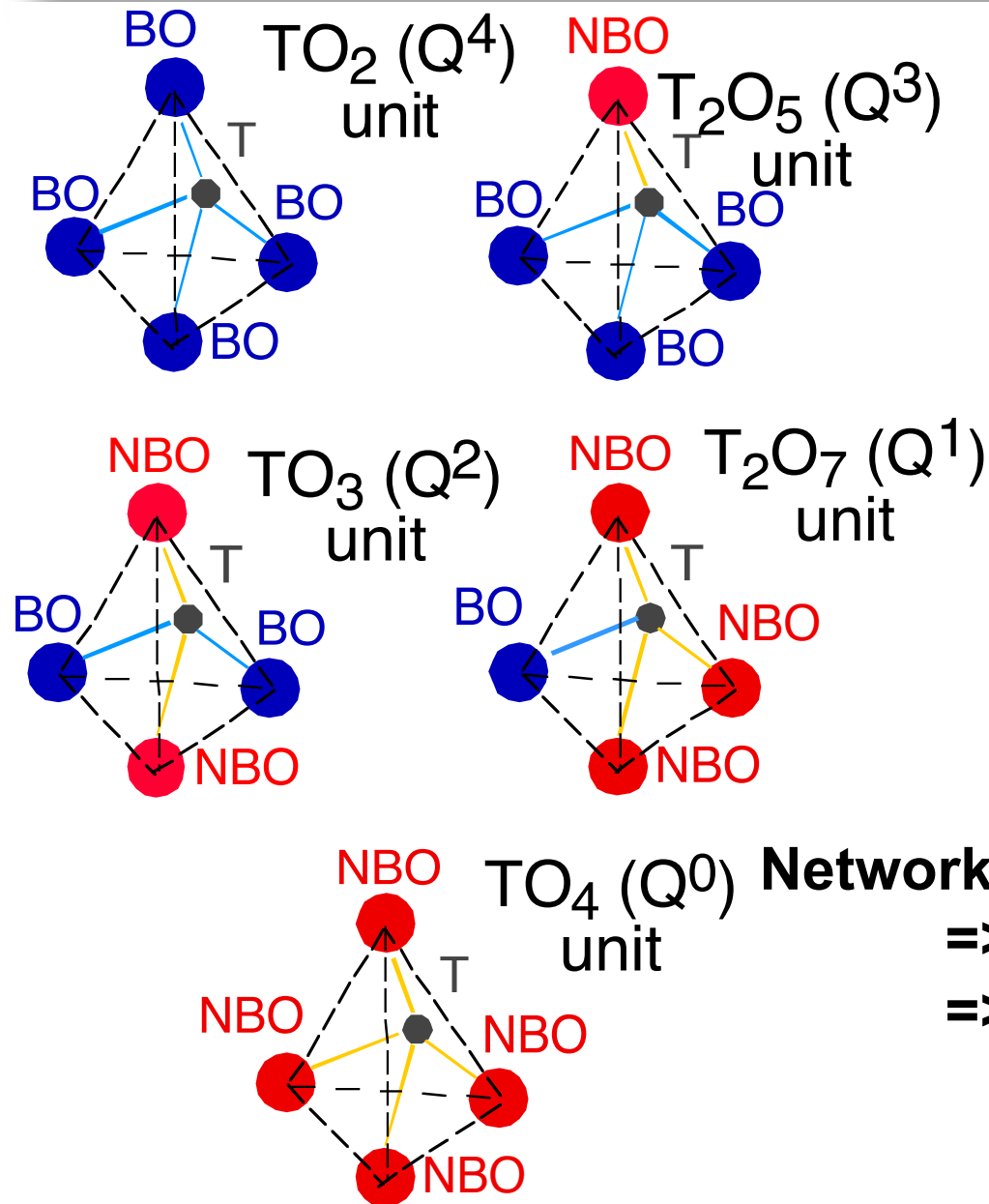
Qⁿ species



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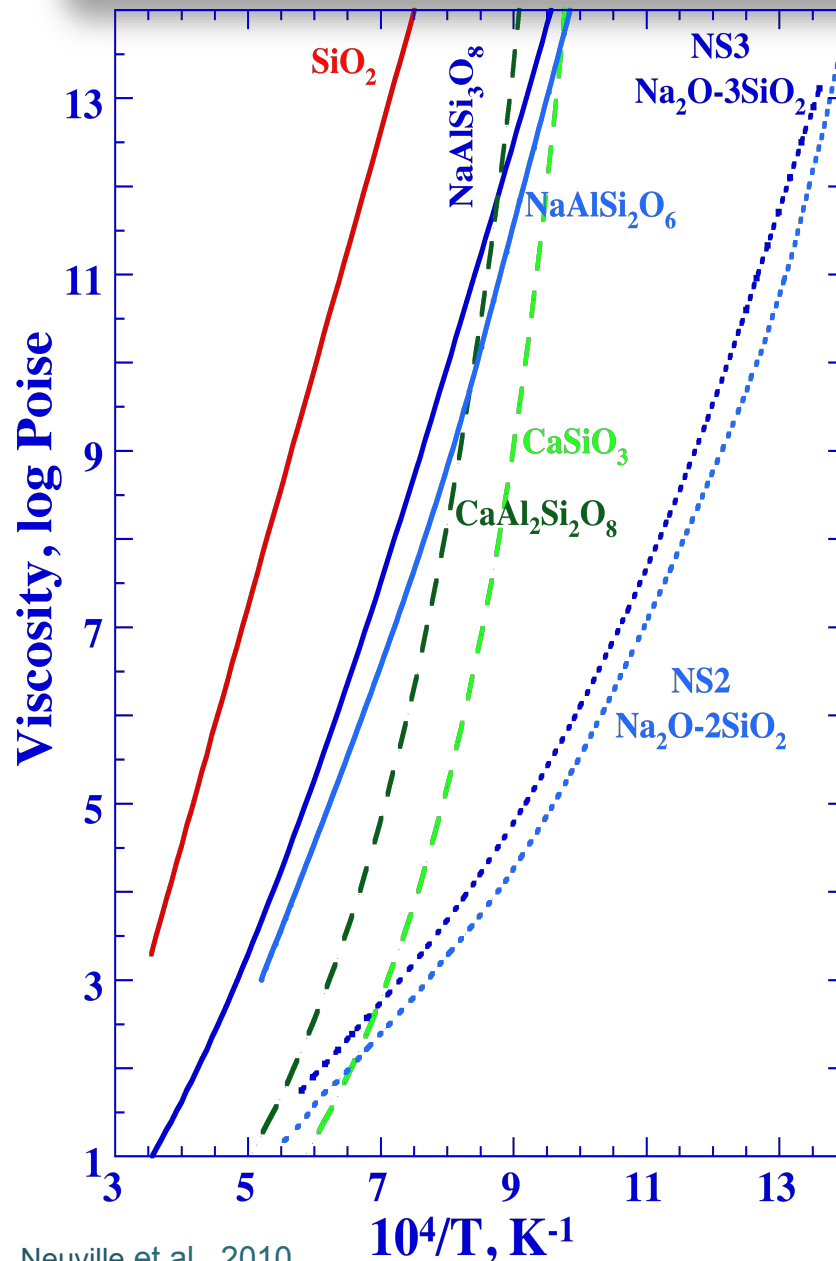


Network modifier with Si⁴⁺

=> Li⁺, Na⁺, K⁺

=> Mg²⁺, Ca²⁺, Sr²⁺, Ba²⁺

How fit viscosity



Arrhenius :

$$\eta(T) = A \cdot \exp(E/RT)$$

$$\Leftrightarrow \log \eta = A + B/T$$

Yes but only for SiO_2 , GeO_2 , NaAlSiO_8 , KAlSiO_8 because activation energy change from 2000 kJ/mol at 1000K up down 300 kJ/mol at 1800K for NS3.

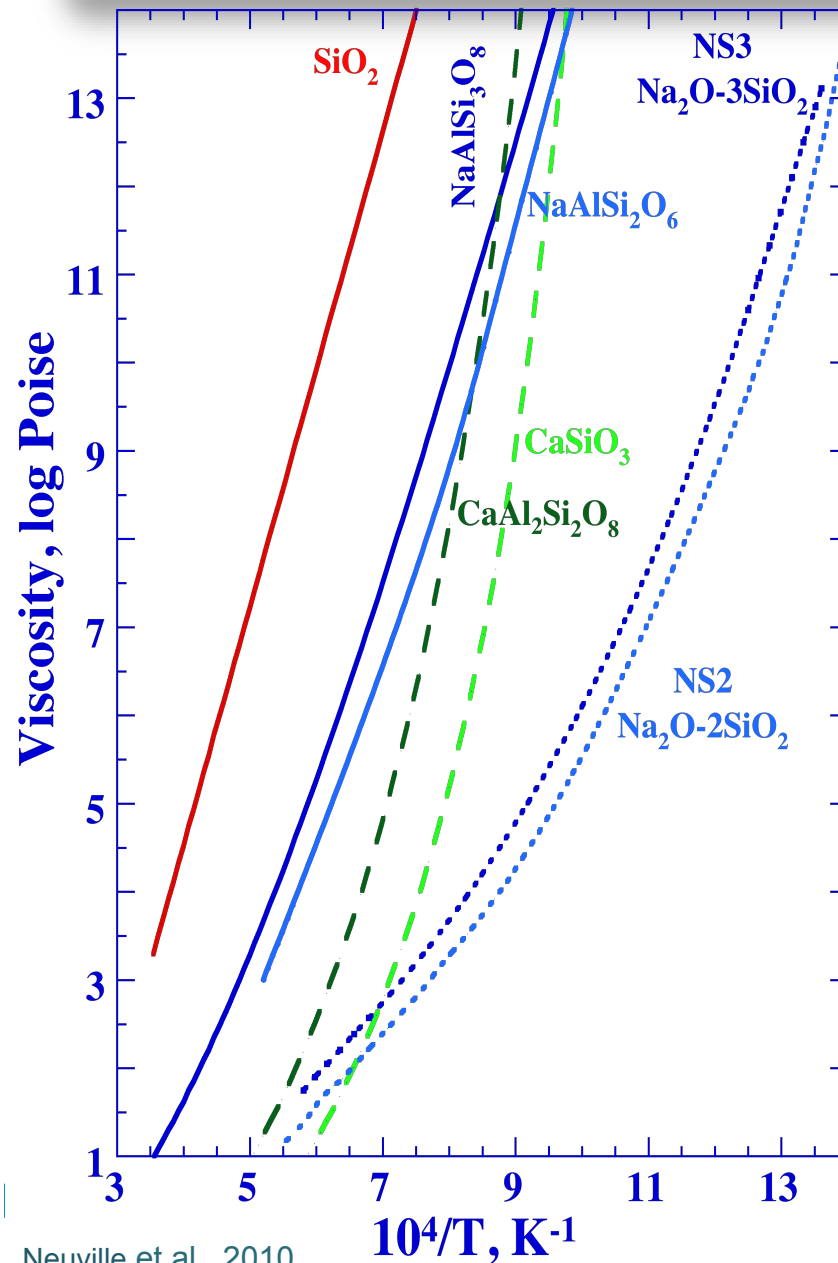
Need TVF equation

$$\log \eta = A_1 + B_1/(T-T_1)$$

But, just a fit



How fit viscosity



$$\eta(T) = A_e \cdot \exp[B_e/TS^{\text{conf}}(T)]$$

Proposed by Adam and Gibbs, 1964

First used to silicate melts by Urbain, 1972, Scherer, 1984, Richet, 1984, Neuville and Richet, 1991....

$$S^{\text{conf}}(T) = S^{\text{conf}}(T_g) + \int_{T_g}^T \frac{C_p^{\text{conf}}}{T} dt$$

$$C_p^{\text{conf}}(T) = C_{pg}(T_g) - C_{pl}(T)$$

Calorimetry measurements
=> Easy

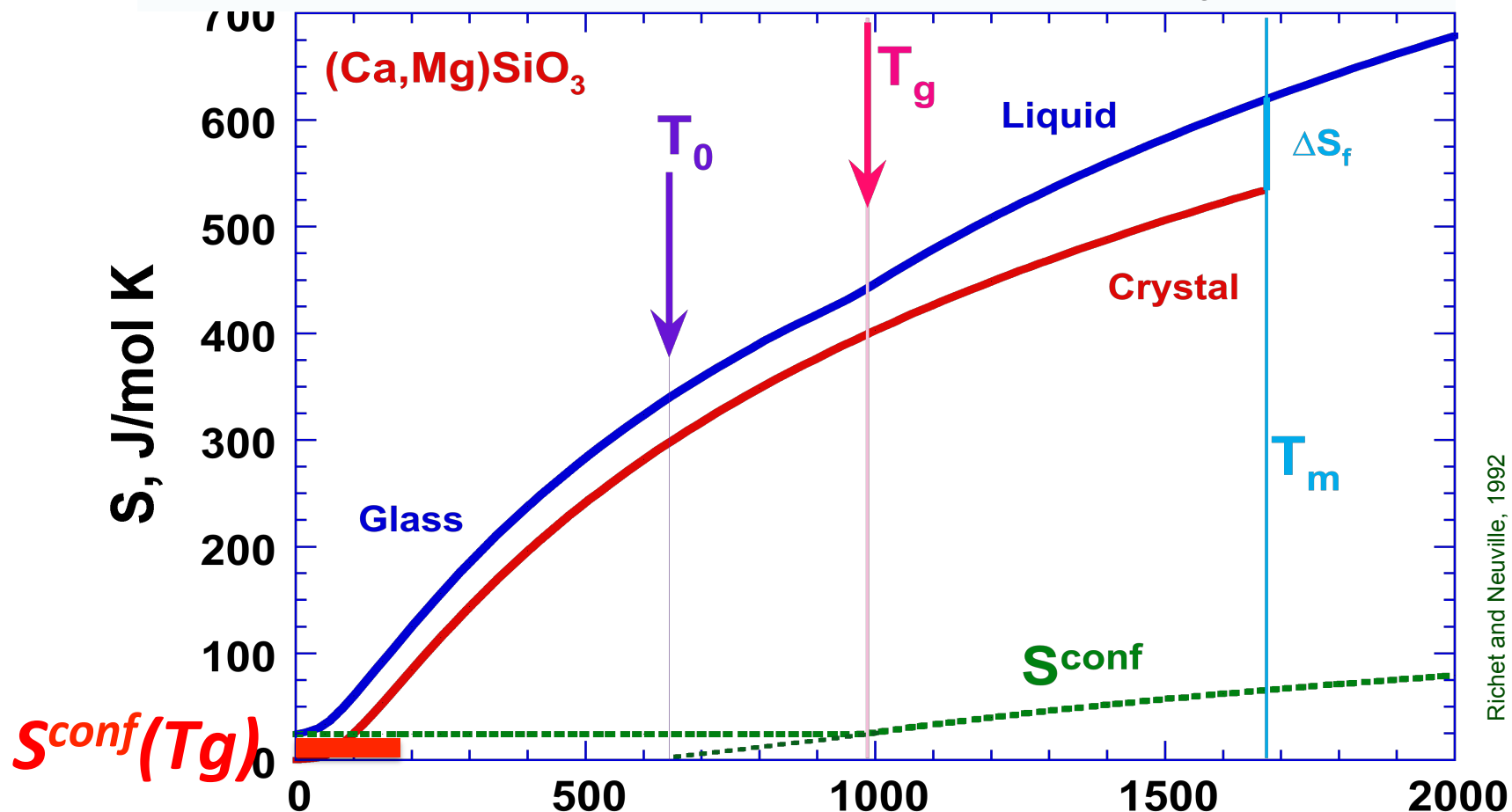
$S^{\text{conf}}(T_g)$
=> What is that ?



Configurational entropy



$$S^{conf}(T_g) = \int_0^{T_m} \frac{C_p^{Crystal}}{T} \cdot dT + \Delta S_f + \int_{T_m}^{T_g} \frac{C_p^{liquid}}{T} \cdot dT + \int_{T_g}^0 \frac{C_p^{glass}}{T} \cdot dT$$



Richet and Neuville, 1992



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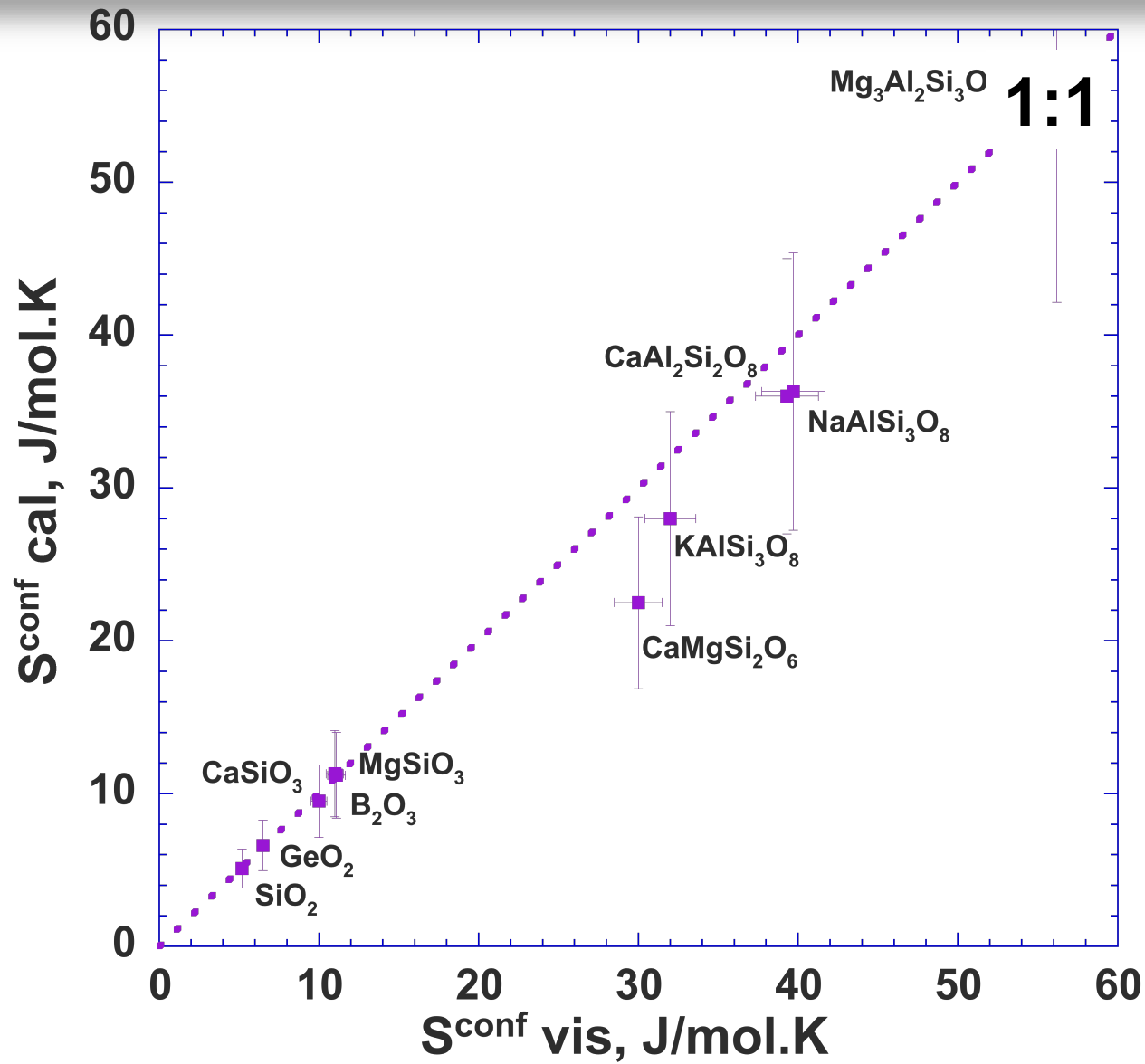
Heat capacity measurements

Adiabatic

DSC

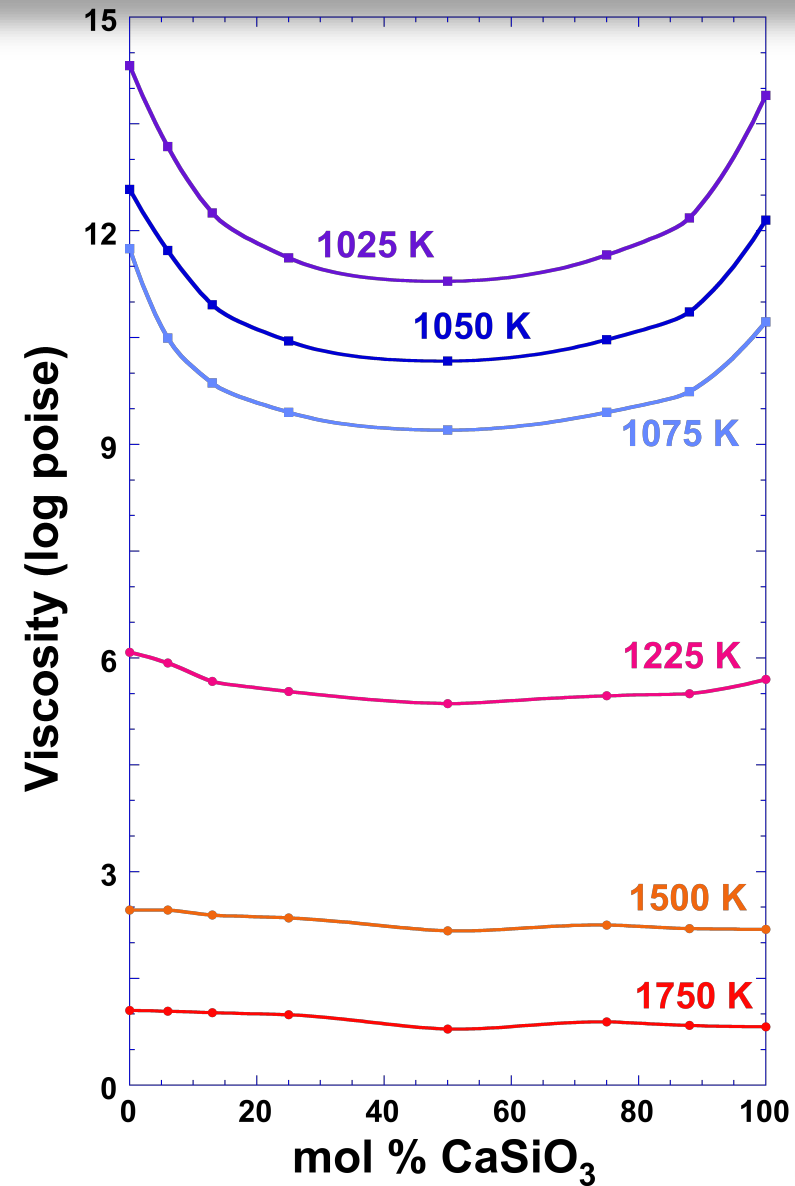
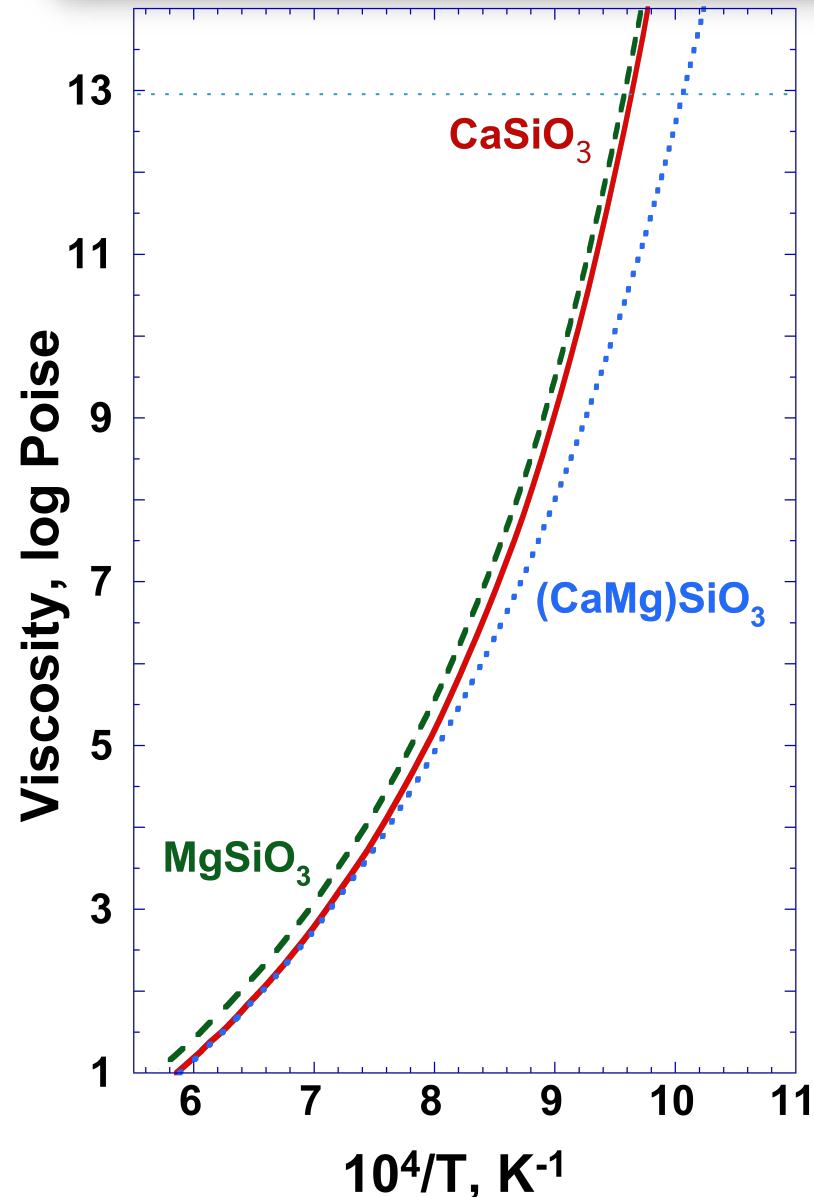
Drop or HT calorimeter

Calorimetric Configurational entropy



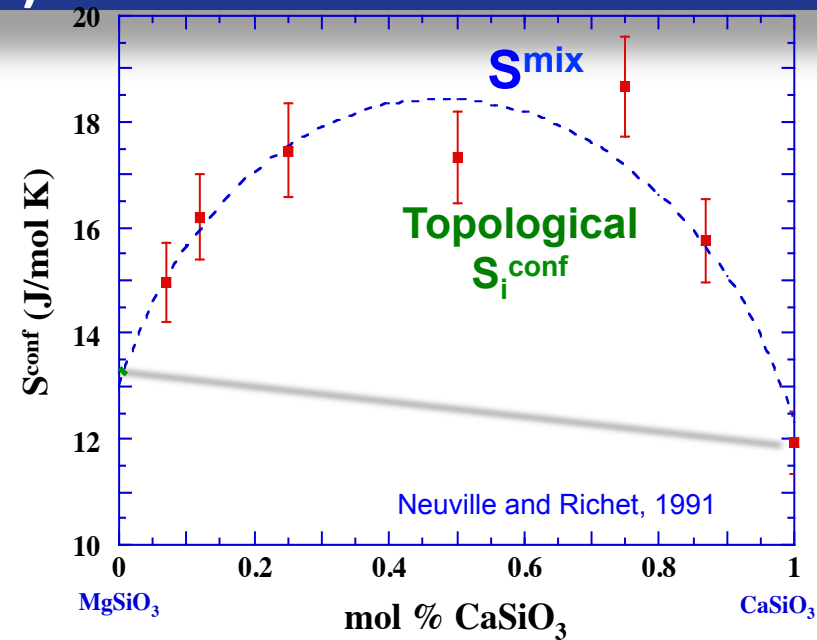
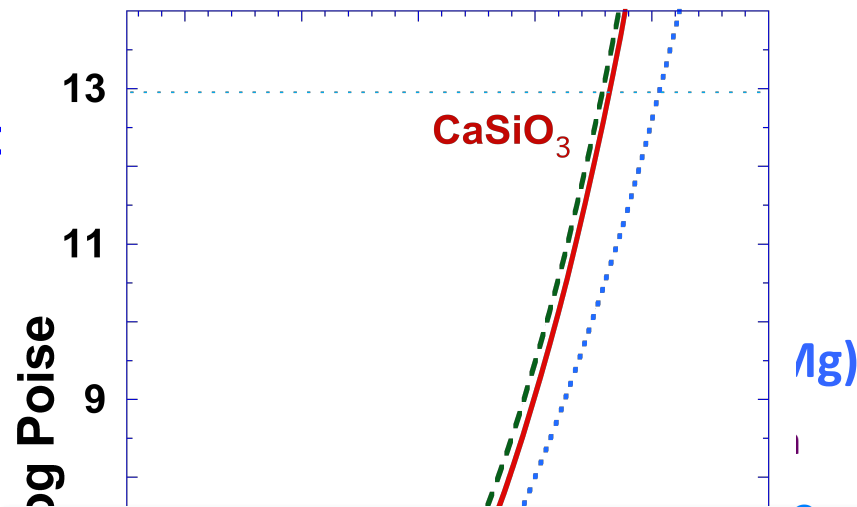
Viscosimetry configurational entropy

Ca/Mg Mixing ?



Entropy theory (Adam et Gibb, 1965)

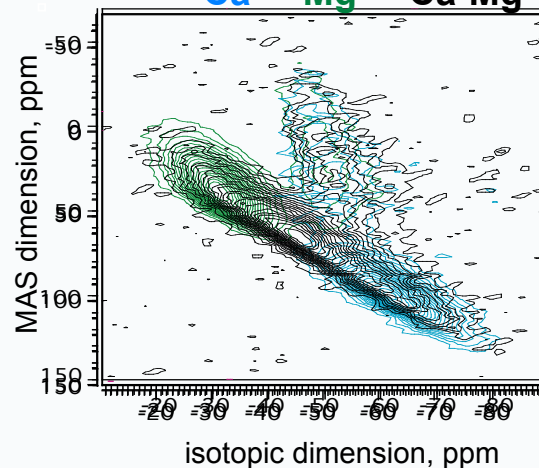
$$\log \eta = A_e + B_e/TS^{\text{conf}}(T)$$



O-NMR

Allwardt and Stebbins 2004

- “viewpoint” of the NBO
- ^{17}O chemical shifts depend strongly on which cations are nearby



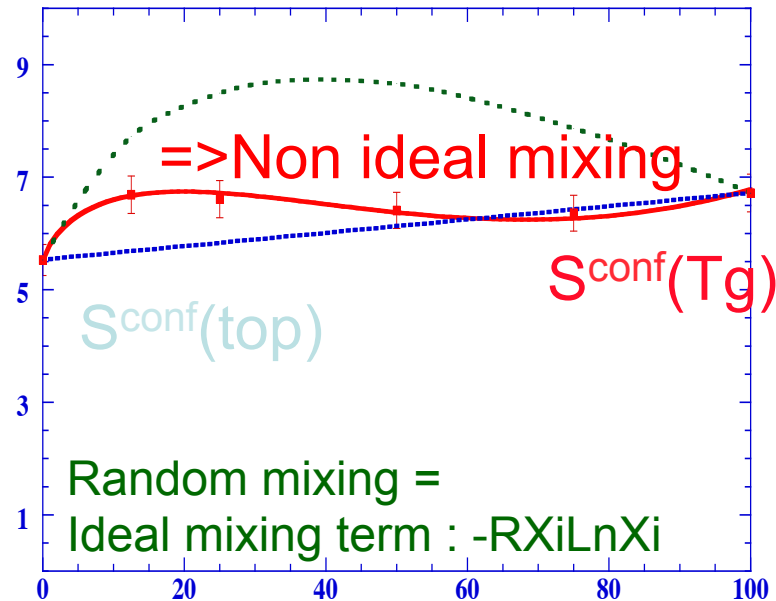
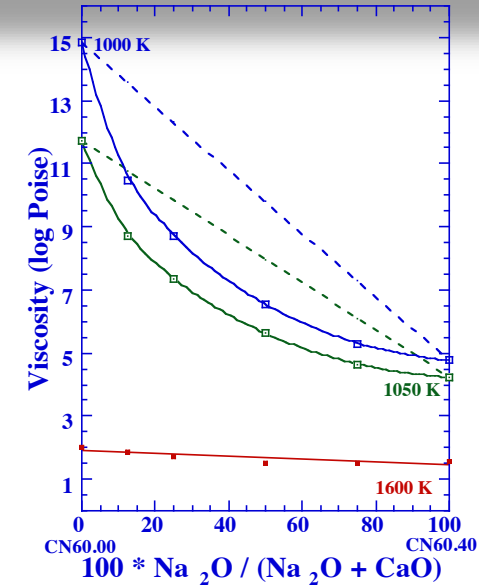
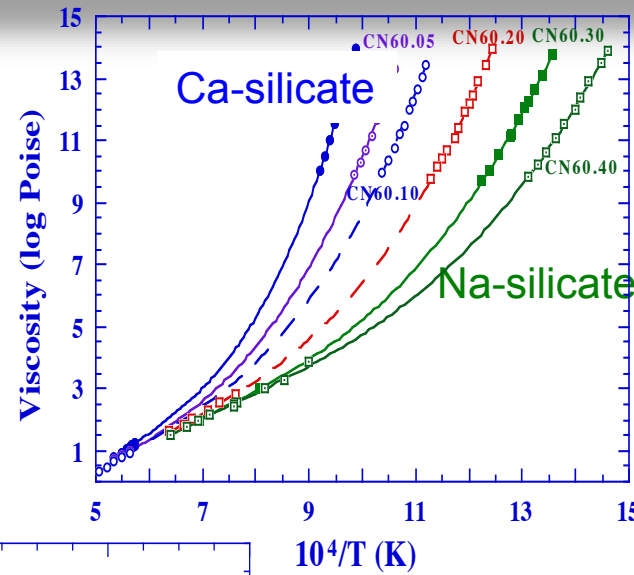
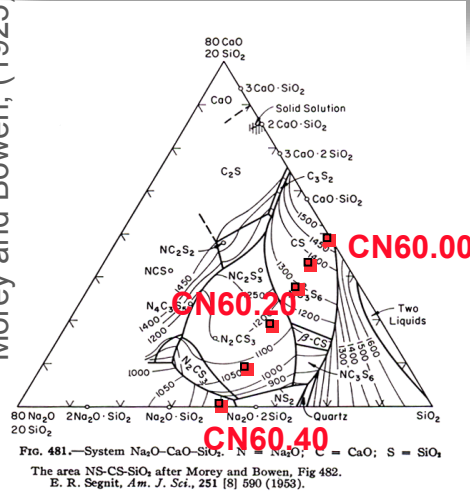
- detailed analyses of spectra support almost random distribution of Ca + Mg around NBO
- size difference of Ca^{2+} and Mg^{2+} is insufficient to cause ordering

$10^4/T, K^{-1}$

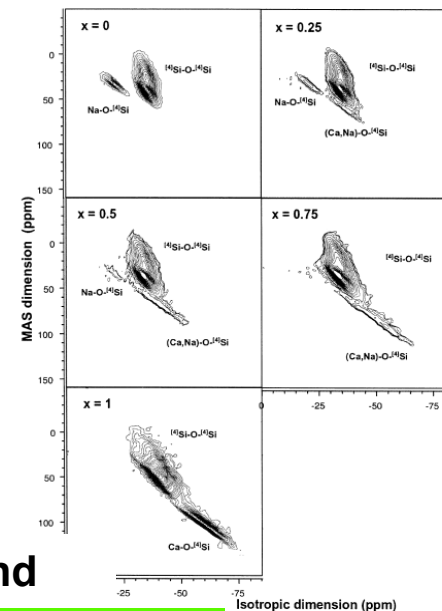
copy: a “picture” of the network structure

CaO-Na₂O-SiO₂ system

Morey and Bowen, (1925)



Ideal mixing does not work between Na/Ca and



Raman spectroscopy (Neuville, 2006) and ¹⁷O NMR (Lee and Stebbins, 2003) show a non random distribution of Na and Ca.

Neuville D.R., (2005) Structure, properties in (Sr, Na) silicate glasses and melts. *Phys Chem Glasses*, 46, 112-119
Neuville D.R. (2006) Viscosity, structure and mixing in (Ca, Na) silicate melts. *Chem. Geol.*, 229, 28-42

The Adam and Gibbs relaxation theory

A_e pre-exponential term
Viscosity at “infinite” temperature

B_e proportional to
the energy
barriers $\Delta\mu$ to
viscous flow

$$\log(\eta) = A_e + \frac{B_e}{T S^{conf}(T)}$$

Viscous flow occurs
through cooperative
rearrangements of
subunits

$$S^{conf}(T_g) + \int_{T_g}^T \frac{C_p^{conf}}{T} dt$$



The Adam and Gibbs relaxation theory

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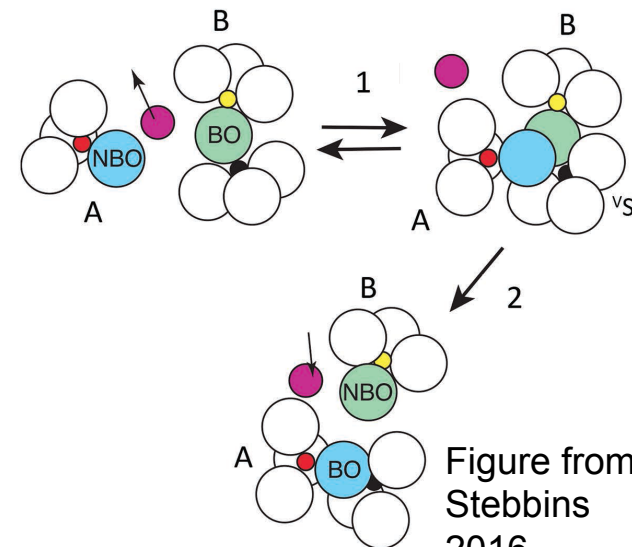
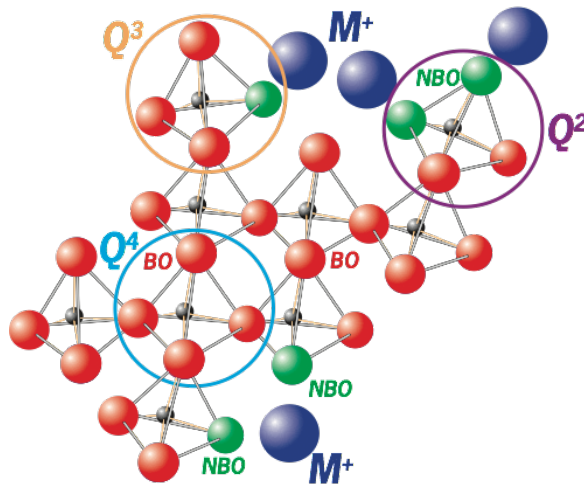


Figure from
Stebbins
2016

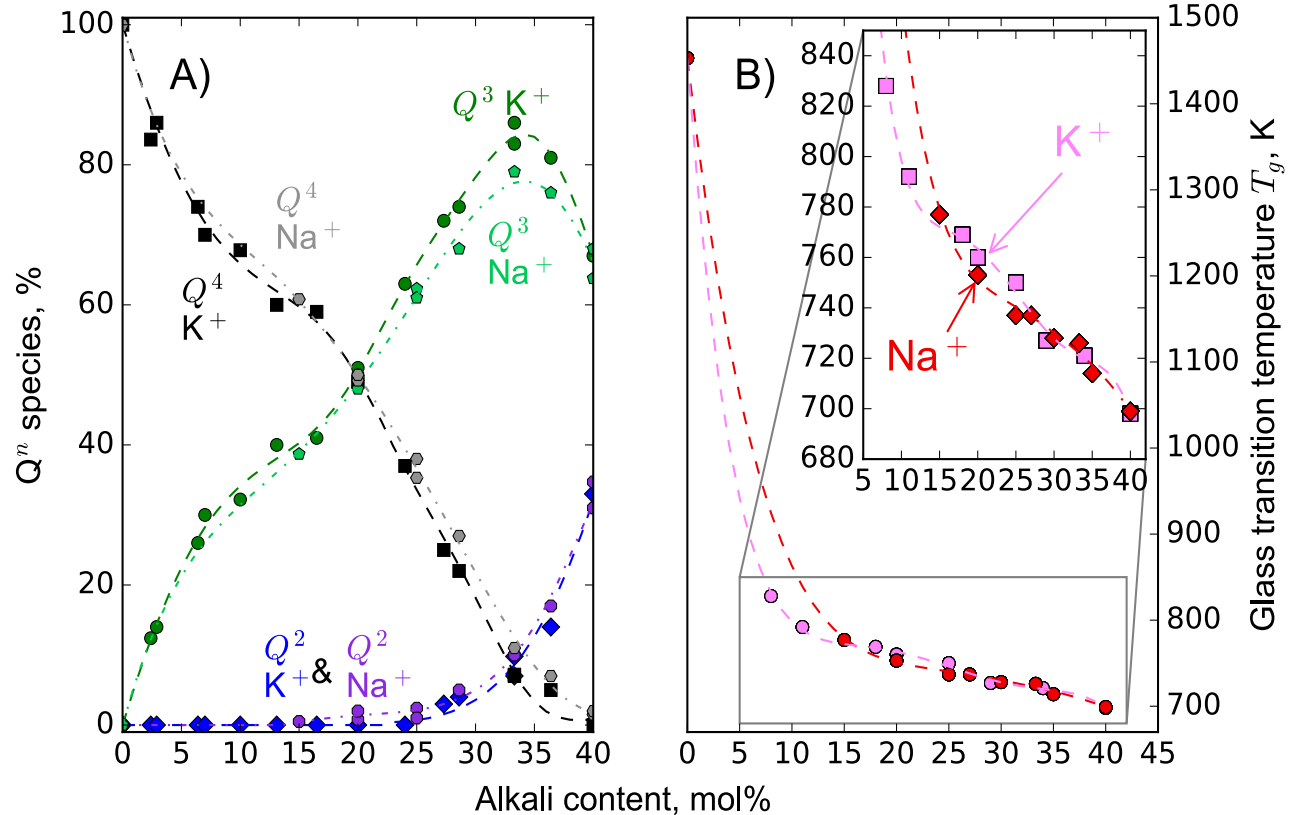
In silicate melts, viscous flow is related to exchange of oxygen by and interactions between tetrahedral SiO_4 and AlO_4 units

Na₂O - K₂O - SiO₂ melts → extensive dataset

- glass and liquid heat capacity (e.g., Richet and Bottinga 1985, Richet 1987)
- glass molecular structure (e.g. from ²⁹Si NMR: Maekawa et al. 1991, Sen and Yougman 2003)
- liquid viscosity (e.g., Bockris 1954, Poole 1949, Sipp and Richet 2002, Neuville 2006)



BO:bridging oxygen
NBO: non-bridging oxygen
M⁺: alkali cation



Linking structure, thermodynamic and rheology:



$$\log(\eta) = A_e + \frac{B_e}{T \left(S^{conf}(T_g) - \int_{T_g}^T \frac{C_p^{conf}}{T} dT \right)}$$



B_e and $S^{conf}(T_g)$ = sum of partial B_e and $S^{conf}(T_g)$ molar values for each tetrahedral species

Na_2SiO_3 and K_2SiO_3 (Na- Q^2 and K- Q^2)

$Na_2Si_2O_5$ and $K_2Si_2O_5$ (Na- Q^3 and K- Q^3)

SiO_2 (Q^4)

Need to add to $S^{conf}(T_g)$:

- ideal mixing of Si between Q^2 , Q^3 and Q^4 units
- ideal mixing of Na and K (Richet, 1984) in network modifier channels

Need to add to B_e :

- the above ideal mixing terms, scaled by two coefficients



- 13 parameters, 326 data points, from 60 to 100 mol% SiO₂
- Minimisation of the least-square criterion using the Ipopt (Wächter and Biegler 2006) and Knitro (Byrd et al. 2006) solvers, and error calculation via non-parametric bootstrapping (20 000 re-sampling done)

$Partial S^{conf}(T_g),$
 $J mol^{-1} K^{-1}$

Na-Q²: 5.8 [4.8 - 6.9]

Na-Q³: 0.9 [0.3 - 1.3]

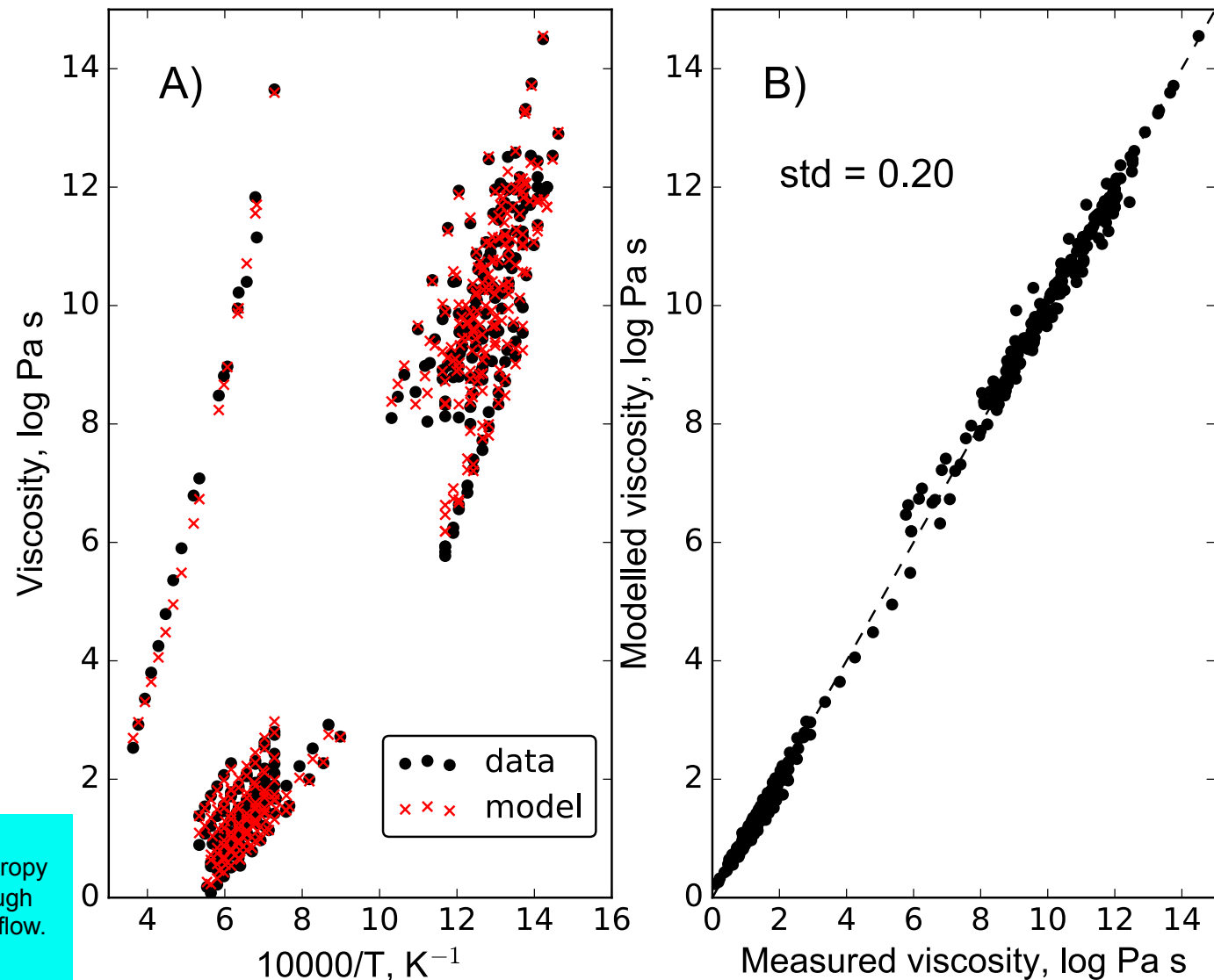
K-Q²: 12.0 [6.8 - 20.4]

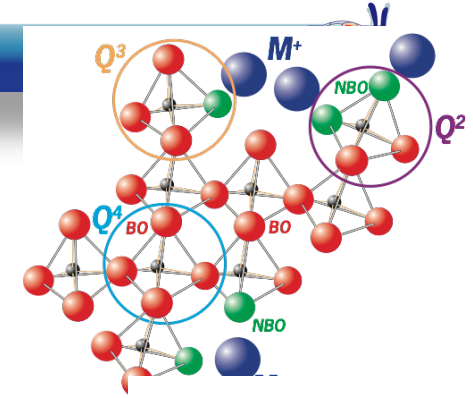
K-Q³: 4.4 [3.3 - 5.4]

Q⁴: 7.9 [7.1 - 8.4]

SiO₂ exp. is 5.1 ± 2

Le Losq Ch., Neuville D.R. (2017)
 Molecular structure, configurational entropy
 and viscosity of silicate melts: link through
 the Adam and Gibbs theory of viscous flow.
 Journal of Non-Crystalline Solids



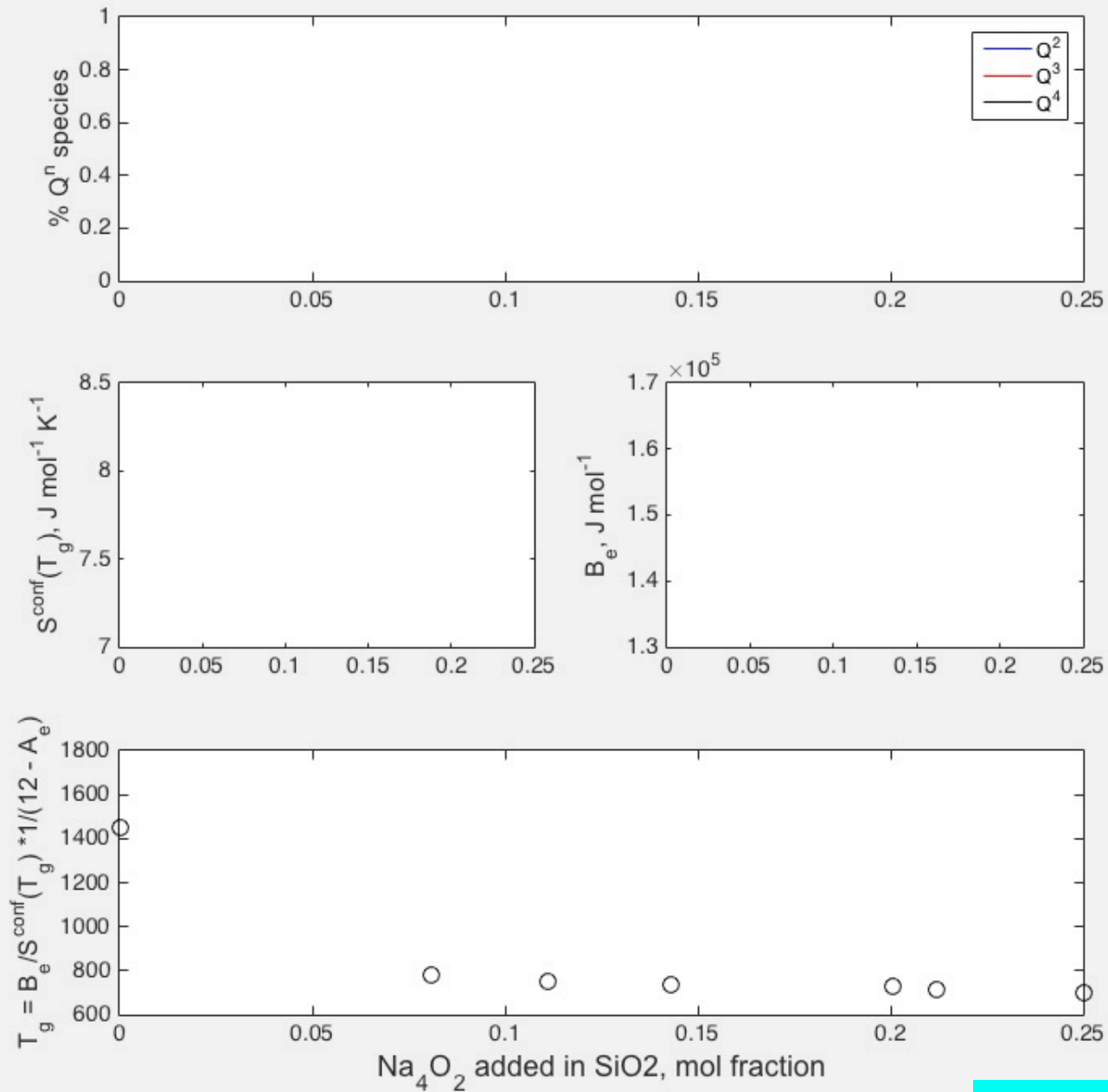


$$\log(\eta) = \frac{B_e}{A_e + \frac{B_e}{T \left(s_{conf}(T_g) + \int_{T_g}^T \frac{C_p^{conf}}{T} dT \right)}}$$

$$A_e + \frac{B_e}{T \left(s_{conf}(T_g) + \int_{T_g}^T \frac{C_p^{conf}}{T} dT \right)}$$

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Example of model prediction along the join Na₂O-SiO₂



M^+ , M^{++} : Network modifier



M^+ , M^{++} : Network modifier => produce non-bridging oxygen,

=> decrease viscosity, T_g , molar volume, ..

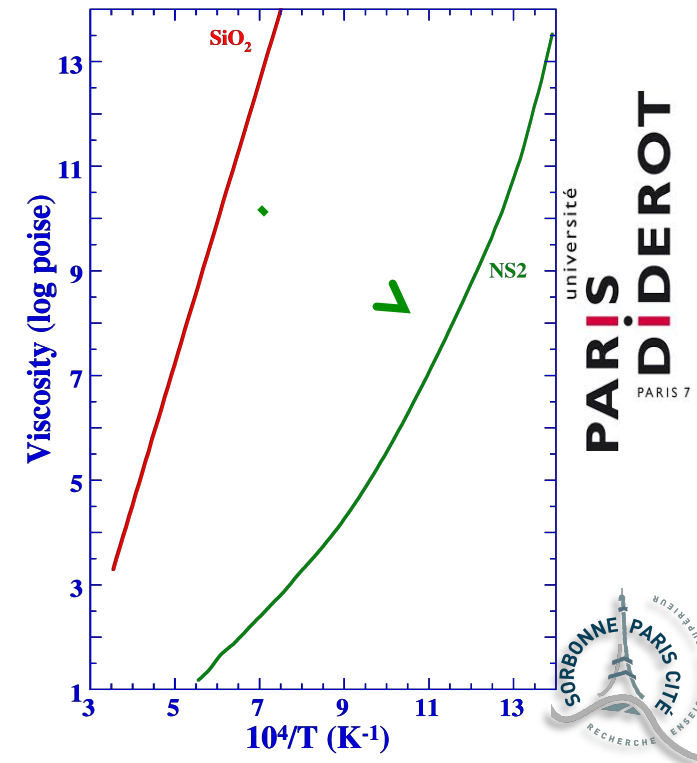
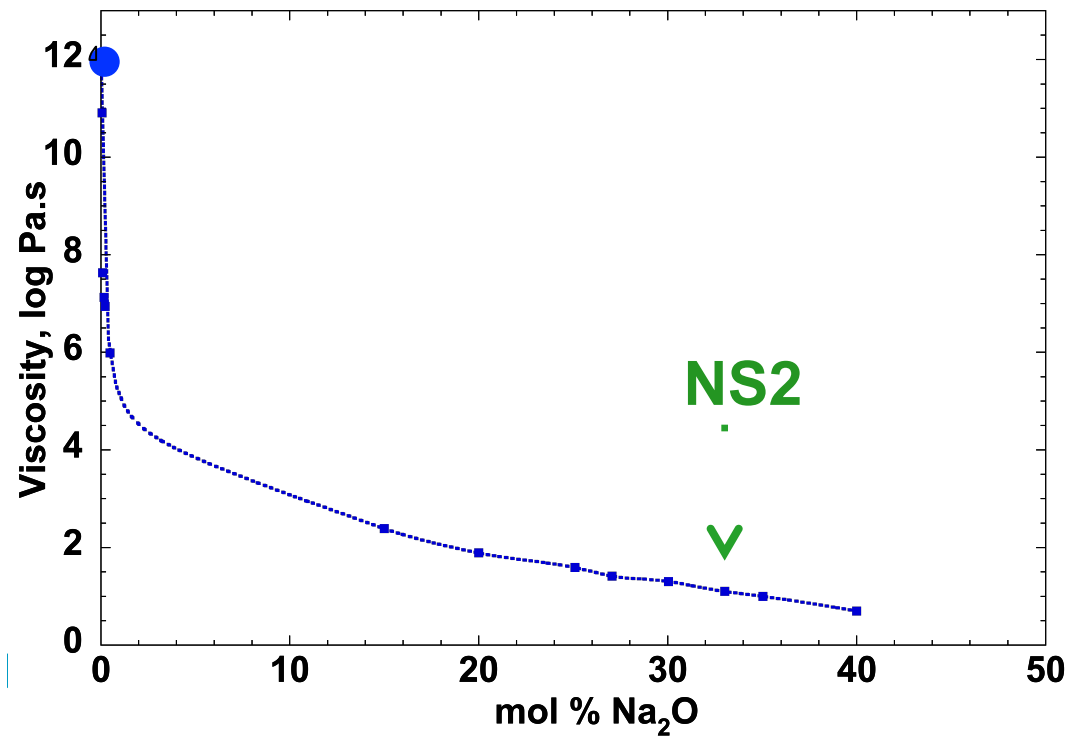
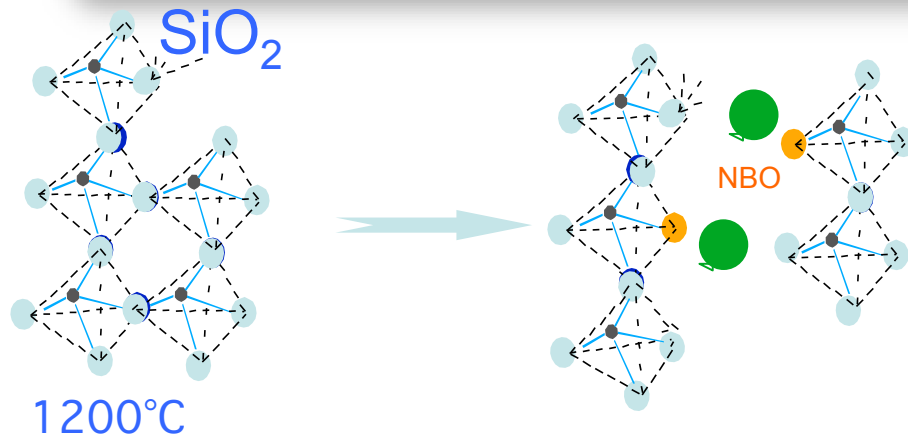
=> increase configurational entropy and disorder....



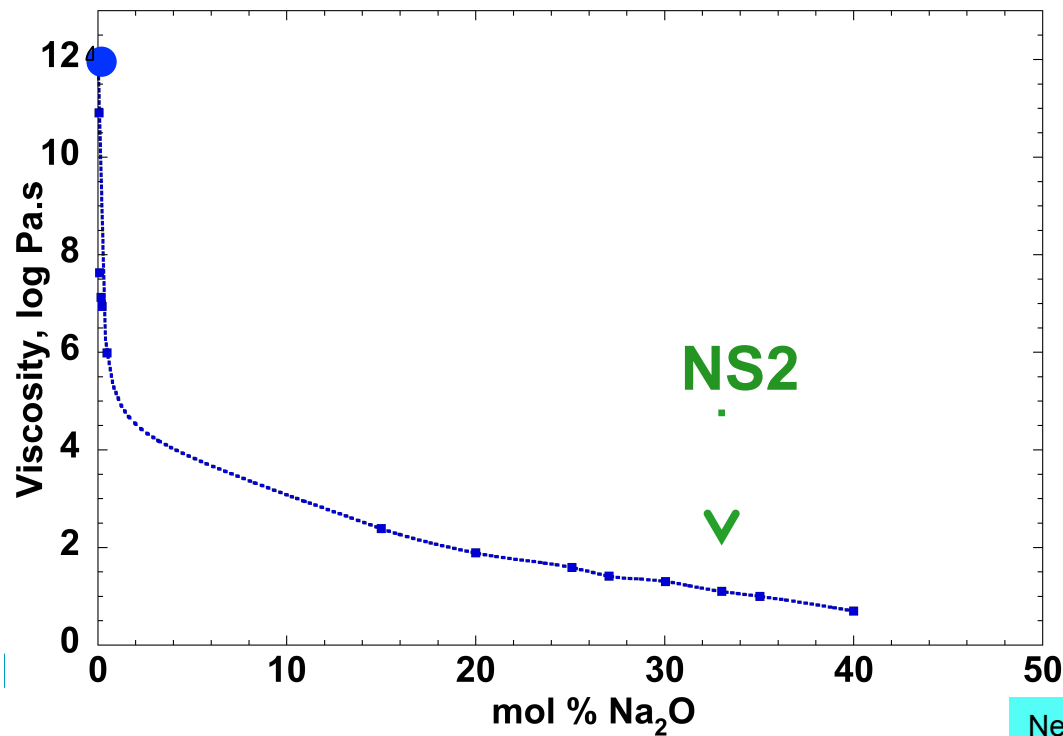
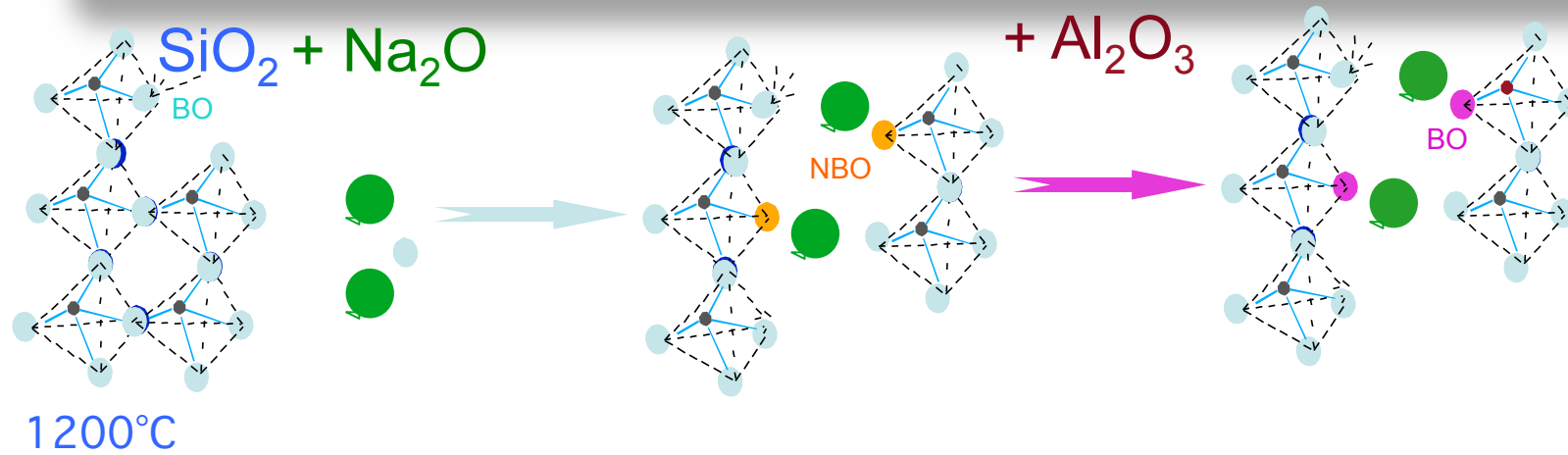
How can be change in charge compensator ?



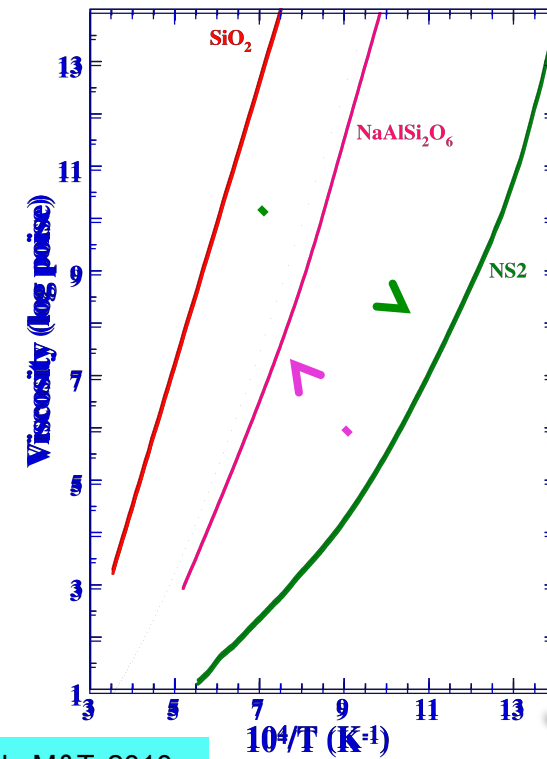
Structure versus properties of silicate melts



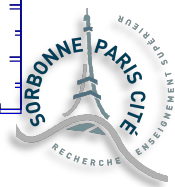
Structure versus properties of silicate melts



Neuville et al., M&T, 2010

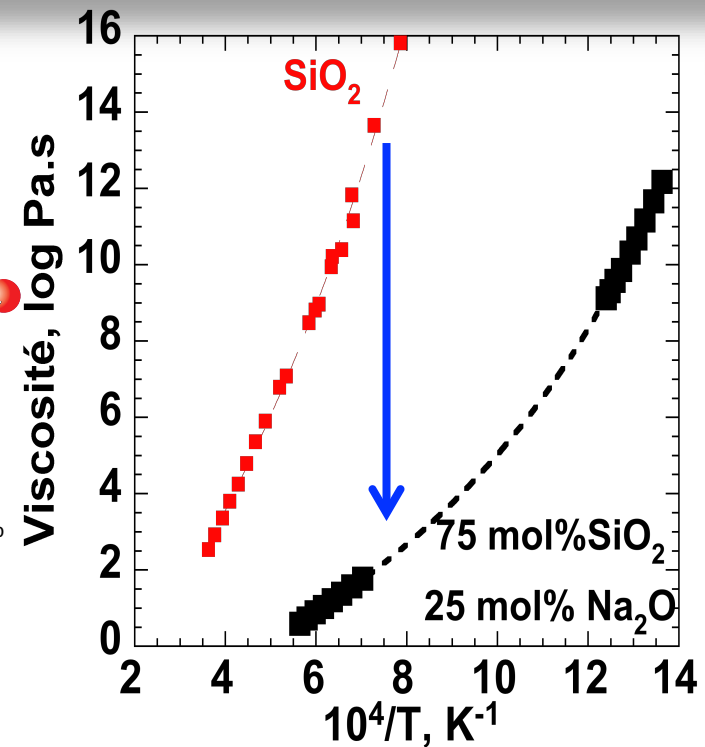
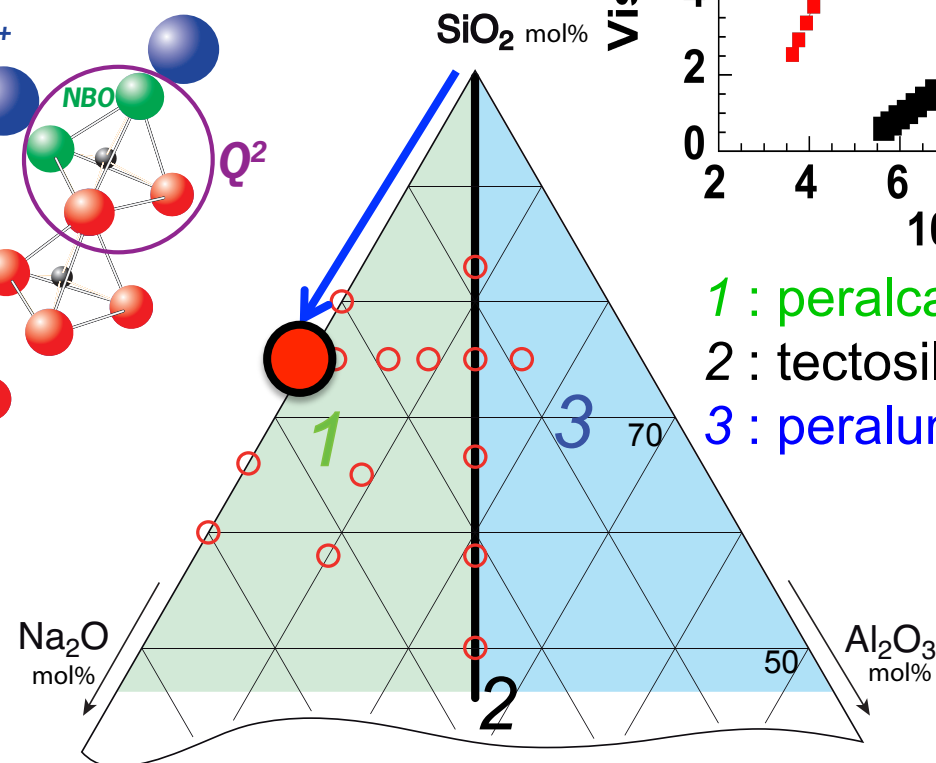
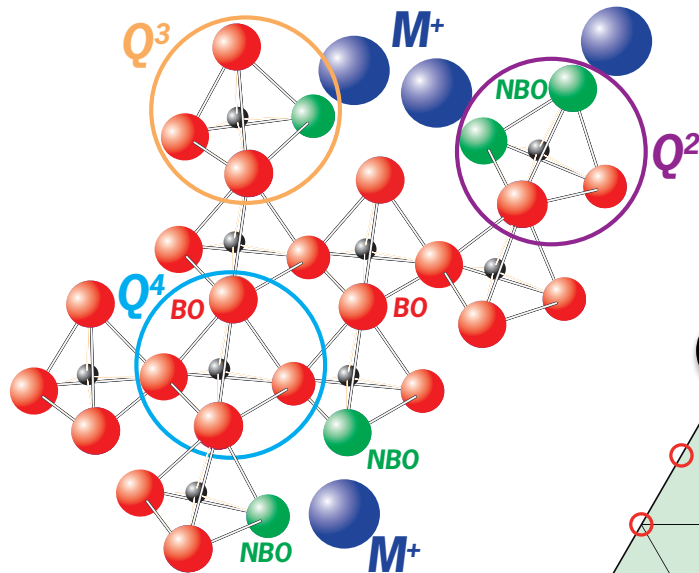
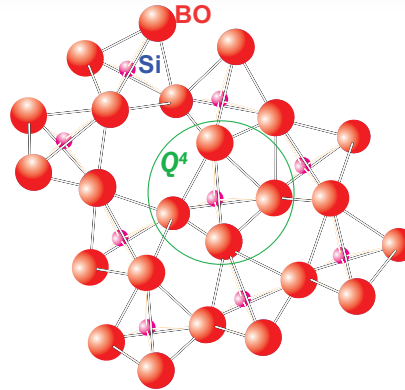


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Adding Na₂O to SiO₂

⇒ Depolymerisation,
formation of NBOs

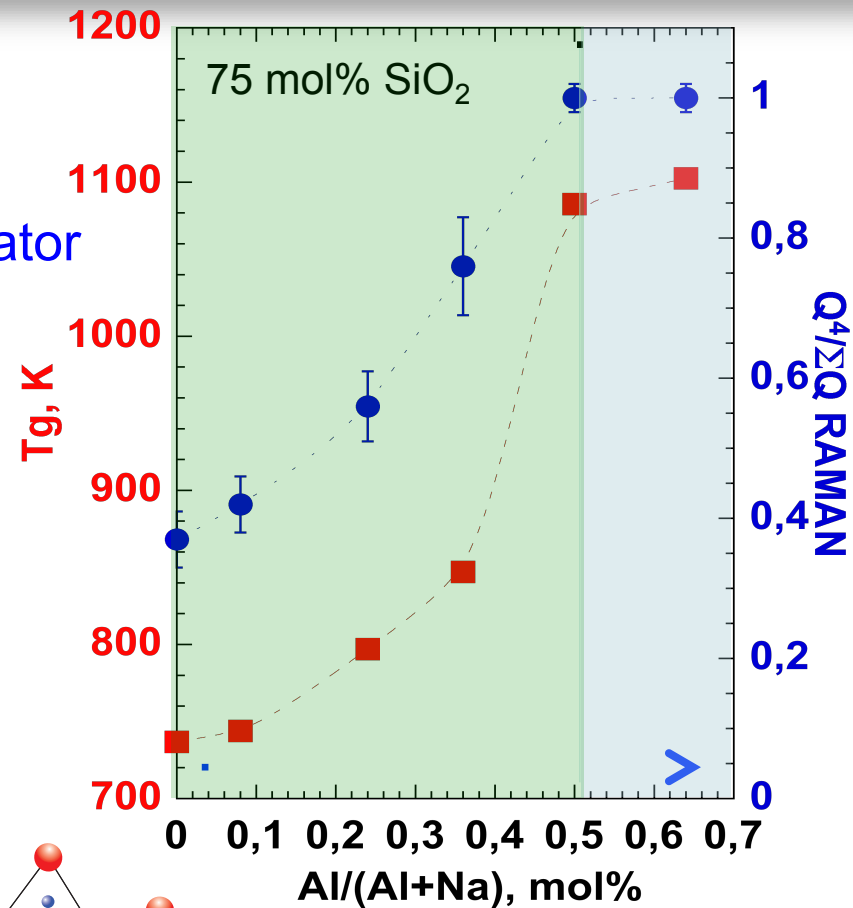
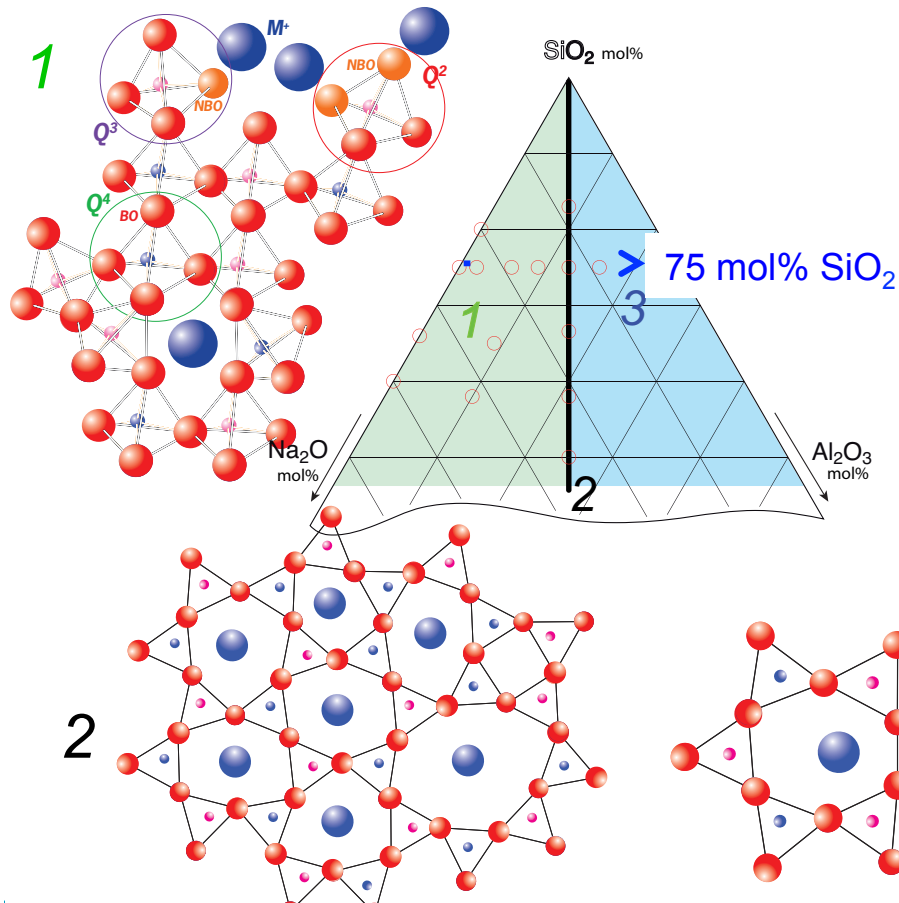


- 1 : peralcalin: Na > Al
- 2 : tectosilicate: Na = Al
- 3 : peraluminous: Na < Al

Aluminium effect

Na_2O substitution by Al_2O_3 :

- ⇒ Polymerization
- ⇒ Change Q^3 in Q^4
- ⇒ Al in Q^4 and Na charge compensator



Al in CN 5 in
peraluminous
domain (3)

Le Losq Ch., Neuville D.R., Florian P., G.S. Henderson and Massiot D. (2014) Role of Al^{3+} on rheology and nano-structural changes of sodium silicate and aluminosilicate glasses and melts. *Geochimica Cosmochimica Acta*, 126, 495-517

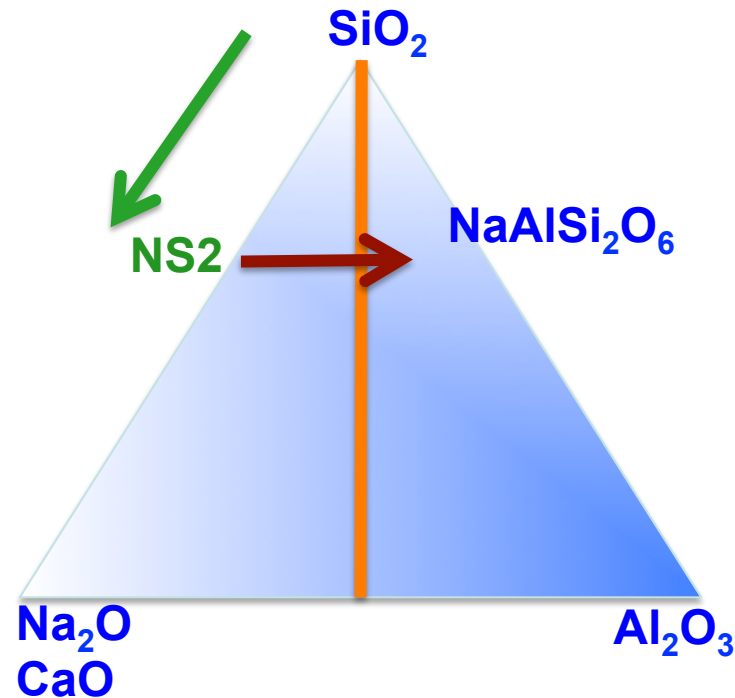
In summary:

Si^{4+} , Al^{3+} network former

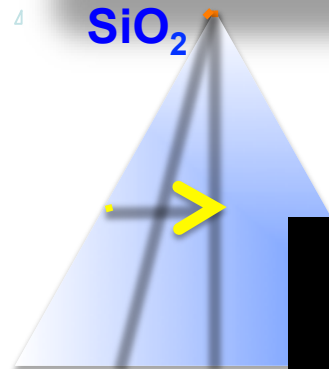
O bridging or non-bridging oxygen

Na^+ , Ca^{++} , network modifier

or charge compensator



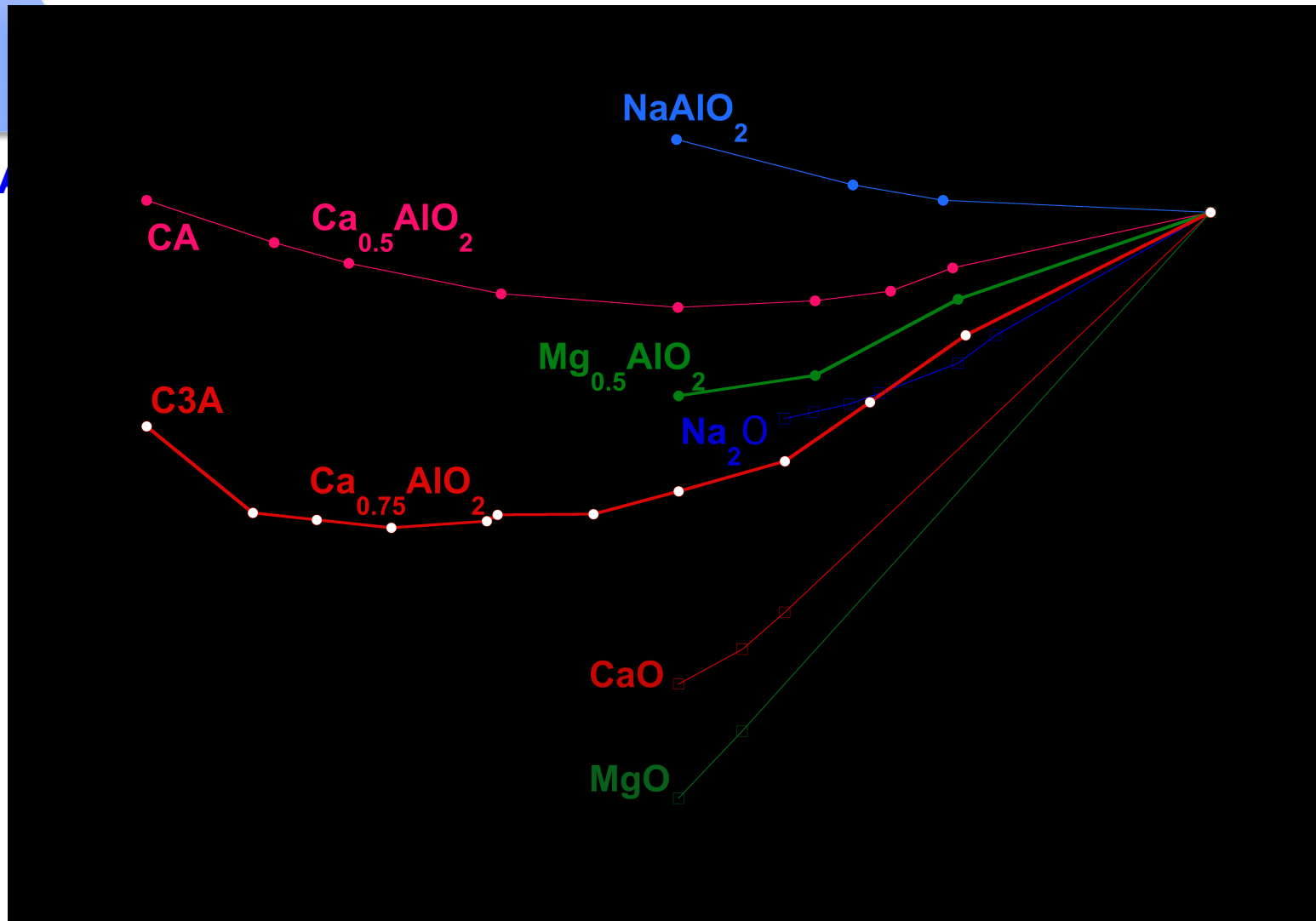
MO-Al₂O₃-SiO₂ Molar volume

Na₂O

CaO

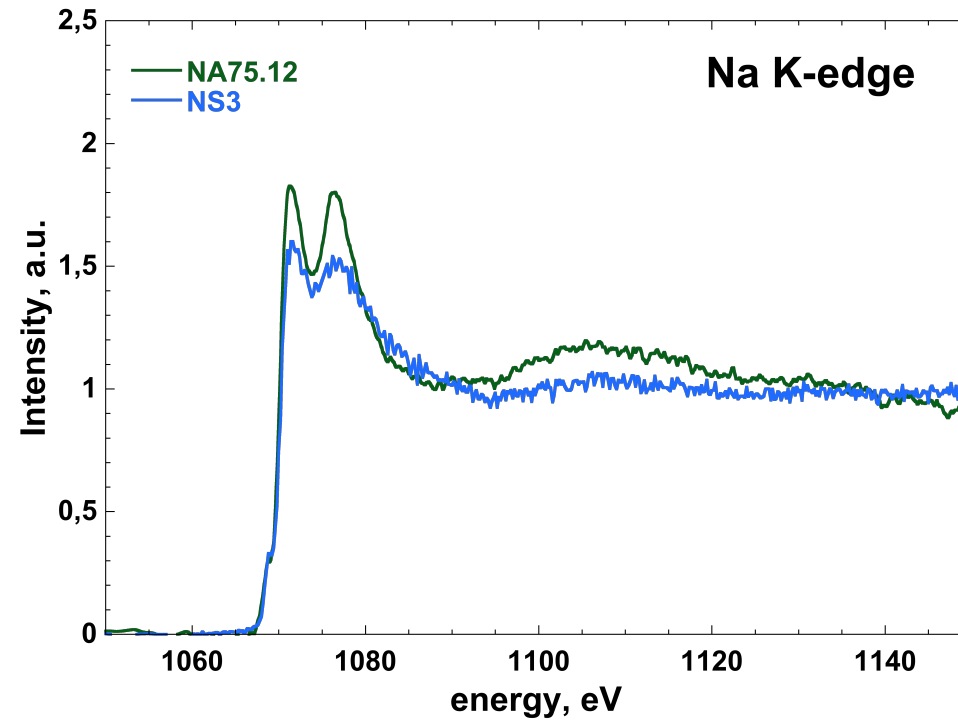
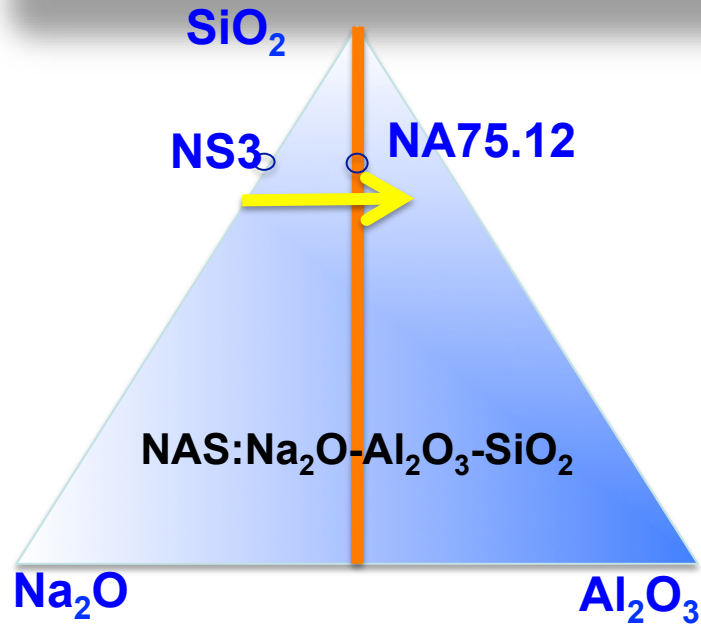
MgO

SiO₄
AlO₄⁻



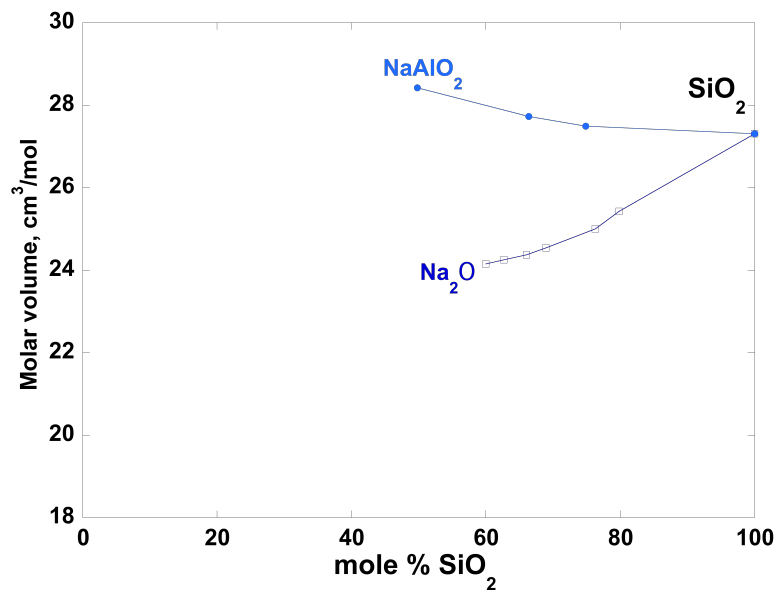


XANES at the Na K-edge



NS3 => $^{[6]}\text{Na}$

NA75.12 => $^{[9]}\text{Na}$



Neuville D.R., Cormier L, R., Flank A.M., Prado R.J. and Lagarde P. (2004) Na K-edge XANES spectra of minerals and glasses. Eur. J Mineral, 16, 809-816.



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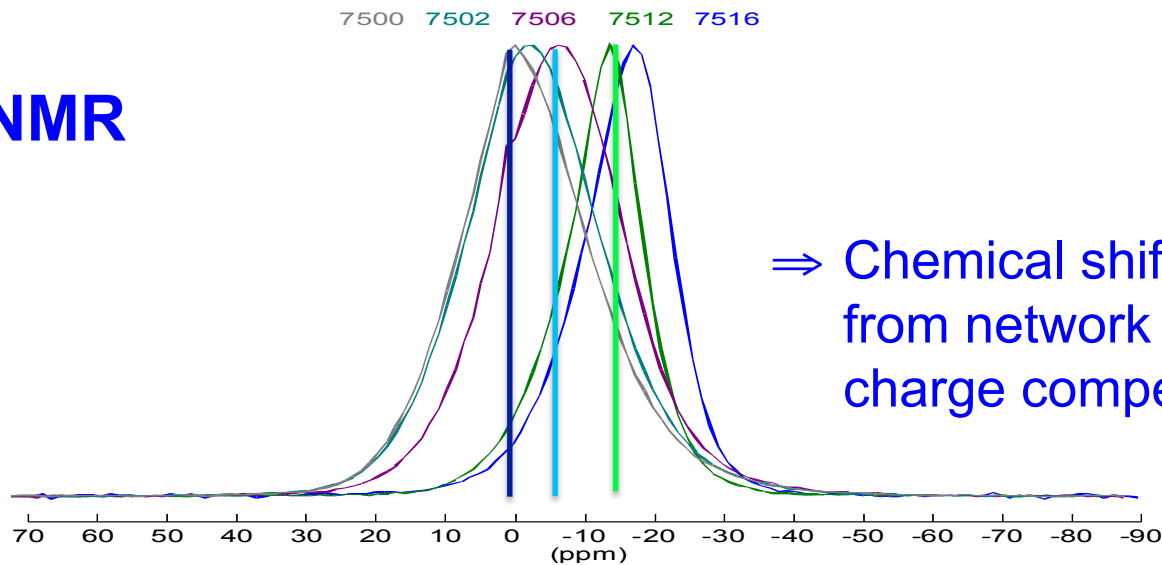
NMR : IR RMN CEMHTI-CNRS, Orléans


 $x=0.67$

Na : Network modifier

SiO₂

Charge compensator region

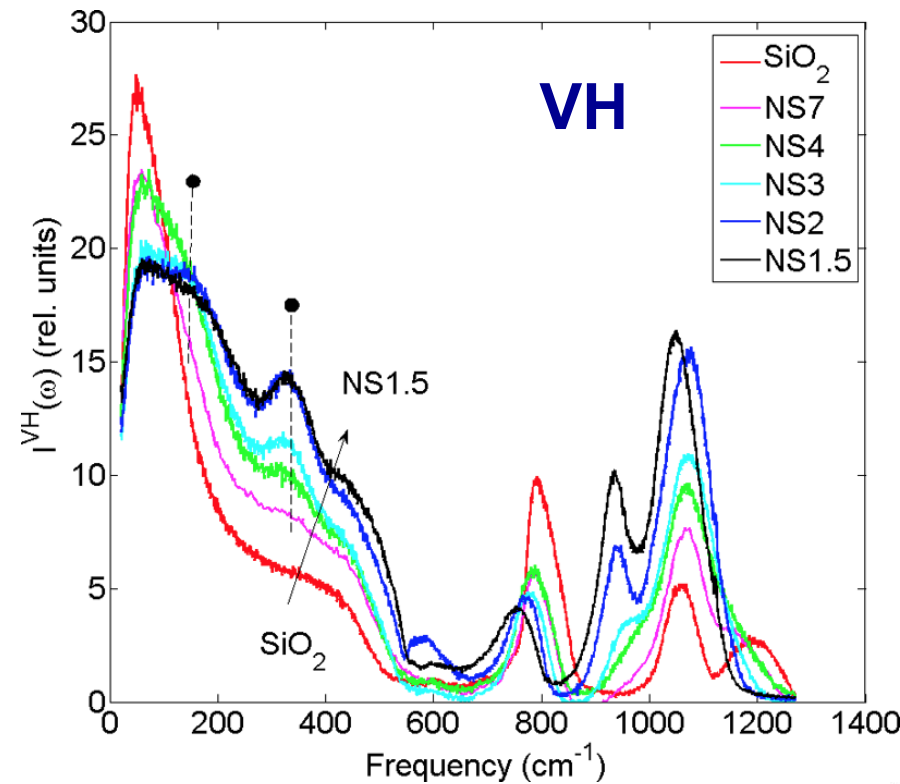
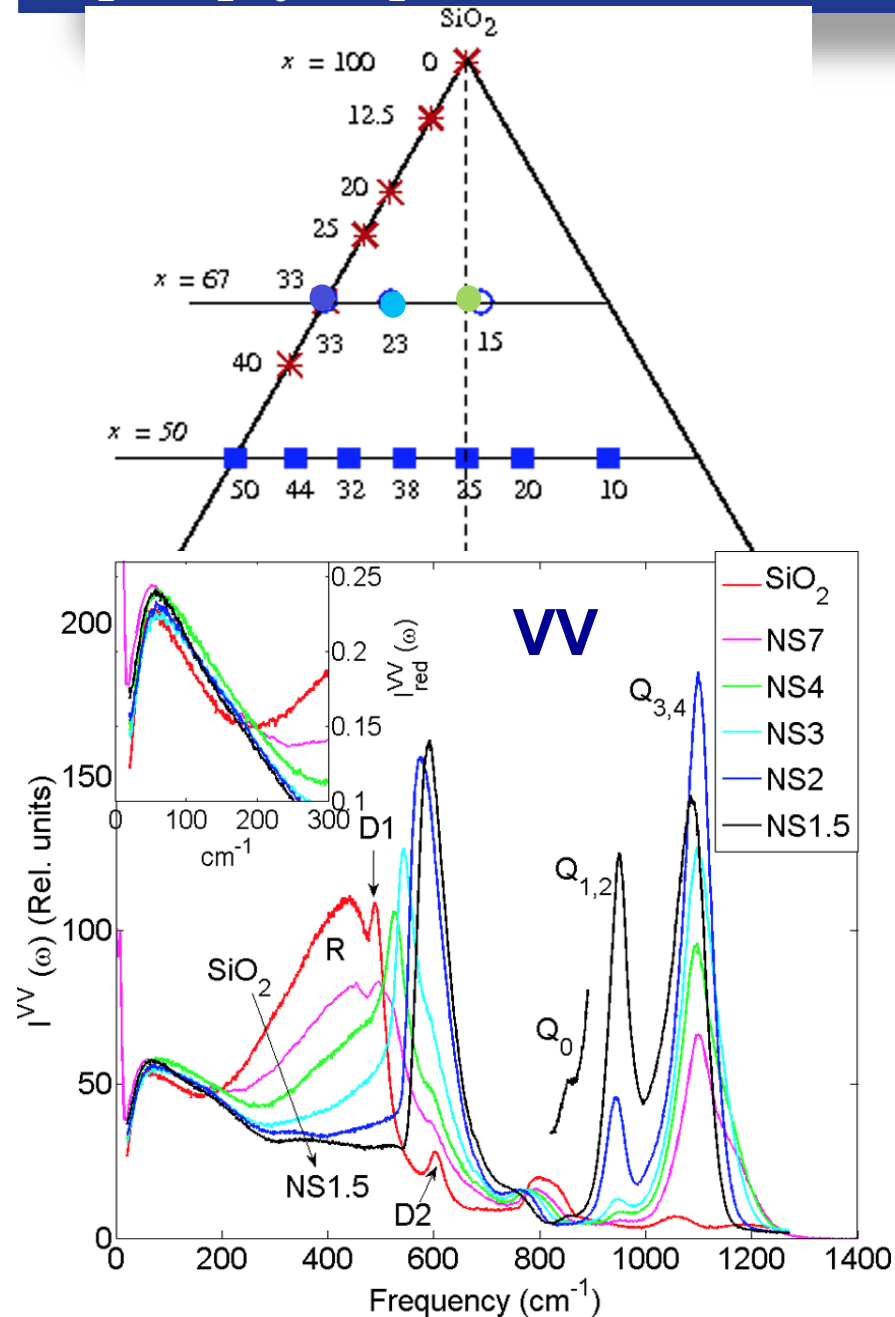
NAS:Na₂O-Al₂O₃-SiO₂²³Na NMR
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Le Losq Ch., Neuville D.R., Florian P., G.S. Henderson and Massiot D. (2014) Role of Al³⁺ on rheology and nano-structural changes of sodium silicate and aluminosilicate glasses and melts. *Geochimica Cosmochimica Acta*, 126, 495-517.



$\text{Na}_2\text{O}-\text{Al}_2\text{O}_3-\text{SiO}_2$

Raman spectrometry VV and VH



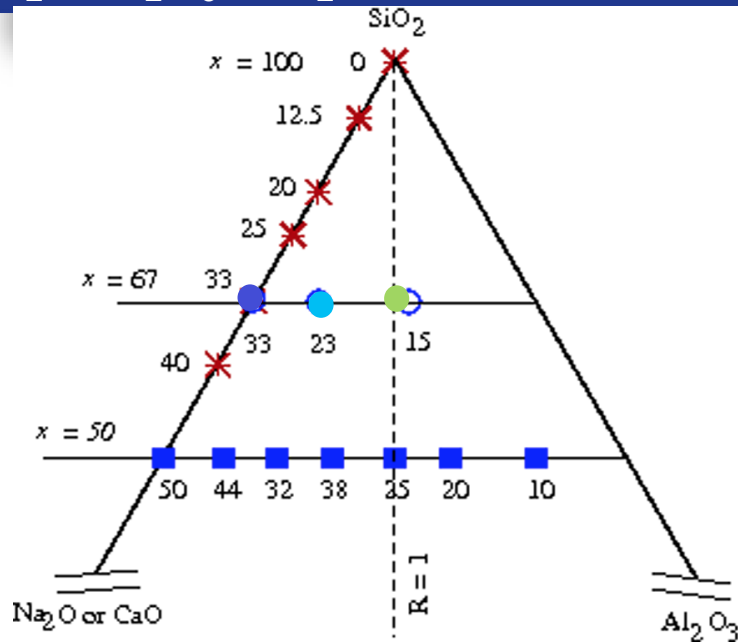
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Hehlen B. and Neuville D.R. (2015) Raman response of network modifier cations in aluminosilicate glasses. The Journal of Physical Chemistry B. 119, 4093–4098.

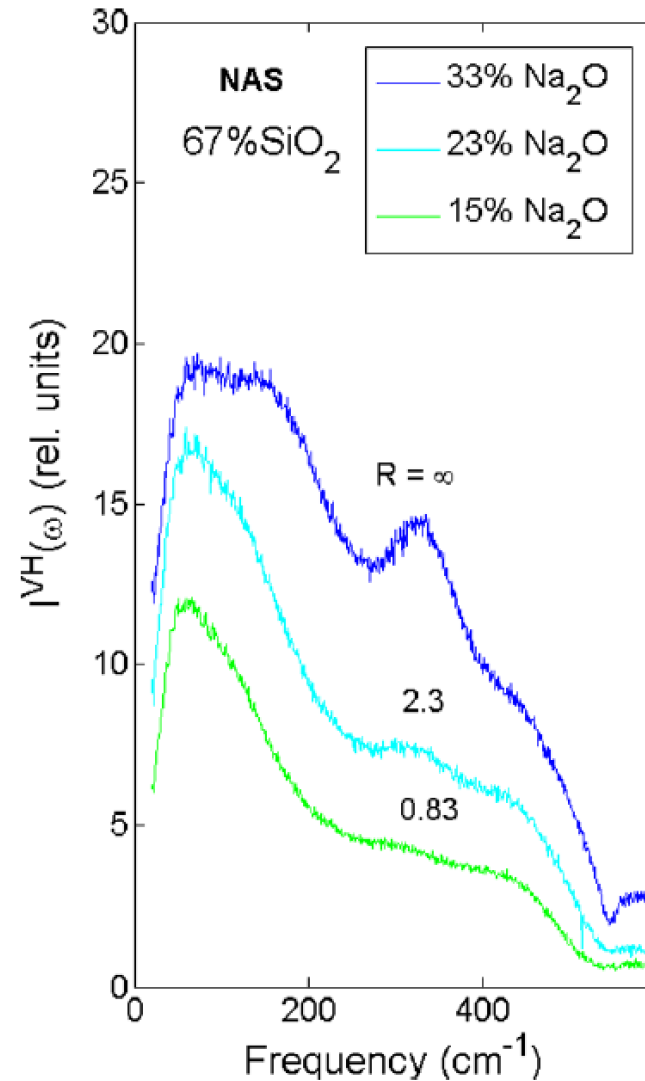
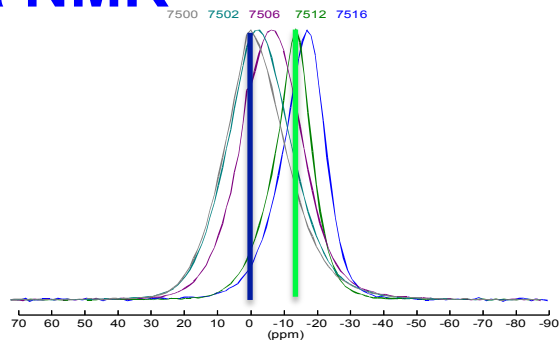


$\text{Na}_2\text{O}-\text{Al}_2\text{O}_3-\text{SiO}_2$

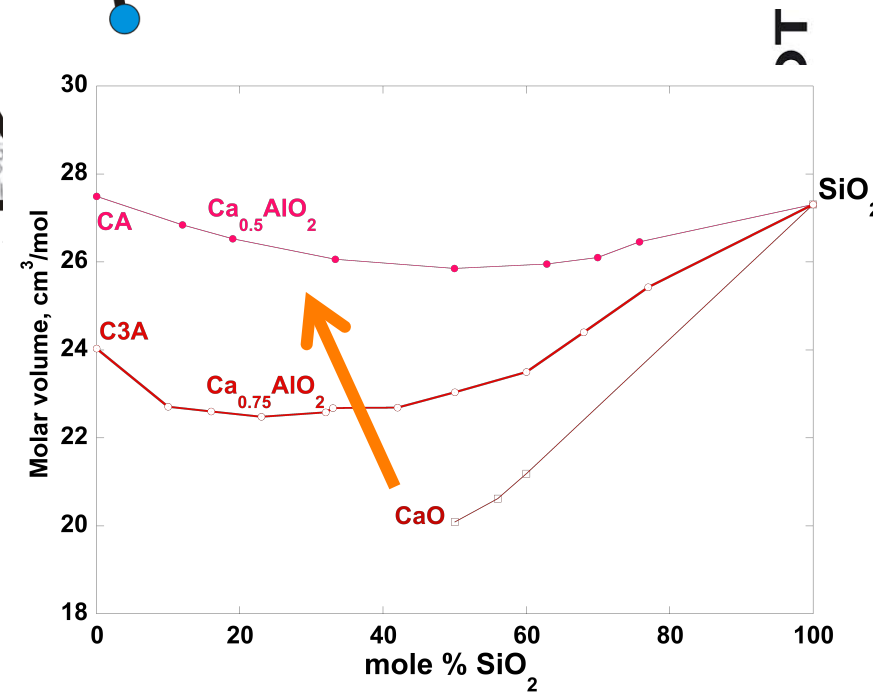
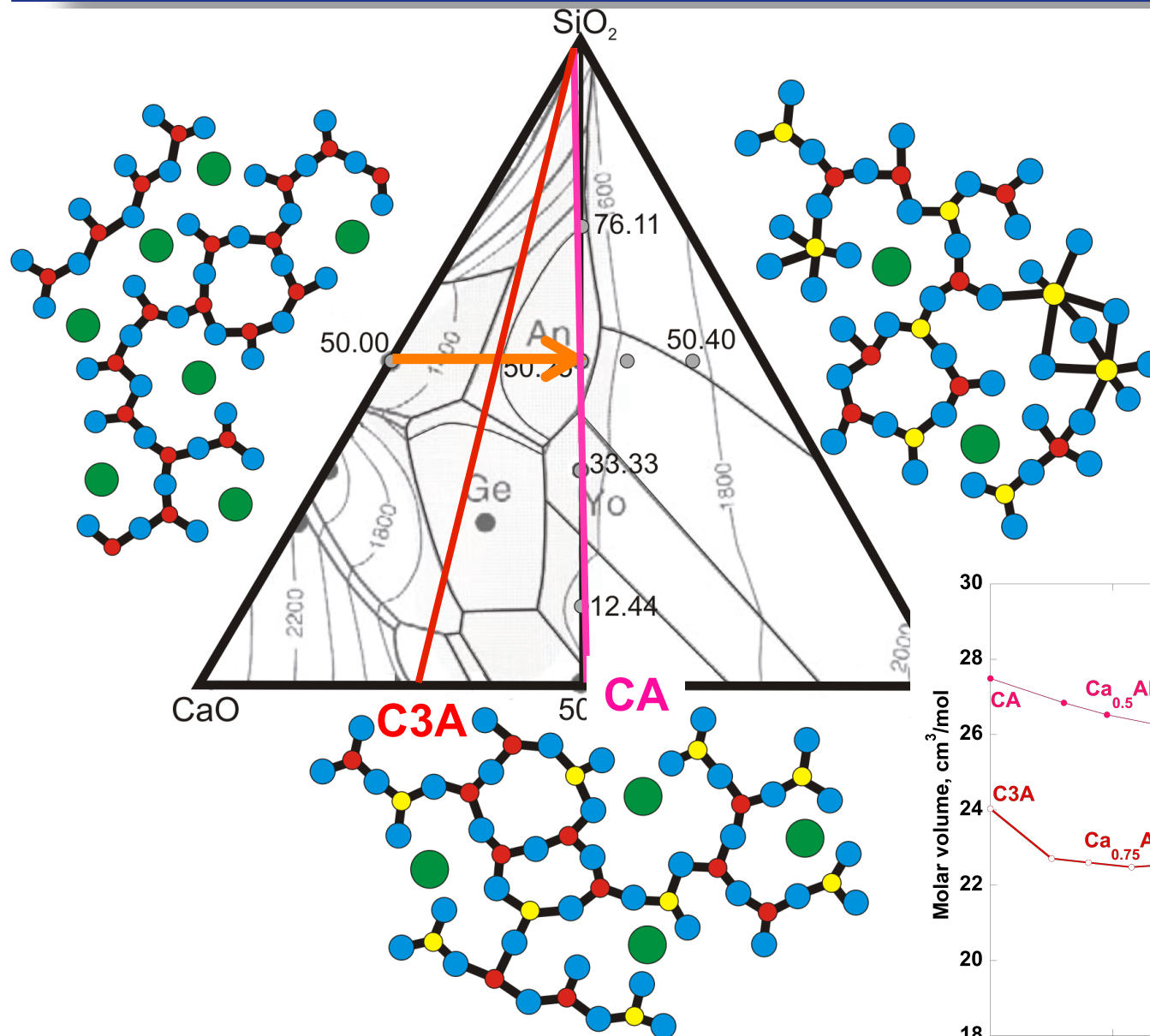
Raman spectrometry VV and VH



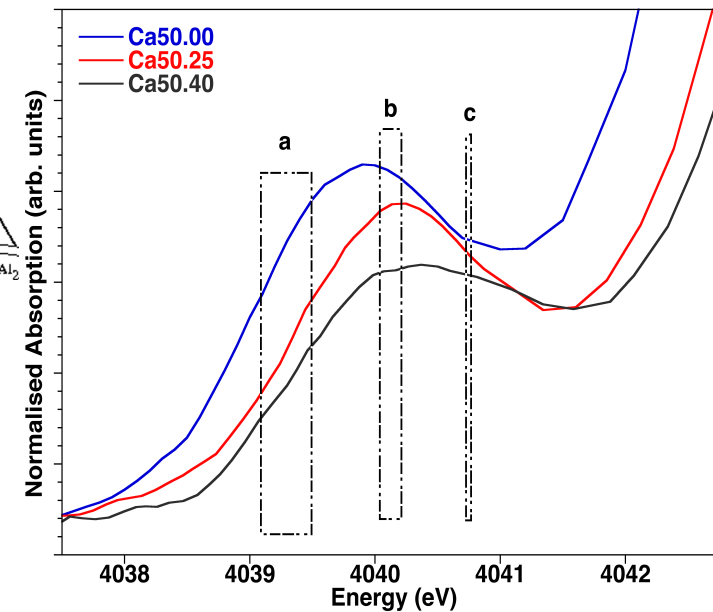
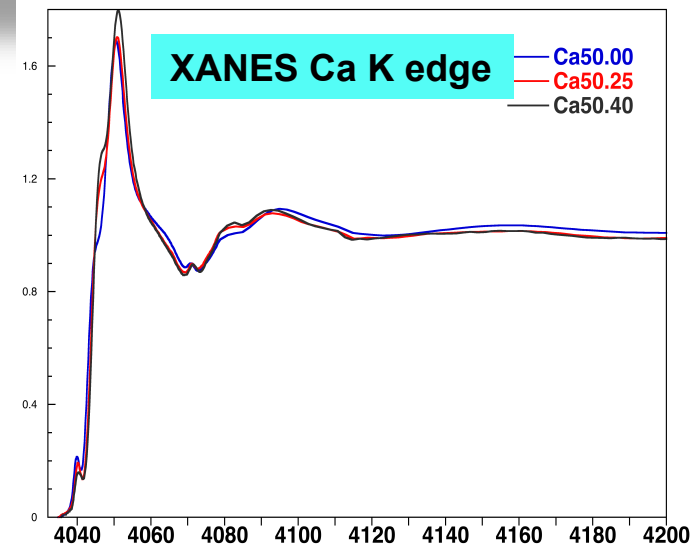
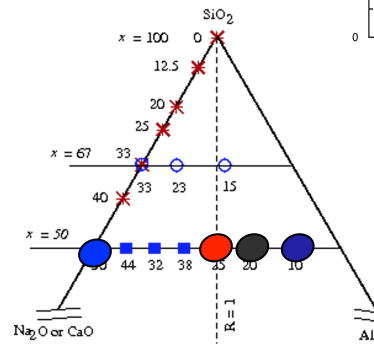
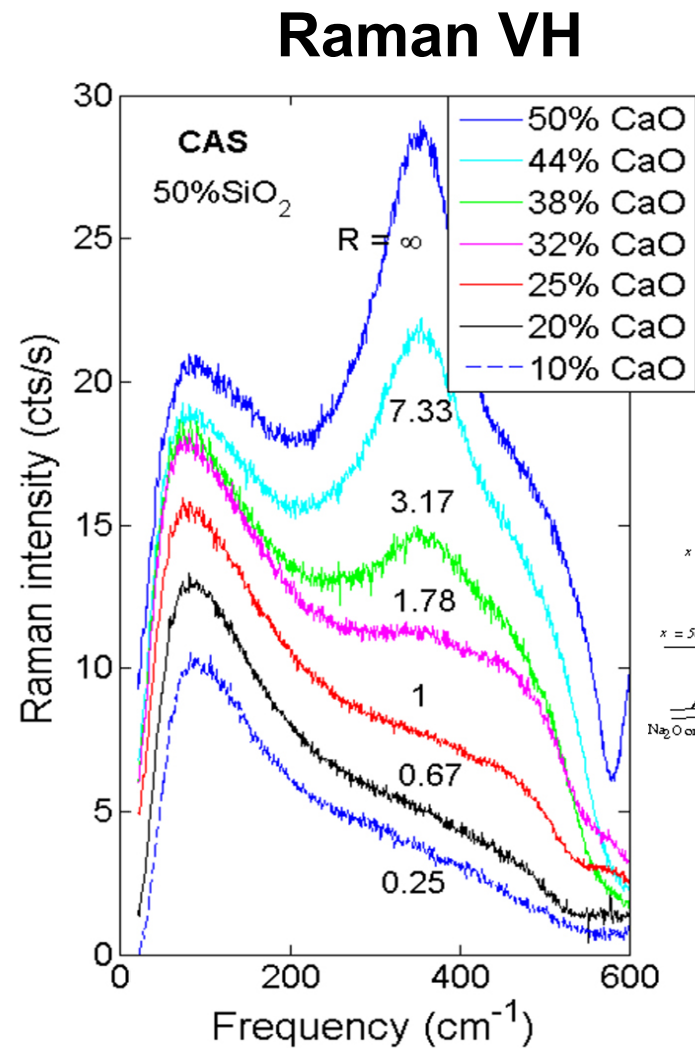
Hehlen B. and Neuville D.R. (2015) Raman response of network modifier cations in aluminosilicate glasses. The Journal of Physical Chemistry B. 119, 4093–4098.

 ^{23}Na NMR

CAS glass system



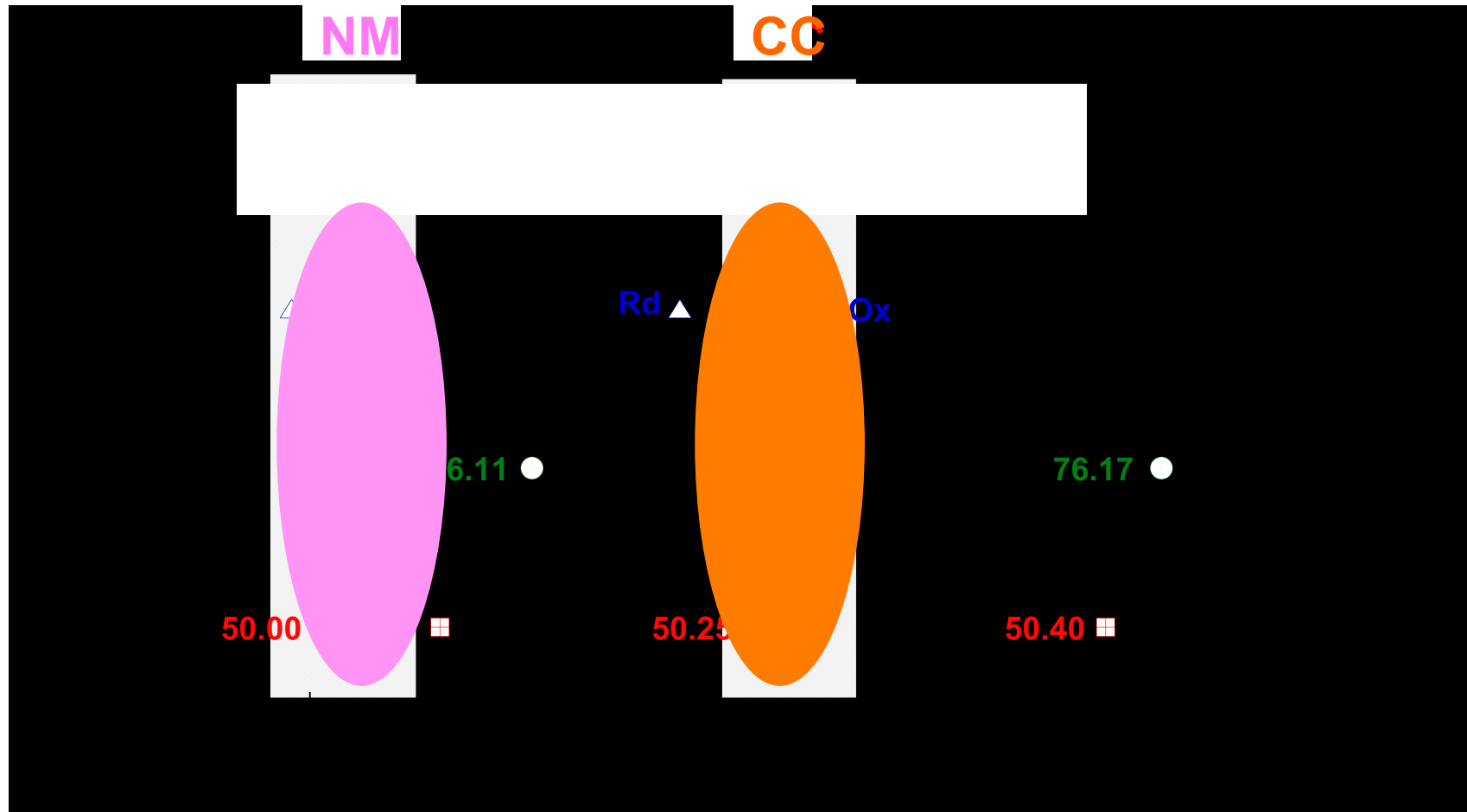
Modifier/compensator state of cations



Hehlen B. and Neuville D.R. (2015) Raman response of network modifier cations in aluminosilicate glasses. The Journal of Physical Chemistry B. 119, 4093–4098.

Cicconi M.R., de Ligny D., Gallo T. M., Neuville D.R. (2016) Ca Neighbors from XANES spectroscopy: a tool to investigate structure, redox and nucleation processes in silicate glasses, melts and crystals. American Mineralogist, 101, 1232-1236.

XANES at the Ca K-edge

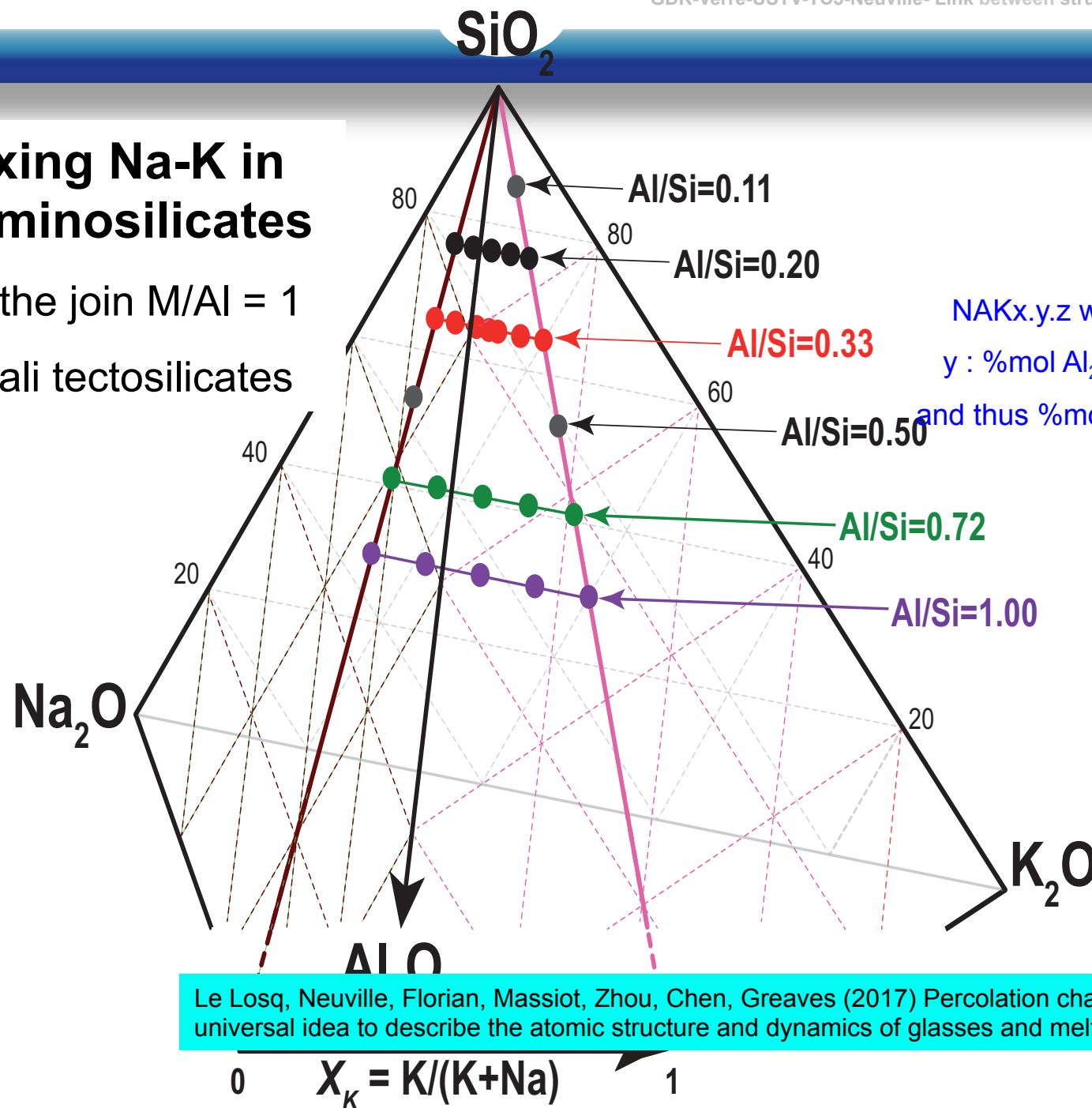


Cicconi M.R., de Ligny D., Gallo T. M., Neuville D.R. (2016) Ca Neighbors from XANES spectroscopy: a tool to investigate structure, redox and nucleation processes in silicate glasses, melts and crystals. American Mineralogist, 101, 1232-1236.

Mixing Na-K in aluminosilicates

On the join $M/Al = 1$

Alkali tectosilicates



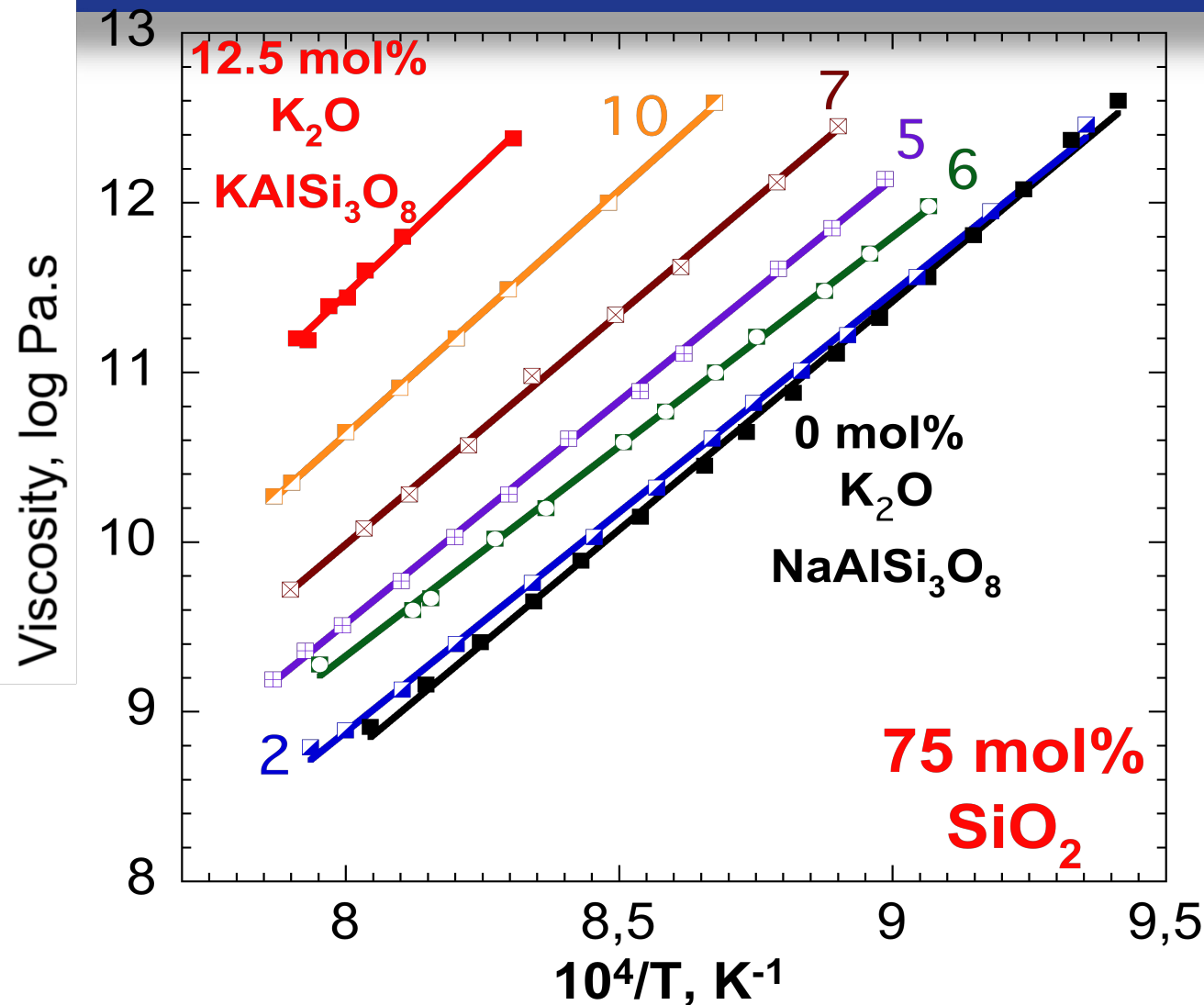
NAK_{x.y.z} with x : %mol SiO₂ ;

y : %mol Al₂O₃ ; z : %mol K₂O

and thus %mol Na₂O = 100-(x+y+z)

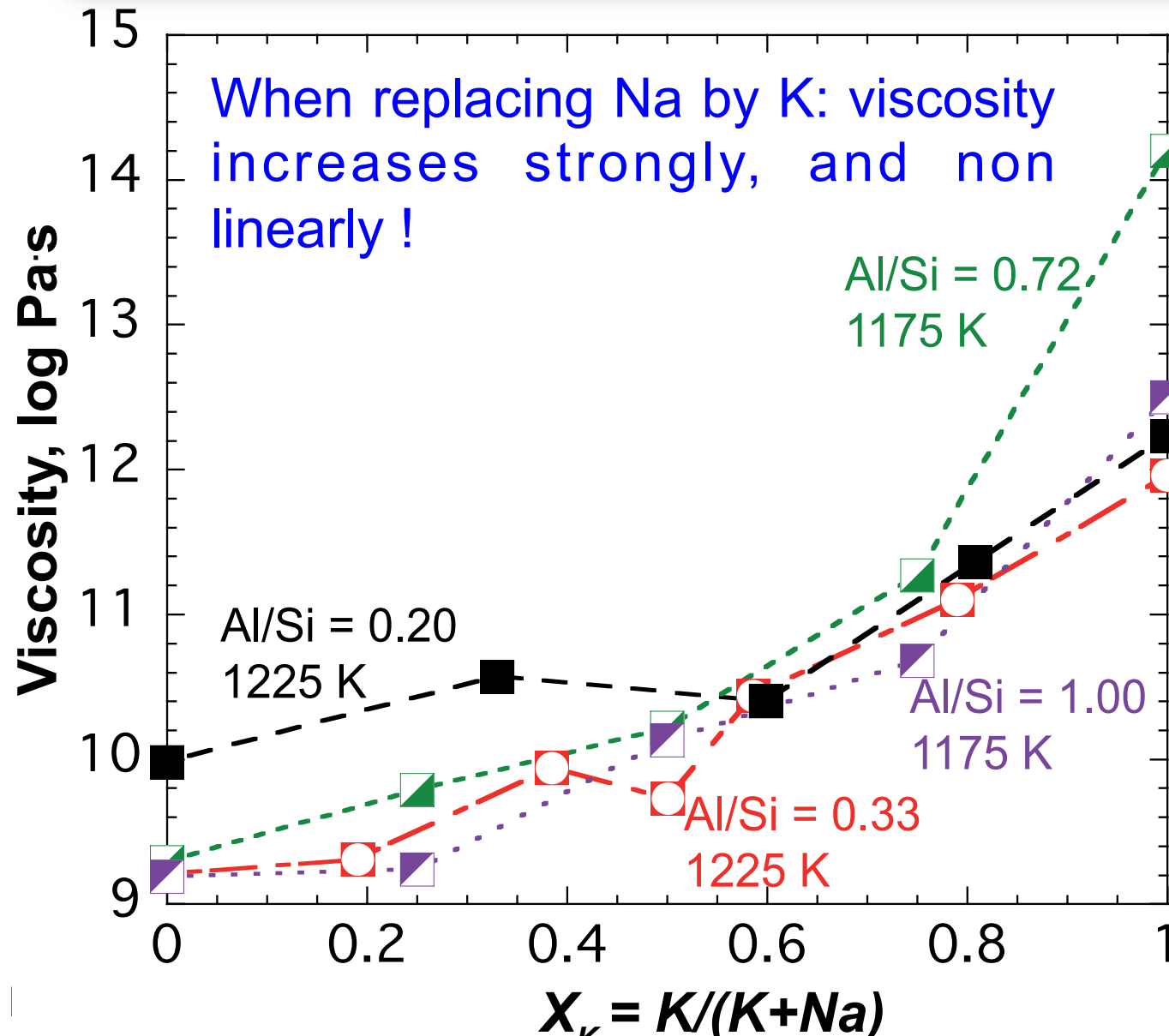
Al/Si=0.11 = 90% S
Al/Si=0.20 = 85% S
Al/Si=0.33 = 75% S
Al/Si=0.50 = 66% S
Al/Si=0.72 = 58% S
Al/Si=1.00 = 50% S

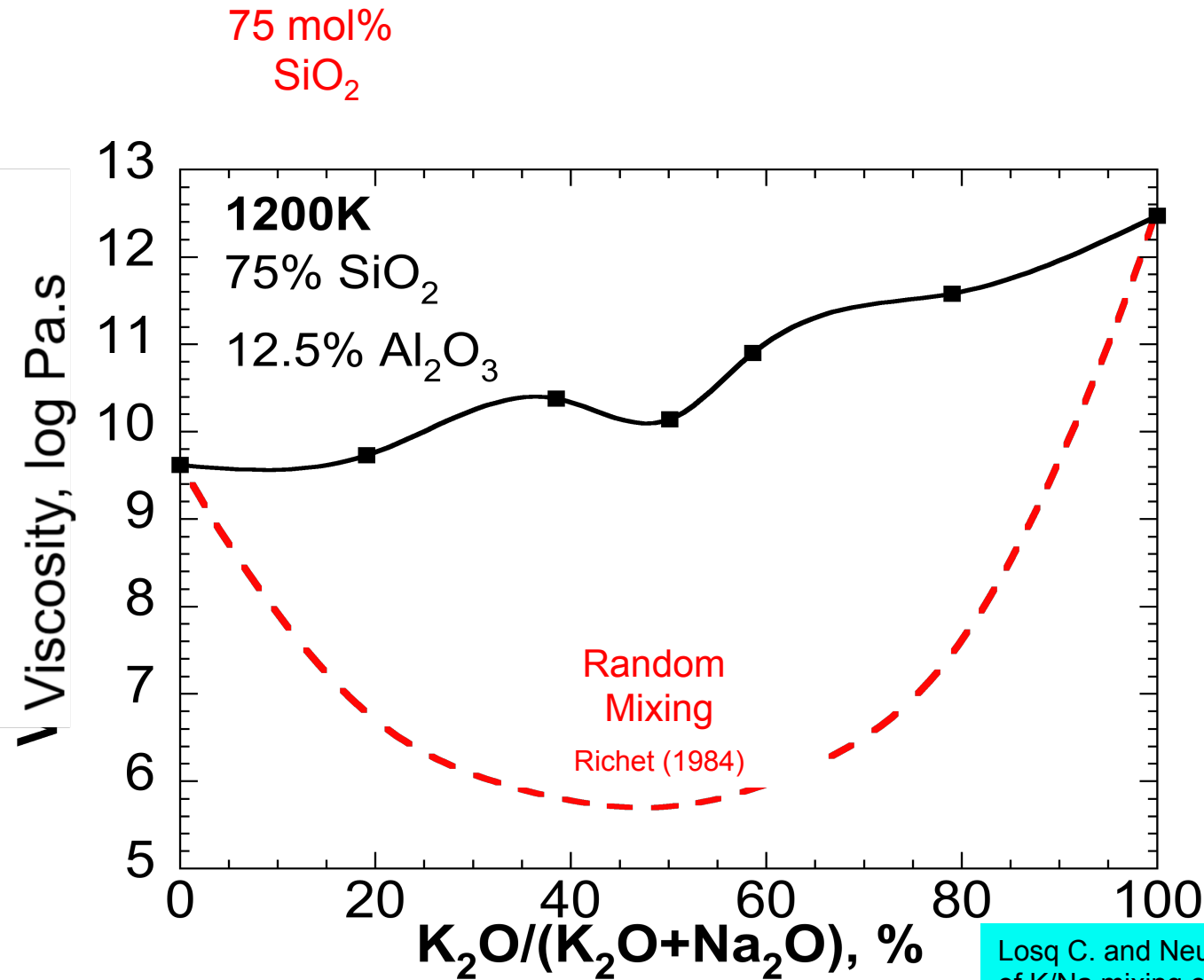
Le Losq, Neuville, Florian, Massiot, Zhou, Chen, Greaves (2017) Percolation channels: a universal idea to describe the atomic structure and dynamics of glasses and melts. PNAS



When replacing Na by K: viscosity increases strongly...

Losq C. and Neuville D.R. (2013) Effect of K/Na mixing on the structure and rheology of tectosilicate silica-rich melts. Chemical Geology, 346, 57-71.



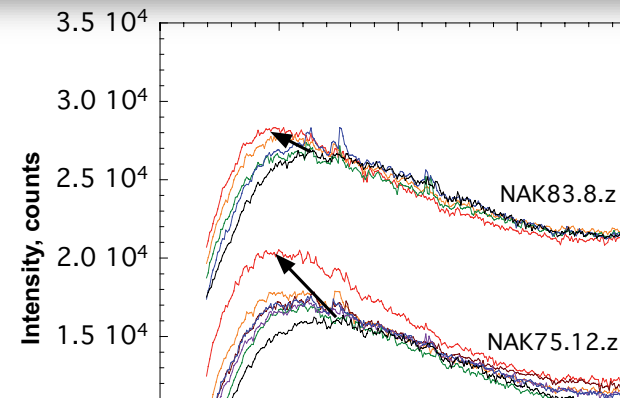
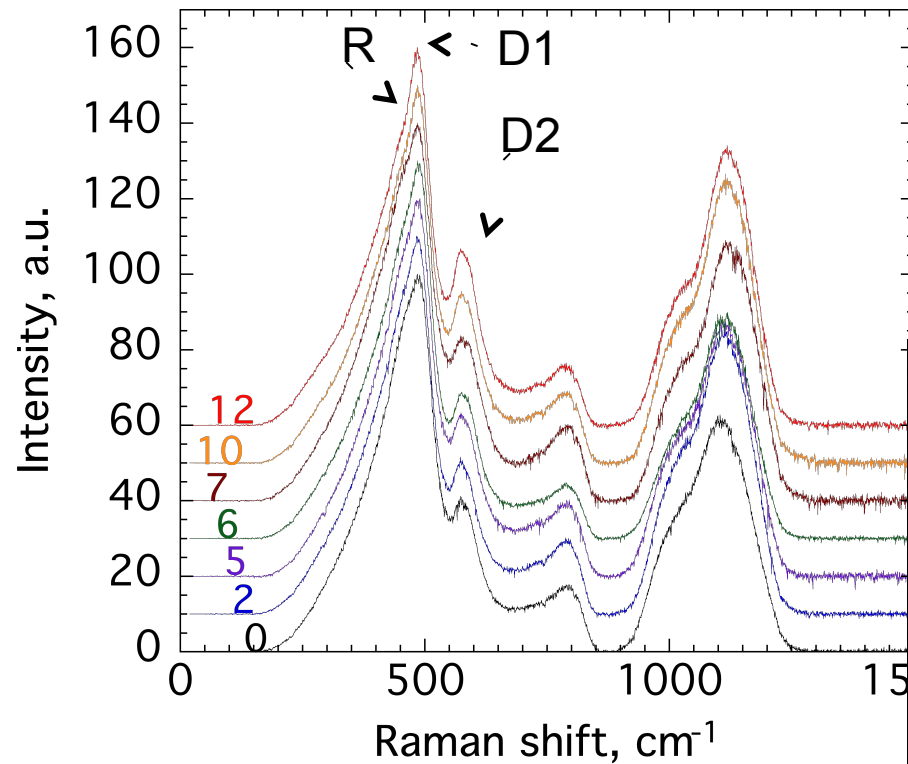
Mixed albite-orthoclase melts $\text{NaAlSi}_3\text{O}_8$ - KAlSi_3O_8 

Na and K do not
mix randomly
⇒ Viscosity,
thermodynamics
⇒ Structure

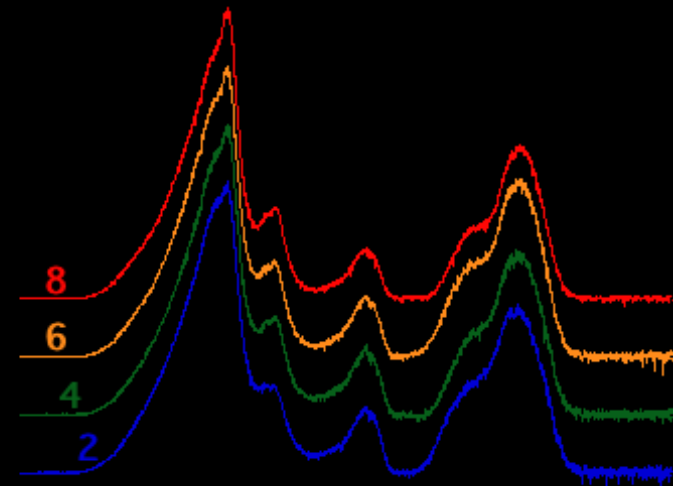


Losq C. and Neuville D.R. (2013) Effect of K/Na mixing on the structure and rheology of tectosilicate silica-rich melts. *Chemical Geology*, 346, 57-71.

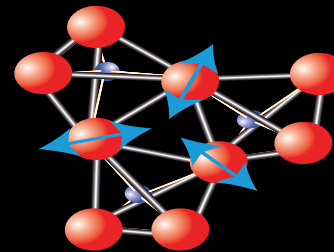
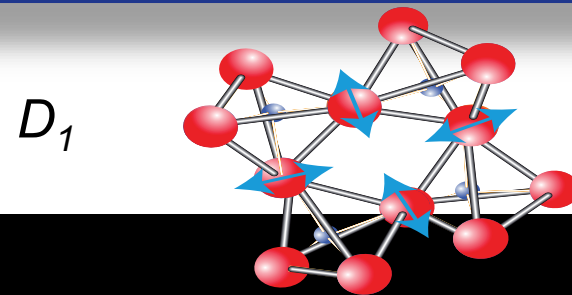




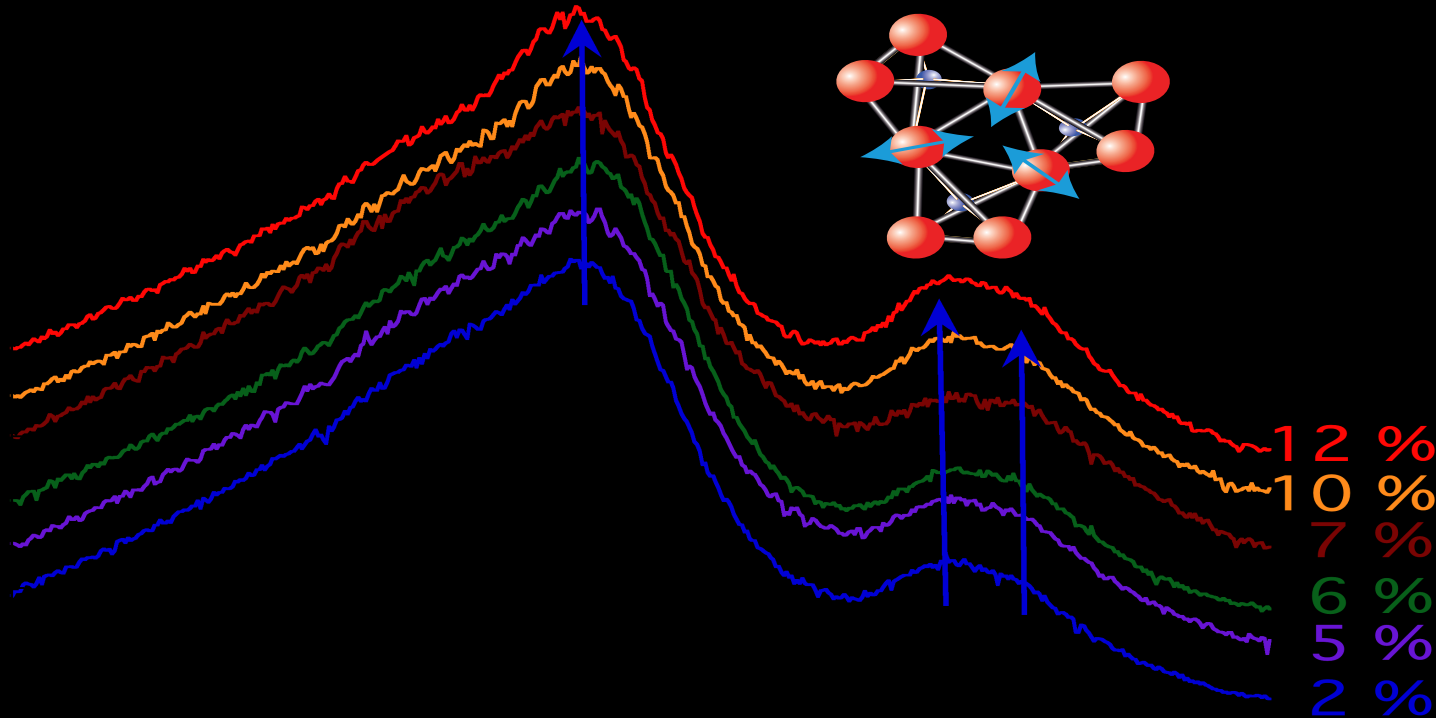
- Boson peak increase in I and decreases in frequency with K like close than SiO₂
- D1 and D2 increase with K
- New D2 band



Losq C. and Neuville D.R. (2013) Effect of K/Na mixing on the structure and rheology of tectosilicate silica-rich melts. Chemical Geology, 346, 57-71.

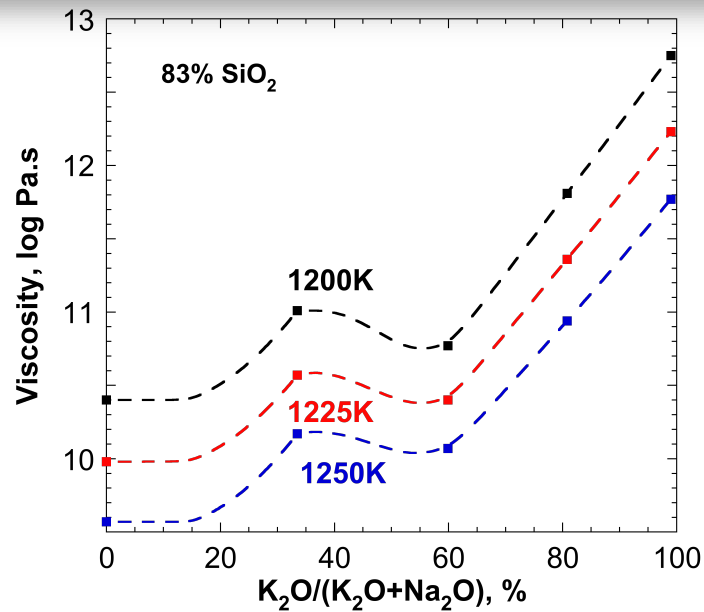


Intensity, a.u.

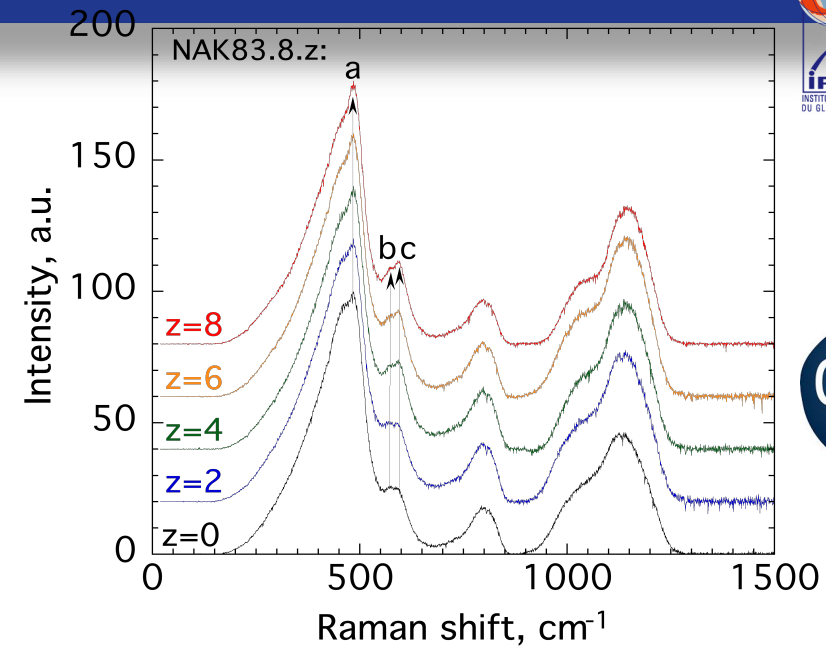
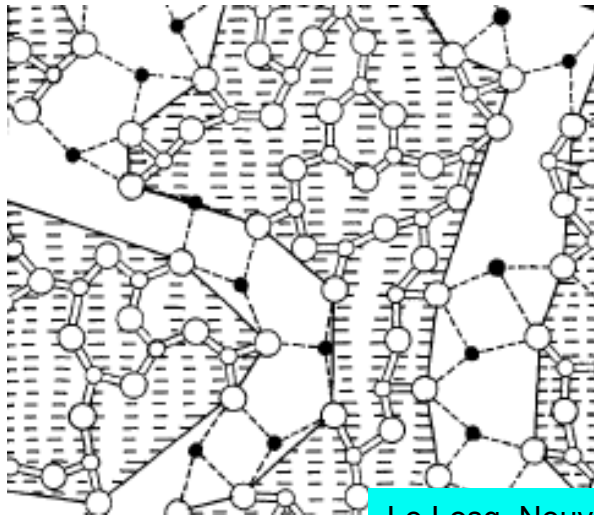


Wavenumber, cm^{-1}

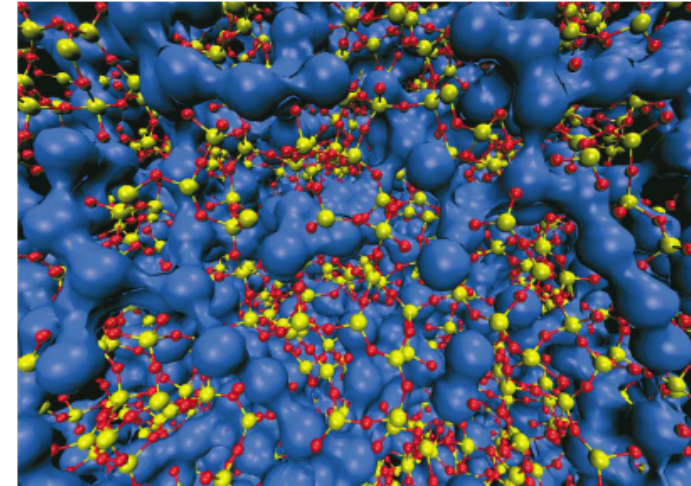
Losq C. and Neuville D.R. (2013) Effect of K/Na mixing on the structure and rheology of tectosilicate silica-rich melts. Chemical Geology, 346, 57-71.



Greaves, 1985: MRN

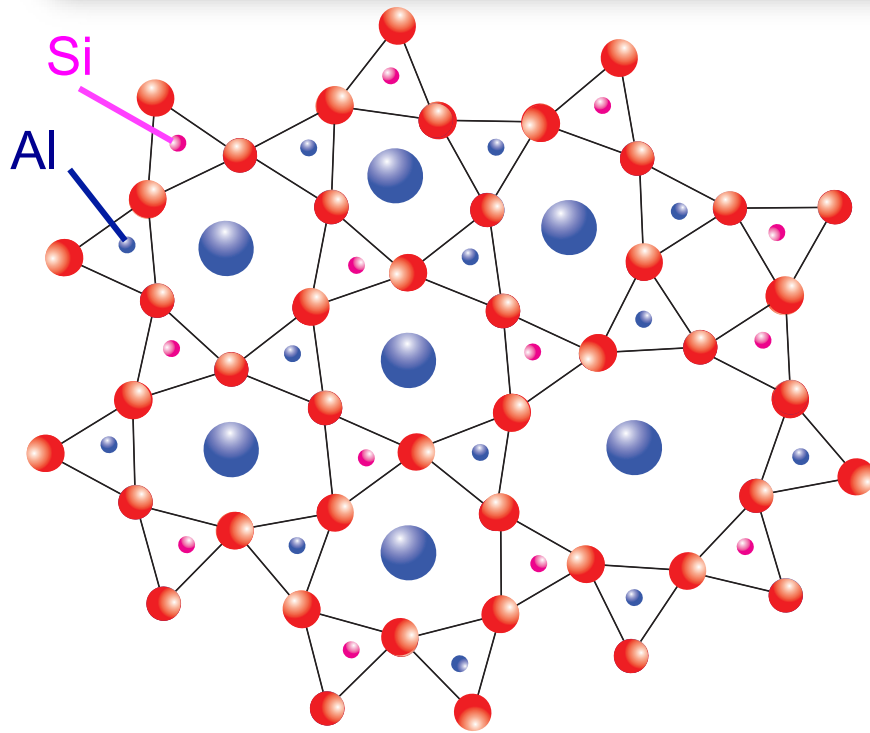


Meyer et al., 2004

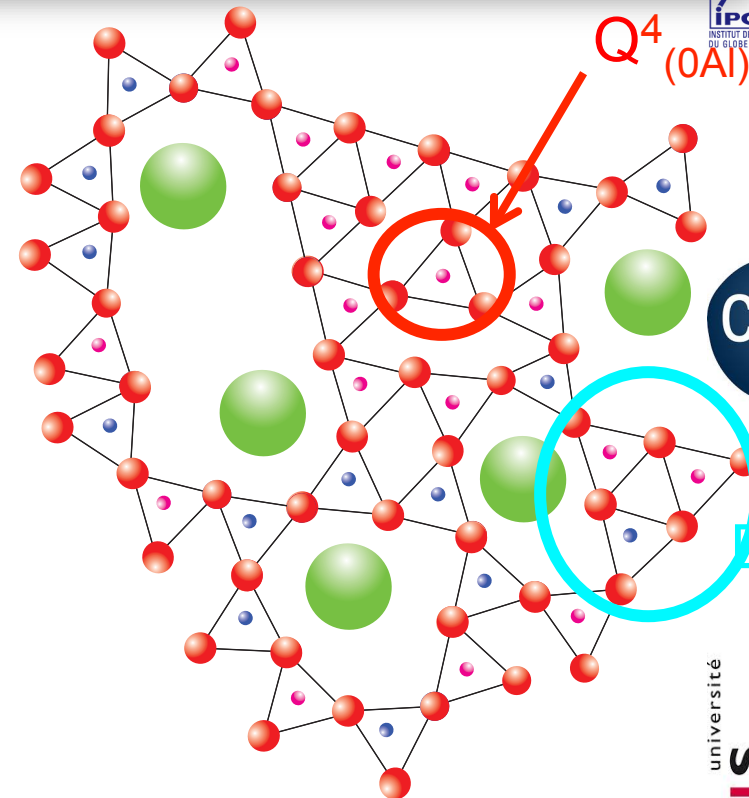


Na tectosilicates

K tectosilicates



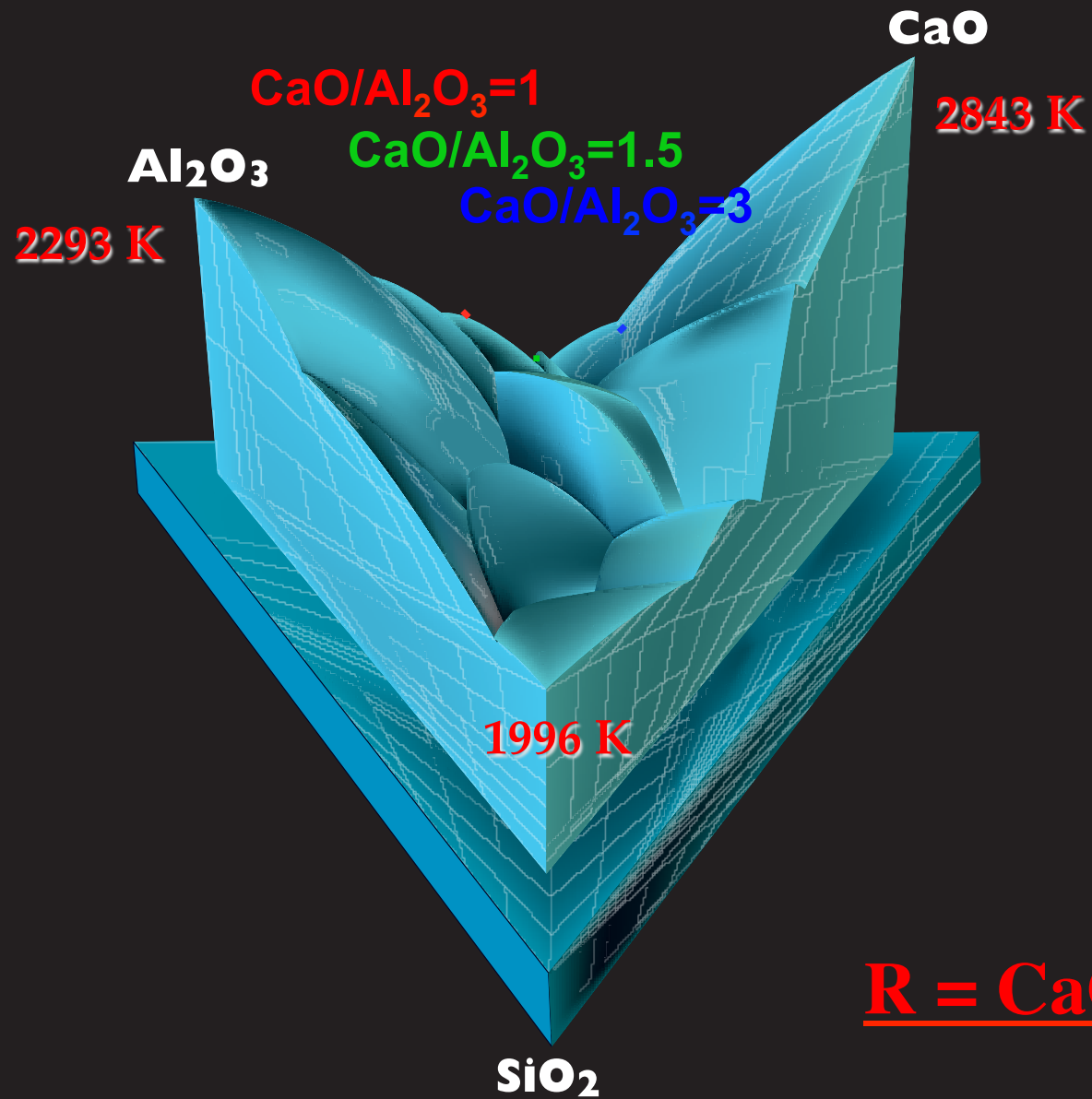
Compensated Continuous
Random Network
From Greaves and Ngai, 1995



We propose a new version:
Compensated Modified
Random Network

Na and K are in different structural positions
⇒ Two different networks
⇒ Non random mixing

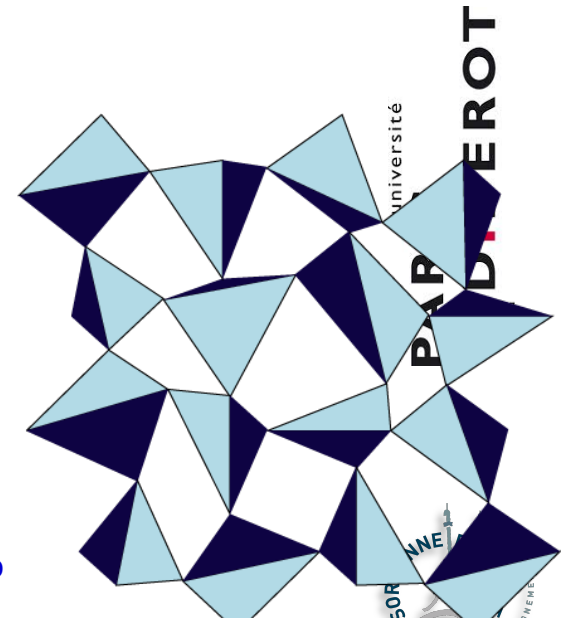
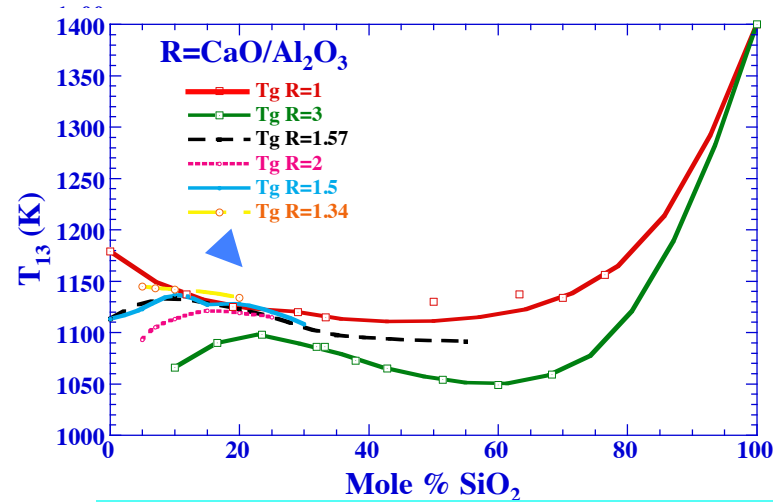
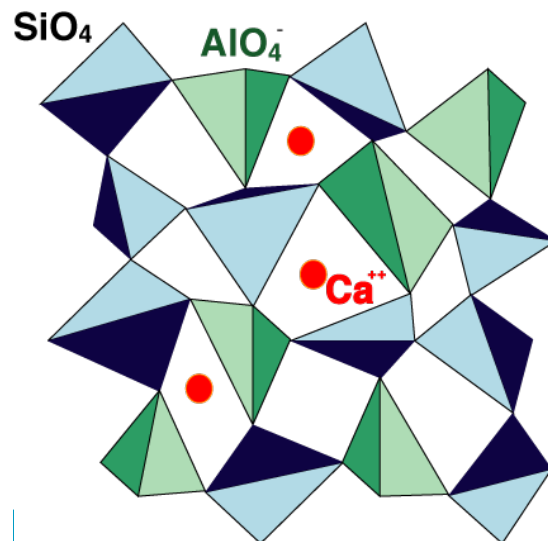
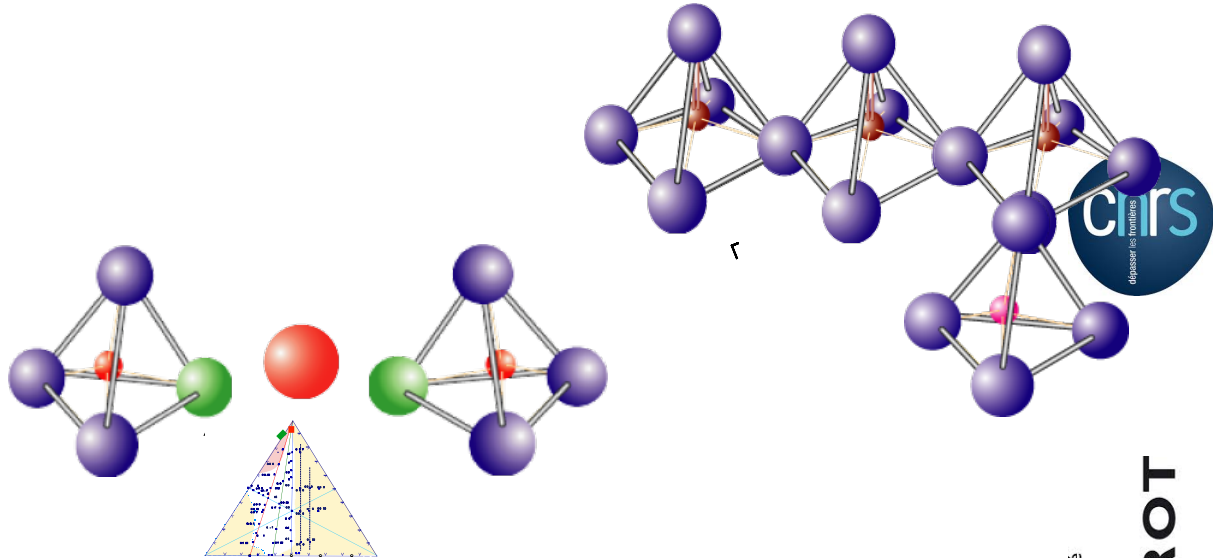
Al³⁺ in silicate glasses and melts

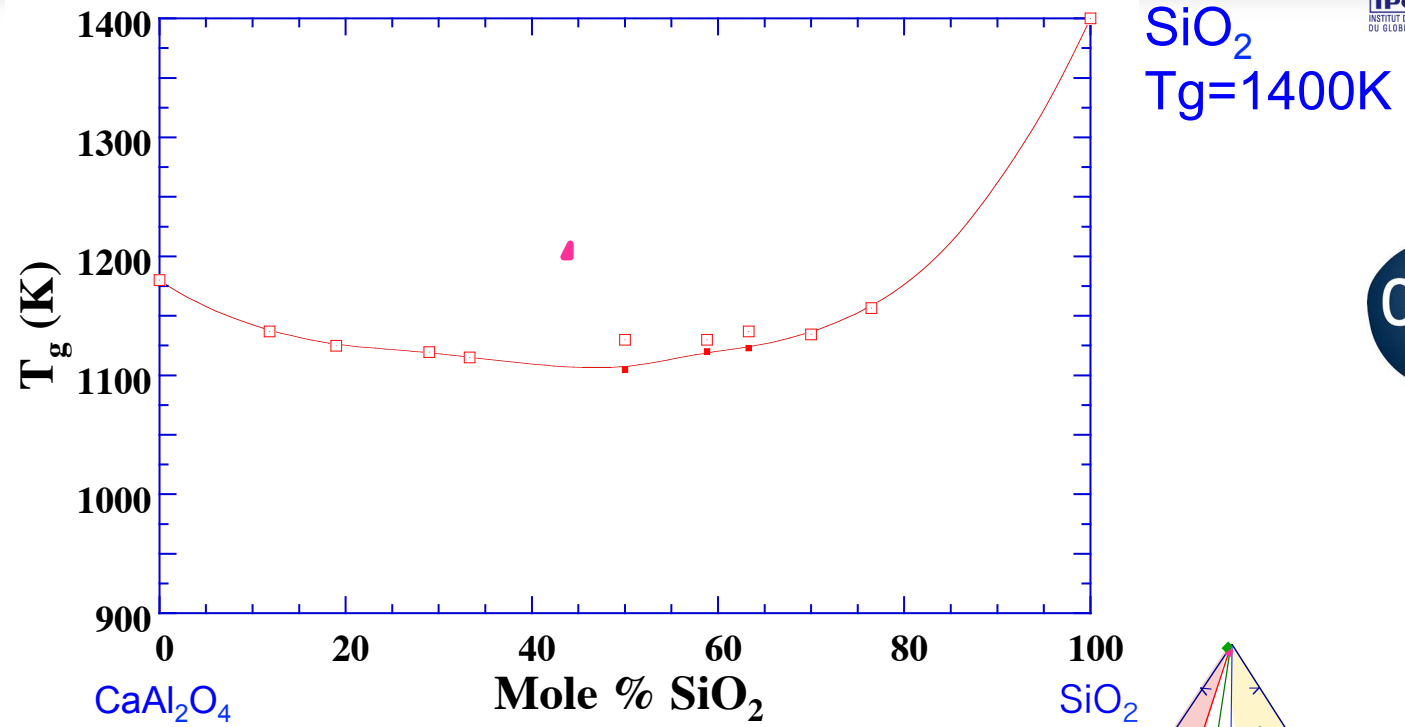


$$\underline{R = \text{CaO}/\text{Al}_2\text{O}_3}$$

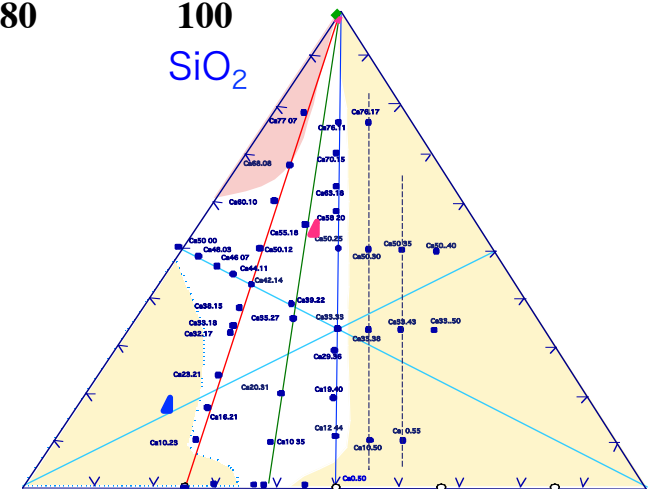
after Rankin, 1915

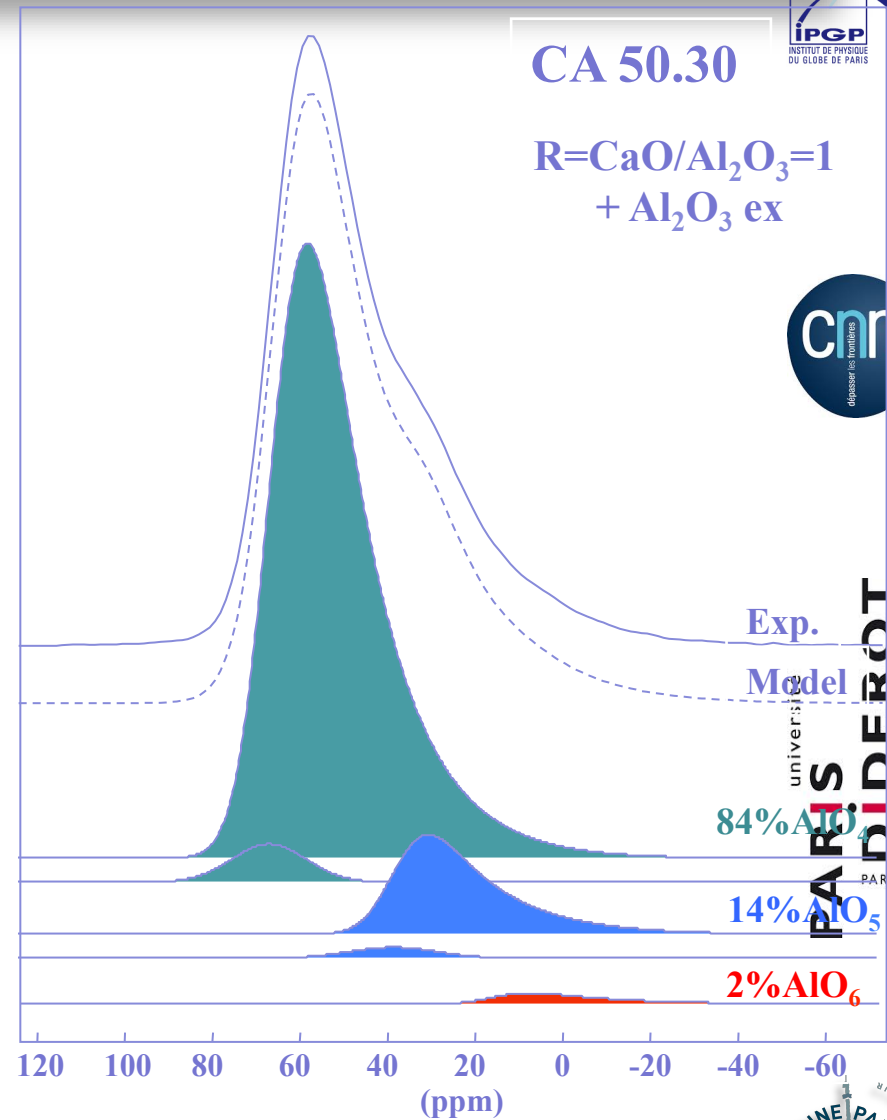
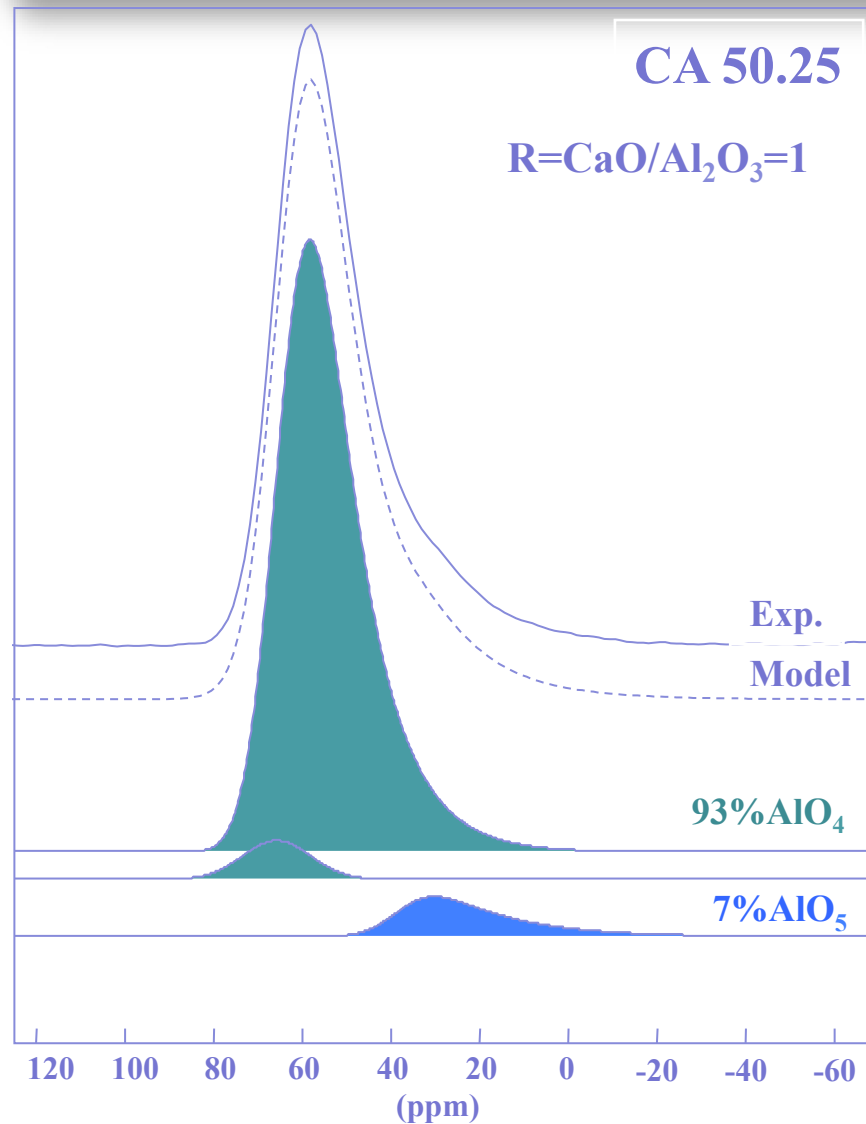
Glass transition temperature





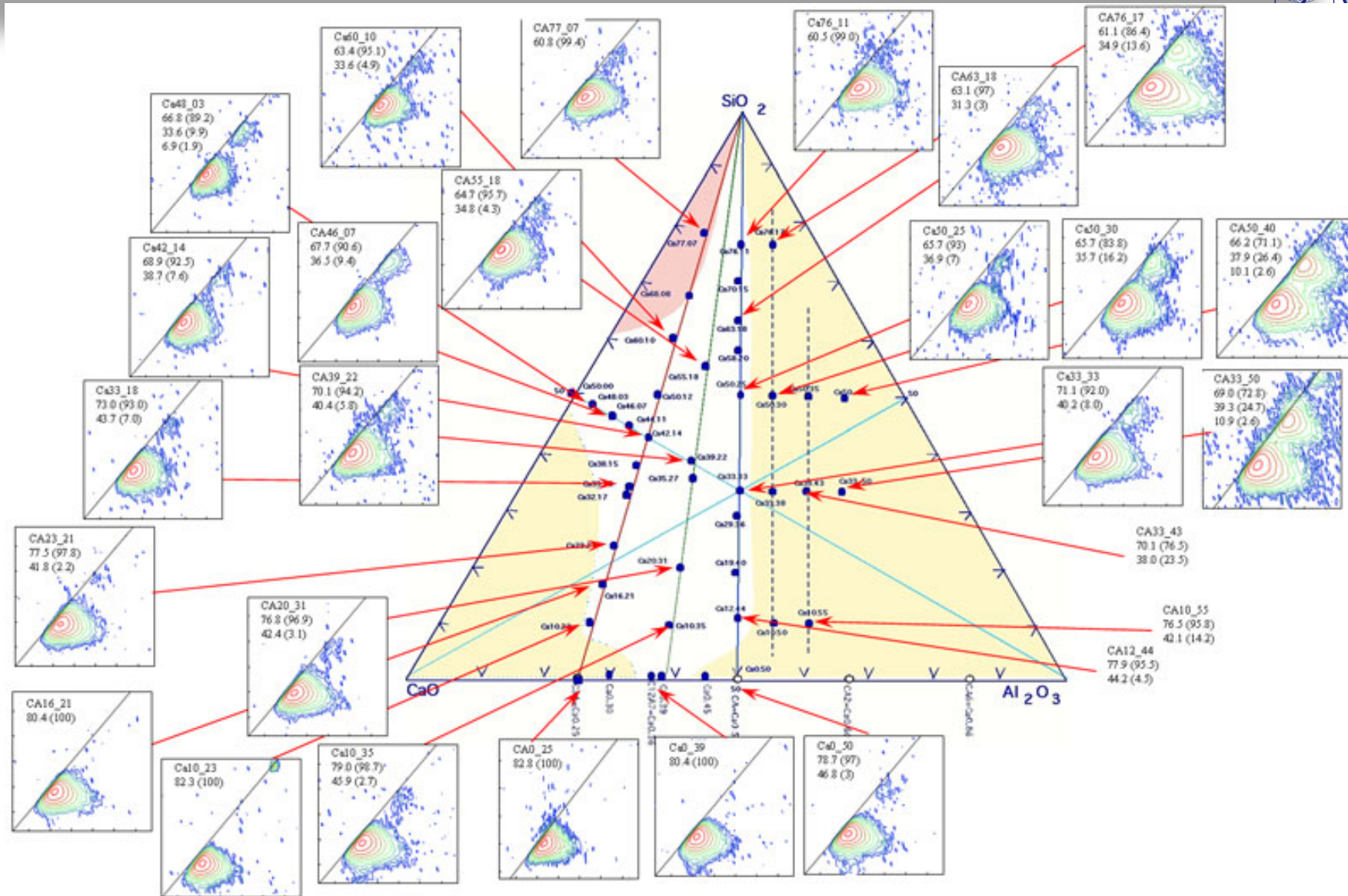
OT



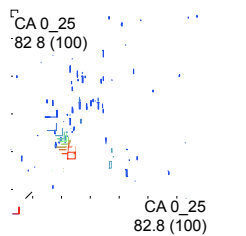
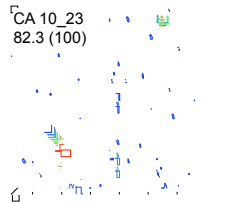
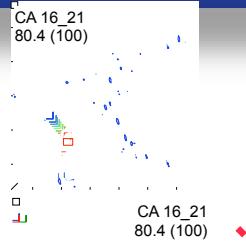
Al³⁺ in silicate glasses and melts

RMN 750MHz, CRMHT, Orléans, ²⁷Al 1D MAS

Al^{3+} in silicate glasses and melts

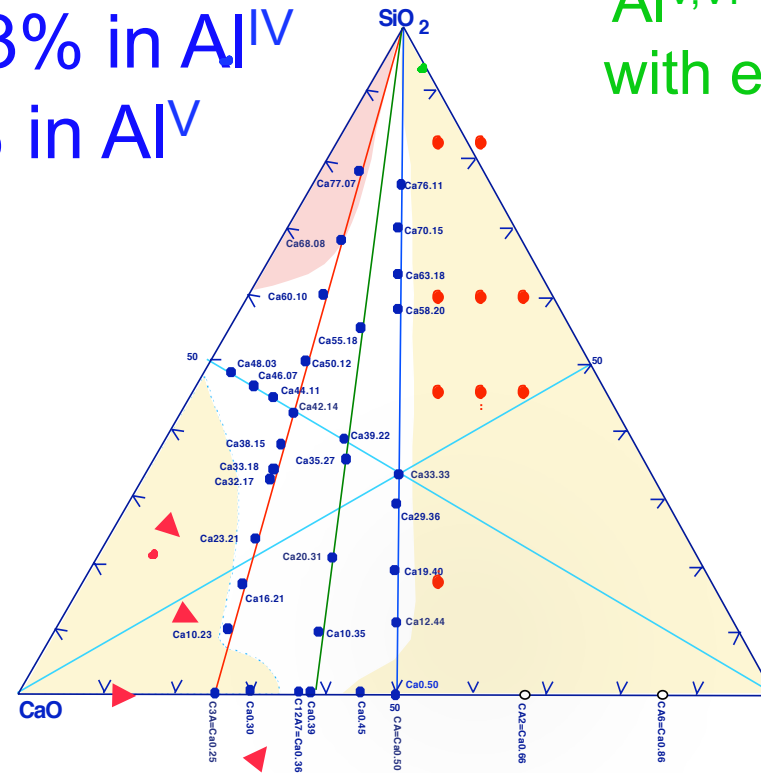


37

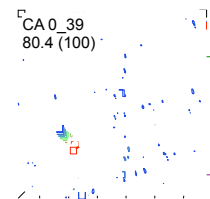
Al³⁺ in silicate glasses and melts

$1 < R < +3$
Al $\Rightarrow$ 93% in Al^{IV}
~7% in Al^V

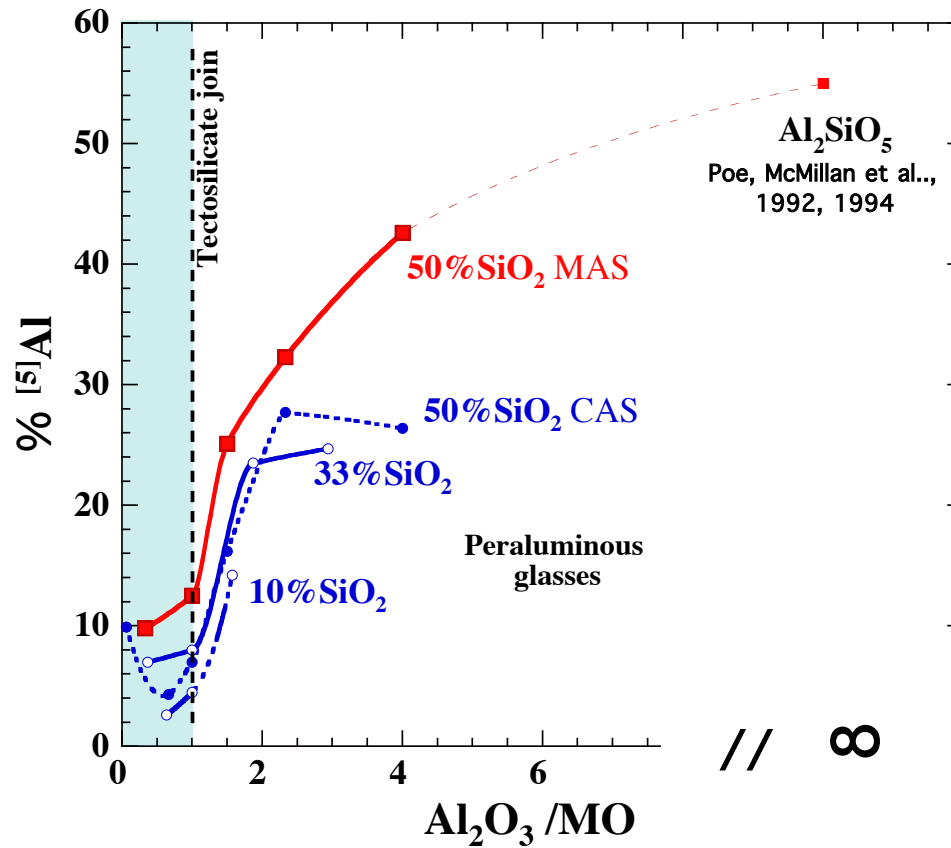
Peraluminous glasses
Al in Al^{IV}, Al^V, Al^{VI}
Al^{V,VI} increase
with excess Al



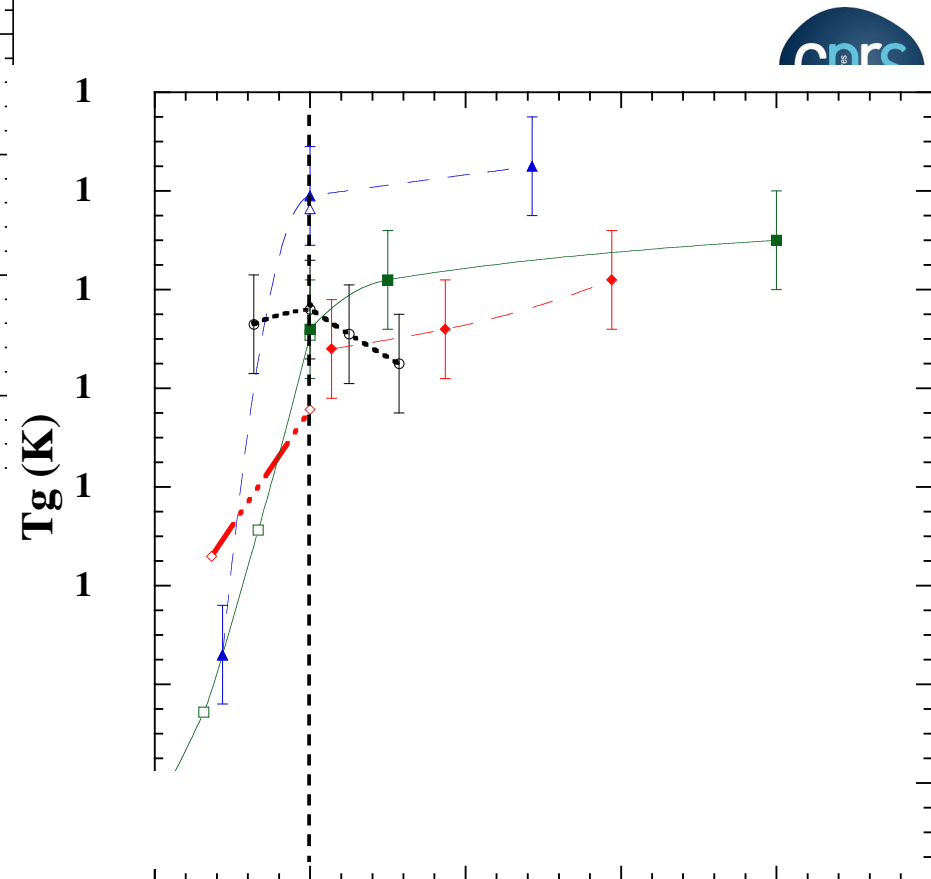
$\Rightarrow$ Al 100% in Al^{IV}
Al in Q² and Q³ species

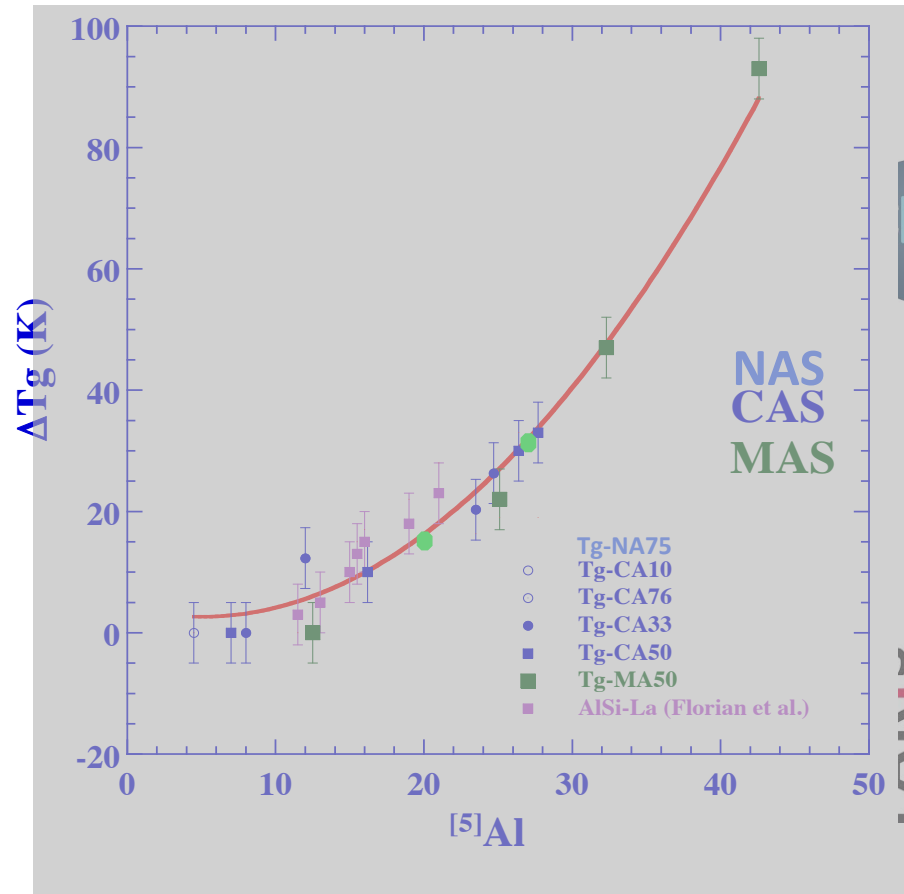


CA 0_39
80.4 (100)



[5]Al increases with Al_2O_3 content for $R=CaO/Al_2O_3 < 1$

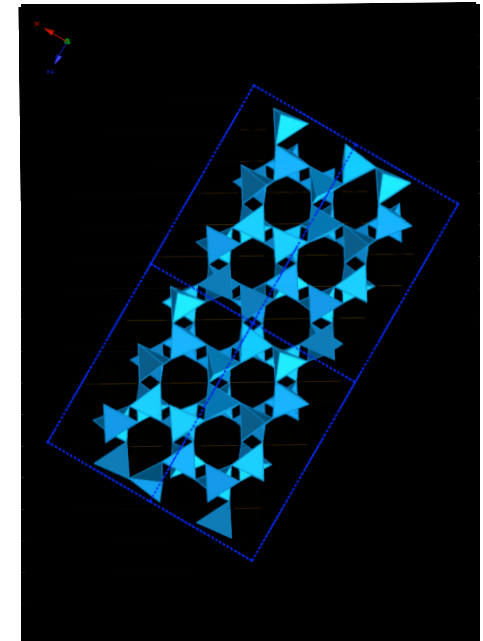
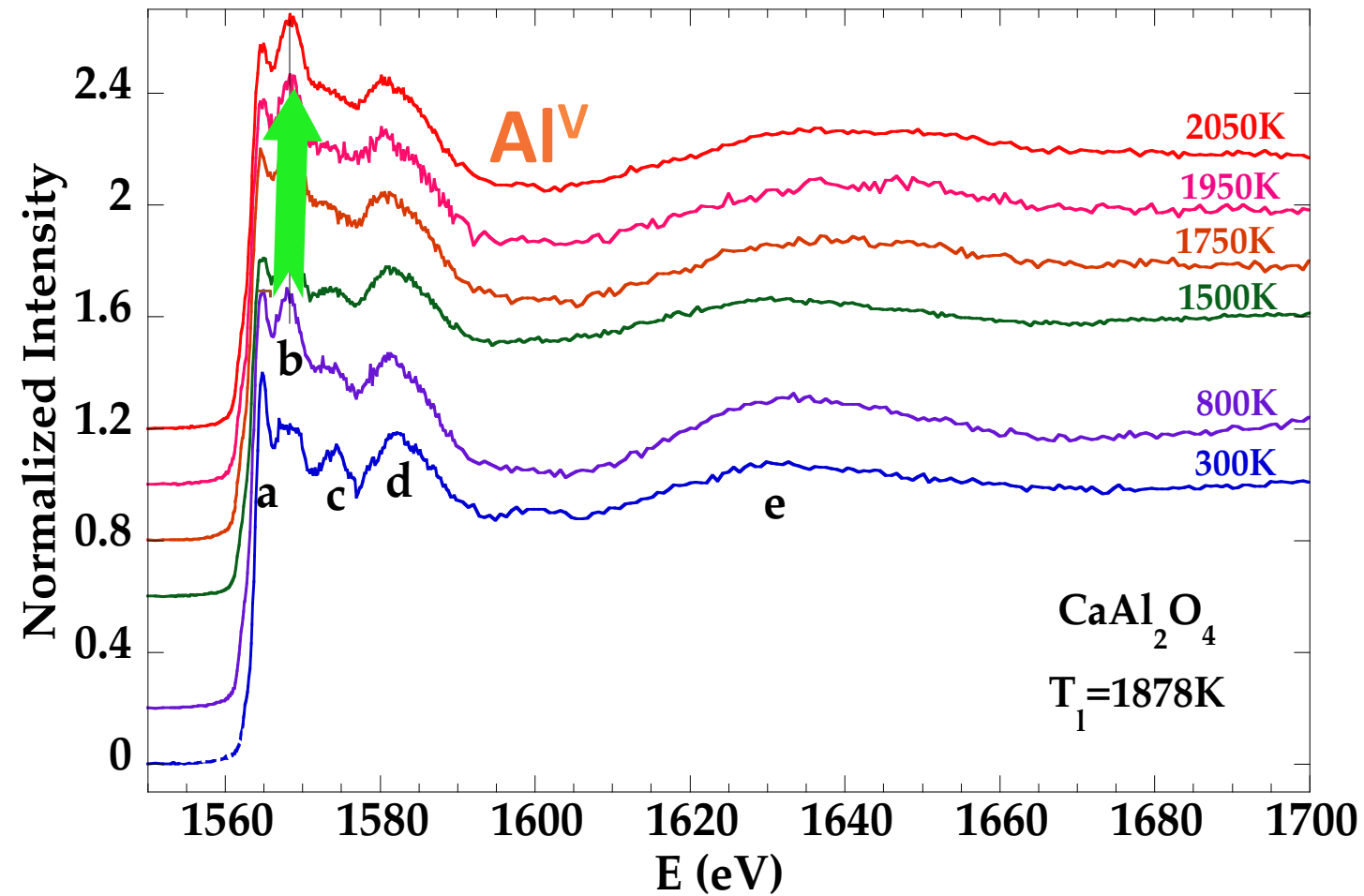


$$\Delta T_g = T_{g_{\text{perAl}}} - T_{g_{\text{tecto}}}$$


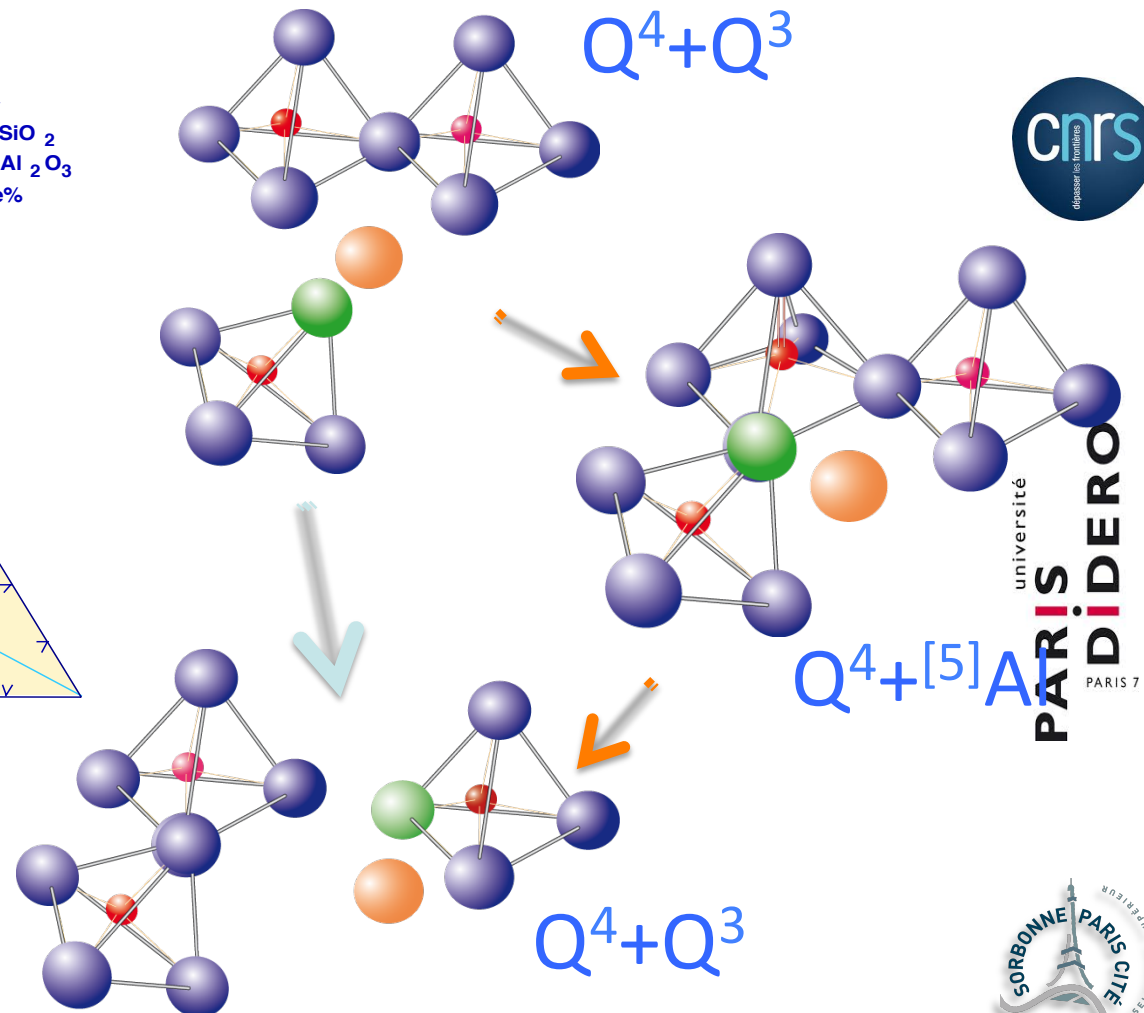
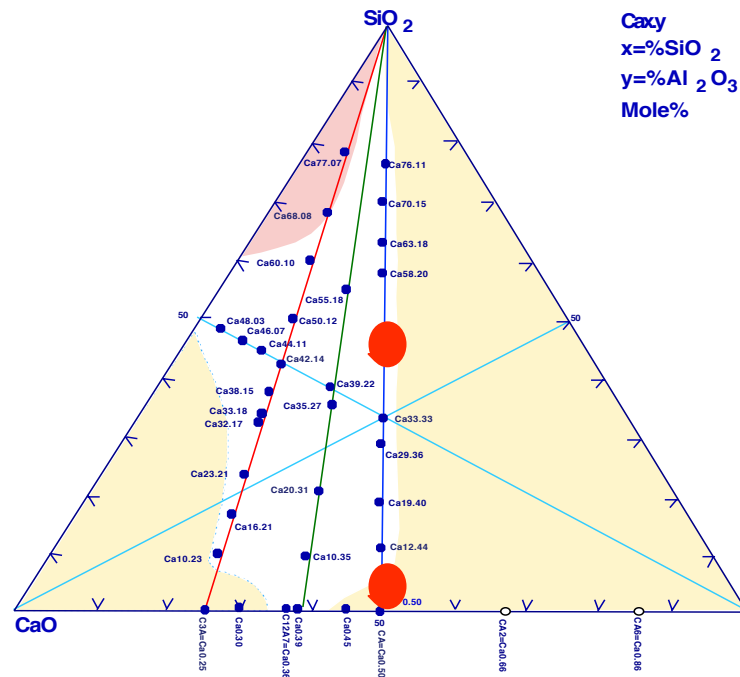
⇒ ^[5]AI can be a strong network former !

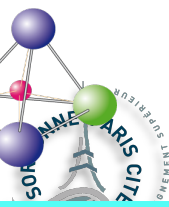
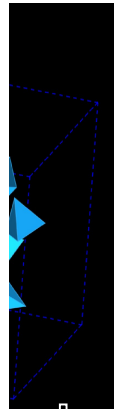
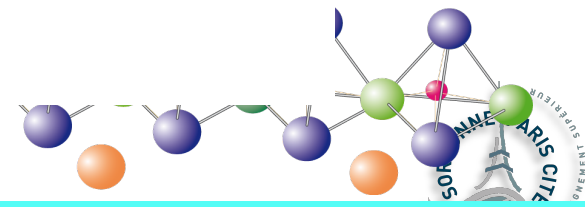
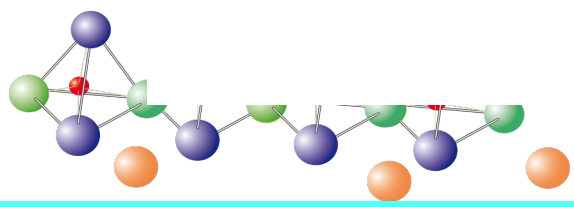
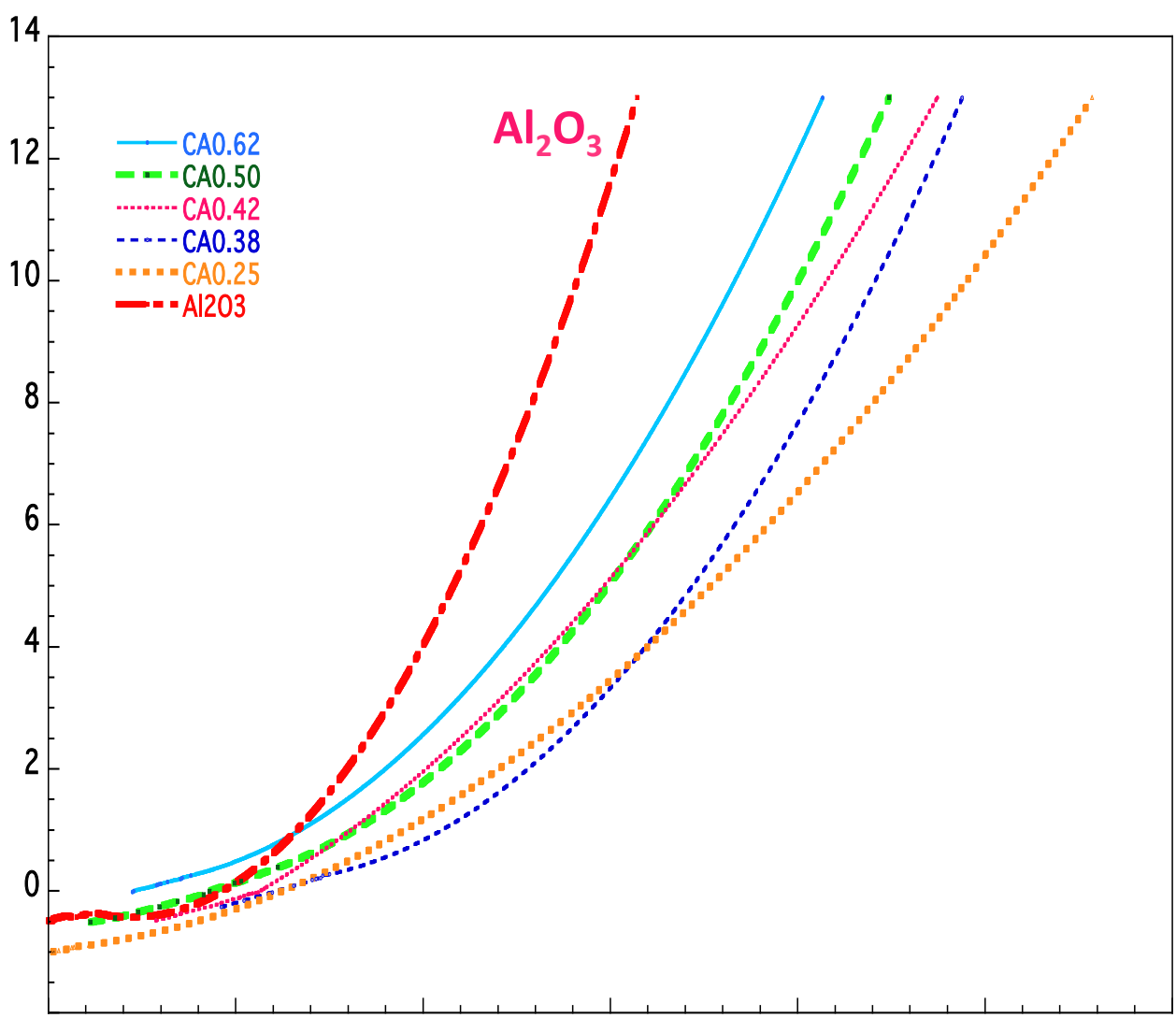
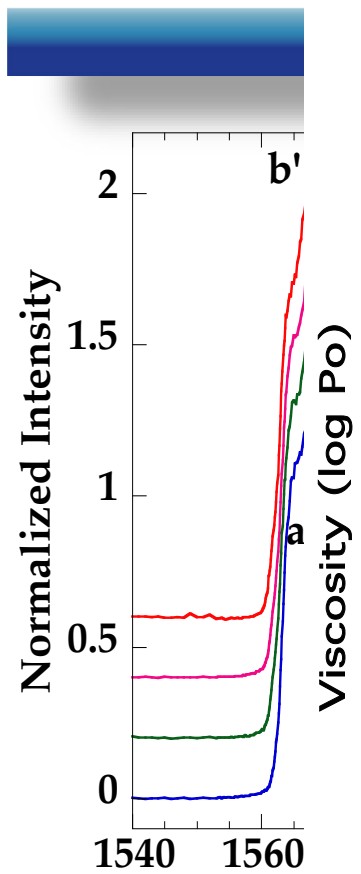
Al in tectosilicate melt ?

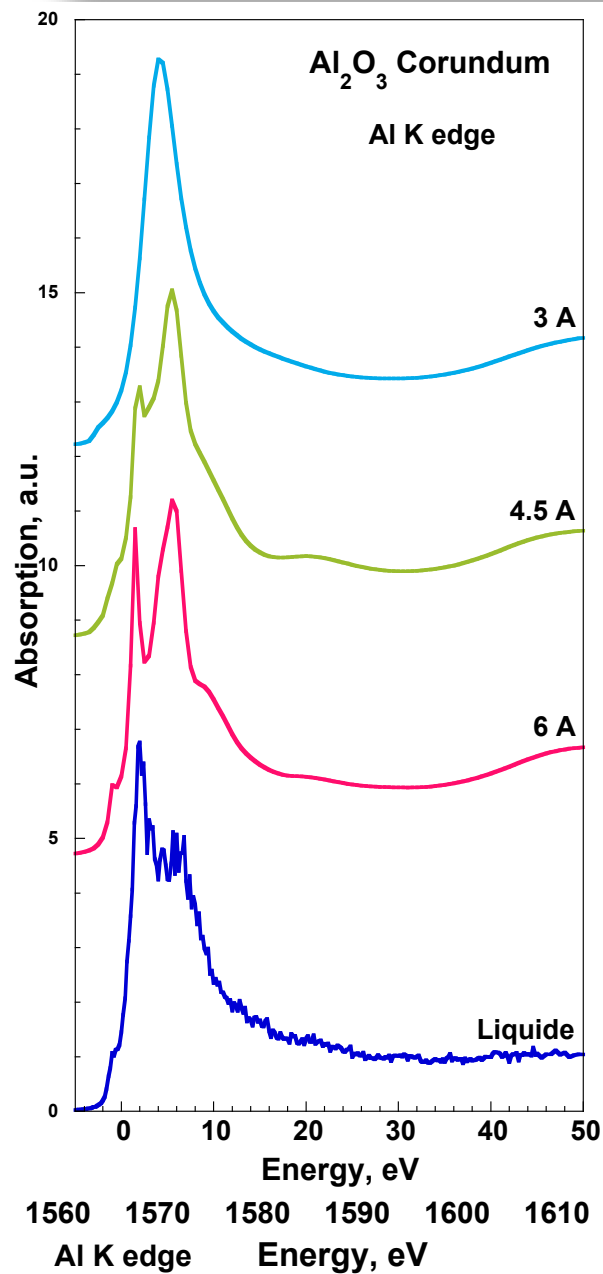
Al K

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Why ^[5]AI?







Al₂O₃ corundum 300K => Al^{VI}

Similar to
NMR-HT
(Florian et al., 1996)

Liquid Al₂O₃

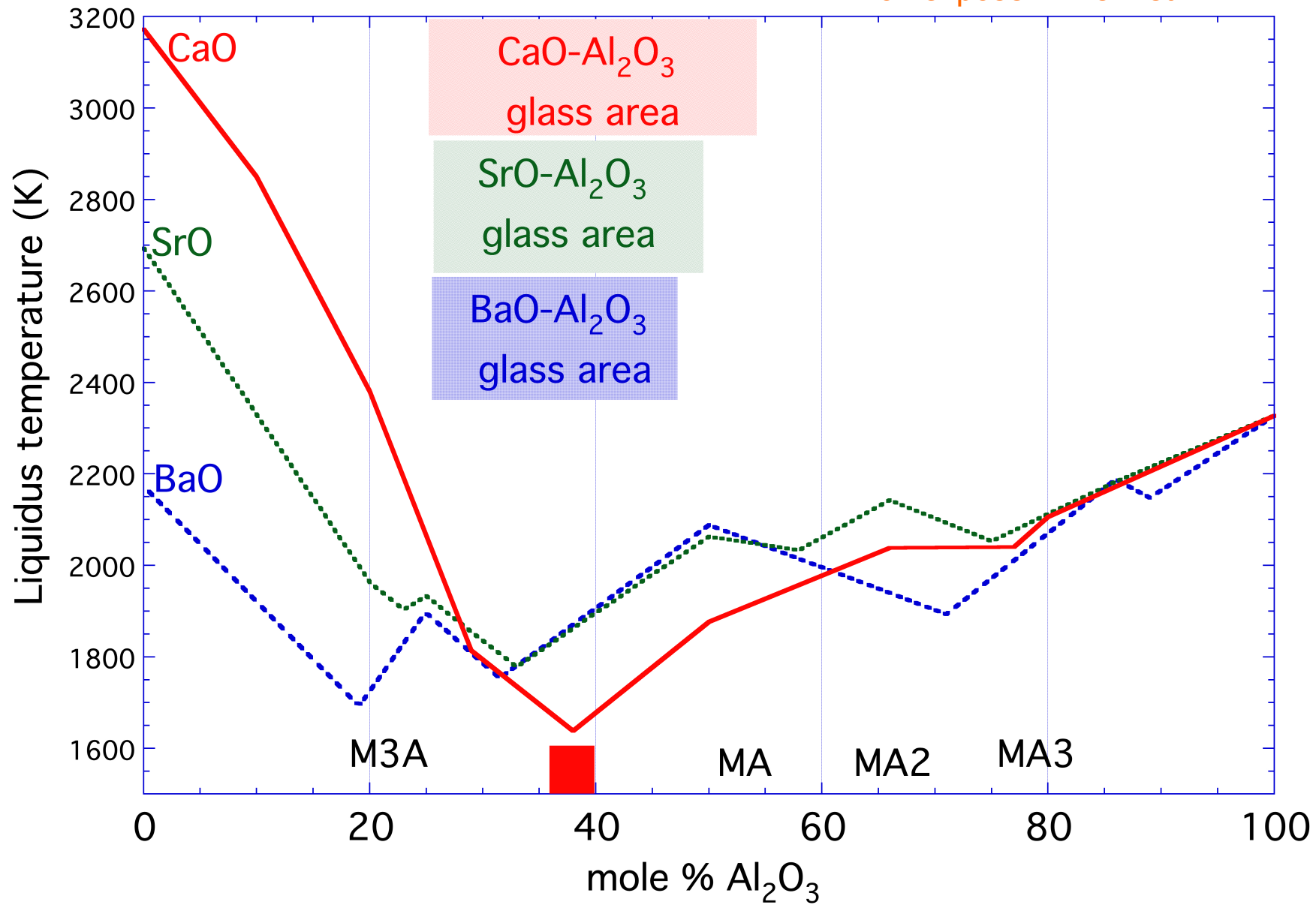
9.4 T
2 lasers
67 mg
2320°C
Experimental
(blue) &
simulation (red)

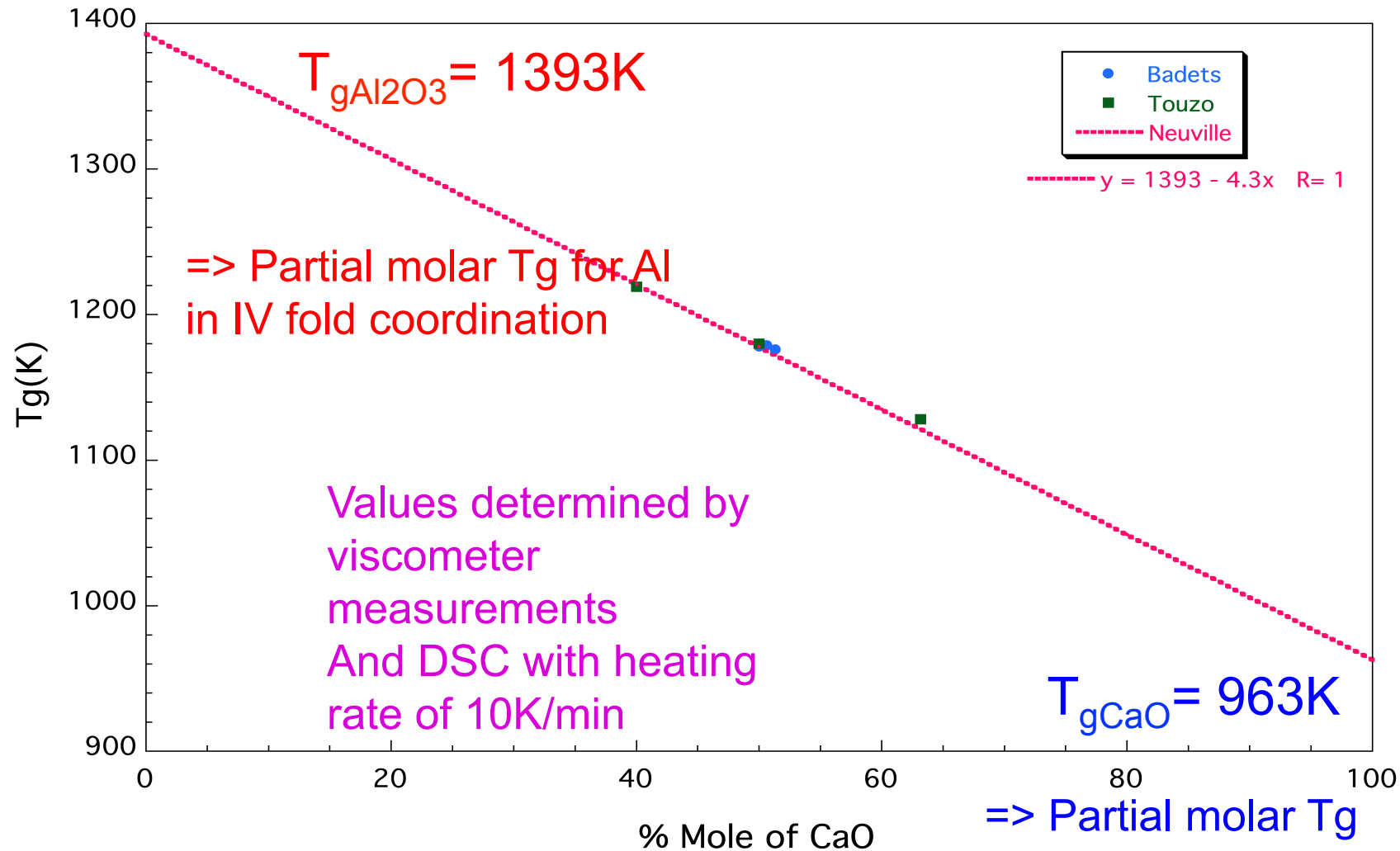
85 80 75 70 65 60 55 50 45 40 35 30
(ppm from Al(NO₃)₃ 1M)

L
S
C
I
T
Y

AS
C
I
T
Y

Voir exposé L. Hennet

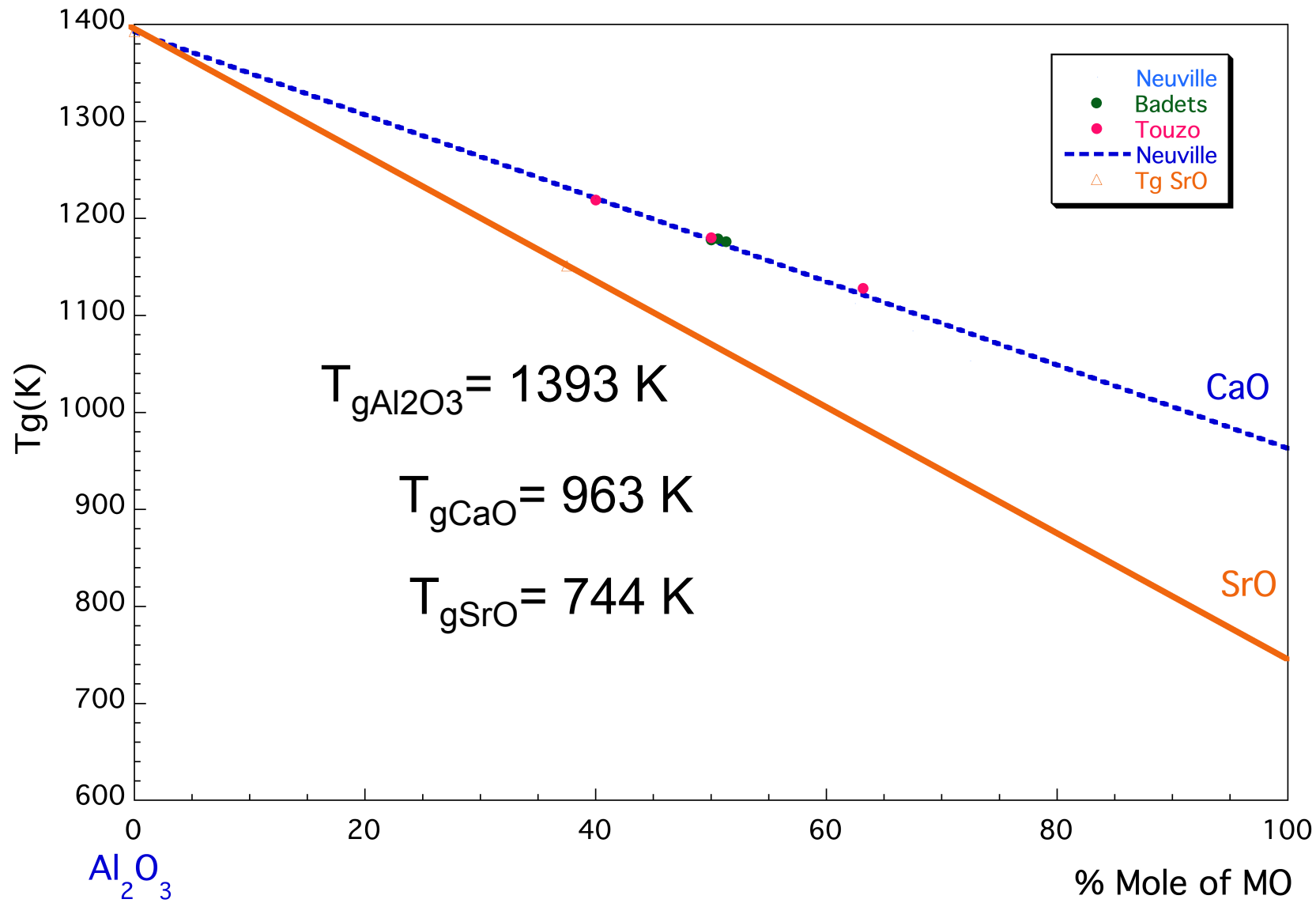




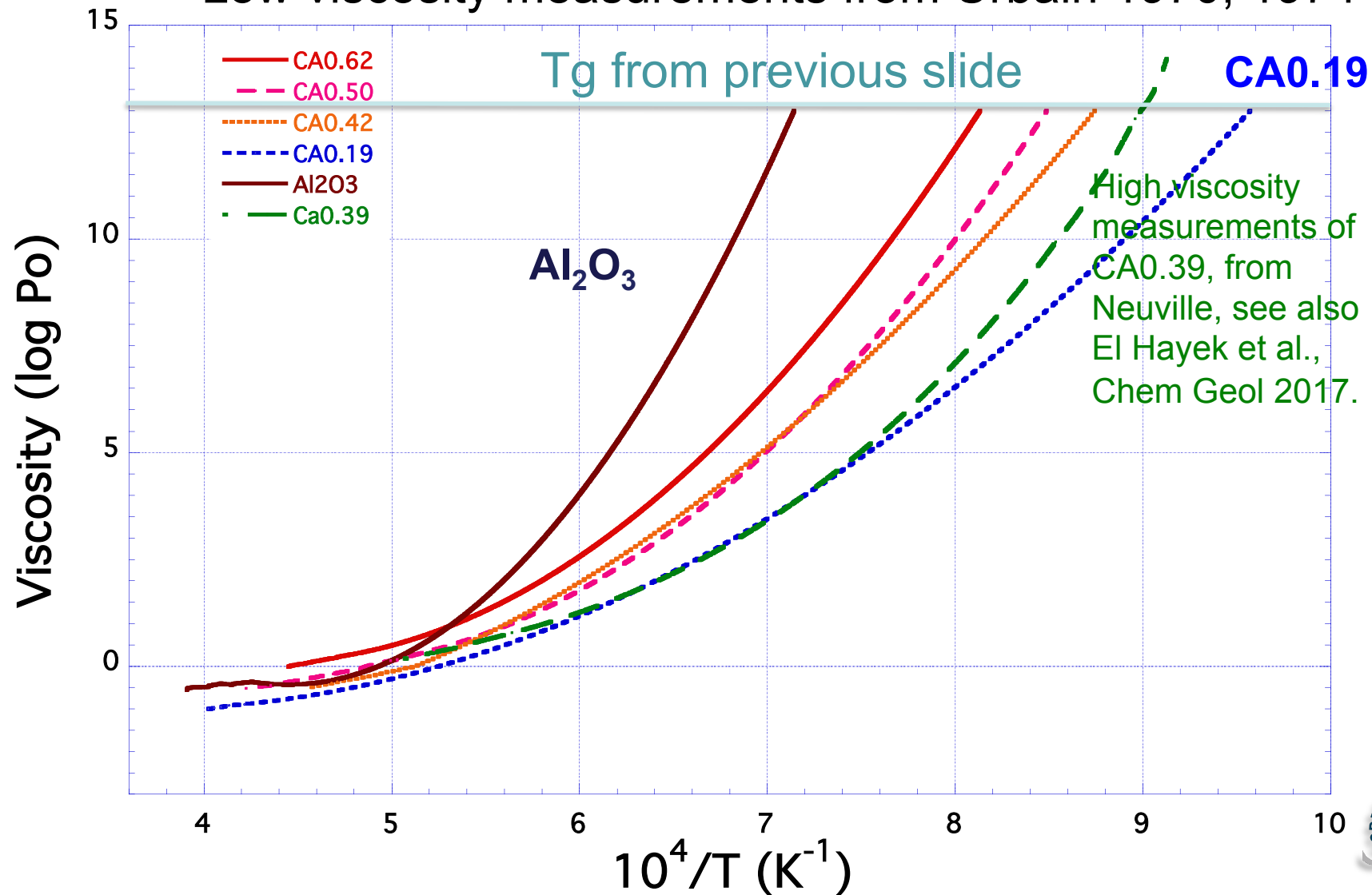
Determination T_{gi} (partial molar)

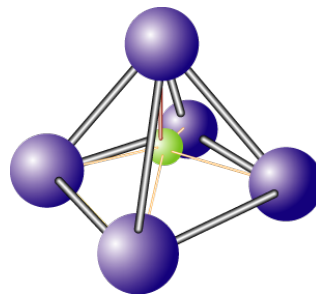
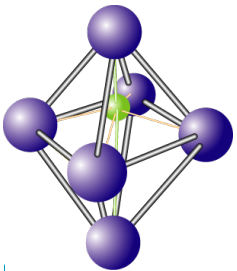
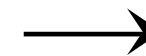
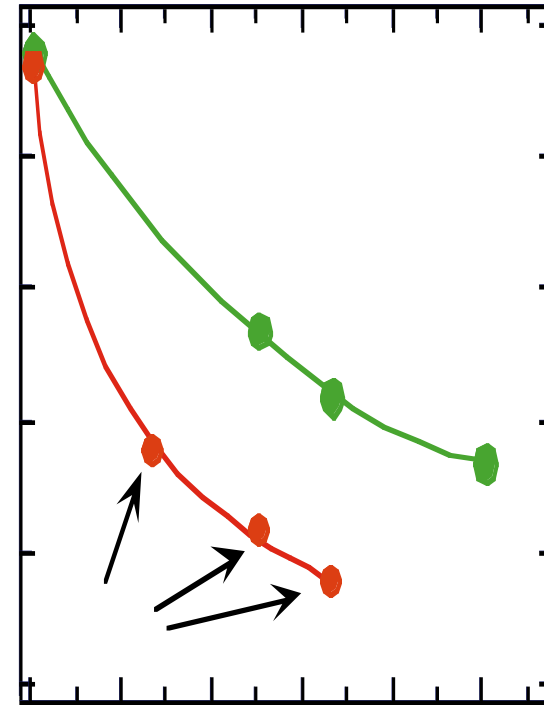
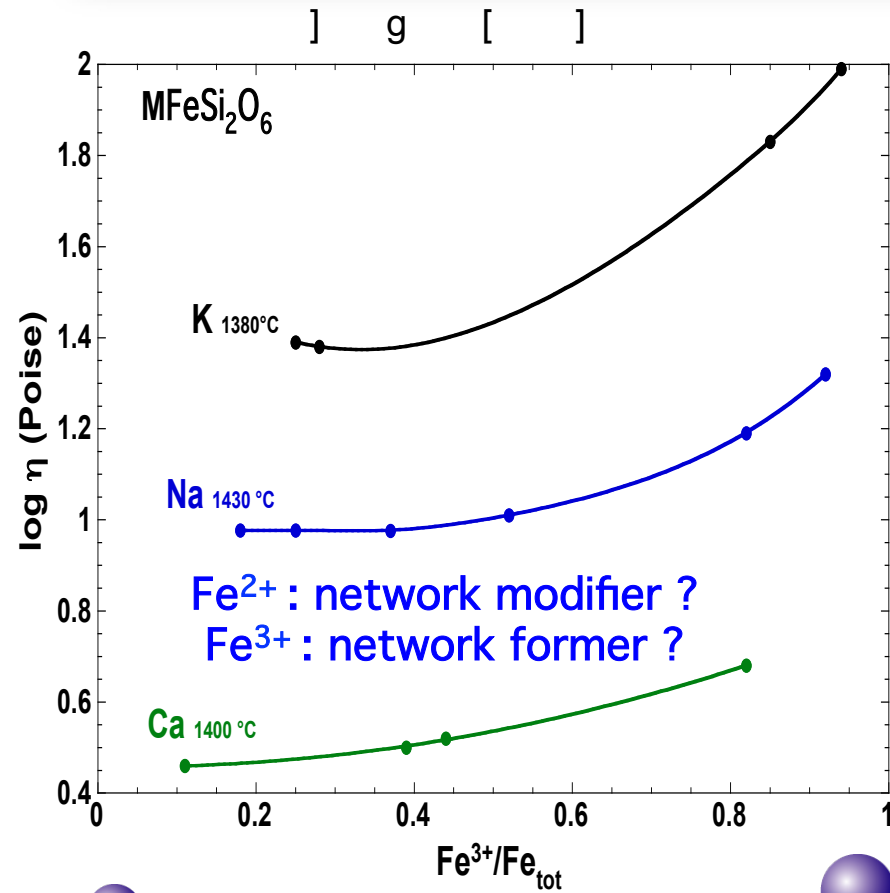
$\Rightarrow$ Partial molar T_g
for Ca in VI fold
coordination

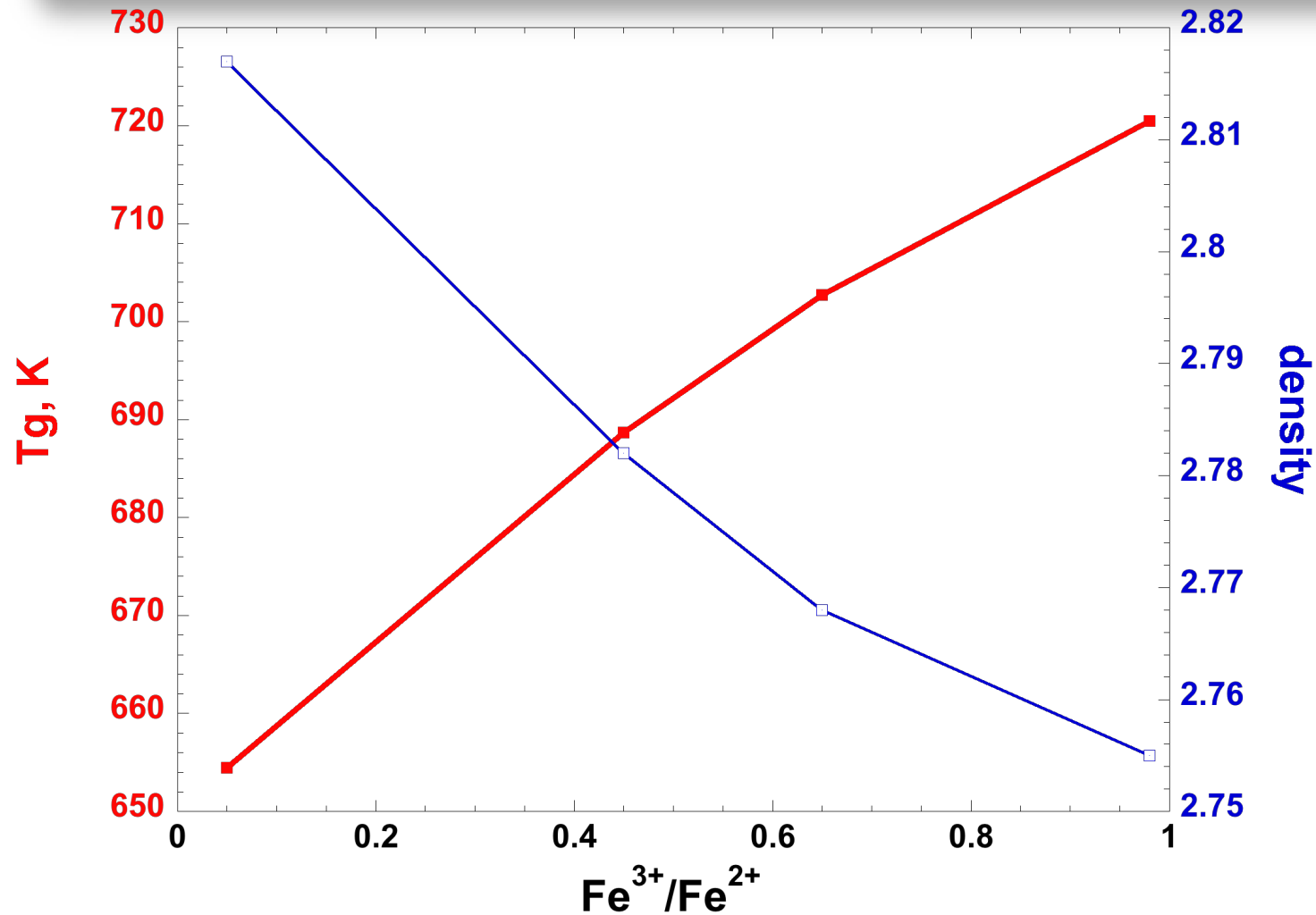
Voir exposé L. Hennet



Low viscosity measurements from Urbain 1970, 1974





$\text{Fe}^{2+}/\text{Fe}^{3+}$ 

Fe^{2+} : network modifier
 Fe^{3+} : network former

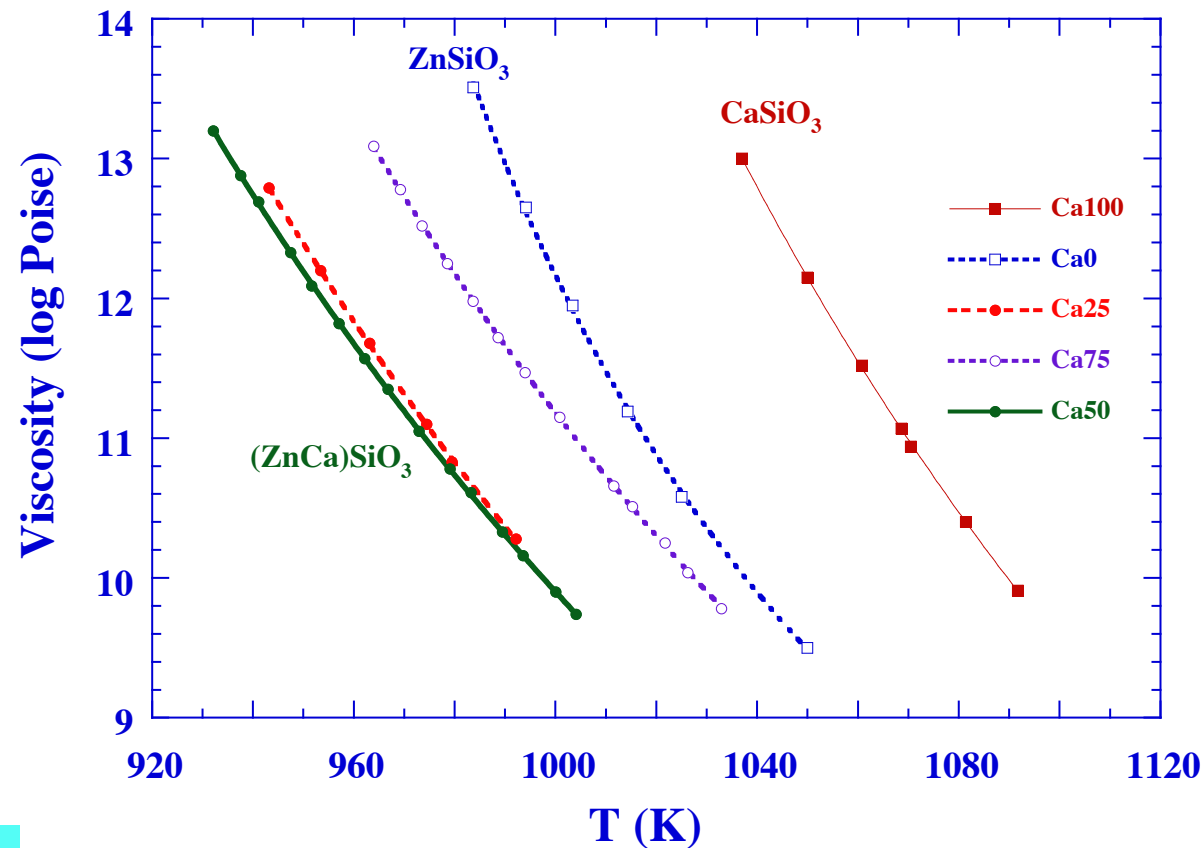
Ca/Mg mixing => ideal mixing



Other systems ?

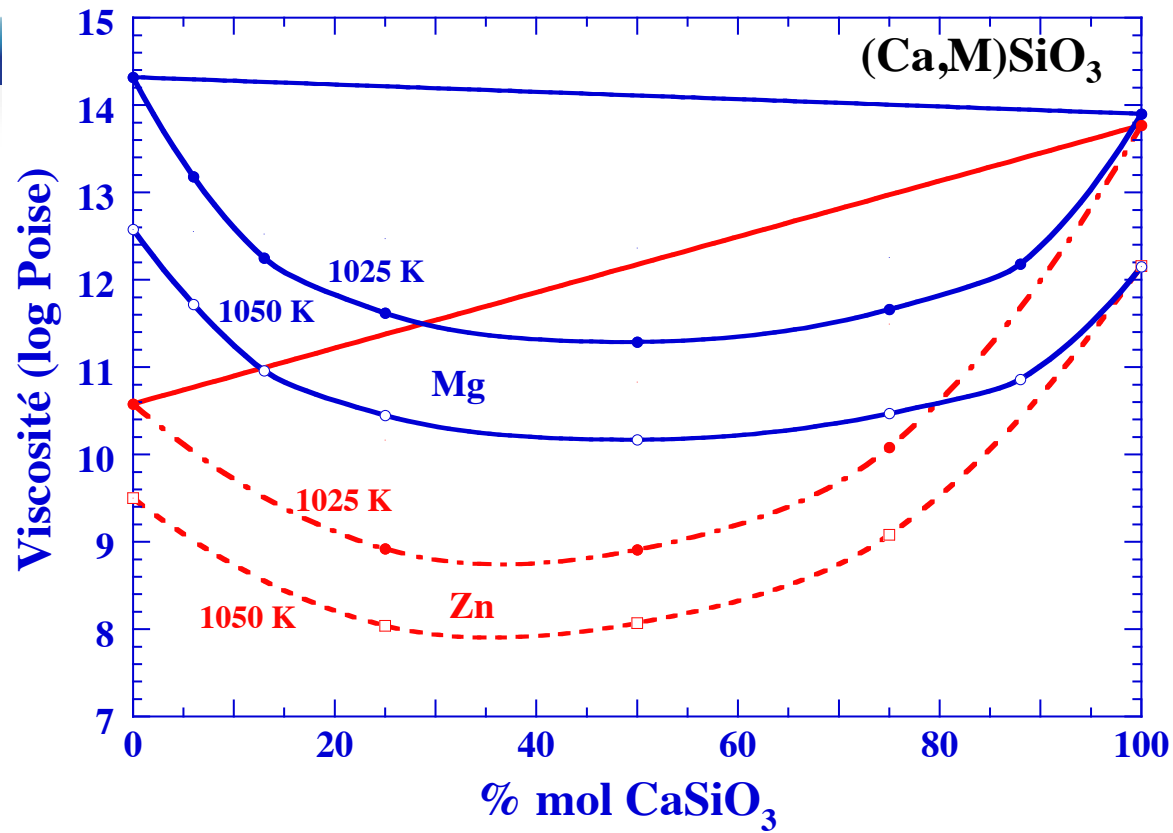
Ca/Zn Substitution?

$(\text{CaZn})\text{SiO}_3$

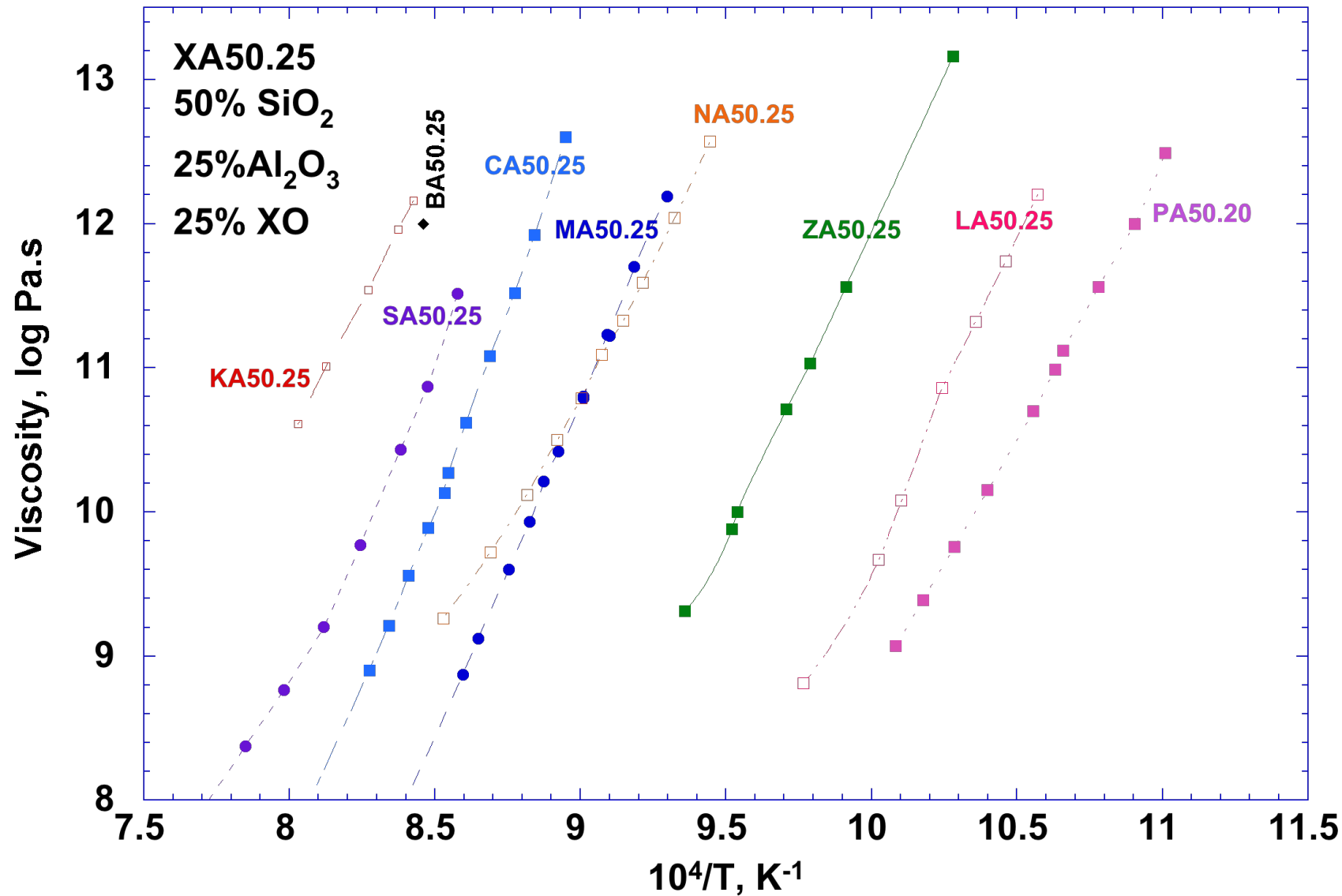


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Zn and Pb with Al...

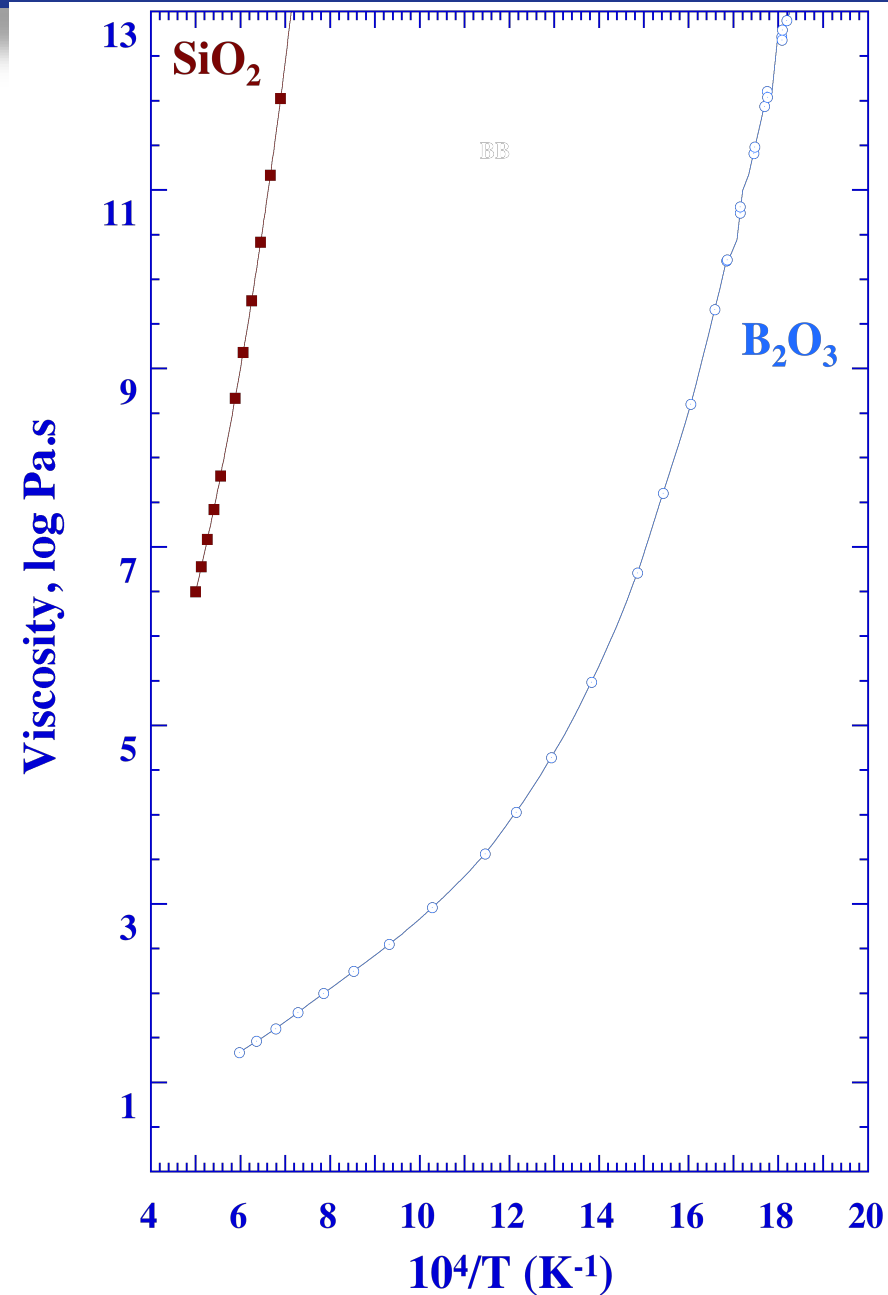


B2O3....



B2O3....

- SiO_4 tetrahedra $\Rightarrow$ strong network former
- BO_3 triangle $\Rightarrow$ soft network former

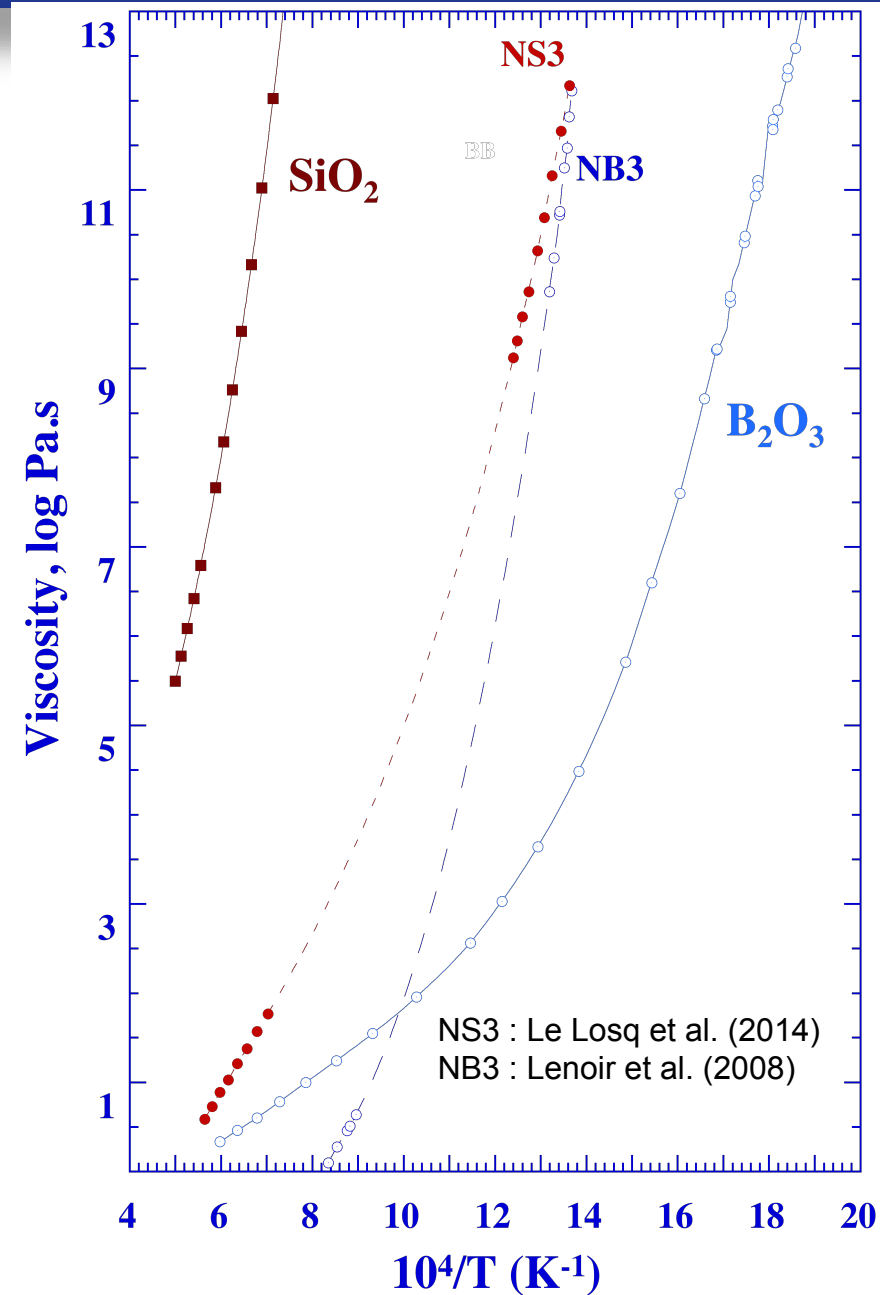


B2O3....

- SiO_4 tetrahedra $\Rightarrow$ strong network former
- BO_3 triangle $\Rightarrow$ soft network former

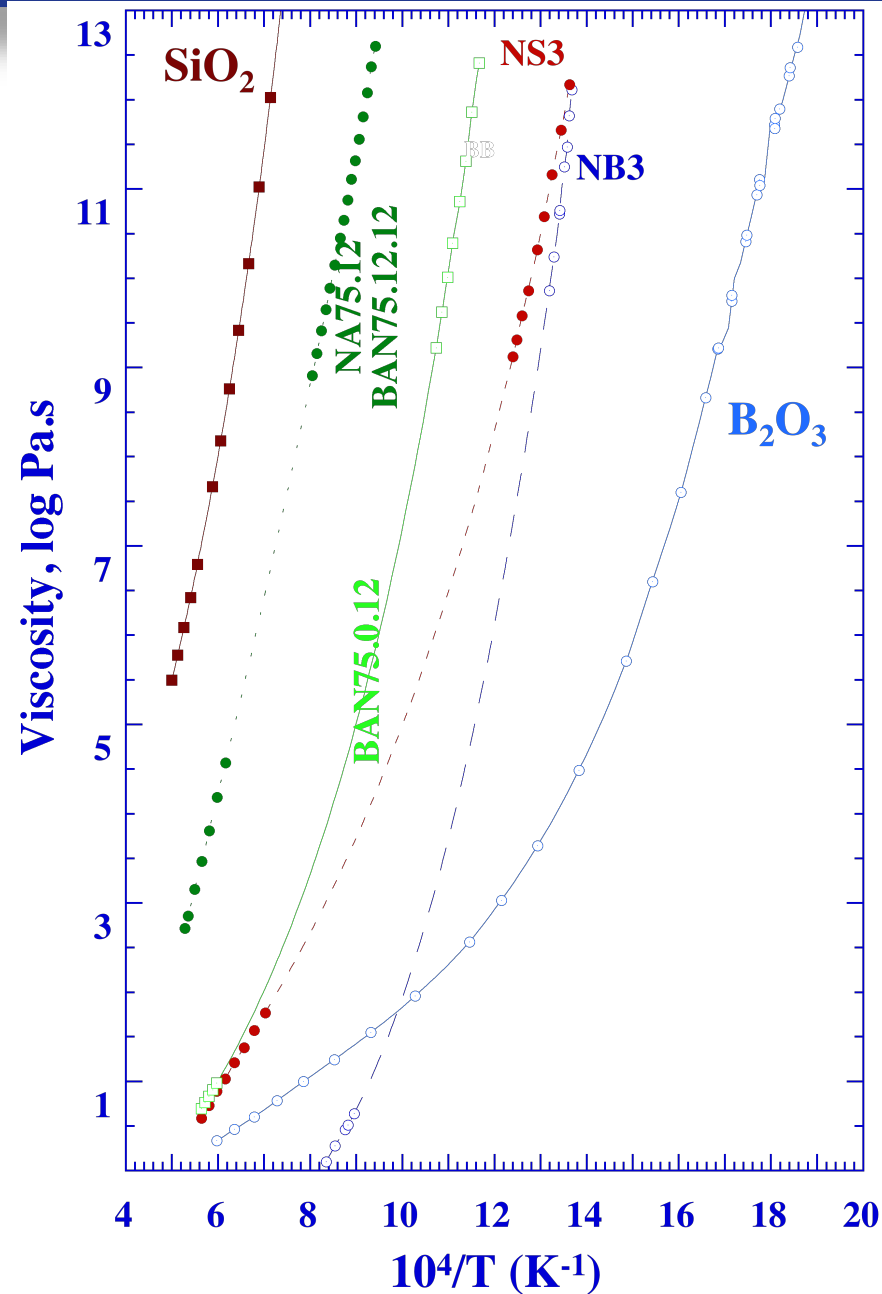
+ Na^+

- BO_4 tetrahedra $\Rightarrow$ network former with M^+



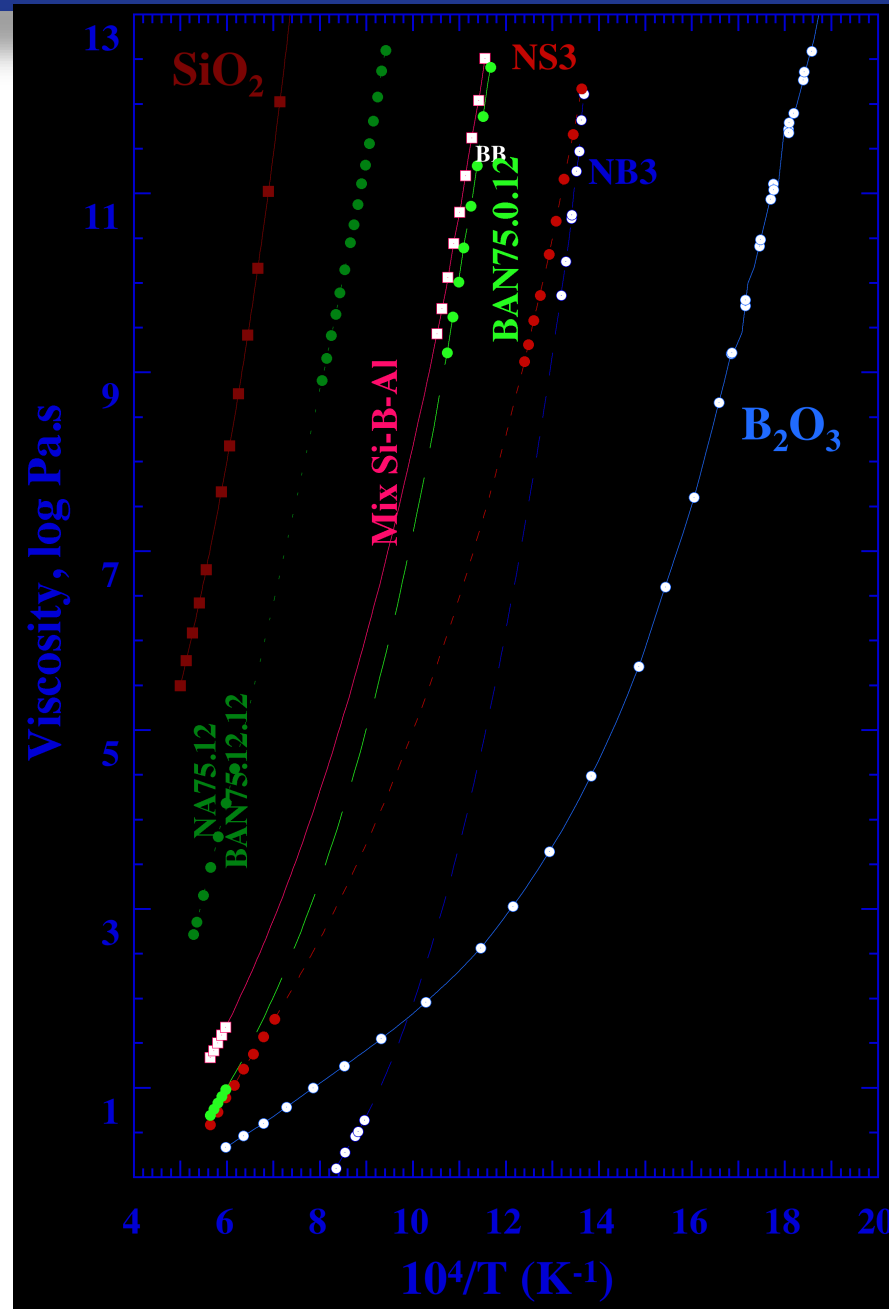
B2O3....

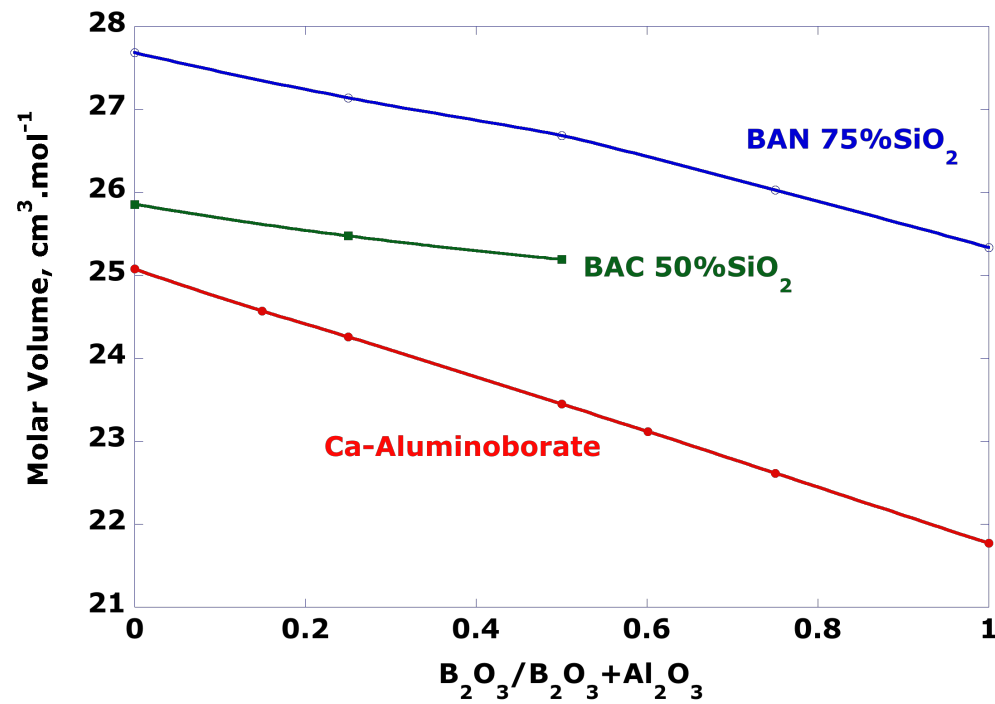
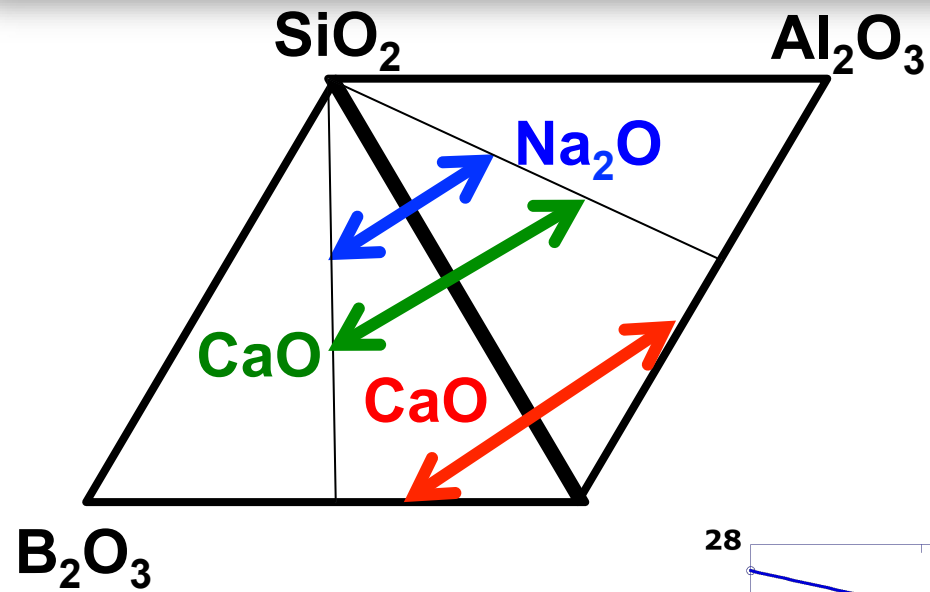
- SiO_4 tetrahedra $\Rightarrow$ strong network former
- BO_3 triangle $\Rightarrow$ soft network former
- BO_4 tetrahedra $\Rightarrow$ network former with M^+
- AlO_4 tetrahedra $\Rightarrow$ network former with M^+



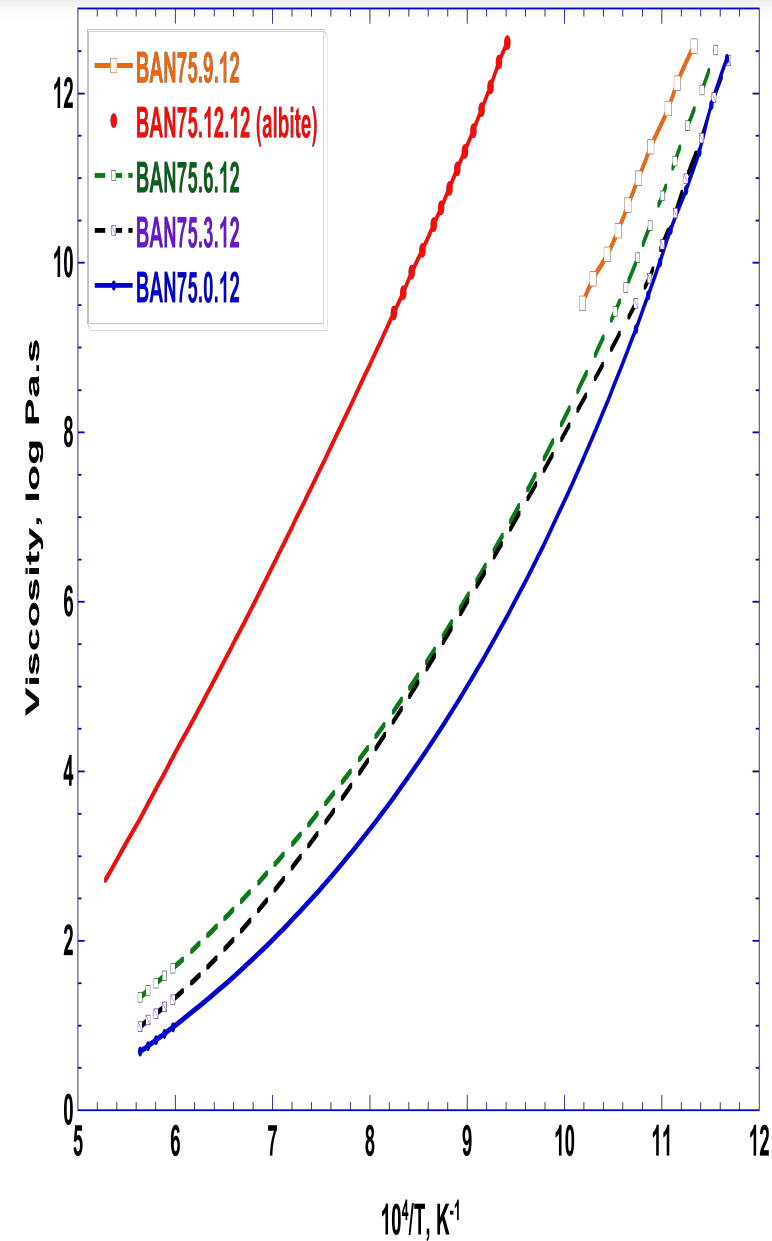
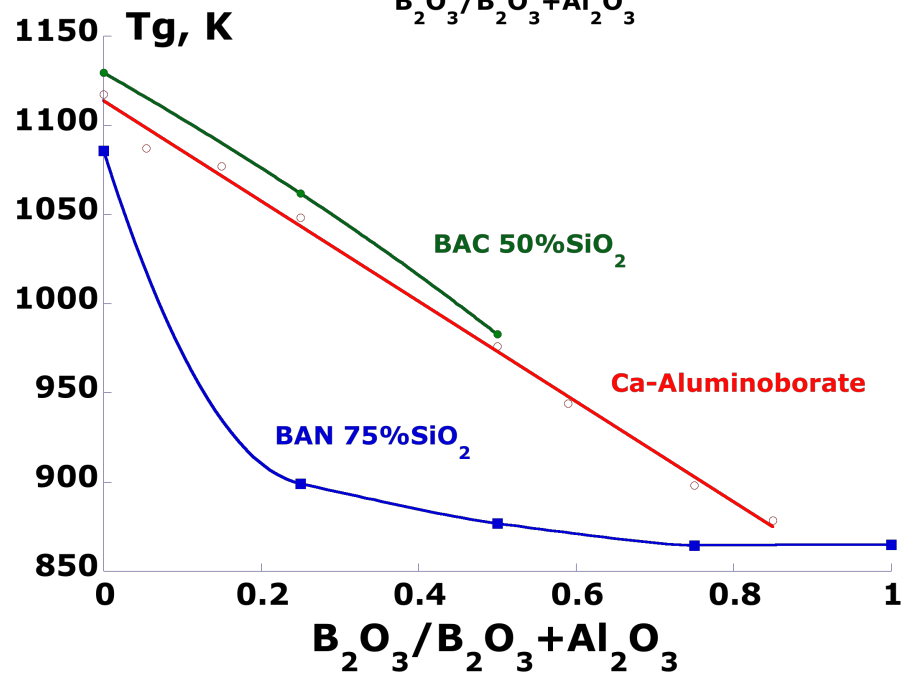
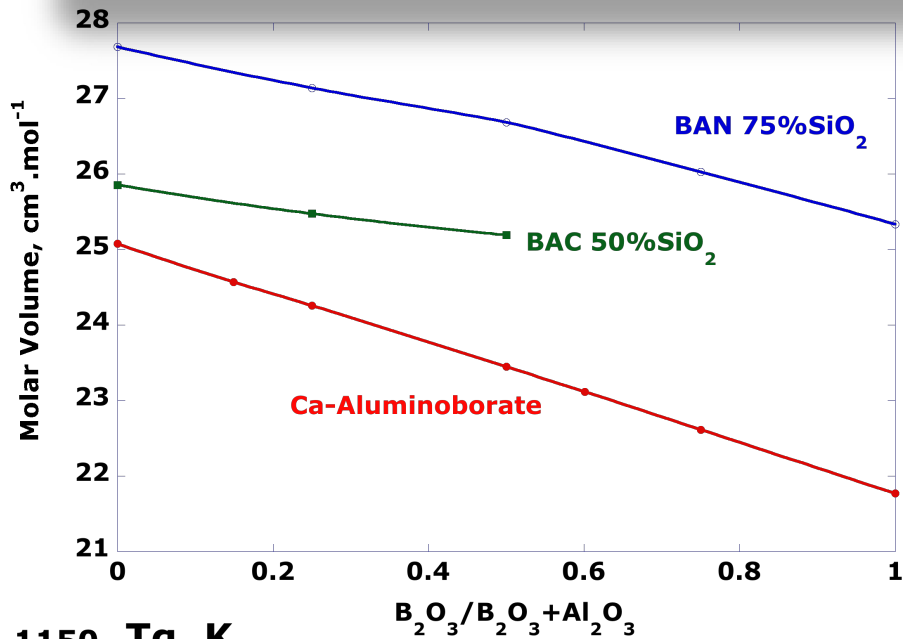
B2O3....

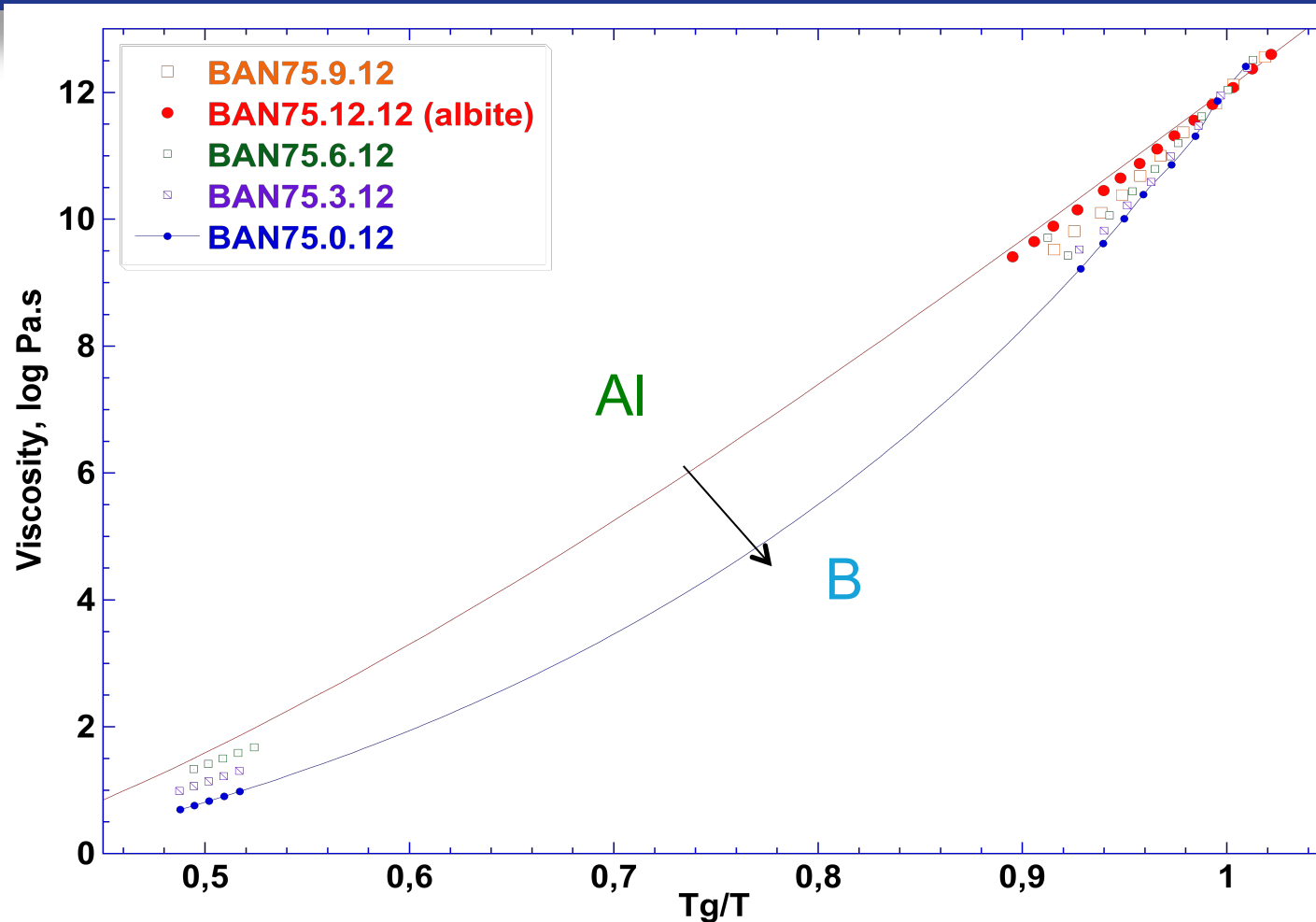
- SiO_4 tetrahedra $\Rightarrow$ strong network former
 - BO_3 triangle $\Rightarrow$ soft network former
 - BO_4 tetrahedra $\Rightarrow$ network former with M^+
 - AlO_4 tetrahedra $\Rightarrow$ network former with M^+
- SiO_4
 - AlO_4 tetrahedra,
 - BO_4 tetrahedra,
 - BO_3 triangle
- $\Rightarrow$ and only one M^+



B₂O₃....

B2O3.... Vm, Viscosity, Tg

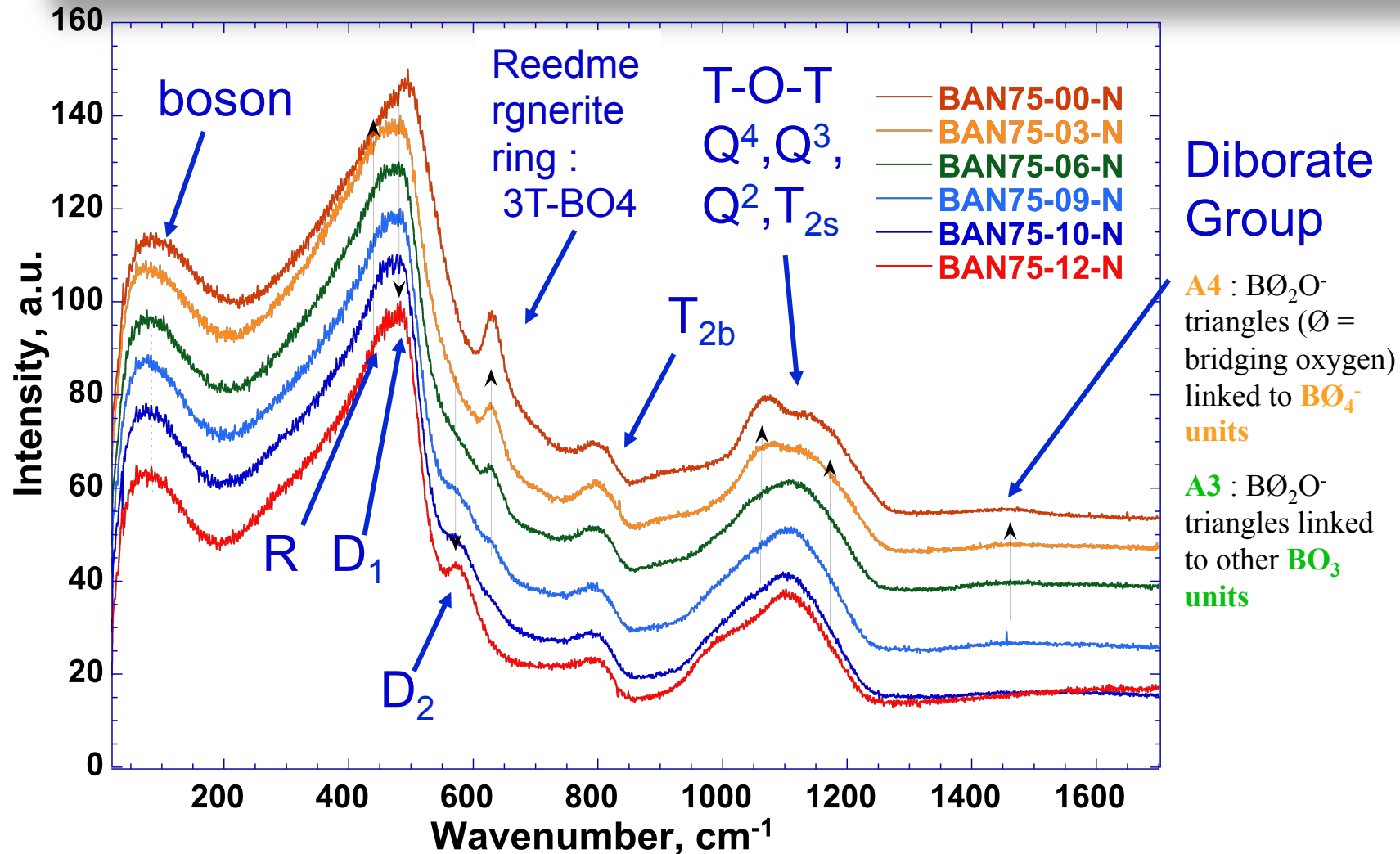




Very strong decrease of V_m , viscosity, T_g and fragility increases with Al/B substitution and using link between viscosity and configurational entropy :

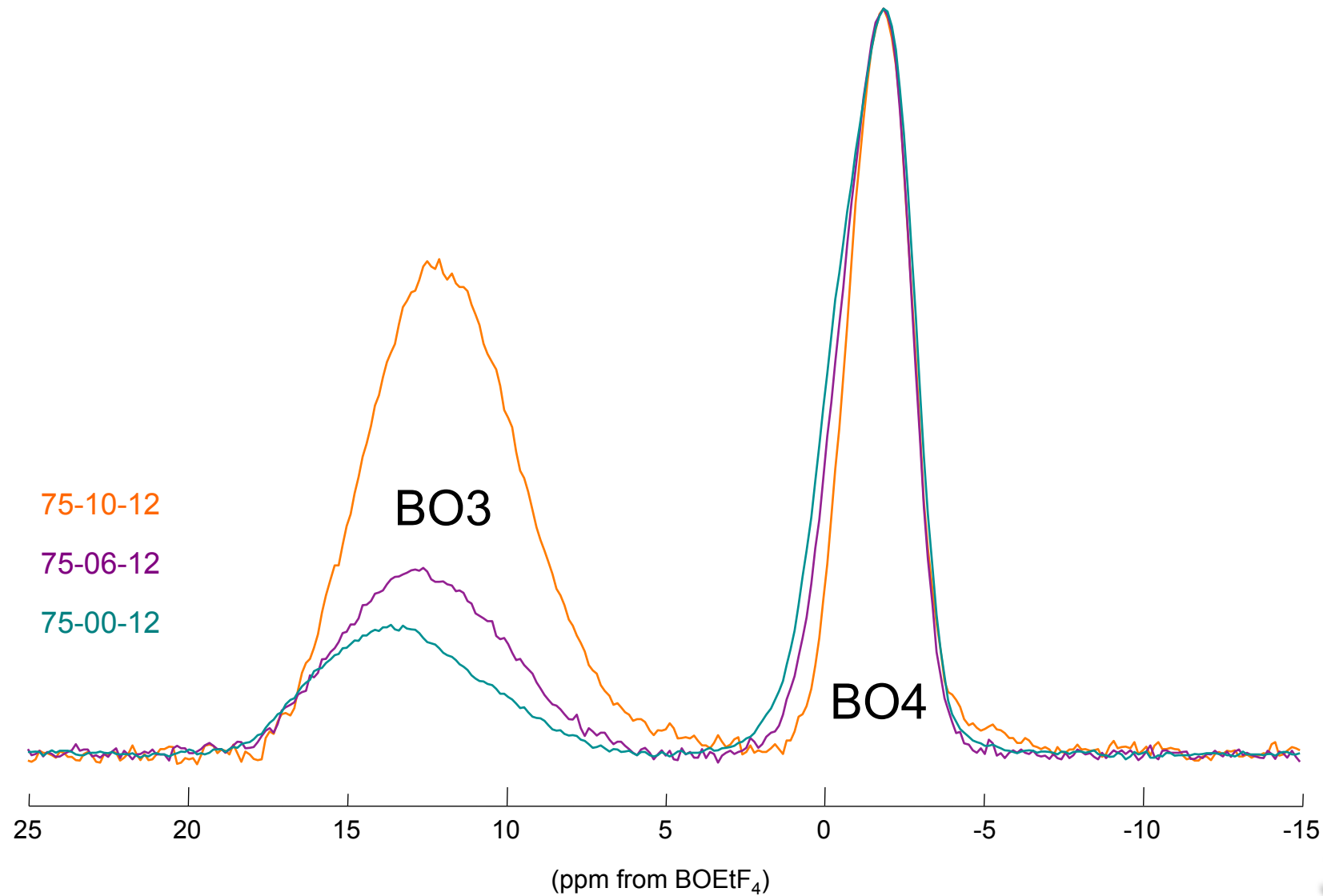
$\text{Log } \eta = A_e + B_e T / S^{\text{conf}}(T)$, we can calculate S^{conf} which goes from 8 J/mol.K for up to 14.5 J/mol.K with Al/B substitution

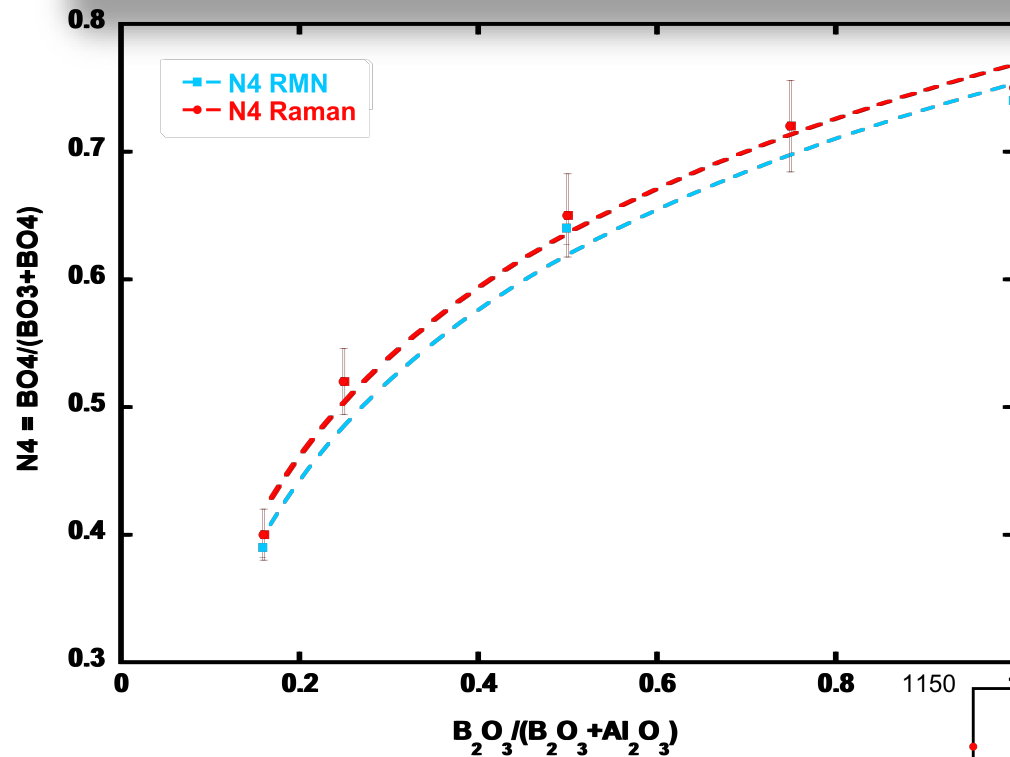
Raman Spectrometry



- With Al/B substitution => New band appear at 615, 920, 1200, 1450 cm⁻¹
- Important change in the T-O-T band vibrations
- No change in the boson peak and D1 and D2 decrease

^{11}B MAS (Echo) NMR in BAN

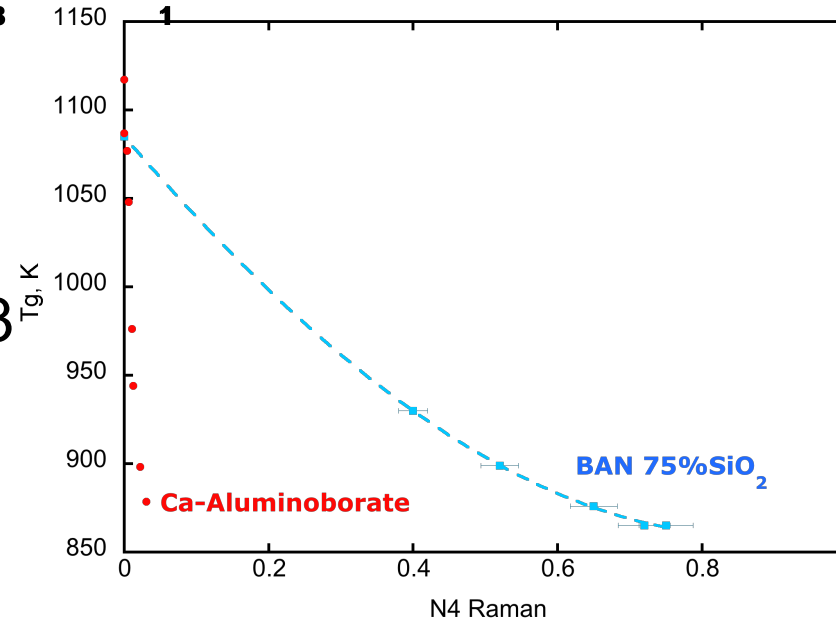




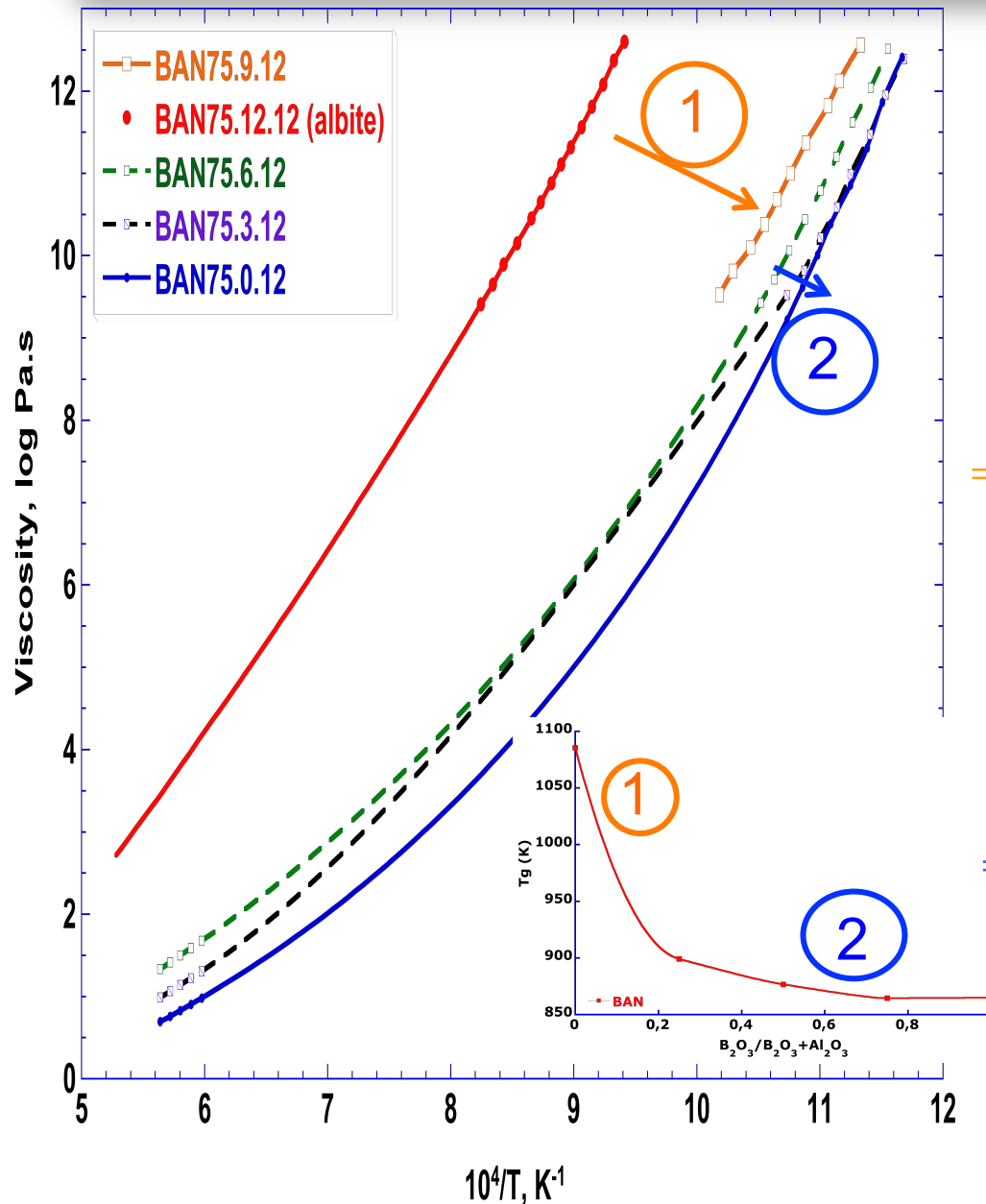
$$N4 = BO4/(BO4+BO3)$$

Similar evolution than
Geisinger et al., GCA 1988

Huge decrease of T_g with BO_3



Assumption for Al/B substitution mechanism ?



But with Al/B substitution BO_3 decreases and $2BO_4$ are present

$\Rightarrow BO_3$ decrease with B_2O_3 increases

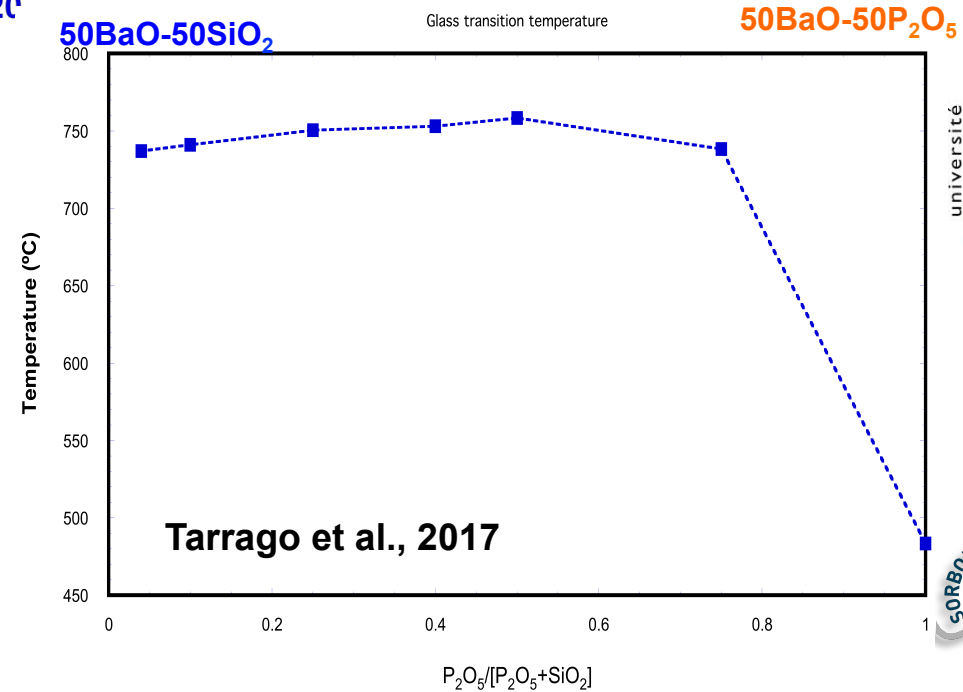
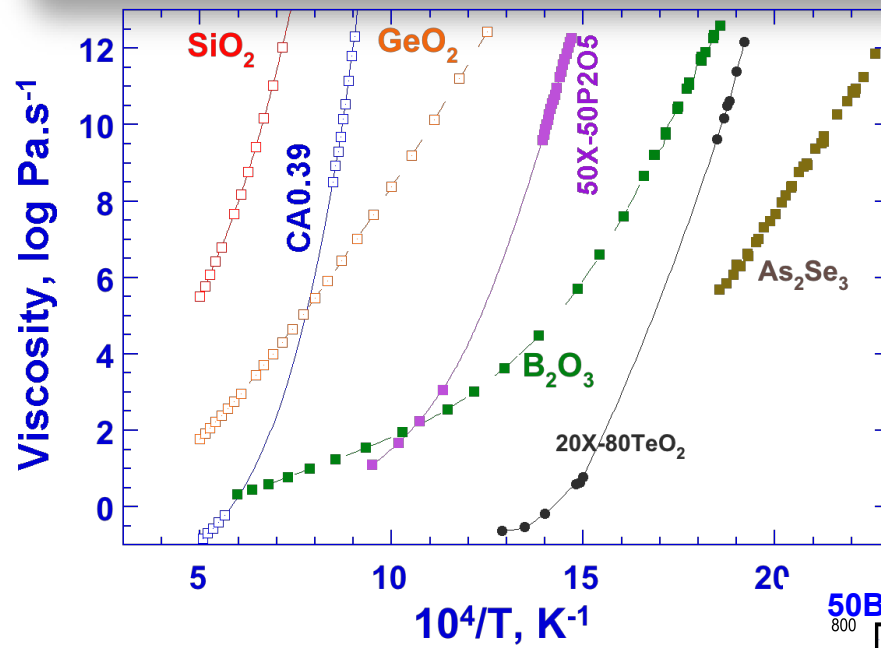
$\Rightarrow$ 1) by substitution Al by B $\Rightarrow$ B is essentially in BO_3 because Al used Na as a charge compensator and BO_3 decreases a lot the viscosity

$\Rightarrow$ 2) With increasing B content $\Rightarrow BO_4$ increases and used Na as a charge compensator $\Rightarrow$ viscosity decreases slowly



P2O5....

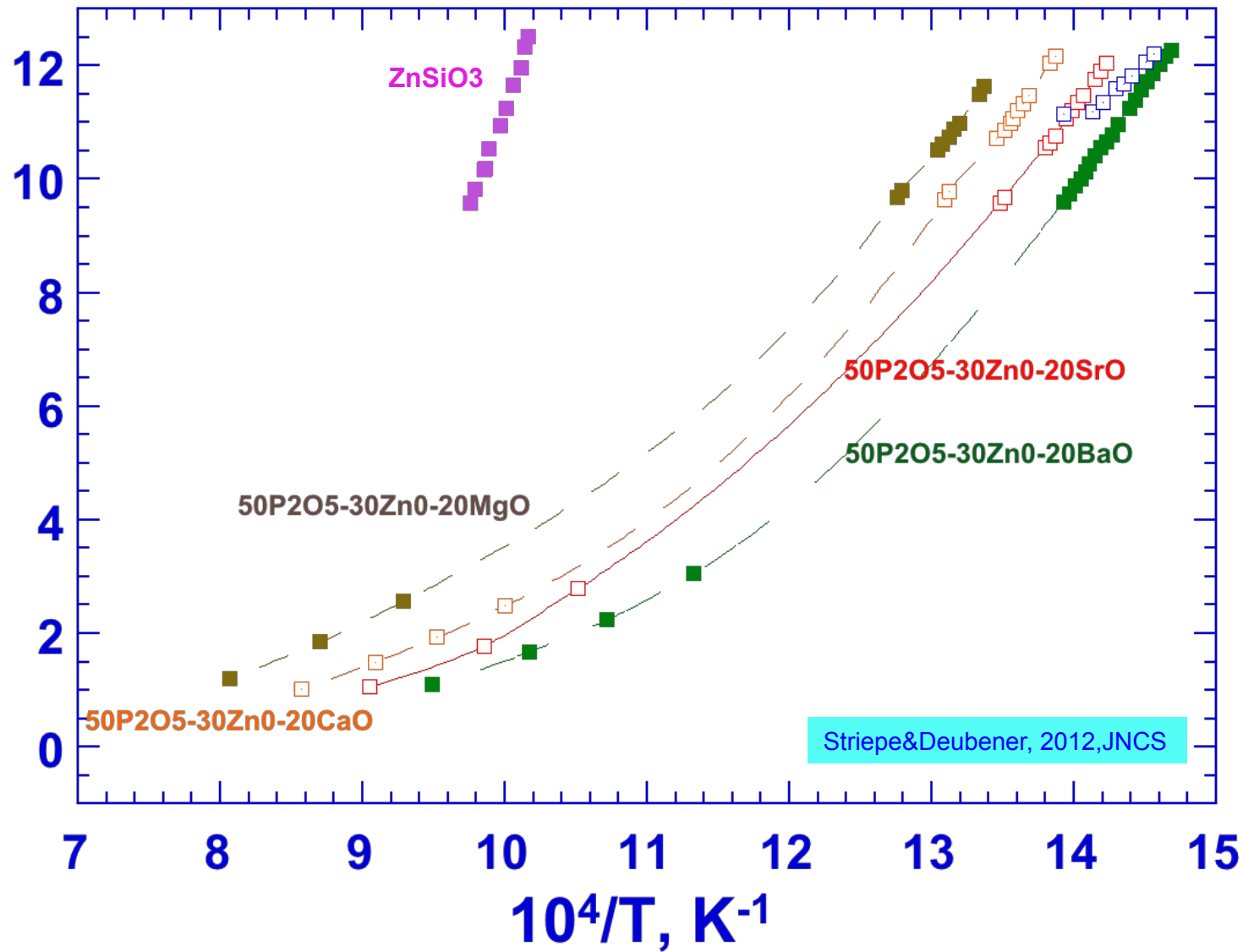


SiO₂-P₂O₅

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Viscosity, log Pa.s⁻¹



TeO_2



Références



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Impact of tellurite-based glass structure on Raman gain

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ABSTRACT

Raman gain efficiency and structure of tellurite glasses in the TeO₂–TaO_{5/2}–ZnO system have been investigated by Raman and IR spectroscopies. It has been found that replacement of TaO_{5/2} by ZnO does not significantly modify the respective proportion of TeO₄, TeO₃ and TeO₃₊₁ entities but induces an important decrease of the TeO₄ structural units Raman cross section. Change of the glass structure at the middle range length scale is proposed to be at the origin of the large decrease of the Raman cross section of the TeO₄ units.

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Table 1

T_g glass transition temperature, T_x onset of the crystallization peak, T_p maximum of the crystallization peak, ΔT glass stability and volumetric weight.

Glass composition	T_g (± 2 °C)	T_x (± 2 °C)	T_p (± 2 °C)	$\Delta T = T_x - T_g$	Volumetric weight ($\pm 0.3\%$ g/cm ³)
90TeO ₂ –10TaO _{5/2}	351	430	465	79	5.81
85TeO ₂ –15TaO _{5/2}	370	489	516	119	5.91
80TeO ₂ –20TaO _{5/2}	387	502	505	115	5.99
80TeO ₂ –15TaO _{5/2} –5ZnO	360	496	537	136	5.89
80TeO ₂ –10TaO _{5/2} –10ZnO	350	478	526	128	5.78
80TeO ₂ –5TaO _{5/2} –15ZnO	335	468	498	133	5.67

80% TeO₂ – 20% ZnO

$T_g = 323^\circ\text{C}$

$T_x > 400^\circ\text{C}$

[I.Savelli 2012]

80% TeO₂ – 20% Na₂O

$T_g = 256^\circ\text{C}$

$T_x = 383^\circ\text{C}$

[A.Santic 2008]

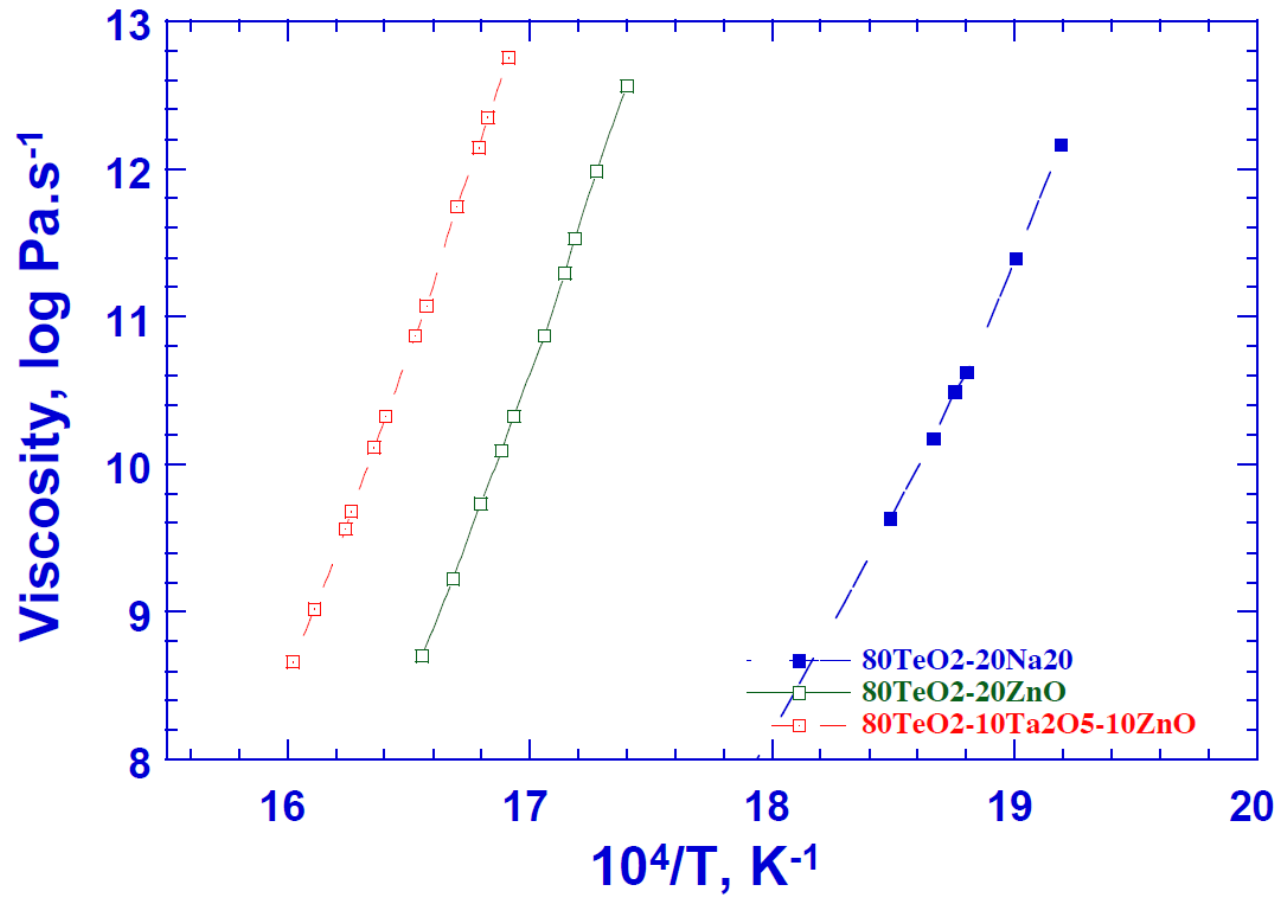


TO

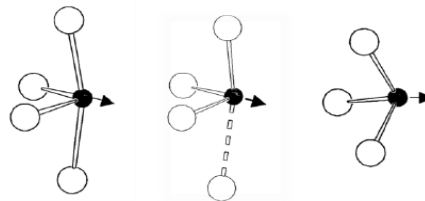
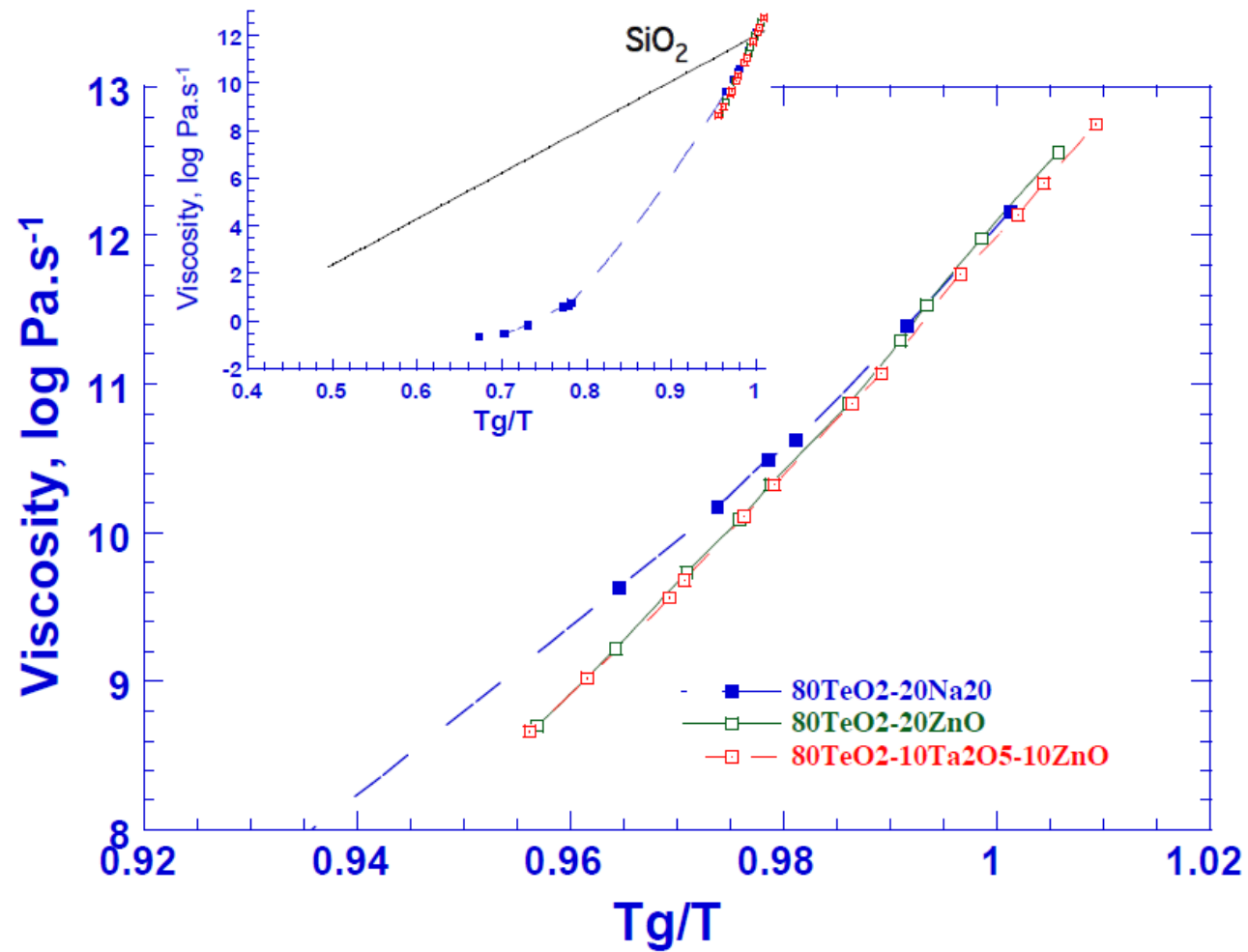


TeO₂....

Viscosity



Fragility



Summarize Oxide glasses



SiO_2 , GeO_2 => strong network former

B_2O_3 (BO_3, BO_4), P_2O_5 , V_2O_5 , TeO_2 => soft network former

Al_2O_3 , Ta_2O_3 : mix... ?

Al^{IV} => network former

Al^{V} => reticulator, need to ensure dynamics at HT

Al^{VI} => generally network modifier... but with P?



Fe_2O_3 => network former
 B_2O_3

As_2Se_3 : similar than

Li_2O , Na_2O , K_2O , MgO , CaO , SrO , BaO , ZnO , FeO
=> network modifier or charge compensator in Si,
Ge, P, B, V systems

PbO : probably network modifier.....

O BO, NBO, free? Other systems ?

