

Structure des verres/liquides boratés et borosilicatés



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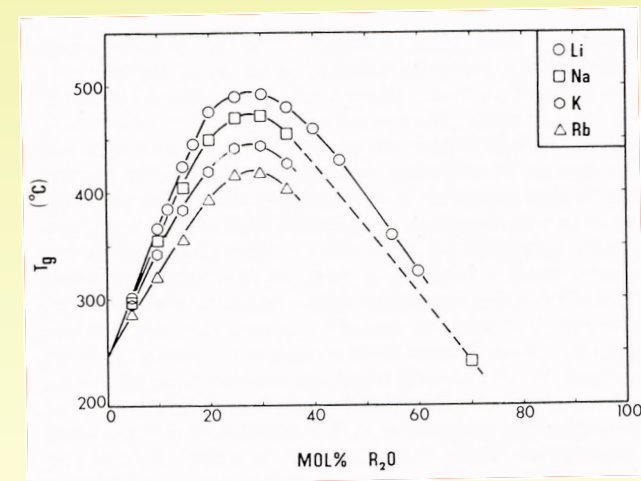
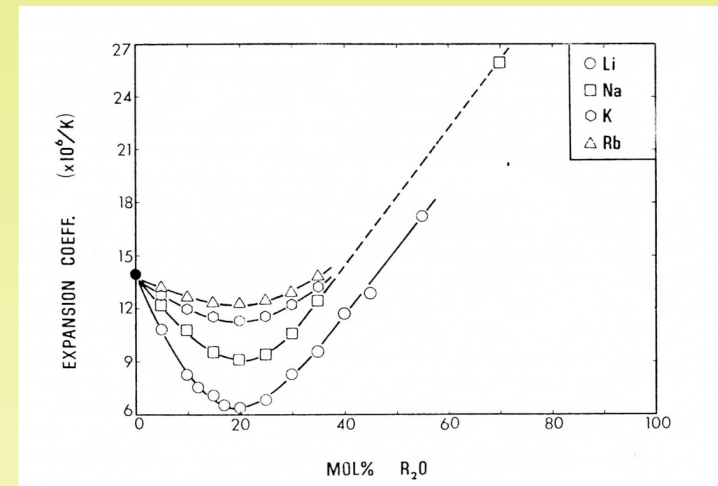
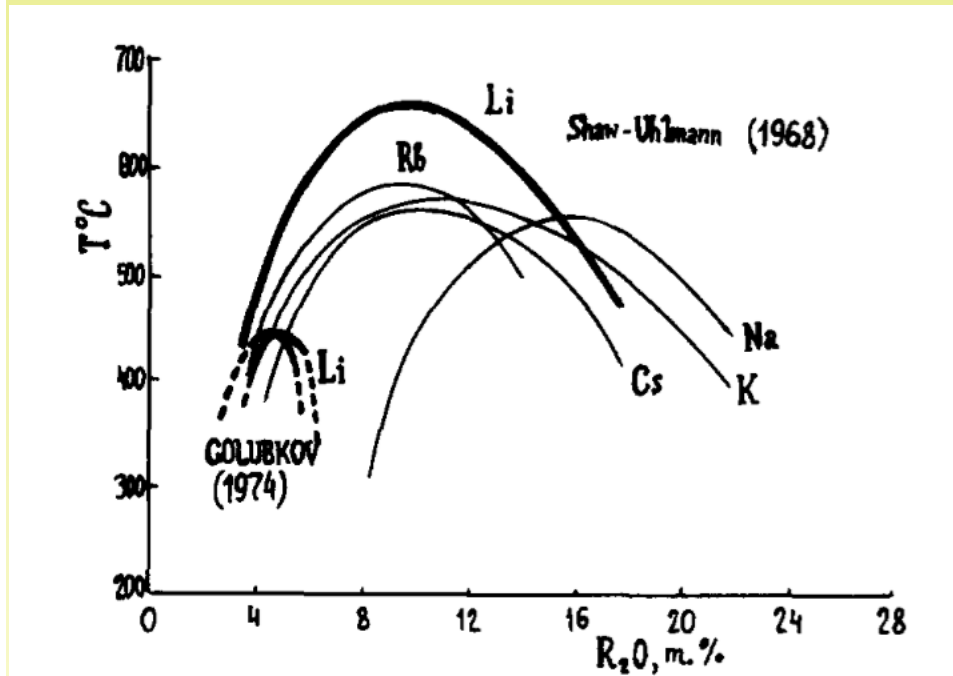
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Alkali Borate

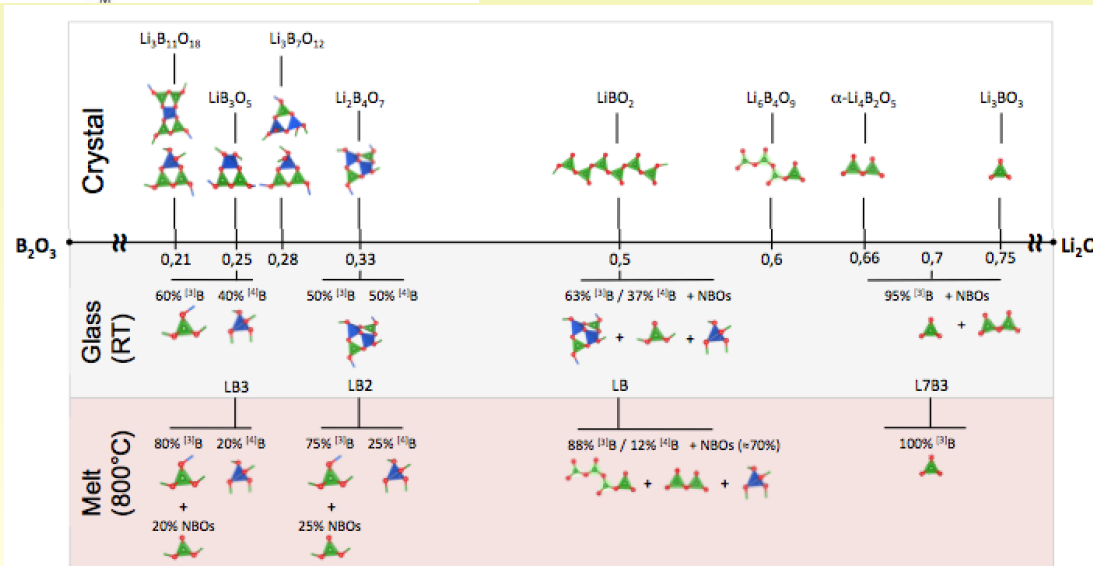
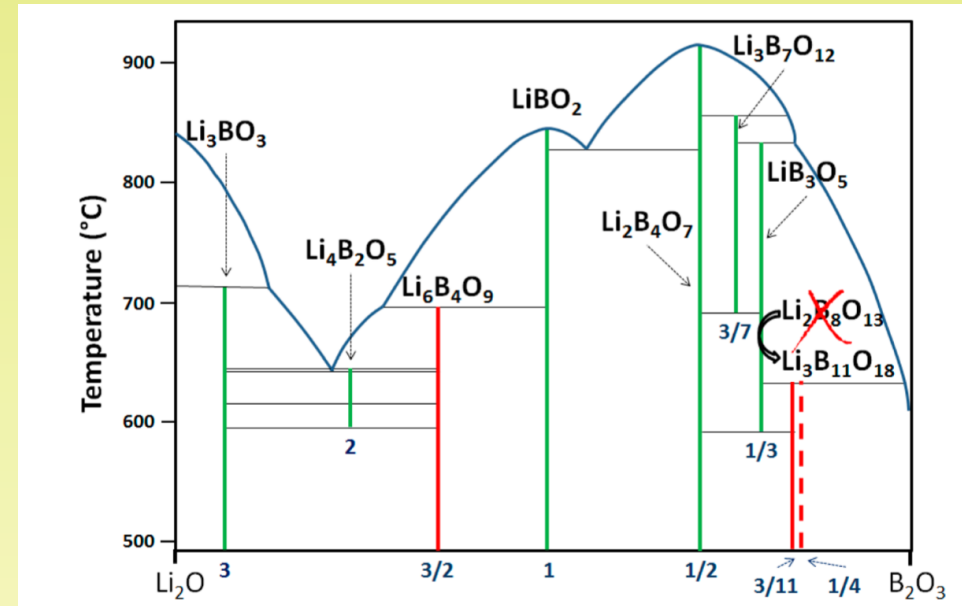
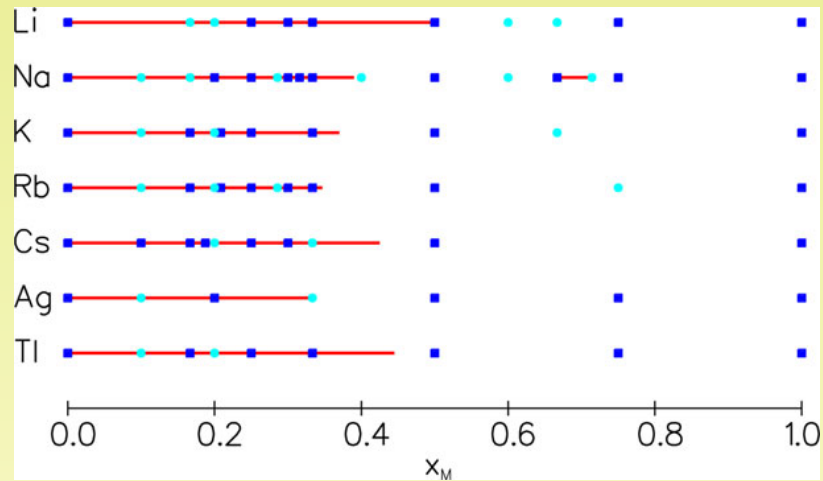
Shelby et al. *J. Am. Ceram. Soc.* **66**, 225 (1982)



Alkali Borate

■ Crystalline structures

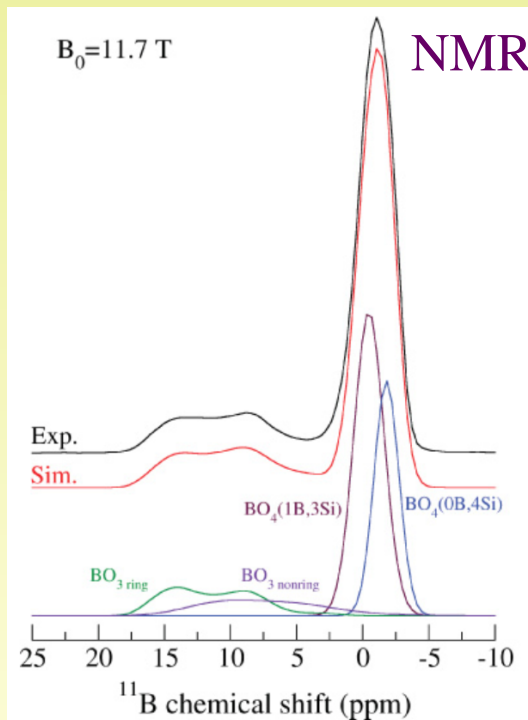
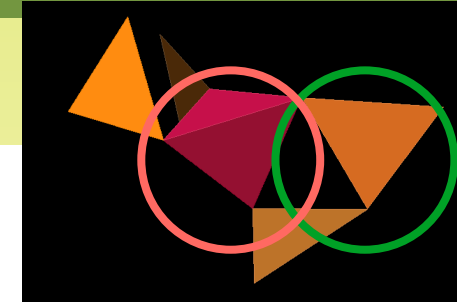
— Glass forming regions



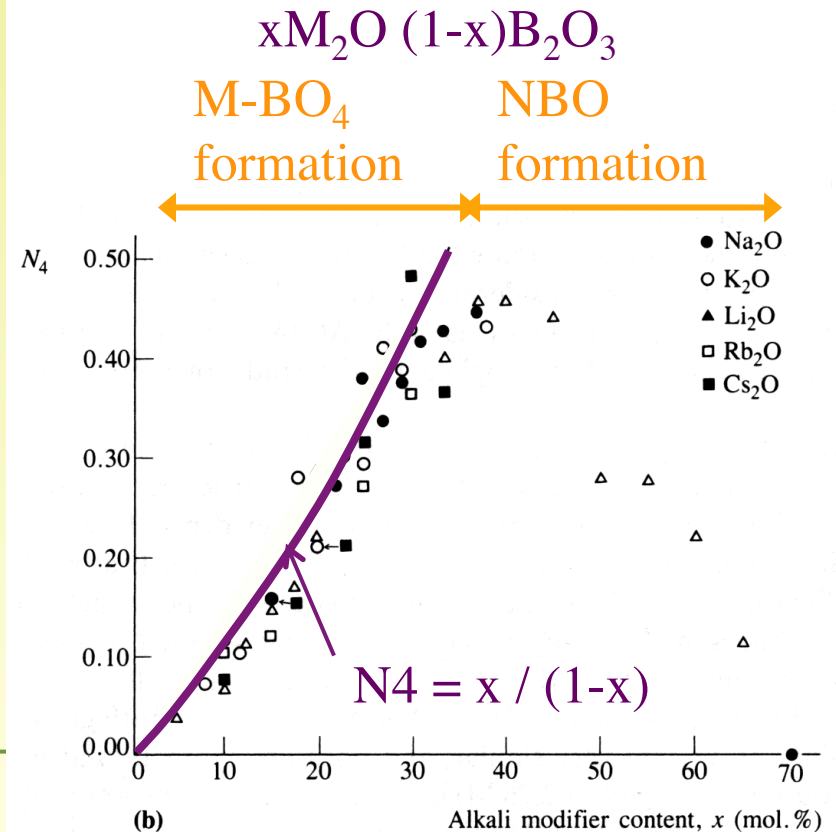
Short range order : Alkali Borate

❖ 2 sites for B

Triangle BO_3 - Tetrahedra BO_4^-
 $d^3_{\text{B-O}} \approx 1.37 \text{ \AA}$ - $d^4_{\text{B-O}} \approx 1.47 \text{ \AA}$



$$N_4 = (\text{Nbr } \text{BO}_4) / (\text{Total nbr B})$$



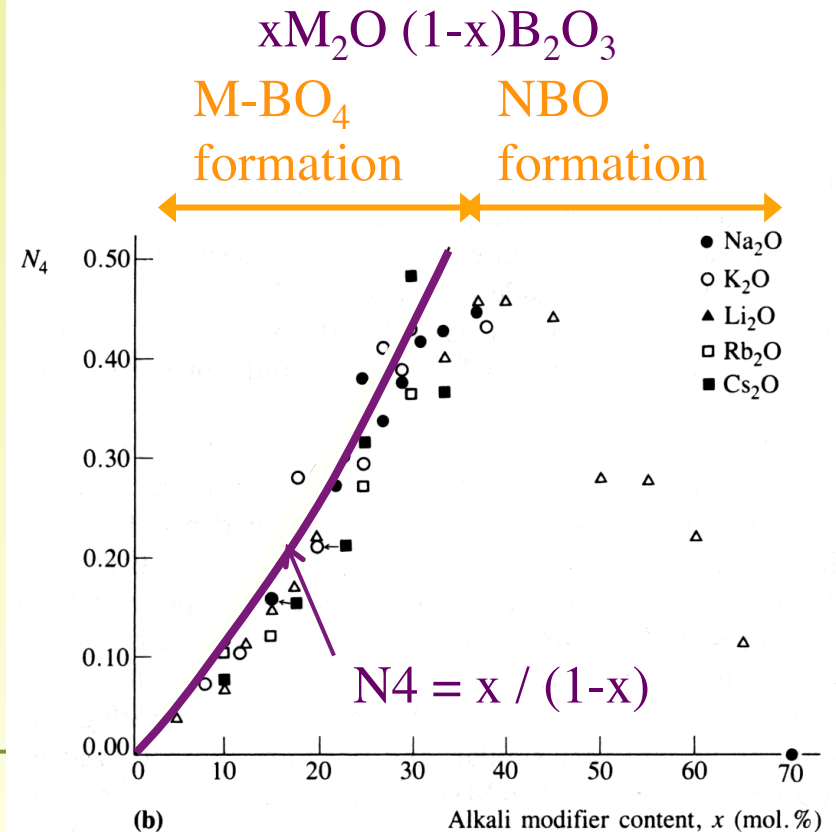
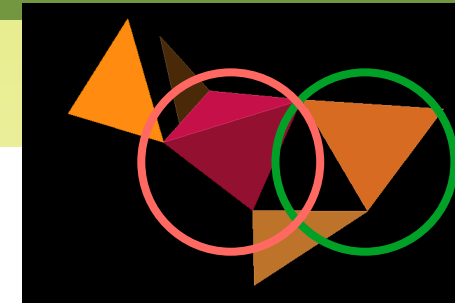
Short range order : Alkali Borate

❖ 2 sites for B

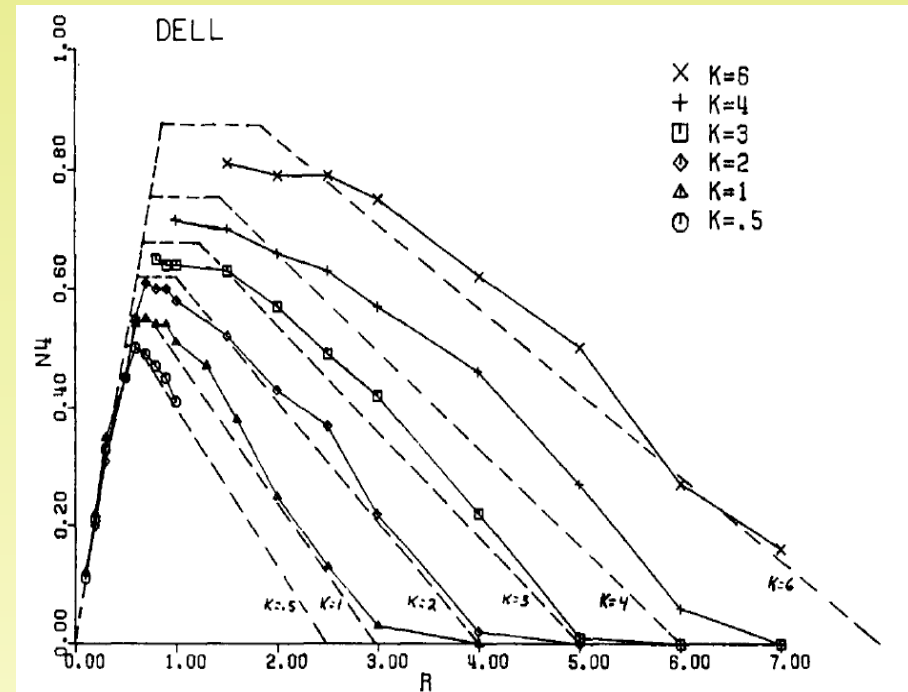
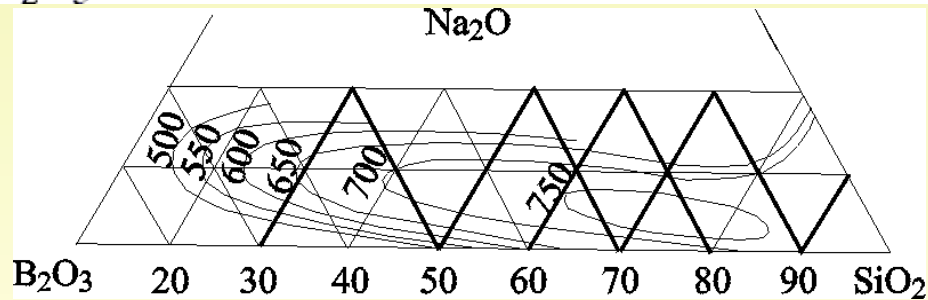
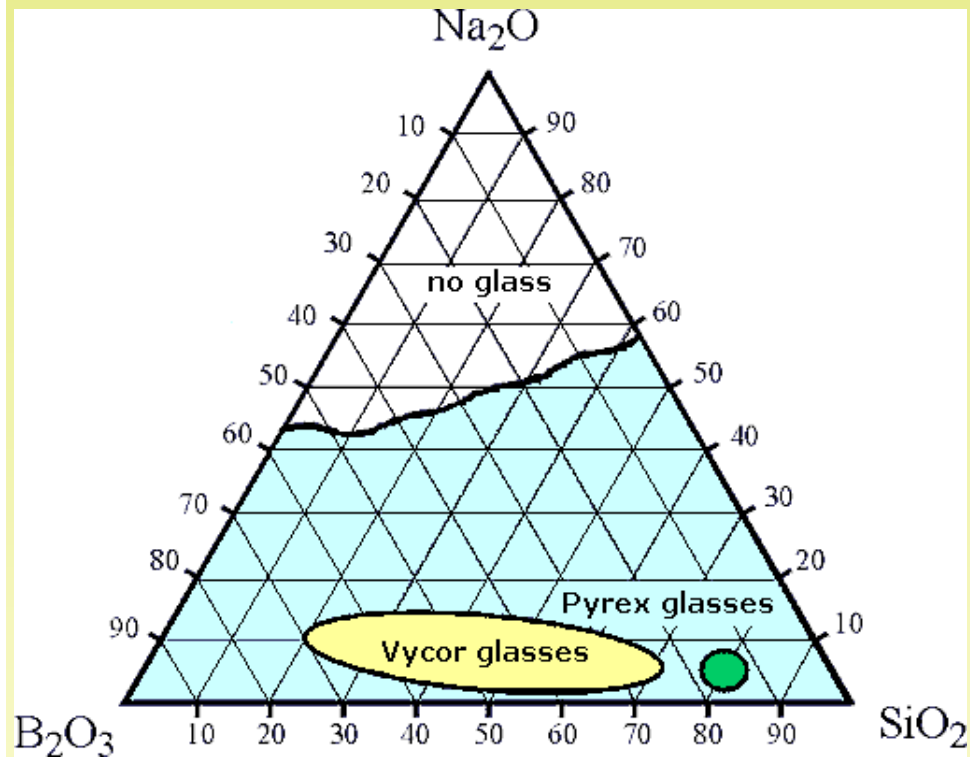
Triangle BO_3 - Tetrahedra BO_4^-
 $d^3_{\text{B-O}} \approx 1.37 \text{ \AA}$ - $d^4_{\text{B-O}} \approx 1.47 \text{ \AA}$

❖ Alkali are modifier near non-bridging oxygens (NBOs) or **charge compensator near BO_4^-**

⇒ Addition d'alcalins dans les borates augmentent la viscosité, T_g plus élevées ≠ silicates



Shor rante order : Borosilicate



*Dell, Bray and Xiao, J. Non-Cryst. Solids
58(1983)1*

<http://www.ptc.tugraz.at/specmag/struct/sbs.htm>
Mazurin, J. Non-Cryst 95&96(1987)71

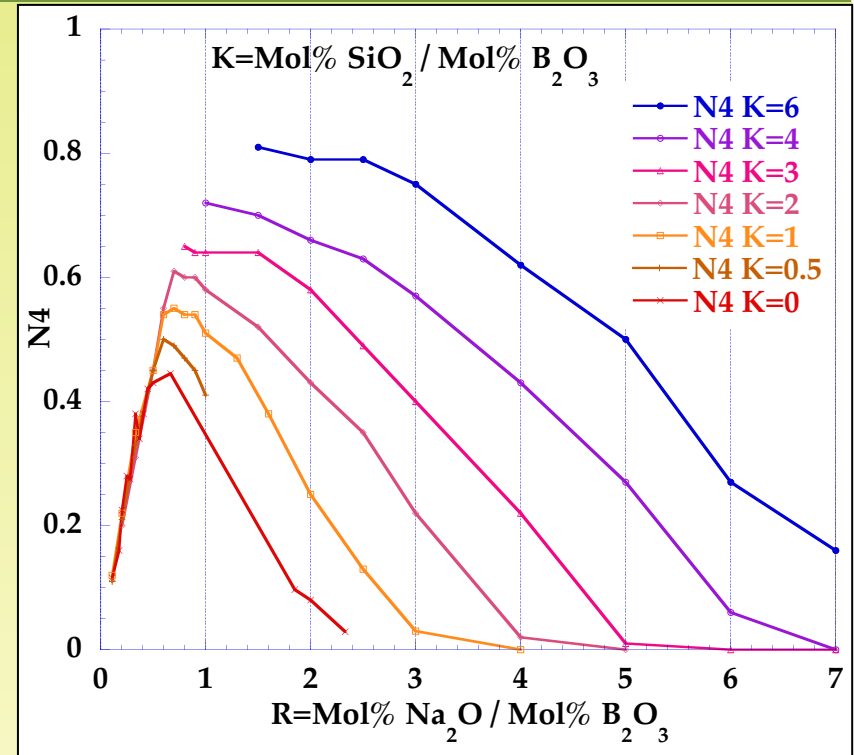
Atelier thermodynamique des verres, 11 octobre 2016

“Dell & Bray” model

$$R = \frac{Na_2O}{B_2O_3}$$

$$K = \frac{SiO_2}{B_2O_3}$$

first stage of creating NBOs :
 $ROSiO_3$ unit or the $ROBO_2$ unit, or both ?



N₄ value relative to R and K

“Dell & Bray” model

Saturation of the silicate network by BO_4 units before formation of NBOs

⇒ In fact important Si/B mixing and more random distribution of alkalis

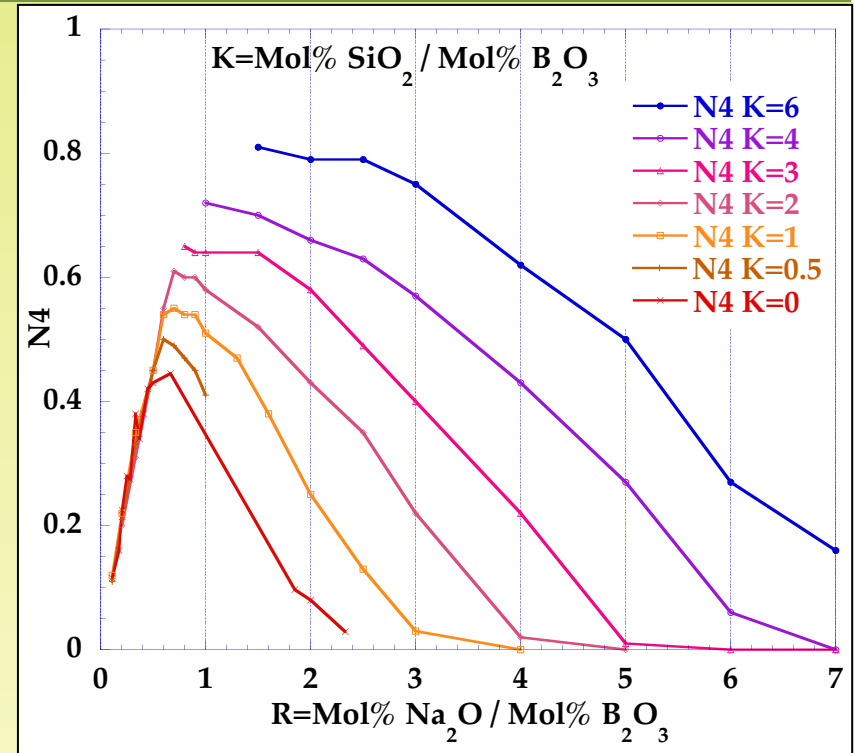
⇒ Revision of the « Dell & Bray » model

Miura et al. J. Non-Cryst. Solids 290(2001)1

Martens & Müller-Warmuth J. Non-Cryst. Solids 265(2000) 167

Zhao et al. J. Non-Cryst Solids 276(2000)122

Manara et al. J. Non-Cryst. Solids 355(2009)2528

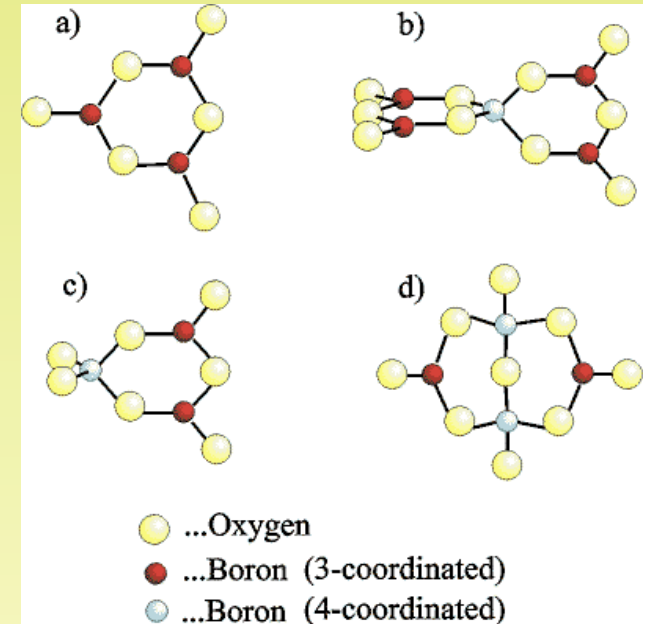
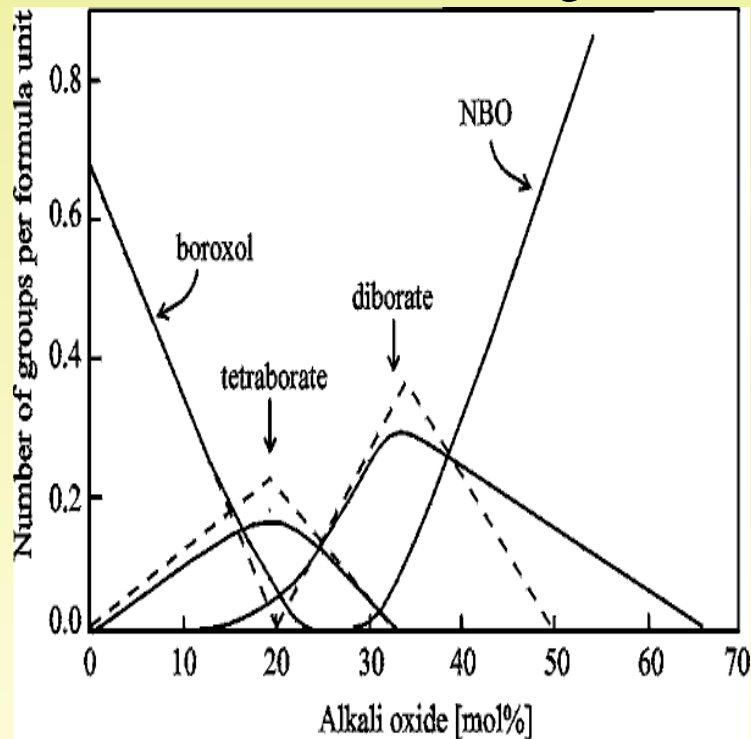


BO_4 formation NBO formation

Medium range order : Alkali borate

❖ « superstructural » units

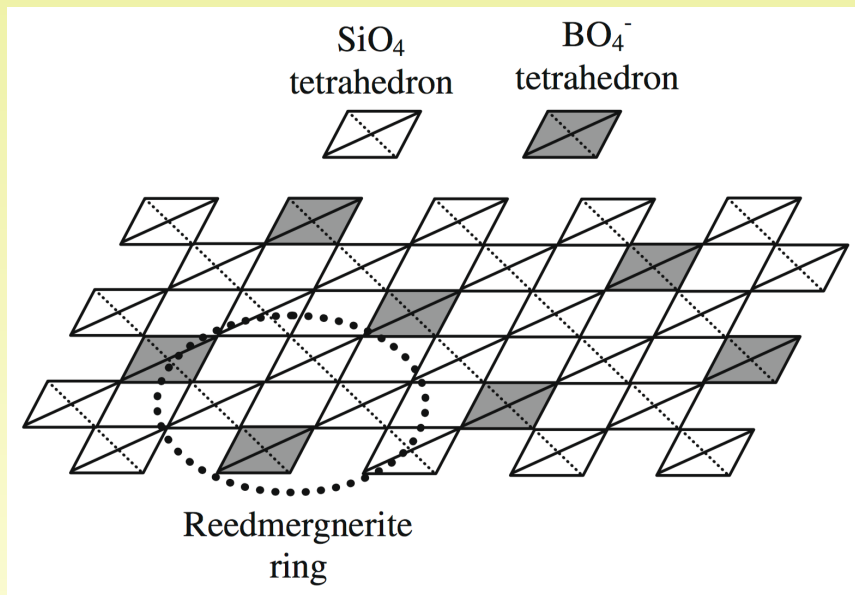
Rigid arrangements of basic units extending over medium range distances



Specific vibrational modes
=> Raman

Medium range order : Borosilicate

❖ Reedmergnerite ring



Proposed in the Dell & Bray model
Not well supported by NMR

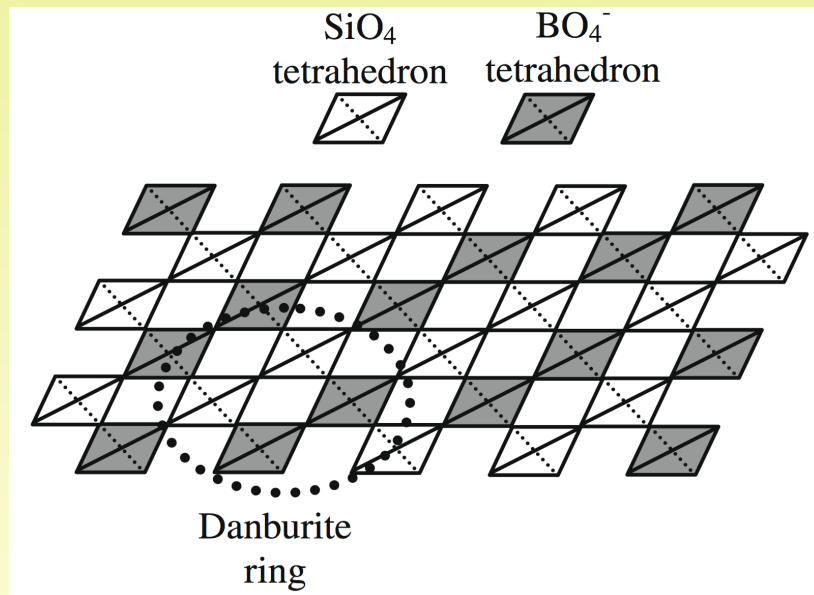
Du & Stebbins J. Phys. Chem. B 107(2003)10063

*Martens & Müller-Warmuth J. Non-Cryst. Solids
265(2000) 167*

Wang & Stebbins JACerS 82(1999)1519

Medium range order : Borosilicate

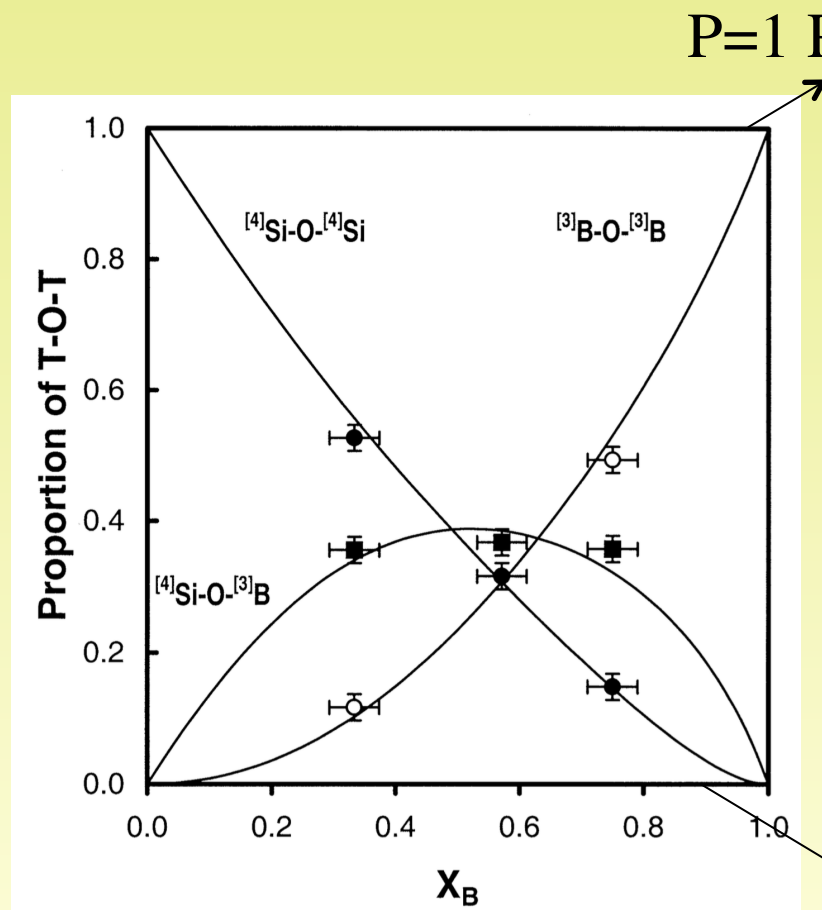
❖ Danburite ring



BO_4 - BO_4 pairs

Bunker et al. Phys. Chem. Glasses 31(1990)30
Manara et al. J. Non-Cryst. Solids 355(2009)2528

Si/B mixing : Borosilicate



P=1 Phase separation

Degree of interdispersion
 $P = 1 \exp [-(2W/kT_f(X_{Si}))]$

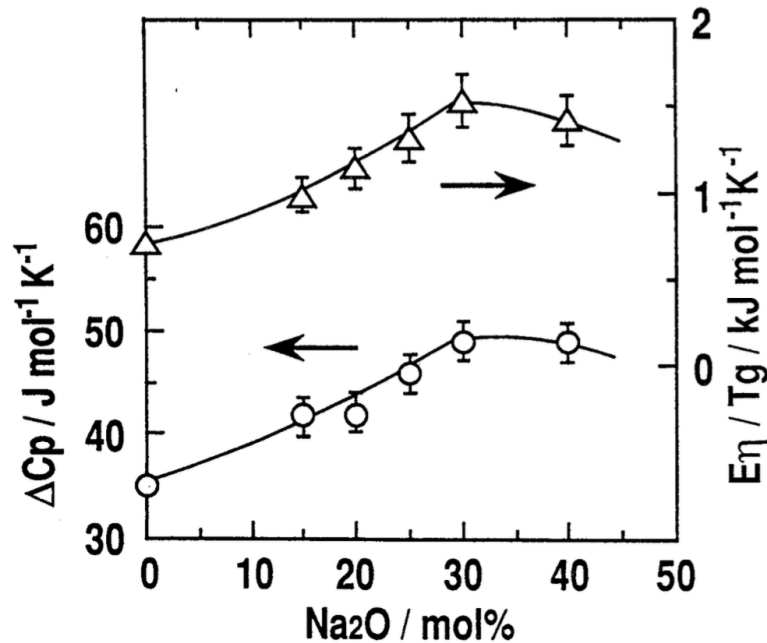
$2W = 5.4 \text{ kJ/mol}$ for $B_2O_3-1.5SiO_2$

$\Rightarrow P \sim 0.62$

\Rightarrow considerable network mixing but
 not fully random

P=0 Random mixing

Structural relaxation of the liquid with T (1/2)



Lee et al. *J. Ceram. Soc. J.* 103(1995)398

$$dH = C_p dT$$

Changes of configurations within the liquid

High C_p^{conf} , FRAGILE liquid

- Dynamic slowing down
- No more evolution of configuration below T_g (glass transition)

$$C_p = C_p^{vib}(T) + C_p^{conf}(T)$$

$$A T < T_g \Rightarrow C_p^{conf}(T) = 0$$

$C_p^{conf}(T)$ more or less high according to the system

- Chemical order
- Structure

Structural relaxation of the liquid with T (2/2)

Properties

C_p^{conf} , α^{conf} , K_T^{conf} , viscosity, transport

Structural changes of the liquid

What are the structural changes that allow the liquid to remain in internal equilibrium when temperature decreases?

Do these changes take part to the glass transition phenomenon?

Network:

Basic forming units

Medium range ordering

Cationic sites:

Coordination of polyhedra

Connection to the network

Neutron structure factors for MB2 glasses

High Q range:

Short distances (BO_4 , BO_3)

Low Q range:

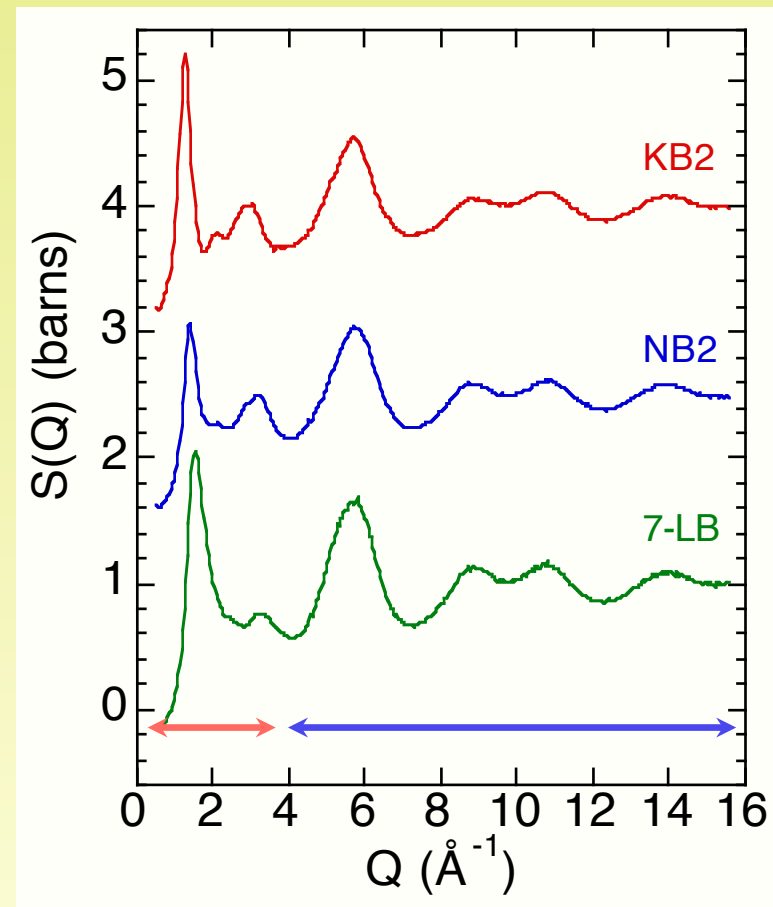
Medium distances (5-8 Å)

KB2: $\text{K}_2\text{O}-2\text{B}_2\text{O}_3$

NB2: $\text{Na}_2\text{O}-2\text{B}_2\text{O}_3$

LB2: ${}^7\text{Li}_2\text{O}-2\text{B}_2\text{O}_3$

Majerus et al., Phys. Rev. B, 67 (2003) 024210



Data obtained on 7C2 diffractometer at
LLB (France)

Neutron structure factors for MB2 liquids

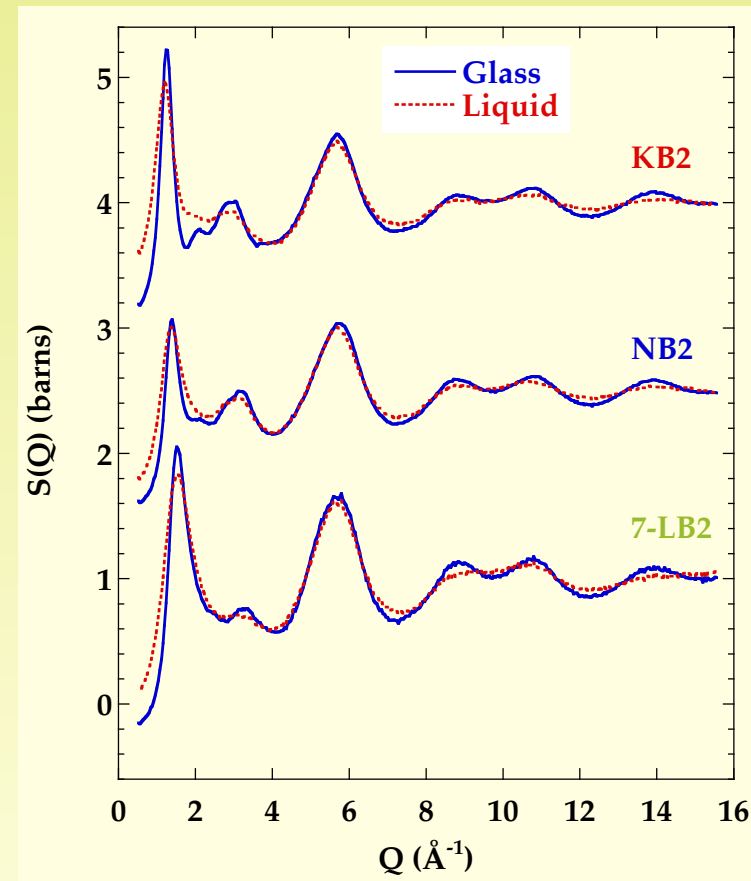
Decrease of the amplitude of oscillations

Less structures at low Q

KB2: 1200 K

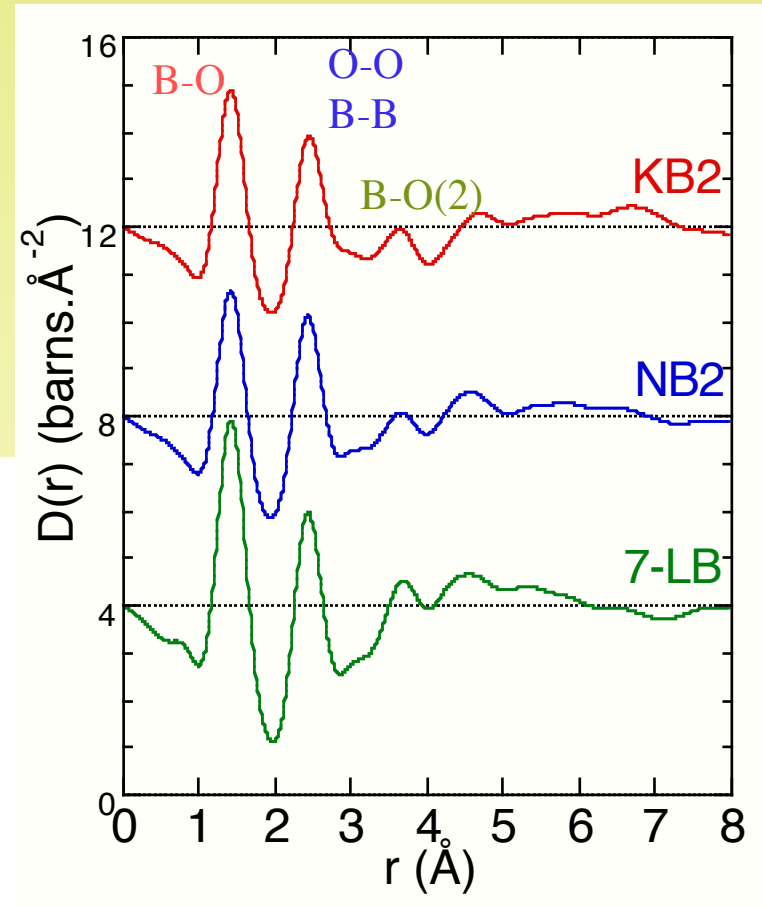
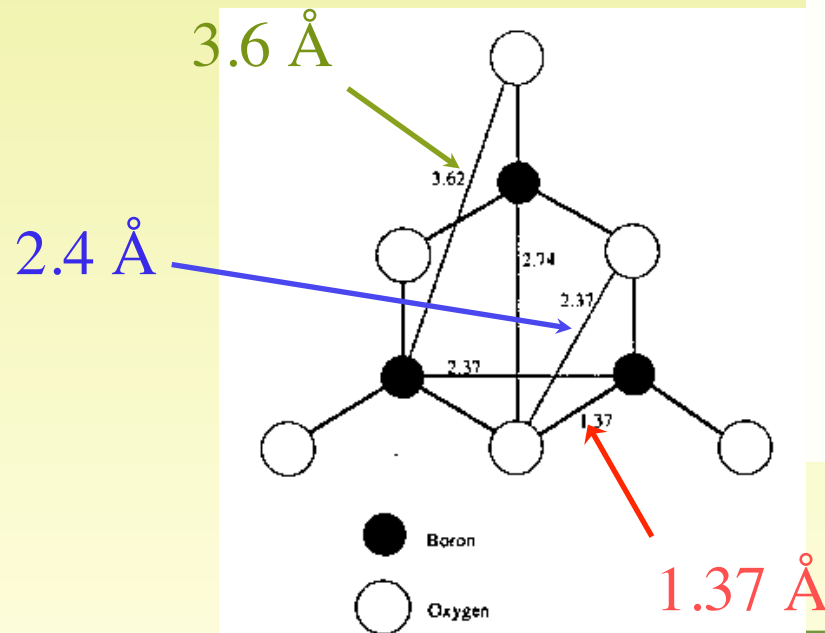
NB2: 1100 K

LB2: 1273 K



Correlation functions for MB2 glasses

First peaks due to the borate network



Correlation functions for MB2 liquids

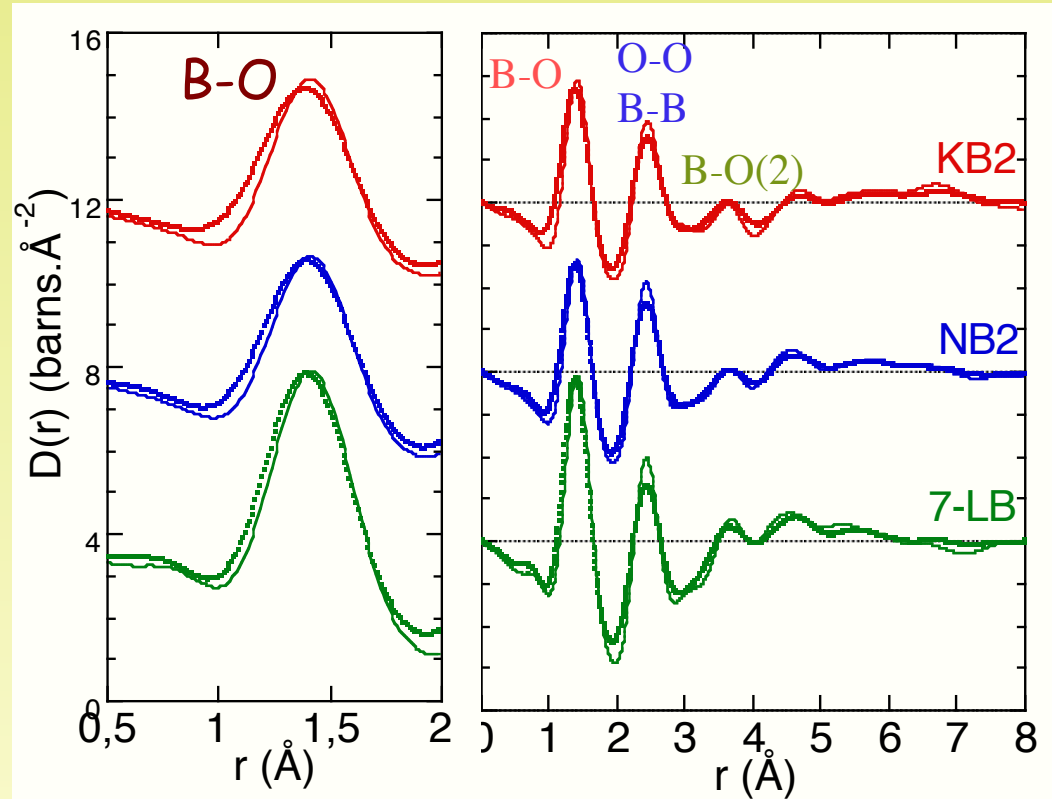
Decrease in intensity
of O-O peak

Disappearance of
structures at 3 Å

Shift of the B-O
peak by -0.01 Å



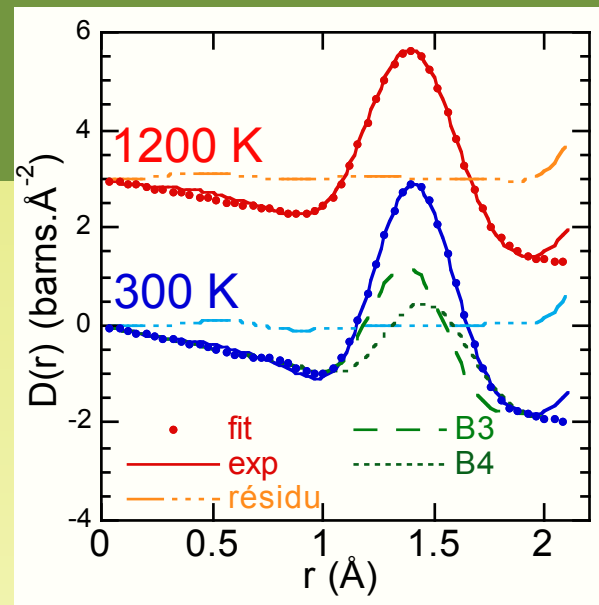
Fits with
Gaussian



Triangle BO_3 - Tetrahedra BO_4^-
 $d^3_{\text{B-O}} \approx 1.37 \text{ \AA}$ - $d^4_{\text{B-O}} \approx 1.47 \text{ \AA}$

Fit of the B-O peak

Decrease of the proportion of BO_4^- in the liquid, by about
45 % to 30 %

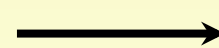
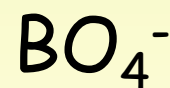


Gaussian fit of the B-O peak

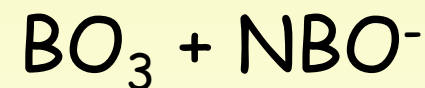
Fraction $\text{BO}_4^- / (\text{BO}_4^- + \text{BO}_3^-)$

	Verre	Liq
LB2	46	30
NB2	43	35
KB2	40	32

Glass

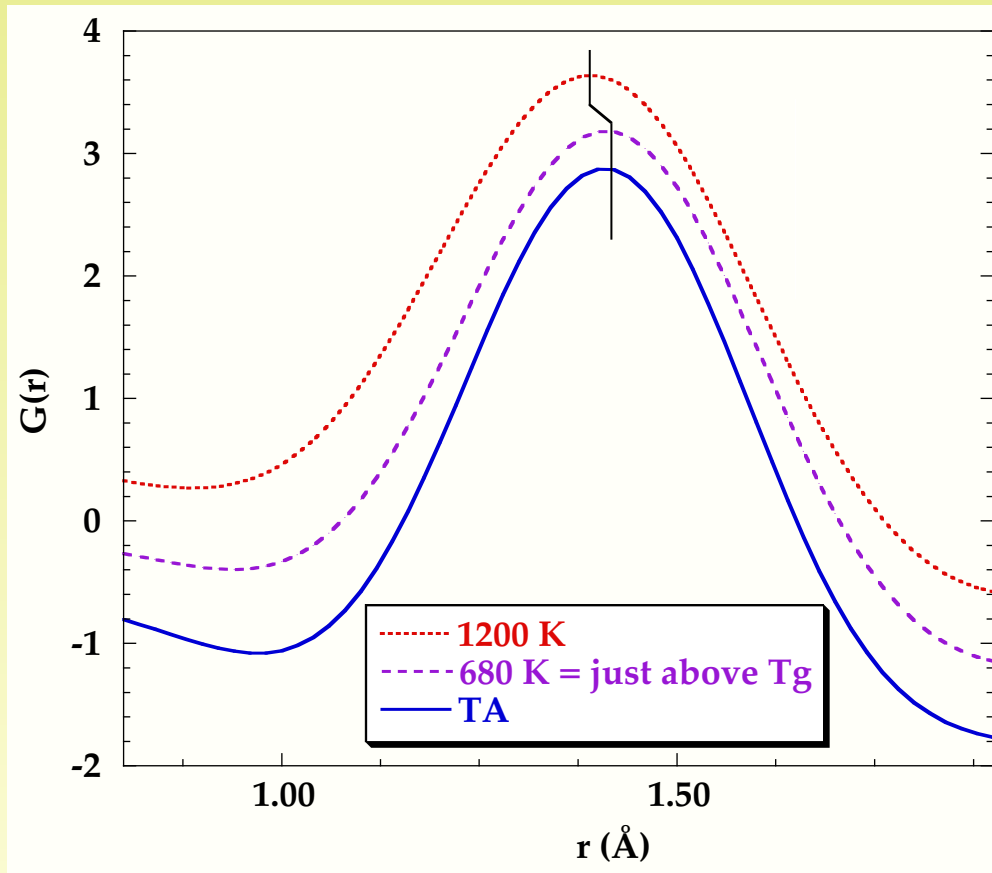


Liquid



Beginning of the BO_4 to BO_3 conversion

B-O contribution



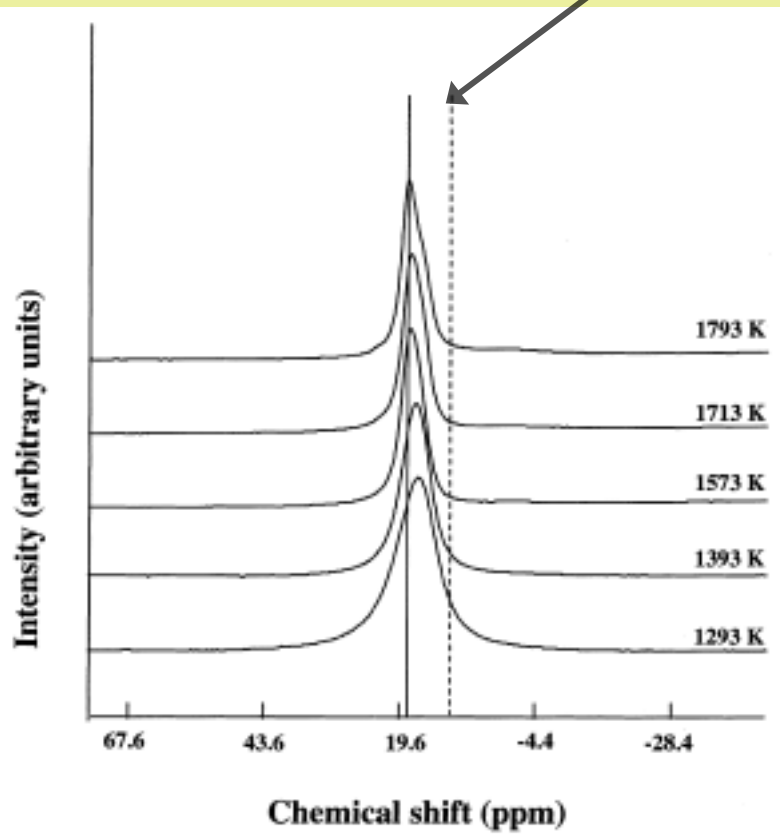
No shift of the B-O peak up to T_g

Coordination change onset above T_g

High-temperature NMR results

Sen, J. Non-Cryst. Solids,
253(1999)84

Position if no change



NMR shows a boron coordination change

but averaging of the two chemical shifts

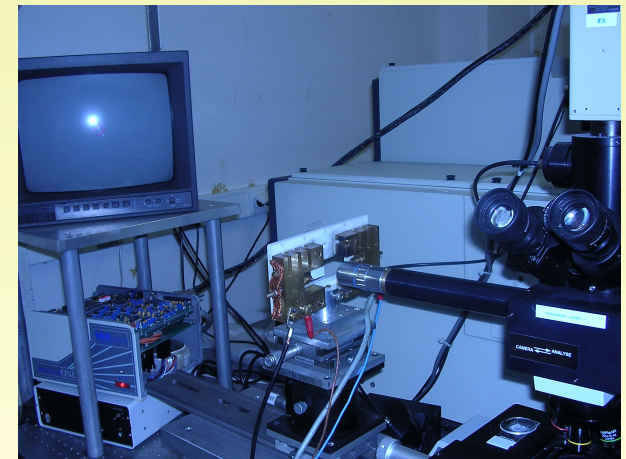
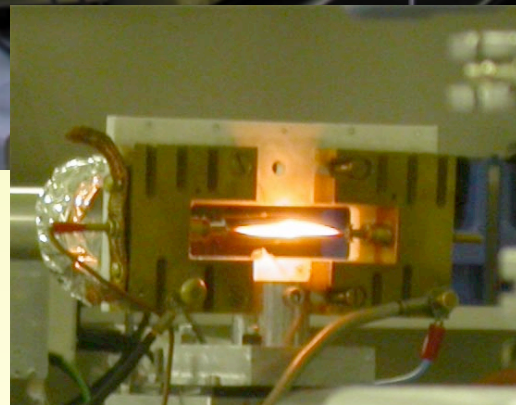
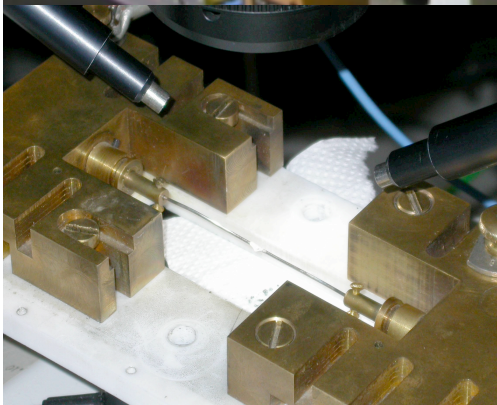
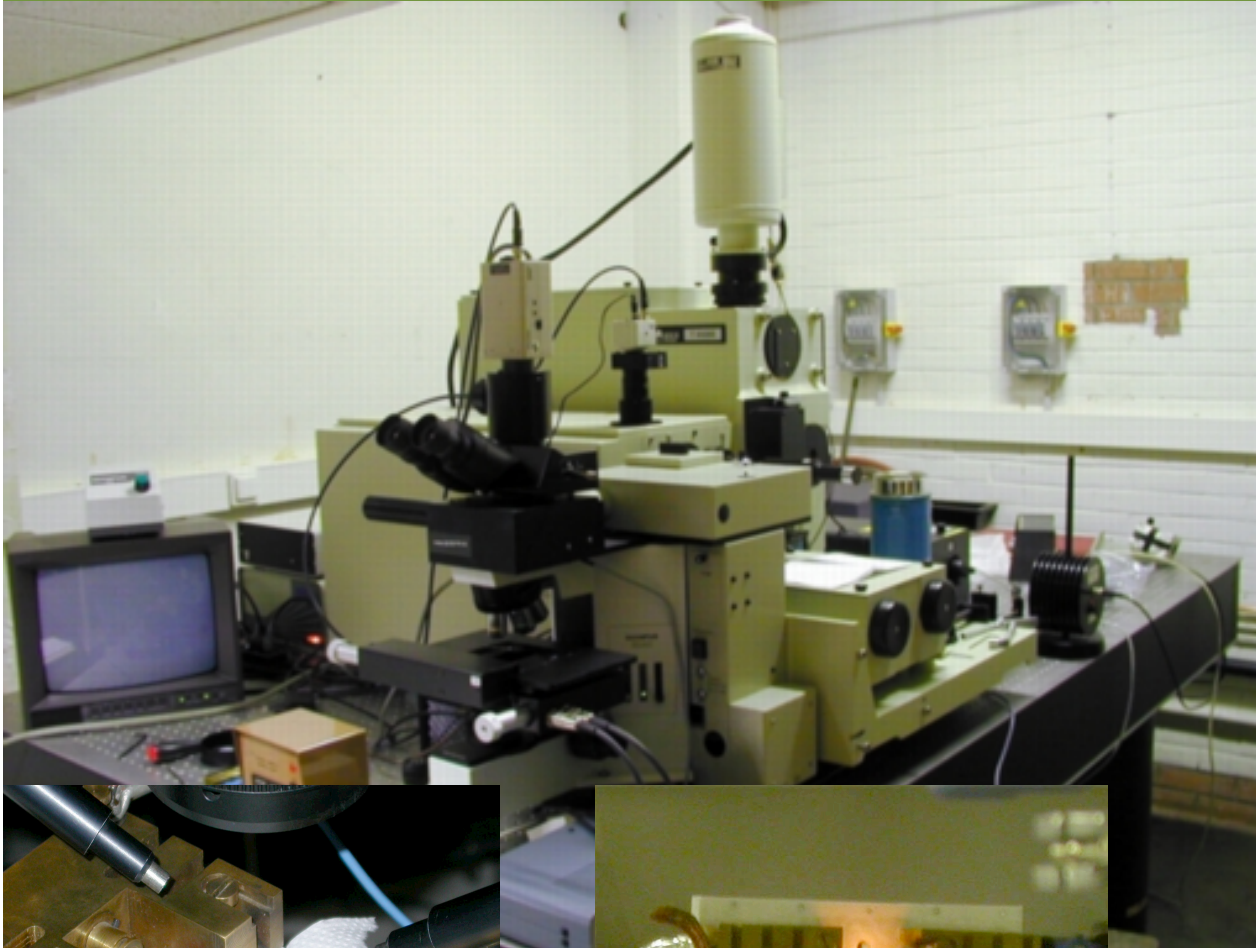
⇒ quantification difficult

Na diborate

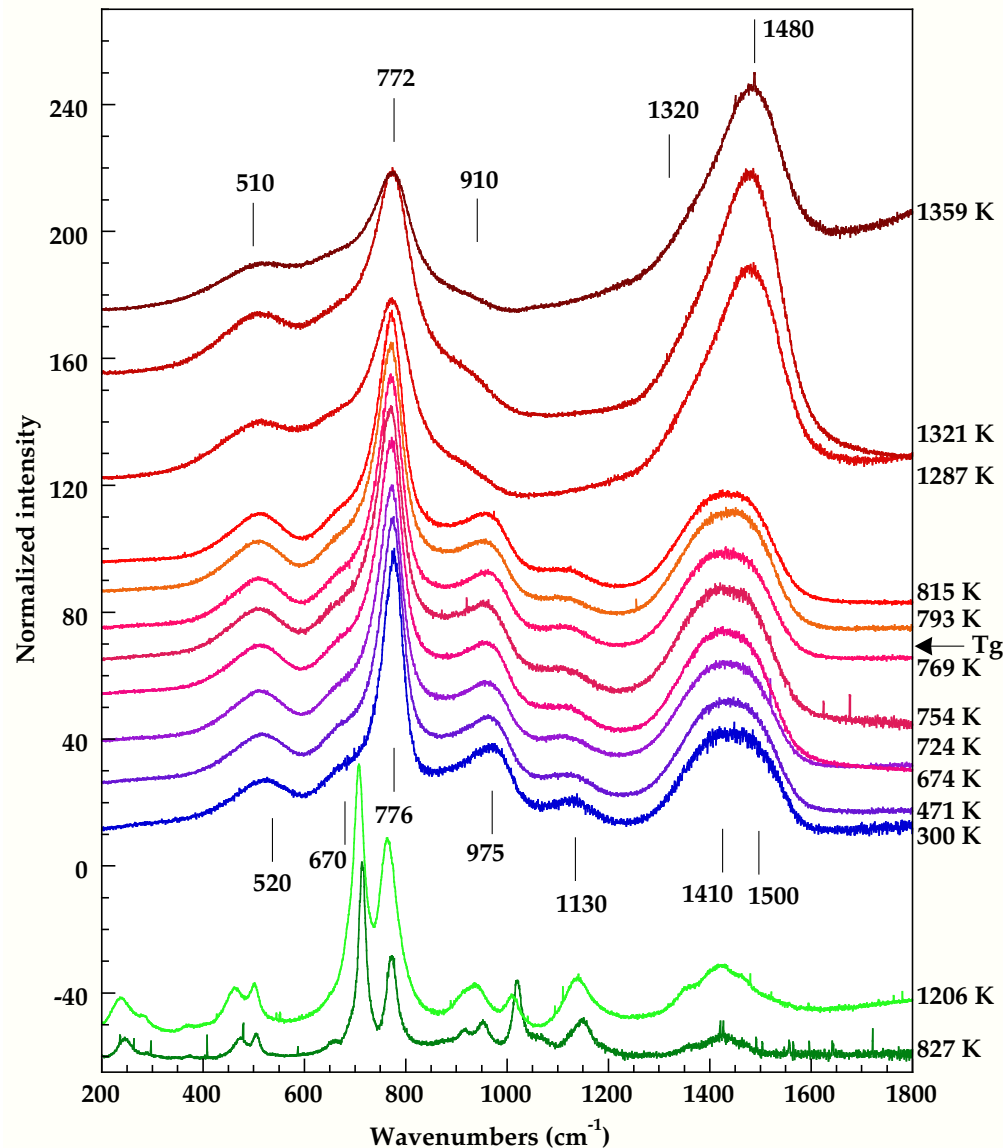
High temperature Raman spectroscopy

Jobin Yvon T 64000
 $\lambda=514 \text{ nm Ar}^+$ laser
CCD detector

Furnace: Pt wire with a hole



High temperature Raman spectra for LB2



Cormier et al., *J. Am. Ceram. Soc.*, 89(2006)13

In the **glass**:

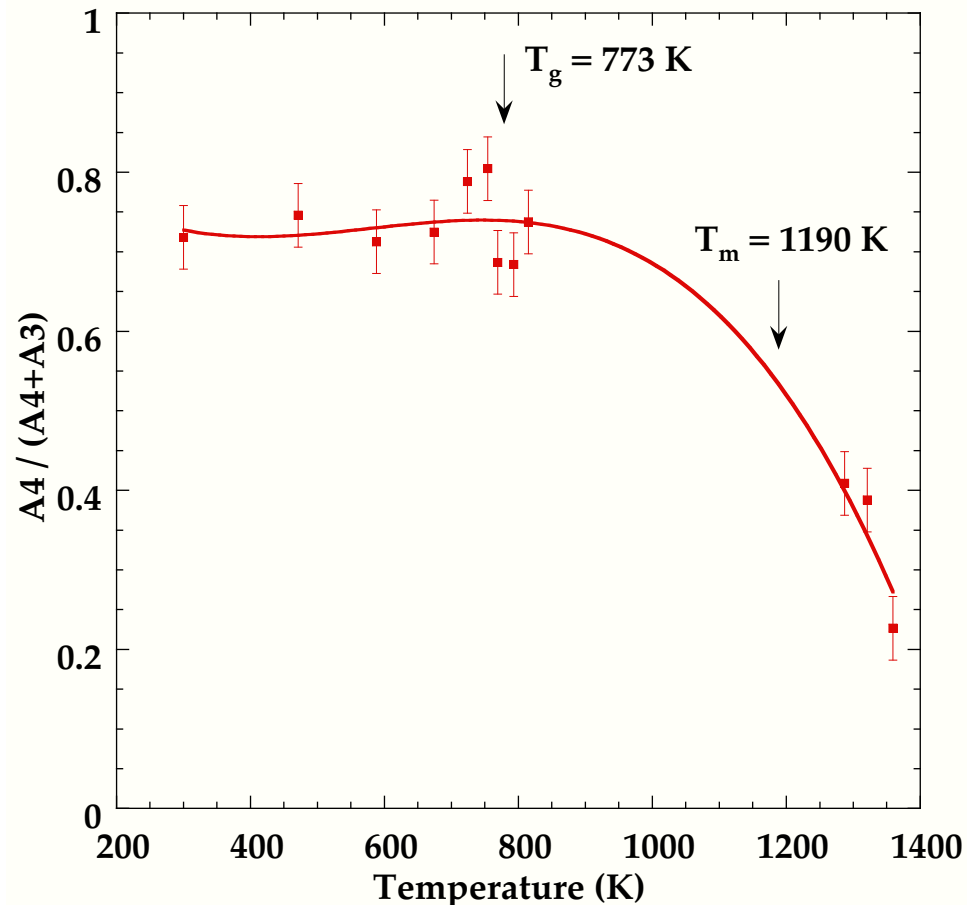
- **Medium frequency bands:**
borate groups containing BO_4 tetrahedra (mainly diborate and tetraborate)
- **High frequency bands:**
due to B-O and B-NBO stretching band at 1410 cm^{-1} related to BO_4 band at 1500 cm^{-1} related to BO_3

In the **liquid**:

- **Medium frequency bands:**
disappearance of the borate groups
- **High frequency bands:**
change in relative intensity of the two bands

⇒ Agree with less BO_4 in the liquid and more NBOs

Proportion of BO_4 determined by Raman



Area ratio of the two high frequency bands (Gaussian fit)

- ⇒ Important decrease of the low frequency band associated with BO_4 tetrahedra
- ⇒ Conversion begins only above T_g
- ⇒ Indirect evaluation

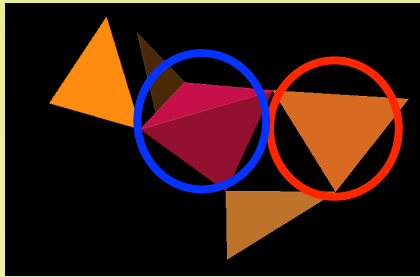
➡ **Détermination qualitative de la coordinence du bore en température**

- Calibration avec une technique quantitative (RMN et neutrons)

K diborate: Akagi et al., J. Non-Cryst. Solids, 293-295 (2001) 471

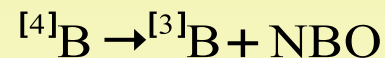
Na diborate: Yano et al., J. Non-Cryst. Solids 321 (2003) 137

High temperature Raman spectra for $x\text{Na}_2\text{O}-(1-x)\text{B}_2\text{O}_3$



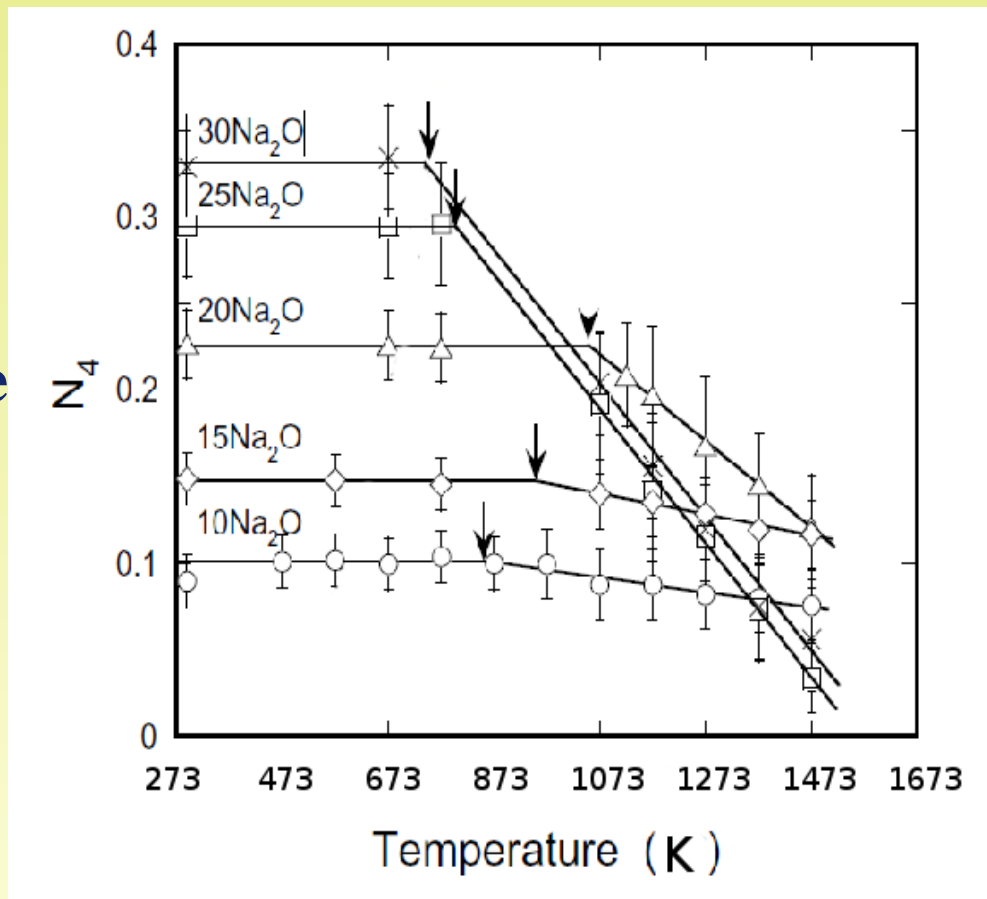
2 boron-site geometry :
 BO_4 tetrahedron BO_3 triangle

With increasing temperature :



$$N_4 = \frac{[4]_{\text{B}}}{[4]_{\text{B}} + [3]_{\text{B}}}$$

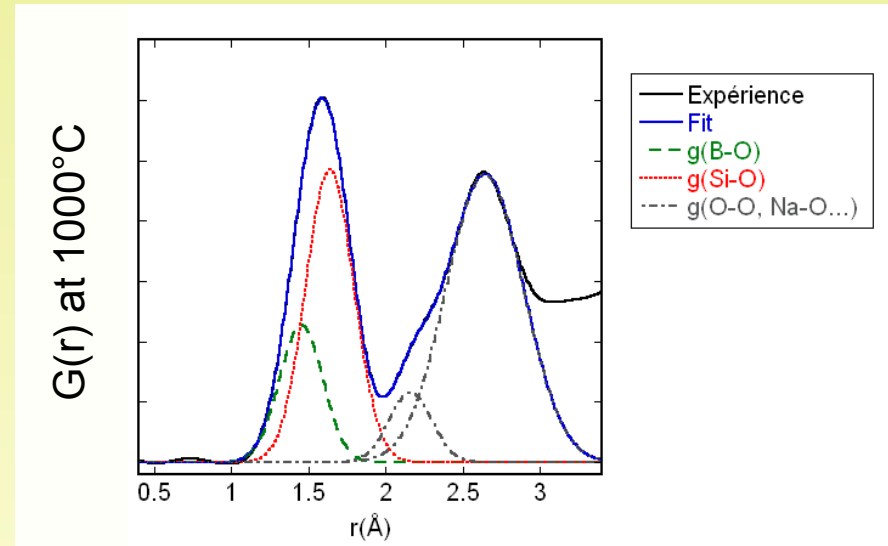
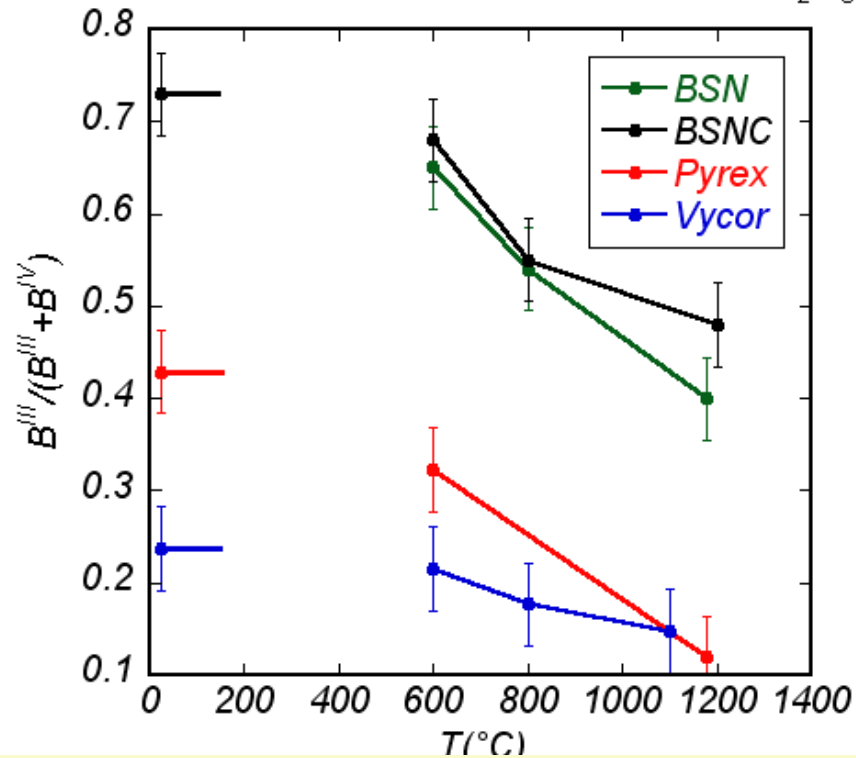
Raman, in situ study of an alkali borate



Yano et al. J. Non-Cryst. Solids 321(2003)147

BO₄ to BO₃ conversion in borosilicates

BSN, BSNC: soda boro-silicated and soda-lime borosilicated glasses with 15%mol B₂O₃

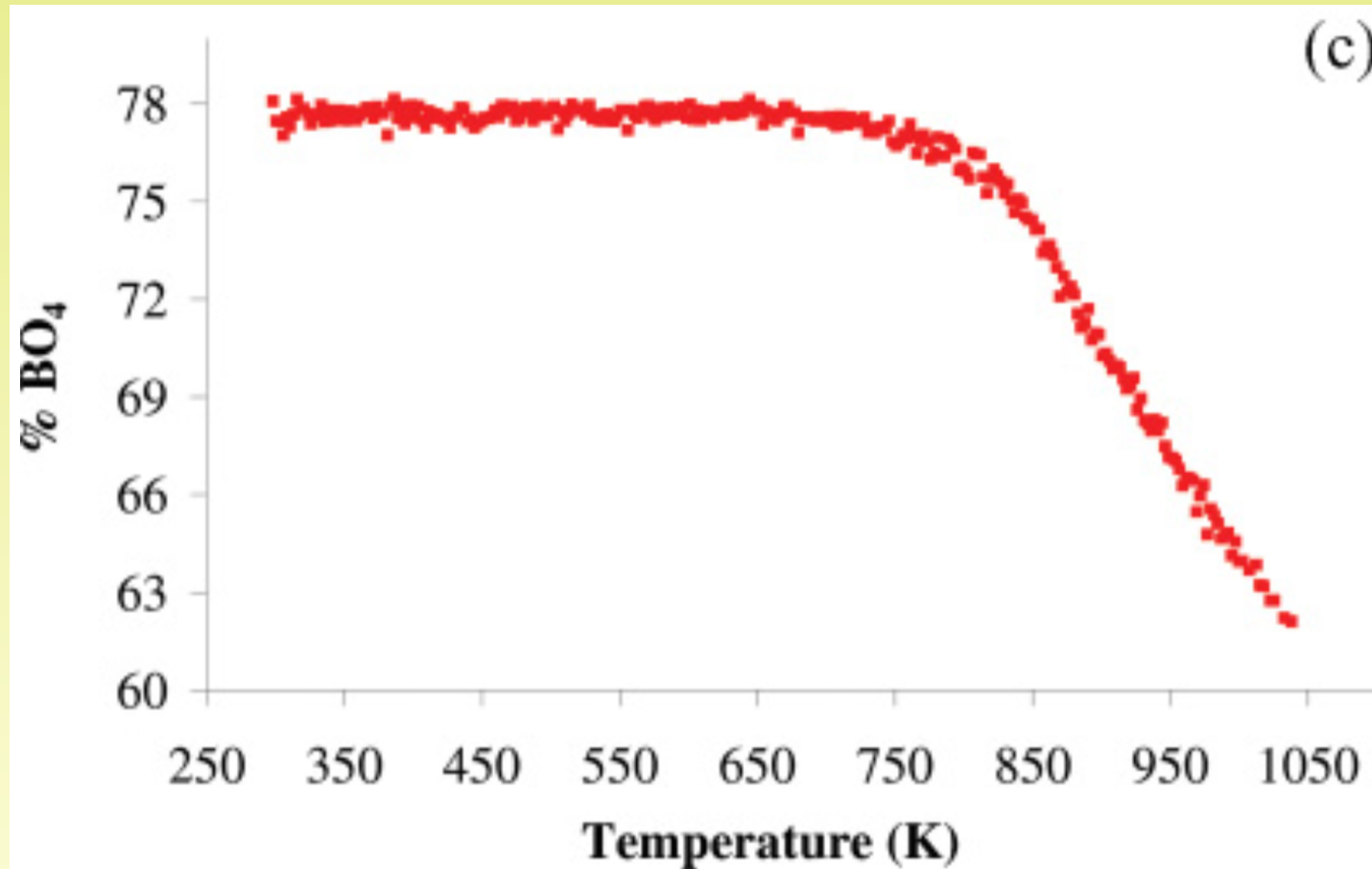


Michel et al. *J. Non-Cryst. Solids* 379(2013)169

Main results:

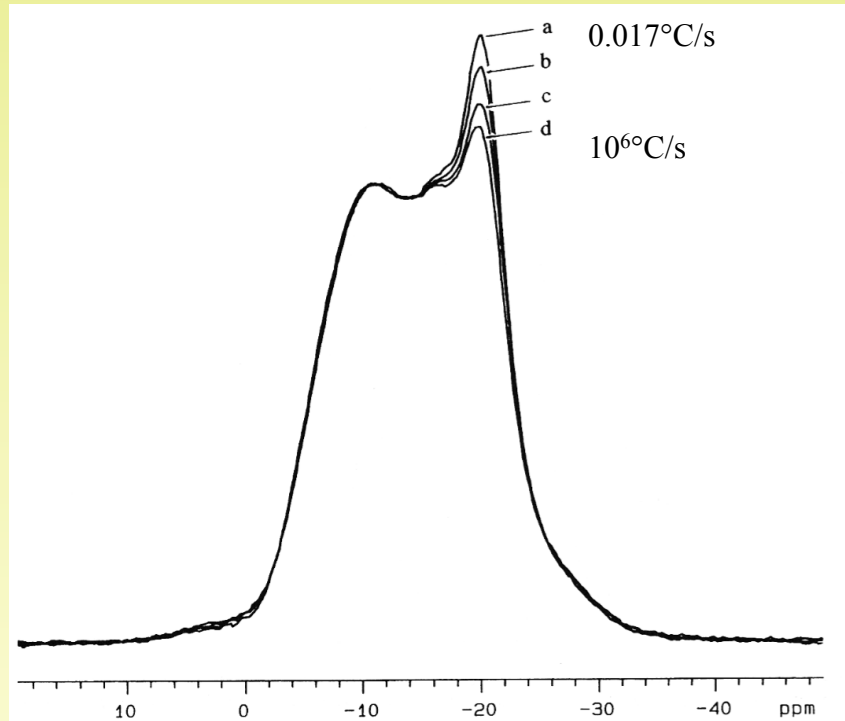
- Initial coordination follows the Dell and Bray model
- Systematic drop of BO₃ with temperature

BO₄ to BO₃ conversion in borosilicates



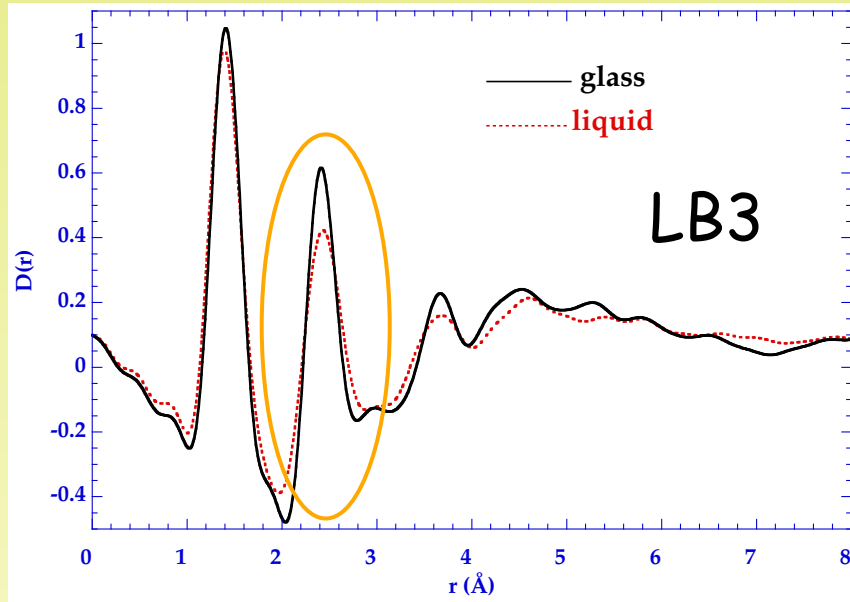
Angeli et al., Physical Review B, 85(2012)54110

Effect of fictive temperature



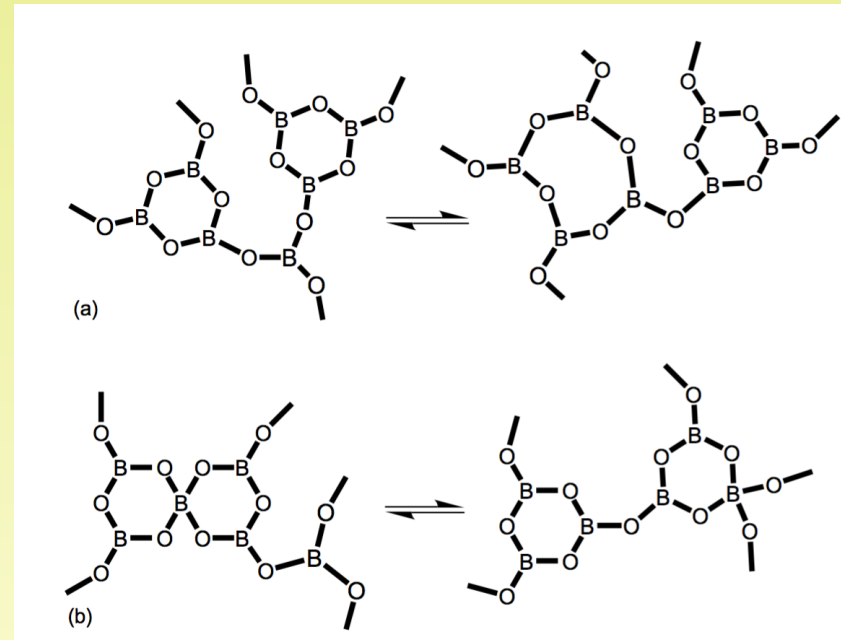
Fraction of BO₄ decreases as the fictive temperature increases

Changes at intermediate range



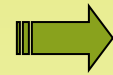
- Broadening of the B-B peak:
- Opening of the network due to more NBO and disappearance of rings, in agreement with Raman:

Cormier et al., J. Am. Ceram. Soc., 89(2006)13



Environment of alkalis ?

Less BO_4 , more NBO



Consequences for the alkalis ?

Study of the environment around Li using neutron diffraction coupled with isotopic substitution technique:

Principle: canceling contributions from the borate network

$$S_6(Q) = \sum_{\substack{\alpha\beta \\ \alpha \neq \text{Li}}} c_\alpha c_\beta b_\alpha b_\beta (S_{\alpha\beta}(Q) - 1) + \sum_{\alpha\text{Li}} c_\alpha c_{\text{Li}} b_\alpha b_{\text{Li}}^6 (S_{\alpha\text{Li}}(Q) - 1)$$
$$S_7(Q) = \sum_{\substack{\alpha\beta \\ \alpha \neq \text{Li}}} c_\alpha c_\beta b_\alpha b_\beta (S_{\alpha\beta}(Q) - 1) + \sum_{\alpha\text{Li}} c_\alpha c_{\text{Li}} b_\alpha b_{\text{Li}}^7 (S_{\alpha\text{Li}}(Q) - 1)$$



We obtain $S_{\text{Li}-\alpha}(Q)$ and $G_{\text{Li}-\alpha}(r)$

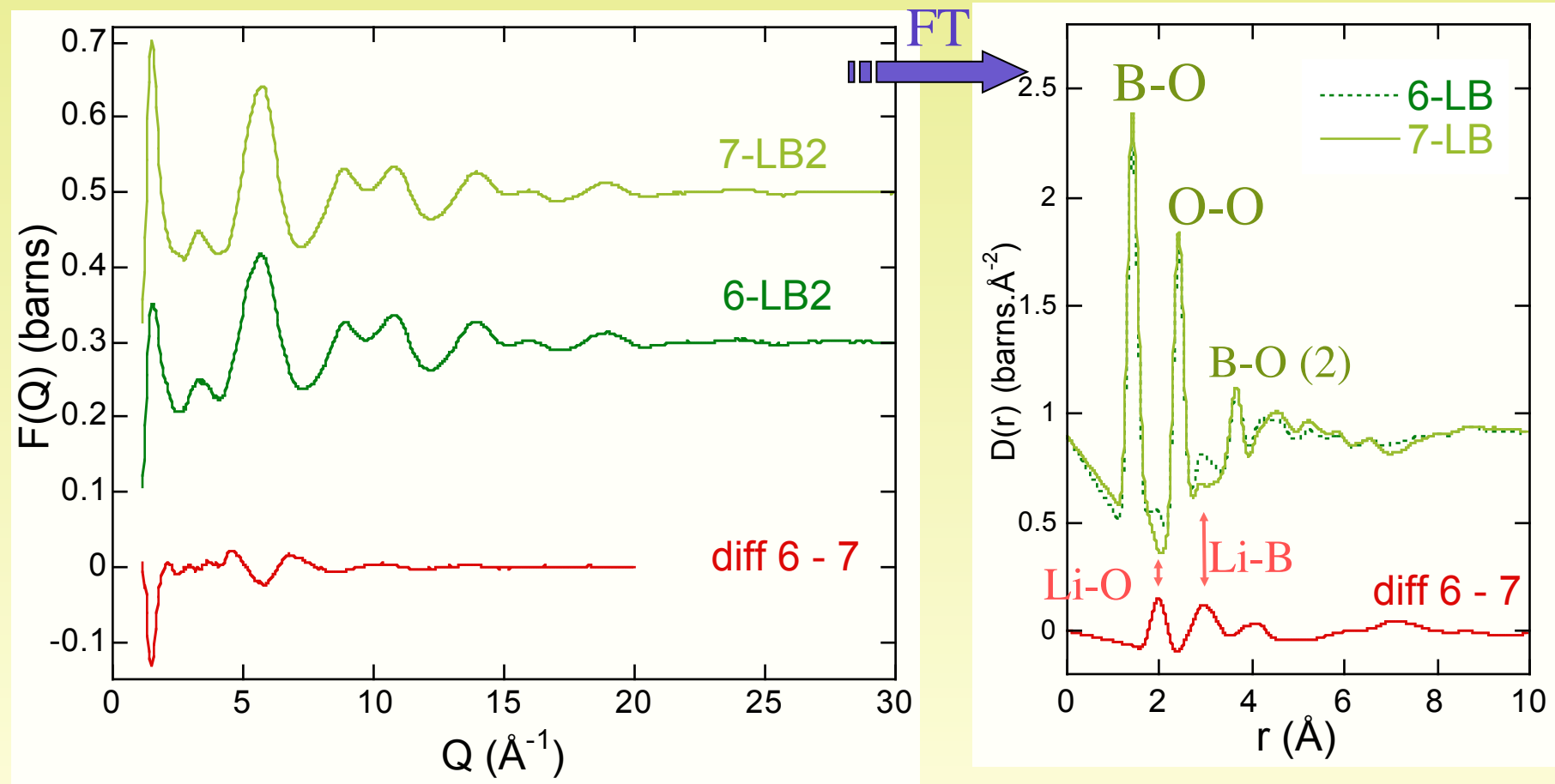
Li site in the glass

Less BO_4 , more NBO



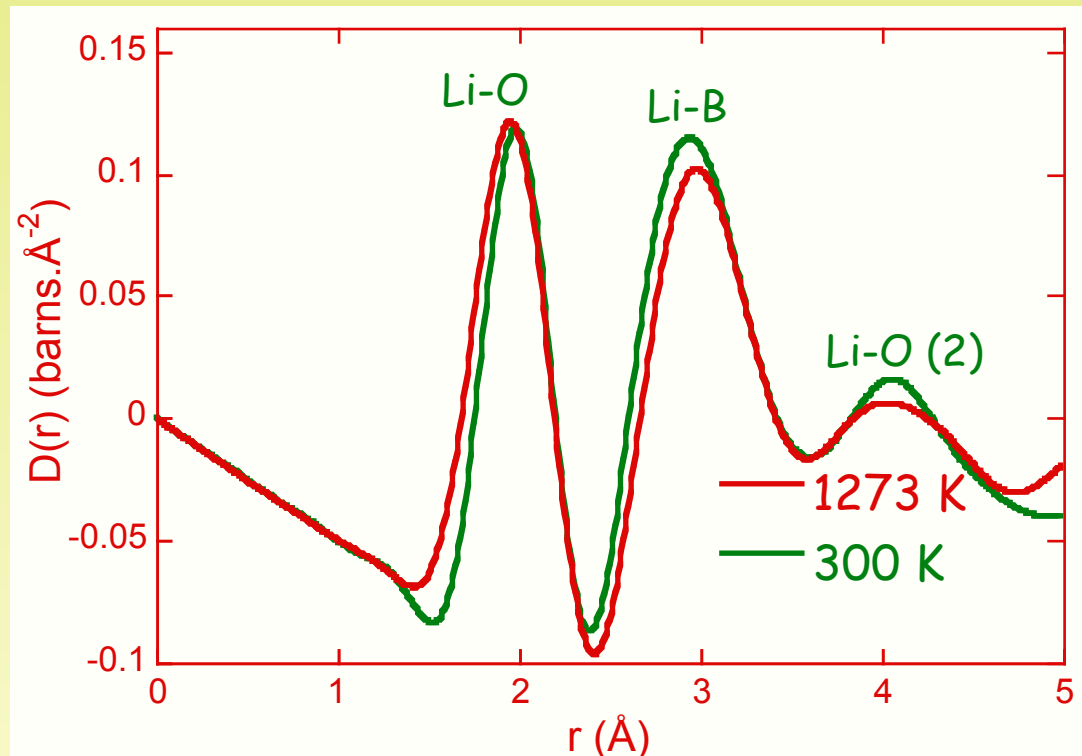
Consequences for the alkali ?

$S_{\text{Li-}\alpha}(Q)$ et $G_{\text{Li-}\alpha}(r)$ obtained by isotopic substitution of Li



Data obtained on SANDALS at
ISIS (UK)

Evolution of the Li site between the glass and the liquid



Data obtained on 7C2 at LLB
(France)

Shortening of the mean
Li-O distance by -0.02\AA

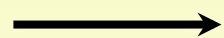
$d_{\text{Li-NBO}} < d_{\text{Li-BO}}$

Higher number of NBO
in the first coordination
sphere of Li in liquids

Glass

MOn

M compensator



Liquid

$(\text{MOn})'$

M modifier

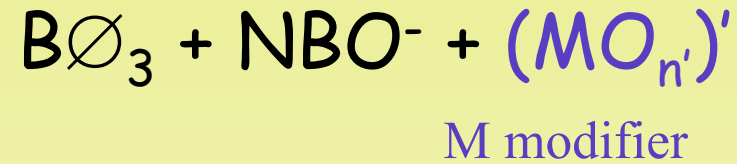
⇒ Modifications of the Li
environment associated with a change
of its structural role in the liquid

Evolution of the Li site between the glass and the liquid

Glass

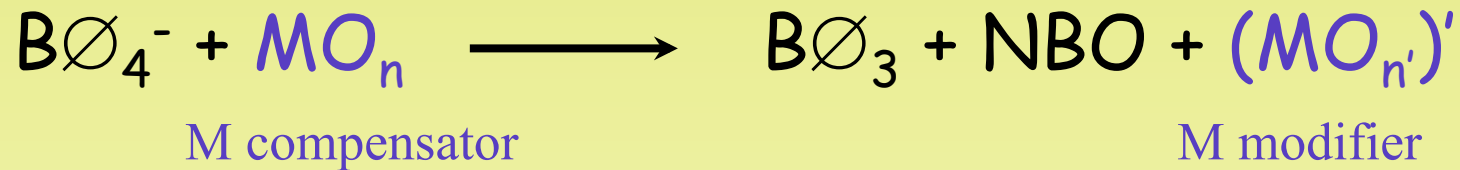


Liquid



⇒ Modifications of the Li environment associated with a change of its structural role in the liquid

Equilibrium of the BO_4/BO_3 conversion in the liquid



Equilibrium constant at temperature i

$$K_i = C \times \frac{X_{\text{BO}_3}^i X_{\text{NBO}}^i}{X_{\text{BO}_4}^i}$$

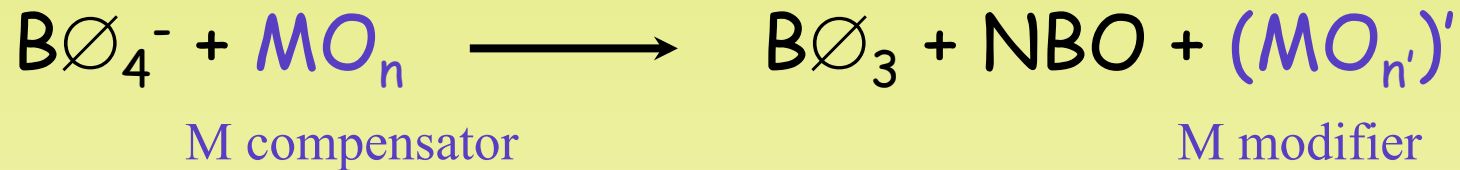
$$X_{\text{BO}_4} = N4$$

$$X_{\text{BO}_3} = 1 - N4$$

$$X_{\text{NBO}} = R - N4$$

C : rapport des coefficients d'activités de chaque espèce, constant avec T

Equilibrium of the BO_4/BO_3 conversion in the liquid



$$K_i = C \times \frac{X_{\text{BO}_3}^i X_{\text{NBO}}^i}{X_{\text{BO}_4}^i}$$

$$X_{\text{BO}_4} = N4$$

$$X_{\text{BO}_3} = 1 - N4$$

$$X_{\text{NBO}} = R - N4$$

Van't Hoff's relation : variation of the equilibrium constant with T and the standard enthalpy difference

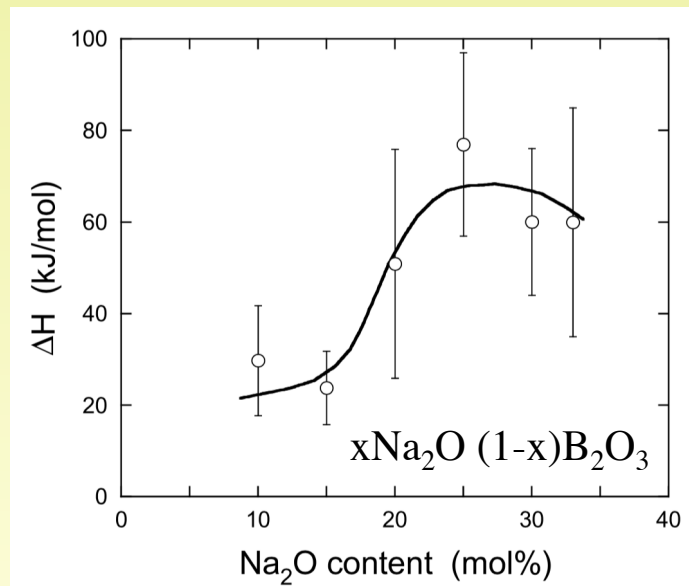
$$\frac{\partial \ln K}{\partial T} = \frac{\partial \left(-\frac{\Delta G^0}{RT} \right)}{\partial T} = \frac{\Delta H^0}{RT^2}$$

Equilibrium of the BO_4/BO_3 conversion in the liquid

Between $T_1=T_g$ and $T_2=T_{\text{liquid}}$

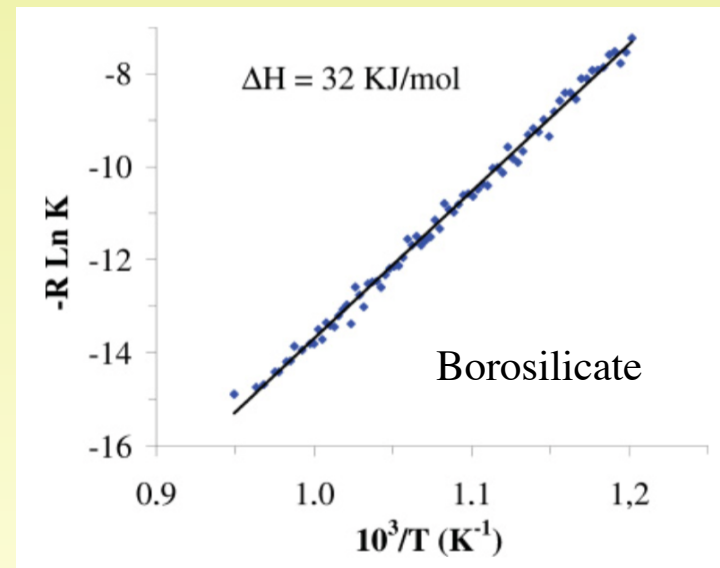
$$\Delta H^0 = -R \frac{\ln K_1 - \ln K_2}{\frac{1}{T_1} - \frac{1}{T_2}}$$

ΔH per mole of B \equiv enthalpy difference if one mole BO_4 converted completely to one mole BO_3



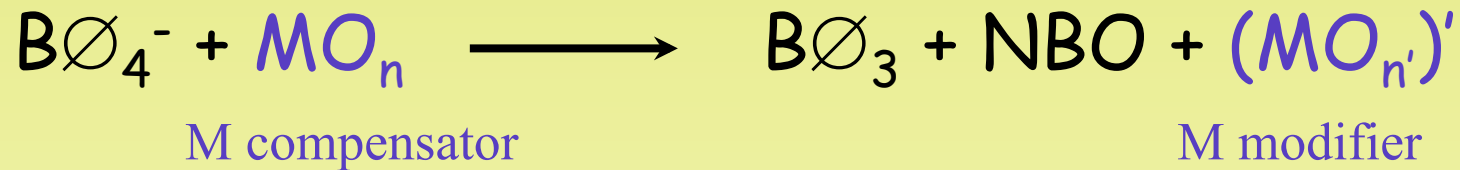
Yano et al. J. Non-Cryst. Solids 321(2003)147

Borosilicate



Angeli et al., Physical Review B, 85(2012)54110

Equilibrium of the BO_4/BO_3 conversion in the liquid



C_p related to the configuration entropy can be estimated

$$C_p = \frac{\Delta H \Delta X}{T_1 - T_2} \quad (\text{J/mol/K})$$

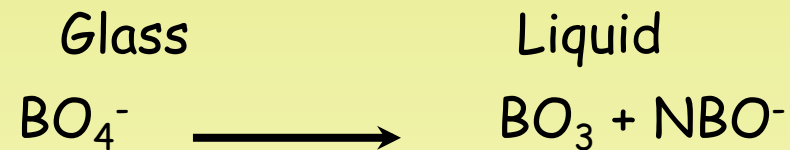
ΔX : number of moles of B converted per mole of glass

Configuration entropy related to structural changes above T_g

⇒ Contribution to the configurational properties ?

B_2O_3 : no coordination change => strong liquid

✓ Boron coordination change



Alkali borate

$$\Delta H = 12-37 \text{ kJ mol}^{-1}$$

ΔC_p of the reaction (loi de Van't Hoff)

$$\Delta C_p \approx 16 \text{ J.mol}^{-1}.\text{K}^{-1} \quad (\Delta C_p^{\text{conf}} = 71 \text{ J.mol}^{-1}.\text{K}^{-1})$$

Cormier et al., J. Am. Ceram. Soc. 89(2006)13

Uhlmann et al., J. Non-Cryst. Solids, 5(1971)426

Borosilicate

$$\Delta H = 32 \text{ kJ mol}^{-1}$$

ΔC_p of the reaction (loi de Van't Hoff)

$$\Delta C_p \approx 2.1 \text{ J.mol}^{-1}.\text{K}^{-1} \quad (\Delta C_p^{\text{conf}} = 8.6 \text{ J.mol}^{-1}.\text{K}^{-1})$$

Angeli et al., Physical Review B, 85(2012)54110

Atelier thermodynamique des verres, 11 octobre 2016

Configuration entropy related to structural changes above T_g

⇒ Contribution to the configurational properties ?

✓ Boroxol ring rupture (*Walrafen et al., J. Chem. Phys., 79(1983)3609*)

$$\Delta H = 27 \text{ kJ mol}^{-1}$$

Both local and medium range changes: two main contributions to the configurational properties but structural changes around Li difficult to estimate